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# The Electric Double Layer Has a Life of Its Own

Céline Merlet<sup>1,2,3</sup>, David Limmer<sup>4</sup>, Mathieu Salanne<sup>1,2</sup>, René van Roij<sup>5</sup>,

Paul A. Madden<sup>6</sup>, David Chandler<sup>7</sup>, and Benjamin Rotenberg<sup>1,2,\*</sup>

<sup>1</sup> Sorbonne Universités, UPMC Univ. Paris 06,  
CNRS, UMR 8234 PHENIX, 75005 Paris, France

<sup>2</sup> Réseau sur le Stockage Electrochimique de l'Energie (RS2E), FR CNRS 3459, France

<sup>3</sup> Department of Chemistry, University of Cambridge, Cambridge CB2 1EW, UK

<sup>4</sup> Princeton Center for Theoretical Science, Princeton, NJ-08544, USA

<sup>5</sup> Institute for Theoretical Physics, Utrecht University, 3584 CE Utrecht, The Netherlands

<sup>6</sup> Department of Materials, University of Oxford, Oxford OX1 3PH, UK

<sup>7</sup> Department of Chemistry, University of California, Berkeley, CA-94720, USA and

\* Corresponding author: benjamin.rotenberg@upmc.fr

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Using molecular dynamics simulations with recently developed importance sampling methods, we show that the differential capacitance of a model ionic liquid based double-layer capacitor exhibits an anomalous dependence on the applied electrical potential. Such behavior is qualitatively incompatible with standard mean-field theories of the electrical double layer, but is consistent with observations made in experiment. The anomalous response results from structural changes induced in the interfacial region of the ionic liquid as it develops a charge density to screen the charge induced on the electrode surface. These structural changes are strongly influenced by the out-of-plane layering of the electrolyte and are multifaceted, including an abrupt local ordering of the ions adsorbed in the plane of the electrode surface, reorientation of molecular ions, and the spontaneous exchange of ions between different layers of the electrolyte close to the electrode surface. The local ordering exhibits signatures of a first-order phase transition, which would indicate a singular charge-density transition in a macroscopic limit.

**Keywords:** Molecular modelling, capacitance, ionic liquids, ordering, structural transition, importance sampling methods

## I. INTRODUCTION

The electric double layer is generally viewed as simply the boundary that interpolates between an electrolyte solution and a metal surface. In contrast, we show here that the interface between an electrode and a model ionic liquid exhibits behaviors distinct from that of the bulk liquid. At a particular voltage, the double layer undergoes an abrupt local ordering transition. The emergent ordered structure is consistent with that of room temperature ionic liquids found on mica surfaces<sup>1</sup>. At another voltage, the interfacial charge density changes abruptly. This change appears to be consistent with observations of anomalous electrical capacitance observed experimentally<sup>2</sup>. Further, we show that this anomaly grows with increasing electrode surface area, suggesting a singular charge-density transition in a macroscopic limit. These nonlinear electrochemical responses, whose molecular origins we elucidate, offer a view of the electric double layer that deviates significantly from standard mean-field models.

Generally, when an electrical potential is applied to an electrode in contact with an electrolyte it induces a local, partial demixing of the cations and anions at the interface. The surface charge is equal and opposite to the net charge in the double layer and thus at a given potential reflects the facility of the interfacial fluid to adopt a structure giving rise to that charge. The canonical model for the interfacial region of an electrolyte at

an electrode surface, the electric double layer, is that of Gouy and Chapman (GC)<sup>3</sup>. This mean-field model applies to a dilute solution of point ions and predicts that the degree of demixing induced by the voltage decreases exponentially with distance from a planar electrode surface and that the differential capacitance of the electrode-electrolyte double-layer increases monotonically with the magnitude of the applied electrode potential. Recently, there has been a great deal of interest in electrochemical interfaces of room temperature ionic liquids (RTILs) driven in part by the potential for constructing supercapacitors that exploit this charge separation to create high power energy storage devices<sup>4</sup>. Far from a collection of noninteracting point charges RTILs are dense mixtures of ions, whose physical properties are largely determined by correlations between ions.

Effects arising from correlations within RTILs at electrochemical interfaces have been observed experimentally and in simulation<sup>5,6</sup>. Specifically, layering has been observed in X-ray reflectivity experiments<sup>7,8</sup>, is detected in surface force and AFM experiments<sup>9-12</sup> and has been examined in a number of computer simulation studies<sup>13-17</sup>. Some of the latter have illustrated how the polarization of the charge density in an ionic liquid in response to an electrode potential is affected by this layering and the strong local coulomb ordering of the ions<sup>13,18</sup>. In-plane ordering of the ions in RTIL has been detected at the liquid-vapor interface by X-ray scattering<sup>19</sup>, on mica surfaces by AFM<sup>1</sup>, and on gold electrodes by STM<sup>20</sup>.

Some simulations<sup>21–24</sup> have reproduced this effect and indicated a connection to changes in the electrode charge density. Using in-situ STM and AFM measurements, Atkin, Endres and co-workers underlined the influence of phenomena such as surface reconstruction (*e.g.* with gold electrodes) and of impurities on the interfacial structure<sup>25,26</sup> and demonstrated the advantage of combining these techniques with electrochemical impedance spectroscopy to investigate these aspects<sup>27</sup>.

Not surprisingly then, the experimentally observed dependence of the differential capacitance on potential has often been found to depart markedly from the predictions of GC theory, often decaying with increasing potential and exhibiting sharp peaks that are interpreted as abrupt changes in the local ionic structuring. Efforts to extend GC theory to include effects from packing of ions have been made at the mean-field level with some success<sup>6</sup>. These account approximately for layering at the interface<sup>28</sup> and the tendency for local coulombic ordering of the ions at high voltage<sup>29–31</sup>. Specifically, Kornyshev and coworkers have demonstrated the effect of these extensions on the differential capacitance, qualitatively reproducing some features observed in RTILs<sup>28,30</sup>. However, correlations in-plane and effects from fluctuations have not been considered, omitting molecular detail and its role in determining the electrochemical properties of these systems. Such features are however essential in conditions where the liquid undergoes large structural reorganization, which as we discuss below, are intimately connected to non-monotonic features of the differential capacitance.

Here we address the correspondence between structure, fluctuations and electrochemical response in RTILs directly by applying novel simulation methods. Such methods are robust and free of assumptions related to the relative importance of correlations that constrain mean-field models. In section II we first describe the considered system and present the theoretical approach to determine the evolution of observables, such as the differential capacitance or structural properties, with applied voltage. Section III then reports the main results of this study.

## II. METHODS

**Microscopic Models** The specific system we consider is a molecular simulation model of liquid butylmethylimidazolium hexafluorophosphate (BMI-PF<sub>6</sub>) bounded by constant voltage graphite electrodes. BMI-PF<sub>6</sub> is an example of a room temperature ionic liquid (RTIL) and has been studied previously<sup>32–34</sup>. Figure 1(a) shows a characteristic snapshot of the model electrochemical cell and Fig. 1(b) shows a characteristic snapshot of the positions of the ions close to the electrode surface with the corresponding induced charges on the electrode surface, which fluctuate as a result of the constant potential constraint. The force field we use to simulate the graphite electrode and the ionic liquid are the same coarse-grained

potentials as used in previous works<sup>32–36</sup>. Similarly, the electrochemical cell is modeled as in previous works<sup>32–34</sup> with each electrode consisting of three fixed graphene layers and subject to two-dimensional periodic boundary conditions<sup>37,38</sup>. The molecular level detail provided by these simulations is evident in Fig. 1(c), where explicit layering within the fluid away from the electrode arises as a consequence of the discrete nature of the material<sup>39</sup>.

**Importance Sampling** In a recent paper, we have introduced a framework to analyze molecular simulation data from constant-electrode potential simulations<sup>41</sup>. We exploit the method of histogram re-weighting<sup>42</sup>, which allows us to study the induced electrode charge as a *continuous* function of the applied potential, rather than by studying it in a series of simulations carried out at discrete values. The re-weighting method also enables us to obtain much better statistical precision than in the previous simulation studies through the utilization of data from multiple simulations. Such techniques are especially necessary when studying collective behavior, as timescales for reorganization make straightforward simulation methods prohibitive.

To generate configurations for this histogram re-weighting technique, molecular dynamics simulations were conducted in the NVE ensemble with a time step of 2 fs. For each simulation, a 200 ps equilibration at  $T = 400$  K is followed by a 10 ns production run from which configurations are sampled every 0.2 ps and the corresponding total charge  $Q_{\text{tot}} = \sigma \times S$  of the electrodes (with  $\sigma$  the surface charge density and  $S$  the surface area) is used to determine the weight of this configuration in the ensemble at an arbitrary potential using the weighted histogram analysis method (WHAM)<sup>42</sup>. Application of WHAM allows us to access statistics that would be unobtainable directly from molecular simulation<sup>41</sup> and to sample the probability distributions of key variables as continuous functions of the applied potential. Simulations were performed for ten potential values ( $\Delta\Psi = 0.0, 0.2, 0.5, 0.75, 1.0, 1.25, 1.5, 1.75, 1.85$  and  $2.0$  V) in order to ensure a good overlap between the histograms for  $Q_{\text{tot}}$ , as required for the histogram re-weighting. The distribution of any property  $A$  is determined as a function of the applied potential from the joint distribution of  $Q_{\text{tot}}$  and  $A$  as

$$P(A|\Delta\Psi) = \int dQ_{\text{tot}} P(Q_{\text{tot}}, A|\Delta\Psi). \quad (1)$$

Specifically, we consider below as observables  $A$ : the surface charge density of the total electrode and that of subsamples of the electrodes, the numbers of anions and cations in the first fluid layers adsorbed on both electrodes, the in-plane structure factors in these layers and the orientation of cations in these layers.

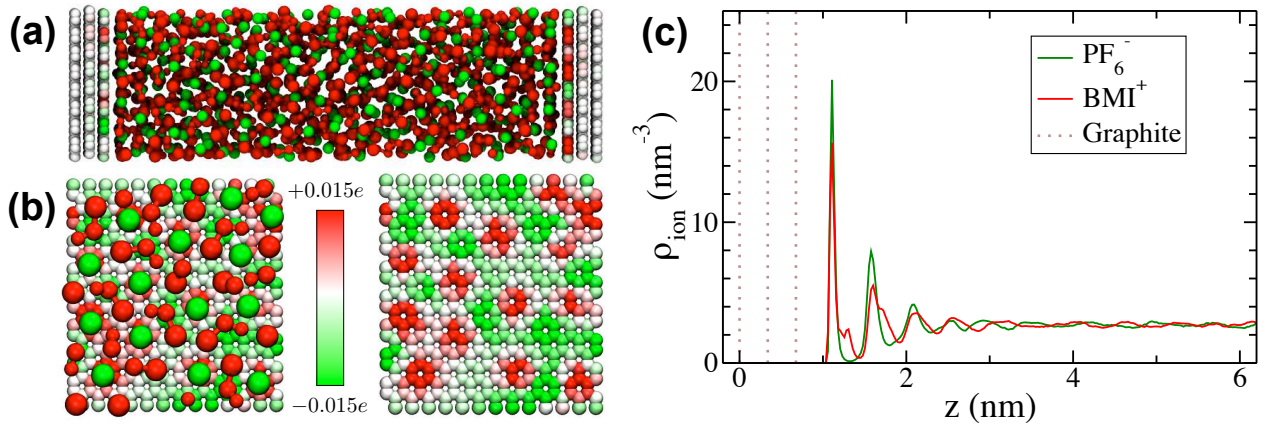


FIG. 1: **Simulated capacitor.** (a) The capacitor consists of liquid butylmethylimidazolium (red) hexafluorophosphate (green) and graphite electrodes (white). The electrode geometry and force field are detailed in Ref.<sup>32</sup>. The force field employs the ion-ion interactions from Roy and Maroncelli<sup>36</sup>, and a constant voltage,  $\Delta\Psi$ , is maintained throughout a molecular dynamics trajectory by adapting the algorithm of Siepmann and Sprik<sup>40</sup>. (b) Polarization of electrode is illustrated with a configuration of the double layer (left), with molecules drawn smaller than space filling, and the underlying graphite electrode, with colors indicative of the instantaneous charge on the carbon atoms. During constant-potential simulations, the charge on carbon atoms fluctuates in response to the thermal motion of the adsorbed liquid. (c) The cation and anion density profiles perpendicular to the electrode surface at  $\Delta\Psi=0\text{V}$ . For the cation the density is plotted for the center of mass.

### III. RESULTS

#### Electrode charge distributions and capacitance

The probability distribution  $P(\sigma|\Delta\Psi)$  of electrode surface charge density  $\sigma$  at various electrode potentials  $\Delta\Psi$  is represented by a contour plot in Fig. 2a. The figure shows the free energy landscape for the system as a function of surface charge density  $\sigma$  and potential  $\Delta\Psi$ , in units of Boltzmann's constant,  $k_B$ , times temperature,  $T$ . It is clear that there are three free energy minima, at  $\Delta\Psi \sim 0.6\text{V}$ ,  $1.2\text{V}$ , and  $1.8\text{V}$ , within the voltage range studied, which correspond to particularly favorable configurations for the interfacial fluid to balance the corresponding surface charge densities of  $\sim 1.0$ ,  $2.6$ ,  $4.1 \mu\text{C}/\text{cm}^2$ . The minima are separated by saddle points, at  $0.9\text{V}$ , for example. A given voltage corresponds to a vertical slice through the free energy surface and Fig. 2b shows sections for three voltages in the  $0.8\text{--}1.0\text{ V}$  range.

The mean-square fluctuations in the electrode surface charge density determine the differential capacitance  $C$  (per unit area) as<sup>43,44</sup>,

$$C \equiv \frac{\partial\langle\sigma\rangle}{\partial\Delta\Psi} = \frac{S}{k_B T} \langle(\delta\sigma)^2\rangle. \quad (2)$$

where  $\delta\sigma = \sigma - \langle\sigma\rangle$ , and the angle brackets denote equilibrium average with fixed voltage  $\Delta\Psi$ . Note that this relation between the response to voltage and the equilibrium fluctuations of the surface charge is not limited to the typical linear response approximation, as both the charge fluctuations and therefore the capacitance are generally potential dependent<sup>41</sup>. As seen in Fig. 3(a),

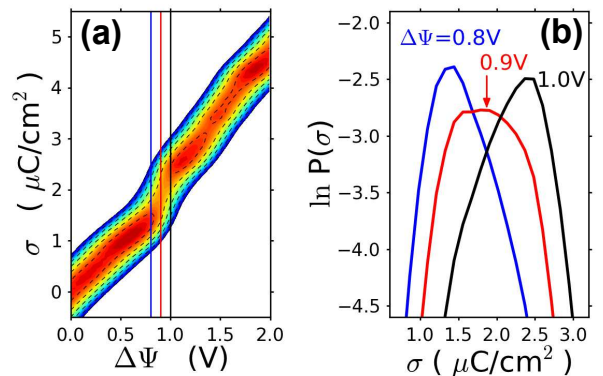


FIG. 2: **Surface charge density distribution.** (a) The probability distribution of the charge density  $\sigma$  on the electrodes is a function of the applied potential  $\Delta\Psi$ . The two-dimensional graph of the distribution employs a logarithmic scale with lines separated by a difference of  $0.5 k_B T$  distribution is graphed as a function of  $\sigma$  in (b). Note the fat tails in the distribution,  $P(\sigma)$ , and note the markedly non-linear shifts with changing voltage.

the capacitance exhibits two anomalous peaks, one near  $\Delta\Psi = 0.9\text{ V}$ , the other near  $\Delta\Psi = 1.5\text{ V}$ . Locating these critical voltages on the free energy surface (Fig. 2(a)) shows that they correspond to the location of saddle points, where fluctuations sample two adjacent free energy minima. Similar features in the differential capacitance have been reported in numerous experimental studies<sup>27,45–49</sup>.

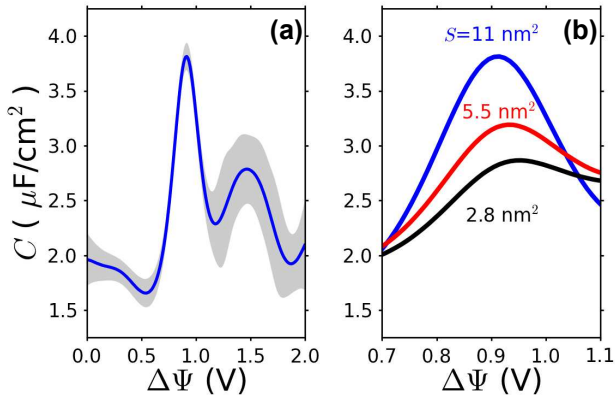


FIG. 3: **Capacitance.** (a) Differential capacitance,  $C$ , computed from molecular simulation as a function of the applied potential  $\Delta\Psi$  with an electrode surface area  $S = 11 \text{ nm}^2$ . Shaded regions indicate the range of statistical uncertainty. (b) Growth of computed capacitance peak with growth of electrode surface area.

The peaks in capacitance arise from correlations within the interfacial layer of fluid. This is evident from the electrode surface charge distributions that exhibit non-Gaussian features characteristic of a first-order phase transition – fat tails at conditions away from phase coexistence and a very flat distribution suggesting incipient bi-modality (limited here by a rather small system size) at conditions of coexistence<sup>50,51</sup>. If the relevant structures in the fluid comprise correlated domains that are small compared to the net surface area, the distribution will be Gaussian because the net surface-charge density will reflect many uncorrelated contributions (the usual Central Limit Theorem argument). Generally, the distribution,  $P(\sigma)$ , shifts according to applied voltage,

$$P(\sigma|\Delta\Psi) \propto P(\sigma|0) \exp(-\sigma S \Delta\Psi/k_B T). \quad (3)$$

In the Gaussian case, vertical sections through the free energy surface,  $-k_B T \ln P(\sigma)$ , will be parabolas with minima at the mean value of  $\sigma$ , and the application of voltage will produce a proportional shift of that mean – the free energy surface would have the form of a regular valley with the position of the minimum tracking the voltage smoothly. This is what has been found previously for simple dielectrics or dilute solutions<sup>41</sup>. If the correlated domains extend across the entire surface, the distribution will be non-Gaussian, and the application of voltage will shift the free energy minimum in a markedly non-linear way as different domain types are stabilized by different voltages. It is clear that the surface-charge distributions found in the simulations correspond most closely to the latter scenario and that, although the non-Gaussian characteristics are a global property of the free energy surface, they will be most pronounced at potentials close to 0.9V.

The size of our system is small, so that the non-Gaussian features are relatively subtle. Nevertheless, the system is large enough to exhibit size dependence of response. In particular, Fig. 3(b) illustrates the growth of the first  $\sim 0.9\text{V}$  of the anomalous peaks in the differential capacitance as the observed electrode surface area is increased. In the absence of a phase transition, the capacitance  $C$  is an intensive property, *i.e.*,  $\langle(\delta\sigma)^2\rangle$  scales inversely with  $S$ . At conditions of a coexistence between two phases, however,  $C$  will grow with system size because cooperativeness at a phase transition extends across the surface. The growth is significant. Ultimately, the capacitance at this voltage should increase linearly with  $S$ , but the system sizes we have been able to study are not yet large enough to reach that scaling regime. The fluctuations we consider might thus foreshadow an unbounded capacitance, a phenomenon that it not predicted by previous theories even beyond mean-field<sup>31</sup>.

While studying larger system sizes in detail is not possible with the computer resources currently available to us, we can nevertheless test the consequences of a putative first order transition. For larger systems, one should expect a bimodal distribution of surface charge density with basins of the charge distribution separated by a free energy barrier growing with system size. We have performed simulations with a system size increased by 100% in the  $x$  and  $y$  directions for voltages close to the estimated coexistence voltage of 0.9V. For these large systems, both the ordered and disordered states are metastable over time scales of 500 ps, and the corresponding charge distributions do not overlap. This behavior is consistent with the expected one for a first order transition. It raises the usual issue for studying phase transitions in molecular simulations: proper sampling requiring long simulation times and enhanced sampling techniques. Such an investigation could benefit from the development of simplified models and analytical theory capable of capturing the physical phenomena which underpin these effects. With these caveats in mind, note that such a transition would also lead to hysteresis in a macroscopic system, and that hysteresis has been observed experimentally for a very similar ionic liquid on graphene using X-ray reflectivity<sup>8</sup>.

The charge density  $\sigma$  is the same on both electrodes (up to a sign change). Its average value and its variance (hence the capacitance, see Eq. 2) reflect the response of the whole capacitor. In order to provide a microscopic interpretation for the observations described above, we have examined several order parameters based upon averages over the ionic positions at each electrode. Given the difference in shape and size between cations and anions, the microscopic structure of the interface is likely to differ at positive and negatively charged electrodes. In the following, results are reported as a function of the variable  $\Psi$  defined as  $+\Delta\Psi/2$  (resp.  $-\Delta\Psi/2$ ) for the positive (resp. negative) electrode when the capacitor is submitted to a voltage  $\Delta\Psi$ . Since in the present system the point of zero charge is close to 0 V, with comparable

capacitances for both interfaces<sup>32</sup>,  $\Psi$  approximately reflects the actual potential drop across each double layer.

**Out-of-plane layering** Figure 1(c) illustrates the density profiles for anions and cations at zero applied potential. Examples of the density profiles at higher potentials for this system have been shown elsewhere<sup>52</sup> and layering similar to that indicated in the figure persists over the full range of potentials studied here. Although the layering is a generic property of dense fluids, the arrangement of the ions within the layers depends on the details of the interionic interactions. In the present case, at 0V, the maxima in the principal peaks of the anion and cation densities coincide, which shows that both ions have roughly the same size and may approach the surface equally closely (the small shoulder appearing on the cation peaks arises from different possible orientations of the cations on the surface). If the ions were of very different size, the cation and anion layers could be displaced from each other, as seen *e.g.* in Ref.<sup>53</sup>. Furthermore, from the areas of corresponding cation and anion peaks for the first two layers it follows that a given layer contains a very similar number of cations and anions (a feature we explore in detail below). This will be favored (at low electrode surface charge, and in the absence of strong attractive interactions of one species with the surface) by the coulombic interactions between the ions which will try to impose local charge neutrality.

**In-plane order** The in-plane anion structure factor for the anions in the first adsorbed layer on a given electrode is

$$S(\mathbf{k}) = \frac{1}{N_-} \sum_{i,j} e^{i\mathbf{k}\cdot\mathbf{r}_{ij}} , \quad (4)$$

where the sum runs over pairs of anions that are found in the first layer of fluid adjacent to the electrode surface and  $\mathbf{k}$  is a wavevector which lies in the plane of the electrode and is commensurate with the periodic boundary conditions. These wavevectors form a discrete lattice and in Fig. 4a we have plotted the average value  $\langle S \rangle$  as a function of  $|\mathbf{k}|$  for two states visited along a trajectory at  $\Delta\Psi = 0V$ . At a given voltage, the interface may explore configurations with varying degrees of in-plane order. The bottom part corresponds to a fluid-like configuration, whereas the top one shows a strong, Bragg-like peak at  $k^* \simeq 0.8 \text{ \AA}^{-1}$ , which corresponds to  $2\pi$  divided by the mean nearest-neighbor separation of the anions in the first layer, and strongly suggests a 2-d lattice-like organization for the anions.

Figures 4b and 4c show snapshots which illustrate such ordered and disordered structures, respectively. The relative contribution of these structures to an averaged property of the interfacial fluid, like the in-plane structure factor, depends on the voltage, as we shall see below. In particular, the lattice-like ordering only persists for a certain range of applied potentials.

The length scales of the underlying graphite play no role in this ordered structure. Indeed, structures like it are inferred from experiments of monolayers of BMI-PF<sub>6</sub>

on gold<sup>20</sup>. This is not to say that other systems cannot show an effect of commensurability between in-plane order in the electrolyte and the surface<sup>23,54</sup>. But the role of graphite in the system studied here is only that of a polarizable metal. Note also that the geometry of the triangular lattice is distinct from that of the (nearly) square electrode that is periodically replicated in the simulation. Thus, the ordered state we find in the simulation is not a finite-size artifact of the simulation geometry.

In Fig. 4d we show a contour plot of the probability distribution of values of  $S(k^*)$  as a function of applied potential. This is again plotted as  $-k_B T \ln P$  so that the surface can be interpreted as a free energy landscape in the structural order parameter  $S(k^*)$ . This plot reveals a substantial basin of stability for the ordered state of the first interfacial layer of fluid for potential differences  $\Delta\Psi$  between about 0.5 and 0.75V (remember that  $\Psi = \pm\Delta\Psi/2$ ). This basin is present on both electrodes, even though the ordered state is less probable on the negative electrode. At both higher and lower voltages the fluid layer is translationally disordered. This observation strongly suggests that the peak in the differential capacitance observed as the potential is increased through 0.9V is associated with the transition between an ordered, lattice-like state of the first layer and the disordered state. The first-order nature of this phase transition, which is evident from free energy surface in  $S(k^*)$ , shows up in the statistics of the electrode surface-charge fluctuations and the resulting peak in the differential capacitance. However, the transition from the low potential, disordered state to the ordered state, which might be expected at about  $\Delta\Psi = 0.35V$ , does not appear to affect the surface charge significantly. Although a transition in first-layer structure is seen in  $S(k)$ , it does not appear to be strongly coupled to the surface charge.

In the present case, the transitions on both electrodes occur approximately at the same voltage  $\Delta\Psi$ . For systems where the difference in size and shape between cations and anions is stronger, or where the fine structure of the electrode plays a more important role, such transitions may occur for different voltages on each electrode. For example, previous work on electrochemical cells involving a molten salt, LiCl, and (100) aluminum electrodes, an order-disorder transition occurred in the first adsorbed layer close to the potential of zero charge<sup>23</sup>. For positive surface charges, an ordered structure, commensurate to the underlying metal structure was stabilized while the structure was liquid-like otherwise. As a result, a large step was observed on the charge vs. potential plot, which should be in principle associated to a large peak in the differential capacitance. Such a peak could not be evidenced since the importance sampling tools discussed here had not been introduced at that time. The asymmetric behavior could be attributed in that case to the large difference in polarizability between the lithium and chloride ions. As soon as the polarization effects (on the ions) were turned off, the ordered structure was unable to form, leading to an almost constant differential capac-

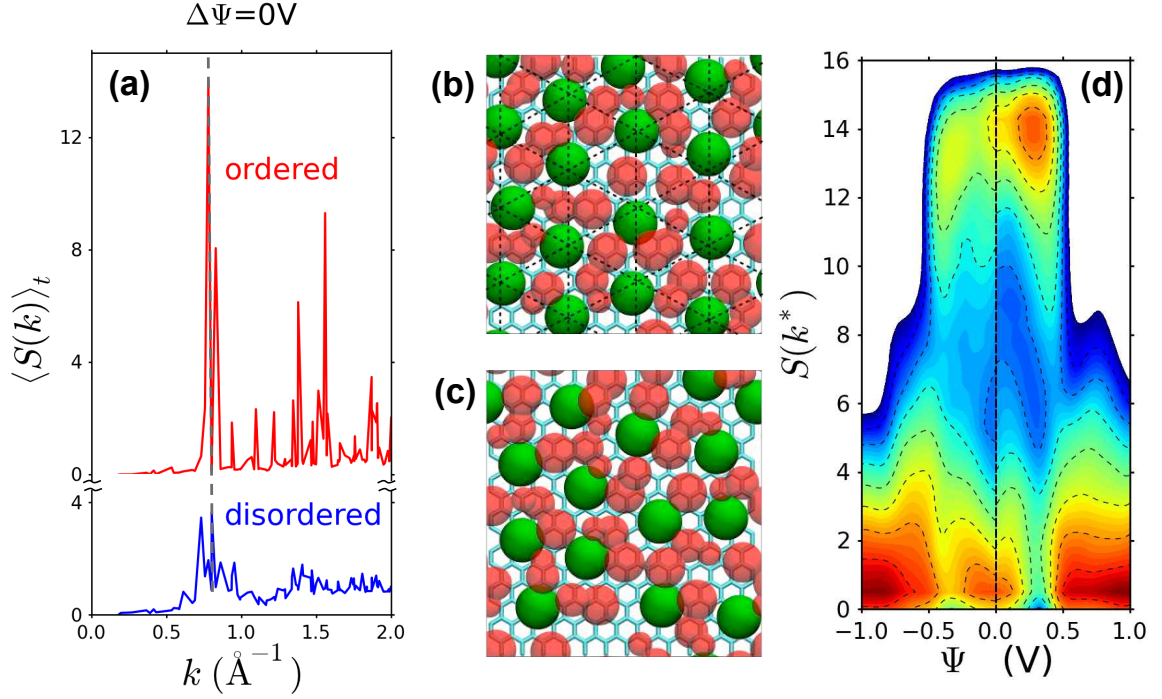


FIG. 4: **In-plane structure of the adsorbed layer** (a) Average anion-anion structure factor  $\langle S(k) \rangle_t$  in the first adsorbed layer on the electrodes, as a function of the norm of the wave vector, for a voltage  $\Delta\Psi = 0\text{V}$ . Structure factors for ordered and disordered states, which are both observed at this particular voltage, are averaged over 1 ns. (b) Configuration illustrating the adsorbed layer in the ordered state. (c) Configuration of a disordered state. (d) Probability distribution of the anion-anion structure factor at the maximum  $S(k^*)$  in the first adsorbed layer on the electrodes, as a function of  $\Psi$ , defined as  $+\Delta\Psi/2$  (resp.  $-\Delta\Psi/2$ ) for the positive (resp. negative) electrode when the capacitor is submitted to a voltage  $\Delta\Psi$ . The probability is reported on a logarithmic scale, with lines separated by a difference of 0.5. Note the presence, on both electrodes, of three distinct basins, indicative of i. a low-voltage phase where the double layer is disordered, ii. an intermediate-voltage phase where the double layer is ordered, and iii. a high-voltage disordered phase where the double layer is charge unbalanced.

itance over the whole simulated potential range<sup>23</sup>.

**Cation orientation and stoichiometry of the layers** The orientation of cations in the first ionic layer on the electrode evolves with the applied potential  $\Delta\Psi$ . This can be measured using angles such as the one between the imidazolium ring-butyl (C1-C3) axis and the normal to the surface, illustrated in Fig. 5(a). The distributions of these angles as a function of  $\Psi$  are reported in Figs. 5(b), 5(c) and 5(d). On both electrodes, the C1-C3 axis remains almost parallel to the surface for all voltages. On the contrary, the distribution of the ring-methyl (C1-C2) orientation displays bimodality, with relative weights of the two configurations that evolve with the applied voltage. The cation may either lie flat on the surface ( $\theta_{12} \sim 90^\circ$ ) or tilt with the methyl group pointing outward ( $\theta_{12} \sim 30^\circ$ ) while the imidazolium ring and butyl group remain on the surface ( $\theta_{13} \sim 90^\circ$ ). At 0 V, both orientations have similar probabilities. With increasing voltage, the fraction of tilted cations increases on the positive electrode and decreases on the negative one. This behavior is consistent with the electrostatic interaction between the positive partial charge of the methyl group

and the electrodes. The fact that the bulkier butyl and imidazolium ring remain closer to the surface may be due to packing effects, their smaller charge density compared to the methyl group and to the stronger attractive dispersion interaction with the wall, in agreement with observations for a model cation consisting of a charged head and a neutral tail<sup>55</sup>.

The value of the electrode surface charge at a given potential reflects the spatial distribution of the induced charge density in the interfacial layer of electrolyte. In order to relate this to the structural changes in the fluid, we have plotted in Fig. 6 the probability distributions for finding a given number of anions ( $N_-$ ) and cations ( $N_+$ ) in the first and second layers of fluid. This is plotted as  $-k_B T \ln P(N)$  so that it can be interpreted as a free-energy landscape. The most striking features of these plots are the pronounced changes in the most probable number of anions in the first layer at both anode and cathode that occur as the voltage is increased. The preferred numbers of both species in the second layer remains roughly constant, at least until the overall cell voltage ( $\Delta\Psi$ ) exceeds 1.3V - corresponding to  $\Psi \sim 0.65\text{V}$ .

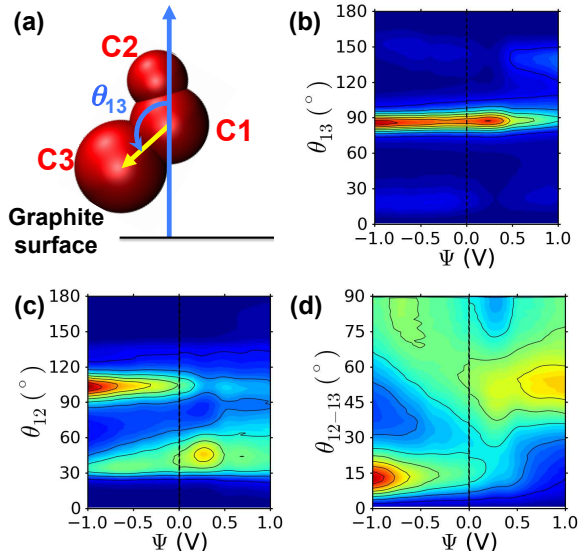


FIG. 5: **Orientation of cations on the electrodes.** (a) Definition of the angle  $\theta_{13}$  between the imidazolium-butyl axis and the normal to the graphite surface (into the fluid). The space-filling representation of the cation illustrates the three sites defined in the force field to model the imidazolium ring (C1) as well as the butyl (C3) and methyl (C2) side chains. The  $\theta_{12}$  angle is defined similarly, while the  $\theta_{12-13}$  angle corresponds to the angle with the normal to the molecular plane. (b) Probability distribution of the angle  $\theta_{13}$  as a function of  $\Psi$ , defined as  $+\Delta\Psi/2$  (resp.  $-\Delta\Psi/2$ ) for the positive (resp. negative) electrode when the capacitor is submitted to a voltage  $\Delta\Psi$ . The probability is reported on a logarithmic scale, with lines separated by a difference of 0.5. (c) and (d) Same as (b) for the angle  $\theta_{12}$  and  $\theta_{12-13}$ .

The preferred cation number in the first layer remains constant until  $\Delta\Psi$  reaches 1.0V and there is initially, close to zero volts, an apparent charge imbalance in the first layer because the number of cations exceeds that of anions.

When the voltage  $\Delta\Psi$  exceeds about 0.35V the number of anions at both electrodes increases so that it balances the number of cations. This occurs because cations become increasingly orientationally ordered by larger applied potentials and this allows enough space for the additional anions to enter the first layer. As the ions become tightly packed in the first layer, the structure factor develops its Bragg-like peak (Fig. 4) showing that the lattice-like arrangement is associated with the ideal stoichiometry. Note that the capacitance (Fig. 3) is a decreasing function of potential between zero and about 0.6V, suggesting that the interfacial fluid becomes increasingly resistant to charging over this range. At  $\Psi$  about 0.5V, there is a sharp increase, over a narrow voltage window, in the number of anions at the positive electrode, which is partially compensated by a smaller decrease in the number of cations. At the cathode, there is a more extended transition at about  $\Psi \sim -0.5V$  which

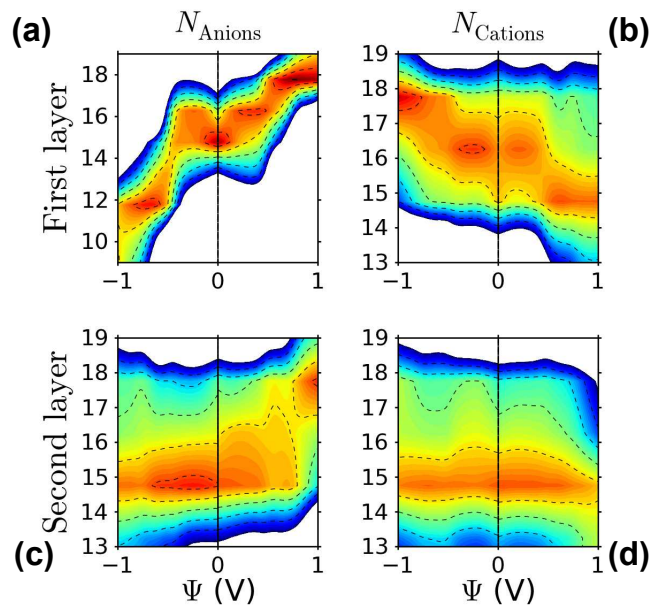


FIG. 6: **Number of ions in the adsorbed layers** Probability distribution of the number of anions  $N_{\text{Anions}} = N_-$  (a,c) and cations  $N_{\text{Cations}} = N_+$  (b,d) in the first (a,b) and second (c,d) fluid layers on both electrodes, as a function of  $\Psi$ , defined as  $+\Delta\Psi/2$  (resp.  $-\Delta\Psi/2$ ) for the positive (resp. negative) electrode when the capacitor is submitted to a voltage  $\Delta\Psi$ . The distribution is reported on a logarithmic scale, with lines separated by a difference of 0.5.

results in still larger changes in the numbers of both anions and cations. These rapid increases in the magnitude of the interfacial charge close to the electrode surface are responsible for the peak in the capacitance close to an overall cell voltage drop of  $\Delta\Psi = 0.9V$ .

From Figs. 4 and 6 it is clear that the nature of the structural transitions in the first layer that are responsible for the capacitance peak at 0.9V are collective transitions between stoichiometric ( $N_+ = N_-$ ), lattice-like ordered states and non-stoichiometric, or partially demixed, disordered states with the latter persisting to higher potentials. A possible interpretation is that the increasing magnitude of the potential drives the like-charged ions out of the first-layer and causes the melting to the non-stoichiometric disordered phase. Notice that the capacitance peak is associated with the *transition* out of the ordered state, rather than the existence of the ordered state itself: from the free energy landscape in Fig. 4 it is clear that the lattice structure is most stable for a voltage  $\Delta\Psi$  of about 0.6-0.75V, substantially lower than that at which the capacitance peak is observed.

At higher potentials, in particular near to the second peak in the capacitance at  $\Delta\Psi \sim 1.45V$  (Fig. 3), the structure factor (Fig. 4) suggests that the interfacial fluid layers are laterally disordered. The ion numbers in the first adsorbed layer on both electrodes are almost unchanged



close to this voltage (Fig. 6). Only in the second adsorbed layer can changes be observed, in particular for the number of anions on the positive electrode, but it is difficult to draw definitive conclusions from our data on the microscopic mechanisms underlying the second peak in capacitance.

#### IV. CONCLUSION

The present work illustrates that the interface between an ionic liquid and a metal electrode can exhibit structures and fluctuations that are not simple reflections of surrounding bulk materials. This rich behavior is absent from a mean-field picture that averages over intraplanar structure. The charge of the electrode is screened by the interfacial fluid and induces subtle changes in its structure. In the present case with ions of comparable sizes but different shapes, small voltages induce changes in the orientation of the anisotropic cations which differ on the two electrodes. In turn, the effective lateral density changes, allowing for the in-plane ordering of the anions and cations in the first adsorbed layer. This in plane order is analogous to that found in experiment<sup>1</sup>. The transition out of this ordered state into a disordered non-stoichiometric state is associated with a large change in the charge distribution across the interfacial region of the electrolyte and leads to a peak in capacitance. While a similar peak in the capacitance had been also noted

in experiment<sup>2</sup>, its origin had been debated<sup>27,56</sup>. This work offers a microscopic explanation, free from the experimental difficulties such as surface reconstruction and defects that plague such experimental studies. Moreover, the presence of ordered states like those we describe here may also be at the origin of the slow capacitive process that have been observed when recording broadband capacitance spectra<sup>27,48</sup>. The structural changes are likely to also induce different mechanical responses. The approach developed in the present work should thus also provide insights into microscopic mechanism for voltage-dependent lubricants<sup>57</sup> and RTIL-based actuators<sup>58</sup>.

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