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Supported and Inserted Monomeric Niobium Oxide Species on/in Silica: A Molecular Picture

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Abstract

The geometry, energetic, and spectroscopic properties of molecular structures of silica-supported niobium oxide catalysts are studied using periodic density functional calculations (DFT) and compared with experimental data. The calculations are done for Nb oxide species inserted or grafted in/on an amorphous hydroxylated silica surface. Different positions of the Nb atom/atoms in the silica structure have been investigated. By means of DFT calculations the geometry and the degree of hydration of Nb oxide species with oxidation state +5 have been studied. The local Nb geometry depends on different parameters such as: the number of Nb-O-Si groups vs Nb-O-H groups; the H bonds formation and the distance between Nb atoms. The interaction between the oxide and silanol groups occurs by formation of Si-O-Nb bonds with chemically and thermally stable Brønsted and Lewis acid sites. UV-vis, Reflection absorption infrared vibrational spectra (RAIRS) as well as various thermodynamic properties have also been investigated in order to get a better insight into the system studied and to provide support to possible experimental studies.

Keywords: Niobium, mesoporous molecular sieves, Ab Initio periodic DFT.

Introduction

Niobium compounds and materials are important catalysts for various reactions¹⁻⁷. They remarkably enhance catalytic activity and prolong catalyst life when small amounts are added to known catalysts. Generally speaking, niobium oxide and mixed oxides including: Nb₂O₅-SiO₂; Nb₂O₅-Al₂O₃, Nb-zeolite, mesoporous Nb-MCM-41, Nb-layer compounds, etc. have redox properties, acidic properties and/or photosensitivities.^{7, 8}

The characterization of the niobium phase on the silica surface has also been the subject of several studies.⁹⁻¹⁵ Niobium can form layers of Nb₂O₅, islands (clusters), oligomers, dimers and monomers on silica.

The synthesis and characterization of niobium(V) oxide grafted on a silica gel surface have been of great interest over the last few years.¹⁶⁻²² Niobia supported catalysts exhibit different properties compared to the bulk compound.^{7, 18} The experimental preparation method, niobia concentration, and thermal treatments may change interactions between niobia and silica.¹⁸ It has been reported that the deposited monolayer phase is stable at elevated temperature (T = 873 K). The Lewis acid sites of the surface are also considerably more stable than those found in the bulk phase.^{16, 23, 24}

The interaction of niobium oxide with other oxides to form a dispersed surface phase has led to the formation of composite oxides with unusual structural and acidic characteristics, which are important as catalysts or catalytic supports. In particular photocatalytic activity has been reported for silica-supported niobium and layered niobate catalysts.^{17, 25, 26}

The interaction between the oxide and silanol groups occurs by formation of Si-O-Nb bonds with chemically and thermally stable Brønsted and Lewis acid sites. These surface acid sites exhibit a great affinity for organic molecules which have carboxylic groups in their structure, forming stable Nb-OOC- bonds. They also play an important role in the interaction with hydrogen peroxide towards the formation of peroxy and superoxy active species²⁷. The metal complexes used in synthesis lead to a well-defined surface. The catalytic behavior changed depending on the structure.^{18, 28}

The dispersion of niobia over large surface area supports favors the enhancement of metal active sites. However, the addition of niobia onto the support decreased the total surface area but also provided less irregularity.²⁹

At low loadings silica supported niobium show monomeric Nb oxide species difficult to characterize due to the spectral overlap of the Nb-O and Si-O vibrations.^{30, 31} In order to shed some light on this complex situations quantum chemical calculations are an adequate investigation tool, as has been proved by our group on similar systems.³²⁻³⁷

An important obstacle to model the silica support is its amorphous structure. Amorphous silica exhibits a diversity of sites, especially in hydrous conditions: there are terminal and geminal silanols, and diversity in H bond connectivity. Moreover, a reliable model needs the presence of the typical siloxane ring size distribution in order to account for the coupled movement of silicon atoms. A representative first principles based model has been proposed earlier for the hydroxylated amorphous silica surface³² where the silanol density and the geminal/terminal ratio is that found on real samples, and accounts for the presence of different types of siloxane rings.¹⁸ This model is used in the present work to explore the grafting of a niobium precursor to the silica surface.

In the present study, a systematic theoretical investigation on molecular models of silica supported niobium oxide is performed. The nature of the niobia species is investigated from an inspection of the optimized geometry and its corresponding total energy. An atomistic thermodynamic model is used to account the effects of temperature on the free energy of the surfaces. This allows estimating the relative stability of the different species as a function of the temperature and hydration conditions. Finally, vibrational spectra are calculated and carefully compared with experimental results.

Methodology

Computational details

All geometry optimizations are performed using *ab initio* plane-wave pseudopotential approach as implemented in VASP^{38, 39}. The Perdew-Burke-Ernzerhof (PBE) functional^{40, 41} has been chosen to perform the periodic DFT calculations with an accuracy on the overall convergence tested elsewhere^{33, 34, 42-44}. The valence electrons are treated explicitly and their interactions with the ionic cores are described by the Projector Augmented-Wave method (PAW)^{45, 46}, which allows to use a low energy cut off equal to 500 eV for the plane-wave basis. The Gamma point is used in the Brillouin-zone integration. The positions of all the atoms in the super cell are relaxed until the

total energy differences decrease below 10^{-4} eV. The atom positions as well as the unit cell have been relaxed.

Vibrational spectra have been calculated for selected surface species within the harmonic approximation. Only the niobium center and its first and second neighbors (O-Si and OH groups) were allowed to move, the support was kept fixed. The Hessian matrix is computed by the finite difference method followed by a diagonalization procedure. The eigenvalues of the resulting matrix lead to the frequency values. The assignment of the vibrational modes is done by inspection of the corresponding eigenvectors.

Model description

The silica model structure is described and characterized in our previous work⁴⁴, (**Fig. 1**), and has been used in the study of vanadium oxide³², chromium oxide^{35, 37}, molybdenum oxide³⁶ and gold^{33, 34} grafting on silica. The niobium precursor is modeled by a $\text{NbO}(\text{OH})_3$ unit. This species is added to the silica unit cell (dimensions $12.77 \times 17.64 \times 25.17 \text{ \AA}^3$) resulting in a niobium coverage of $0.44 \text{ atoms nm}^{-2}$. The coverage is used to confront the results obtained for the high coverage Nb sample used in the work of Burcham et al.¹³ ($0.33 \text{ atoms nm}^{-2}$) and to the 10% Nb loaded MCM-41-sample of Gao et al.³¹ ($0.26 \text{ atoms nm}^{-2}$).

In the cases when two Nb atoms are inserted on the silica surface, the unit cell dimensions were doubled in order to keep the Nb loading constant: $12.47 \times 34.93 \times 24.52 \text{ \AA}^3$.

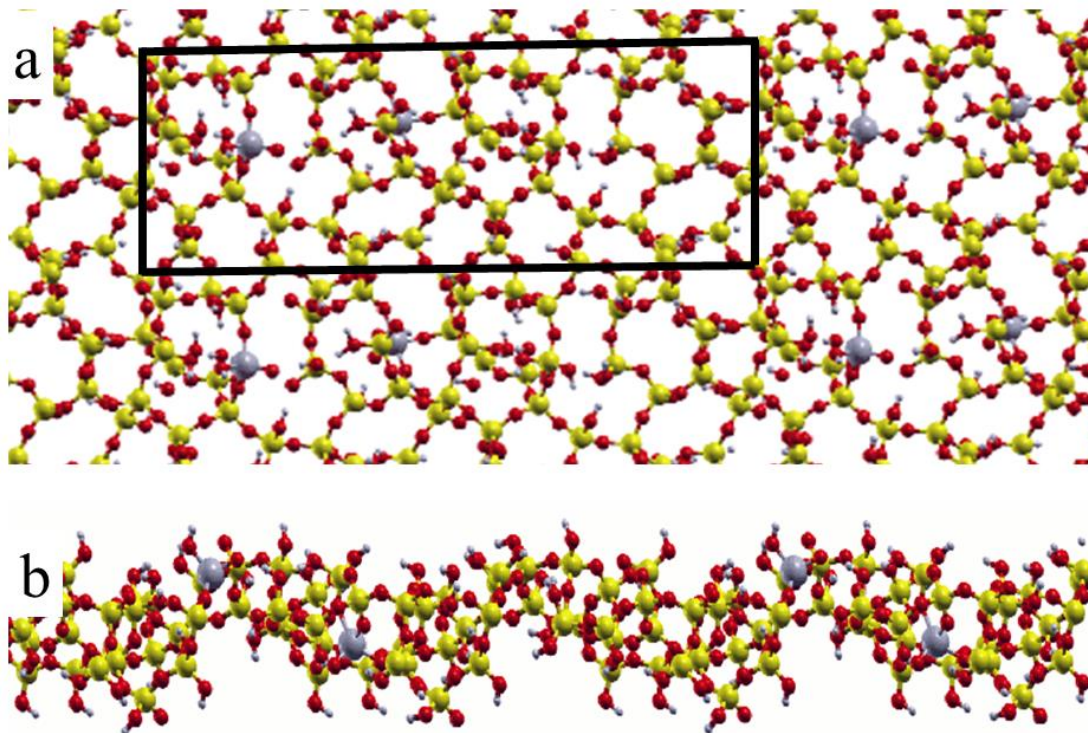
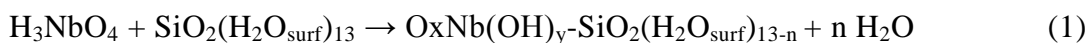


Figure 1. Model for the amorphous silica surface used in this work, for two supported Nb oxide species grafted onto silica surface showing the double unit cell dimensions, a: top view and b: side view of Structure 1. Color code: H: white; O: light red; Si: light orange; Nb: light gray.

The model precursor H_3NbO_4 is grafted by dehydration of surface silanols, following the reaction:



where n is the number of water molecules eliminated and varies between 1 and 3, x is the number of Nb=O groups and, y is the number of Nb-OH groups, with the condition that $x + y = 4 - n$, due to the oxidation state of niobium. The oxidation state of Nb is in our models always equal to +5, and is not mentioned further in the text. Theoretically up

to four silanols may be involved in the reaction yielding different modes of grafting: mono, di, tri and tetra.

Several geometrical configurations are systematically investigated. Structures involving the different silanol types: isolated (Si-OH), vicinal (HO-Si-O-Si-OH), geminal (HO-Si-OH) and isolated (two Si-OH groups not directly connected), on the surface were considered. Due to the flexibility of the silica surface, these species can be easily accommodated, and only the most important and interesting are discussed here.

Experimental conditions

Preparation of materials

Mesostructured cellular foam (MCF) was prepared by one-pot synthesis method according to Schmidt-Winkel et al.⁴⁷ and Margolese et al.⁴⁸, whereas NbMCF was obtained *via* the modified method shown in ^{49, 50}. Firstly, Pluronic P123 (poly(ethylene glycol)-block-poly(propylene glycol)-block-poly(ethylene glycol)-block) (8 g, 1.4 mmol) was dissolved in 300 cm³ of 0.7 M HCl solution at 308-313 K. Secondly, 1,3,5-trimethylbenzene (Aldrich) (4g, 33.28 mmol) and NH₄F (Aldrich) (0.0934 g, 2.52 mmol) were added under vigorous stirring. Following 1 h of stirring, TEOS (Fluka) (17.054 g, 81.99 mmol) was added. The final mixture was stirred at 308-313 K for 20 h and then transferred into a polypropylene bottle (500 cm³) and heated at 373 K under static conditions for 24 h. The solid product was recovered by filtration, washed with distilled water and dried at room temperature. The template was removed from the as-synthesized material by calcination at 773 K for 8 h under static conditions. For the preparation of NbMCF (in one pot method – isomorphous Nb substitution) ammonium niobate(V) oxalate hydrate (C₄H₄NNbO₉ (Aldrich)) was also added to the gel while keeping Si/Nb molar ratio = 30 (10 min after TEOS admission).

Niobium was also included by the wetness impregnation of MCF (postsynthesis grafting; 5 wt% of Nb) with ammonium niobate(V) oxalate hydrate. The obtained material is labelled as Nb/MCF.

Niobosilicate used as pillars for MCM-22 zeolite (obtained according to ⁵¹) was prepared by mixing of TEOS and niobium(V) ethoxide (Aldrich) with the swelled (by

hexadecyltrimethylammonium chloride) zeolite (MCM-22 : TEOS : Nb-ethoxide = 1 : 5 : 0.19). The mixture was stirred at 363 K for 24 h, filtered and washed with distilled water, dried at 333 K and finally calcined at 813 K for 5 h.

UV–Vis Spectroscopy

UV–Vis spectra were recorded on a Varian-Cary 300 Scan UV–visible spectrophotometer with an integrated sphere CA-30I. Catalysts, dehydrated at 673 K for 2 h, in the form of powders were placed into a cell equipped with a quartz window. The Kubelka-Munk function (F(R)) was used to convert reflectance measurements into equivalent absorption spectra using the reflectance of SPECTRALON as a reference.

Infrared measurements

Infrared spectra were recorded with a Bruker Vector 22 FTIR spectrometer using an in situ cell. Samples were pressed under low pressure into a thin wafers of ca. 8 mg cm⁻² and placed inside the vacuum cell. Catalysts were evacuated at 623 K during 2 h.

Results and Discussion

Geometry and Energetics

Nb(V)oxide species grafted on SiO₂ surface

Nb oxide can be grafted on different sites on the silica surface. A detailed study of the all the possibilities has been performed for Mo oxide³⁶. From this study it was concluded that the selected site, also used for V and Cr, is representative as a mean environment at the surface, with the possibilities to graft the oxide on up to three silanol groups. The adsorption process on Silica of other transition metal precursors such as for Ni has been investigated as well.^{52, 53}

Figure 2 shows (structure: A) mono-, (structure: B) di-, (structure: C and D) tri- and (structure: E) tetra-grafted niobium oxide species on silica support. For mono- and di-grafted niobia, the vicinal silanol sites are preferred, in agreement with earlier studies on

the reactivity of silanol groups.³⁶ Tri- and tetra-grafted species need the presence of three and four neighboring silanol sites, respectively. The preference of mono-grafted species to the vicinal HO-Si-O-Si-OH sites may be due to the formation of H-bonds between the Nb-OH groups and surface silanol groups which stabilize the conformations.

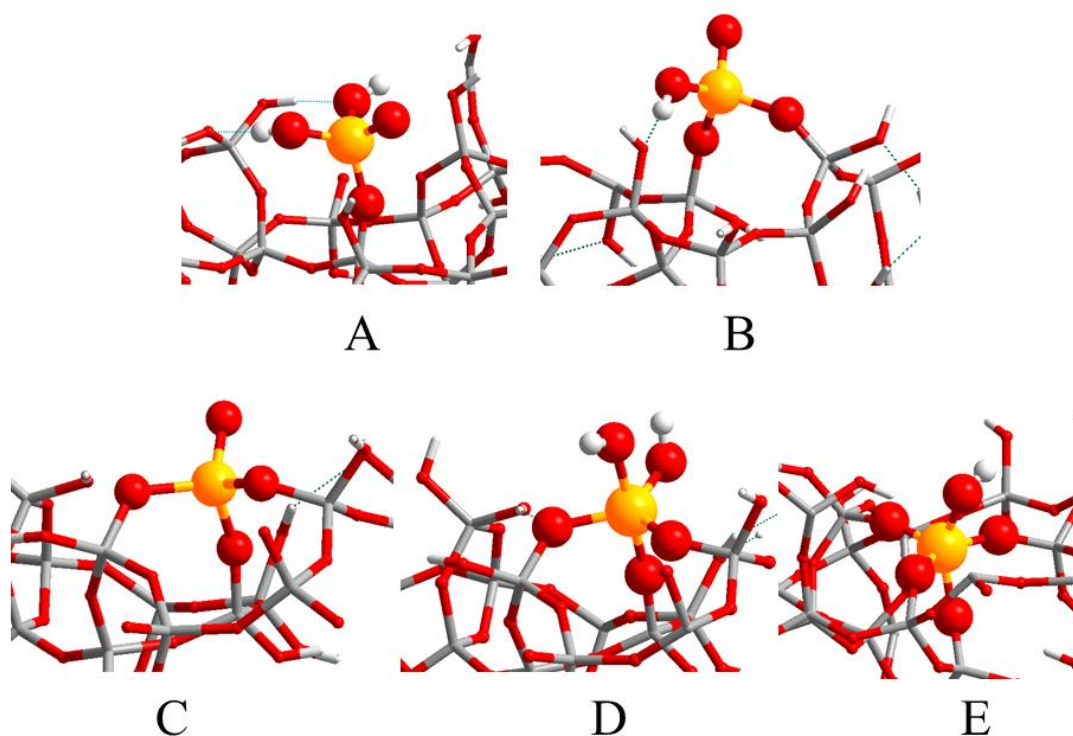


Figure 2. Optimized molecular models for the Nb-oxide species grafted onto silica surface. Color code: H: white; O, red; Si: gray; Nb: orange.

According to the calculated energy values for reaction (**eq. 1**) shown in **Table 1**, the mono- and tetra-grafted niobium species are the highest exothermic configurations, species A and E. However it should be noted that the tetra-grafted species (E) are the same as the substituted one (Si by Nb). Since geometrically the probability to have four silanol groups close together in such a configuration is negligible on amorphous silica surface it should be considered as substituted site and for clarity in this paper it will be called the substituted form. The other properties should be considered if silica ordered

mesoporous molecular sieves (e.g. SBA-125, MCF) are used as supports for niobium species. For such structures (e.g. MCF molecular foams containing cavities) the formation of tetra-grafted species is probable during grafting and isomorphous substitution as well. Experimentally, by the use of UV-Vis spectroscopy (**Fig. 3**) one cannot distinguish the coordination of niobium incorporated into MCF structure by impregnation and by the substitution during the synthesis (Nb/MCF and NbMCF, respectively). In both cases two UV bands are observed, at ca 220 and 260 nm. Both come from charge transfer from oxygen to niobium atoms (at low coordination states; tetra and penta) located at the different surroundings. The higher coordination state (octahedral) typical of bulk niobium oxide should give the band at ca 330 nm. Such a band is not visible in the spectra presented in **Fig. 3**. One can assign the observed UV bands to structures A and E in the theoretical models, i.e. the most stable structures.

The other picture appears when amorphous niobosilica is prepared and located as pillars in MCM-22 zeolite. UV-Vis spectrum (**Fig. 4**) reveals only one UV band centered at ca 220 nm. In the case of amorphous silica surface (not structured in cavities like in MCF) the probability to have four silanol groups close together in the configuration E is negligible. Therefore the observed band can be assign to niobium mono grafted in the configuration A.

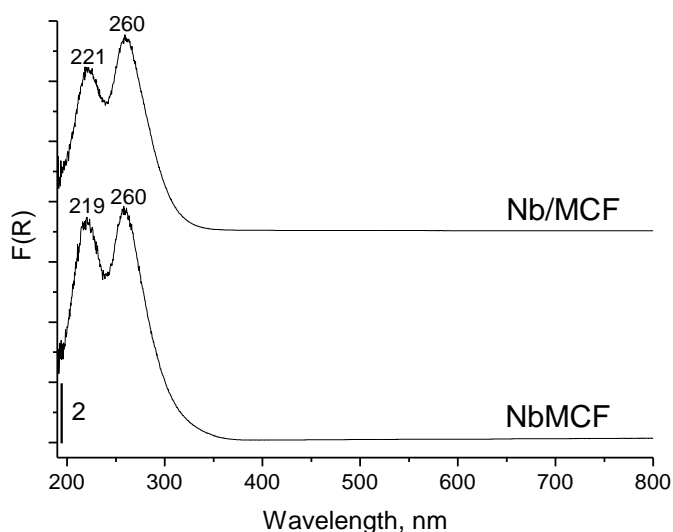


Figure 3. UV-Vis spectra of niobium containing MCF obtained by the impregnation (Nb/MCF) and isomorphous substitution during the synthesis (NbMCF)

Table 1. Reaction energy at 0 K in vacuum, calculated using equation (1) for the grafting of different Nb and V oxide models. (Values in eV)

Structure	H-Bonds	ΔE_{react} For Niobium (this work)	ΔE_{react} for Vanadium ³²
A-V: Surface + H ₃ NbO ₄ - 1H ₂ O	NbOH...O(H)Si Nb(H)O...HOSi	-0.77	-0.66
A-G	-	-0.34	-0.30
A-I	-	-0.45	-0.44
B-V: Surface + H ₃ NbO ₄ - 2H ₂ O	-	-0.25	-0.25
B-G	Nb=O...H NbOH...OHSi	-0.51	-0.24
C: Surface + H ₃ NbO ₄ - 2H ₂ O	-	-0.38	+0.95
D: Surface + H ₃ NbO ₄ - 3H ₂ O	NbOH...(Nb)OH	+0.45	-
E: Surface + H ₃ NbO ₄ - 1H ₂ O (substituted)	HO...HOSi	-0.95	-

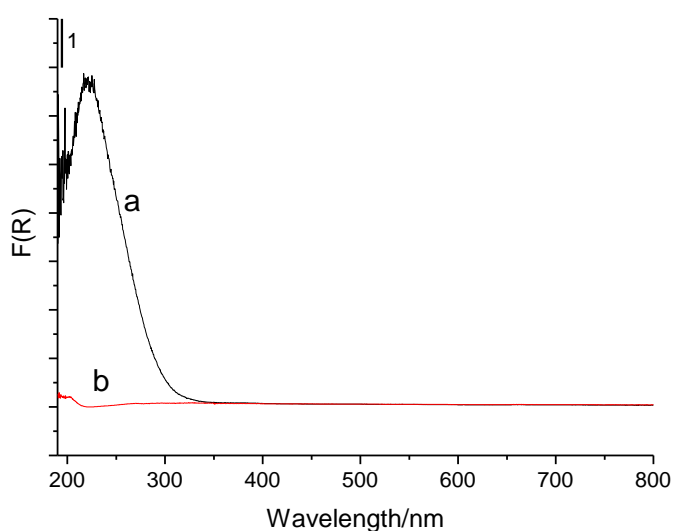


Figure 4. UV-Vis spectra of MCM-22 pillared with niobiosilica (a) and silica (b).

The least stable structures are di- and tri-oxo-grafted species: B, C, and D.

Tri-grafted species need the presence of three neighboring silanol sites and no preference for a particular configuration is found. Species with bridging hydroxyl Nb-OH-Si groups and di-oxo species (O=Nb=O) are clearly disfavored (not shown).

The bond lengths and bond angles have been calculated both for one and two Nb atoms grafted on silica surface (See **Table 2**). The optimized structures have a Nb=O bond length of 1.76 – 1.79 Å for one Nb atom and a value of 1.77 Å for two Nb atoms grafted on silica surface, which is very close to the gas phase Nb₂O₅ molecule (1.779 Å)⁵⁴ The niobyl bond is almost not influenced by the environment; only a small shortening can be noticed with increasing number of Nb-O-Si links. The bridging Nb-OSi bond distances (1.85 - 2.13 Å) and the bond angle Nb-O-Si (between 126.4° – 171.4° for one Nb atom, and 133.6° - 150.3° for two Nb atoms), are independent of the nature of the grafting site. For the di- and tri- grafted sites one extra angle of around 125° is observed. The Nb-OH bonds are in the range of 1.92 – 2.03 Å for one Nb atom and 1.86 - 1.93 Å for two Nb atoms. The values obtained for the calculated Nb=O bonds are in good agreement with the bond distance derived from RAMAN spectra⁵⁵.

Table 2. Calculated selected geometrical parameters in niobia/silica systems (distances d in Å, angles α in degrees).

Distance d angle α	Nb ¹	Nb ²	Literature
d(Nb=O)	1.76 – 1.79	1.77	1.77 ⁵⁴
d(Nb-OSi)	1.88 – 2.04	1.85 – 2.13	–
d(NbO-Si)	1.63 – 1.65	1.61 – 1.74	–
d(Nb-OH)	1.92 – 2.03	1.86 – 1.93	–
d(Nb-Nb)	-	5.08 – 12.7	–
α (Nb-O-Si)	126.4 – 171.4	133.6 – 150.3	–
α (Nb-O-H)	115.0 – 136.9	117.1 – 132.5	–

¹ One Nb atom per unit cell.

² Two Nb atoms in double unit cell with different inter-atomic Nb-Nb distances.

Grafting of niobium oxide on silica surface has a relatively small effect on the silica framework. The largest distortion is observed for the bridging oxygen sites (Nb-OSi). Initially silica hydroxyl groups are displaced by 0.39 Å after grafting for one Nb atom and by 0.48 Å for the case of two Nb atoms inserted onto silica surface. For the case when two Nb atoms are grafted to the silica surface, the NbO-Si bond is slightly displaced by 0.1 Å from their initial positions. The Si-O-Si angles range from 139.8° to 155.5° for the niobium silica surface which is close to the values of 130° to 180° for Si-O-Si angles of silica surfaces²³. The Nb-O-Si angles range from 133.6° to 150.3°.

In summary, the overall silica geometry is only slightly modified upon Nb oxide grafting. The same result was obtained for the grafting of vanadium oxides on silica surfaces³².

Besides grafting one can also insert Nb into the silica framework. Experimentally it was shown by Ziolk et al.⁵⁶ that the Nb species inserted into ordered mesoporous silica of MCM-41 type can form Lewis acid and base pairs after dehydration. In order to investigate the molecular structure of this particular situation, two Nb oxide sites were studied as a function of the distance between them. **Table 3** and **Figure 5** show the relative energies for the case when two Nb atoms are inserted into the silica surface.

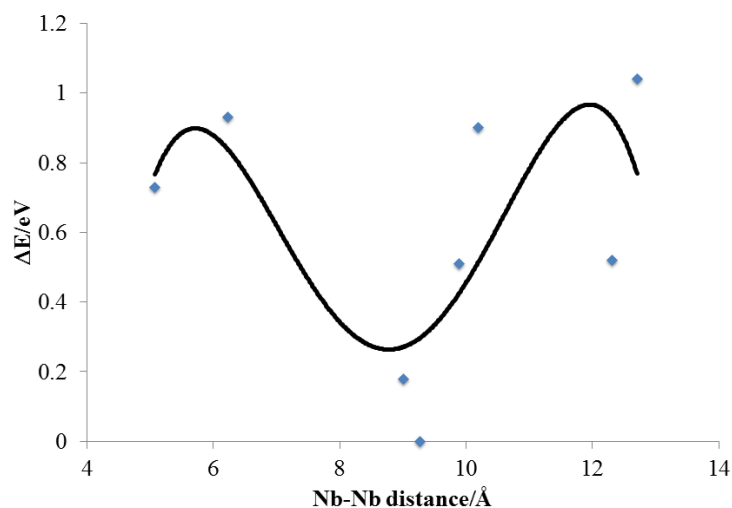


Figure 5. Relative energy between the most stable structures with two supported Nb oxide species grafted onto silica surface (Structure 1) vs. Nb-Nb distance in the silica model. A trend line is drawn in black.

Table 3. Relative Energies for the double unit cell with two Nb atoms. Energies in eV and distances in Å, the index nR of Nb corresponds to the size of the siloxane ring in which the substitution has been performed (n = number of Si sites).

Structure	ΔE	Nb...Nb Distances	Structure Description	H-Bonds
1	0.00	9.27	Nb _{6R} + Nb _{5R}	Nb=O...H
2	0.18	9.01	Nb _{6R} + Nb _{4R} (1 SiOH)	NbOH...OHSi
3	0.51	9.89	Nb _{6R} + Nb _{4R} (2 SiOH)	NbOH...OHSi
4	0.52	12.31	Nb _{6R} + Nb _{4R} (1 SiOH)	NbOH...OHSi
5	0.73	5.08	Nb _{6R} + Nb _{4R} (2 SiOH)	SiOH...(Nb)OH...OHSi (2 H-bonds)
6	0.90	10.20	Nb _{6R} + Nb _{4R} (2 SiOH)	SiOH...(Nb)OH...OHSi (2 H-bonds)
7	0.93	6.23	Nb _{6R} + Nb _{4R} (2 SiOH)	-
8	1.04	12.71	Nb _{6R} + Nb _{4R} (2 SiOH)	-

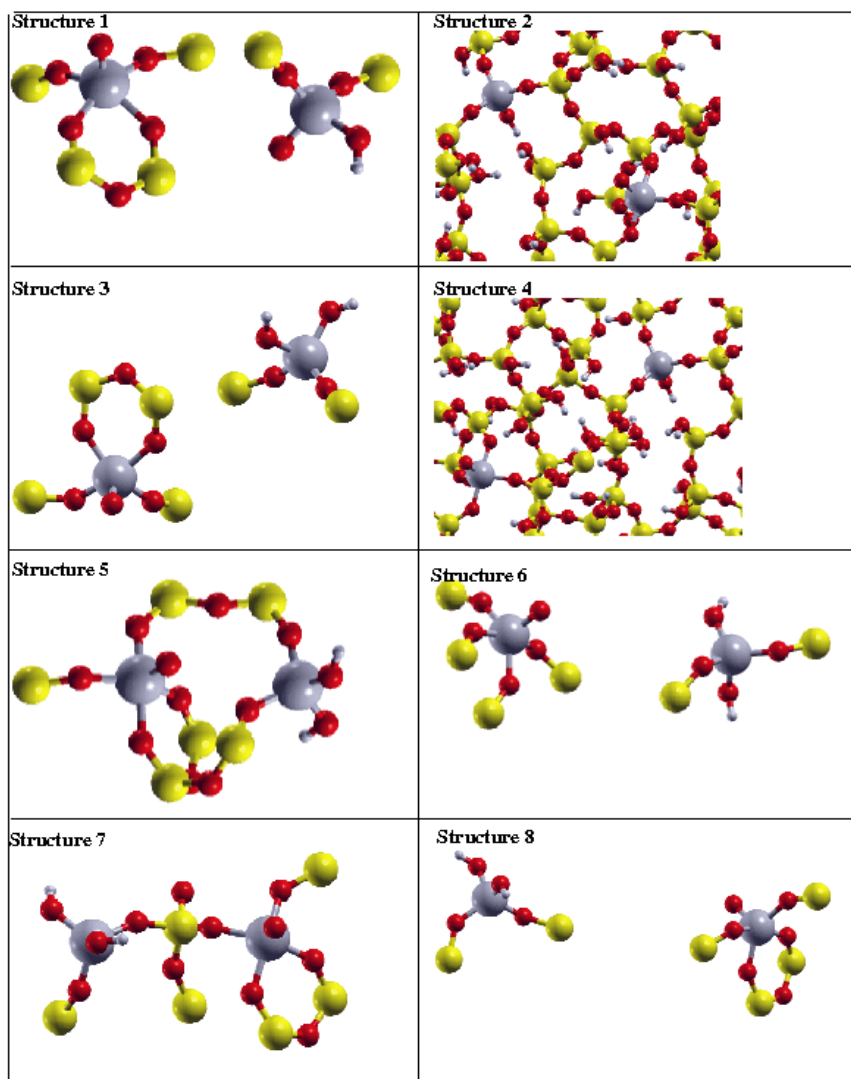


Figure 6. Most stable geometries for the structures with two supported Nb oxide species grafted onto silica surface. Color code: H: white; O: light red; Si: light orange; Nb: light gray.

The most stable structure (See **Figure 6**) is obtained when Nb is substituted in a siloxane ring containing 6 T atoms and 5 T atoms ($\text{Nb}_{6\text{R}} + \text{Nb}_{5\text{R}}$ is formed). The next stable structures are observed for the formation of $\text{Nb}_{6\text{R}} + \text{Nb}_{4\text{R}}$. The number of Nb-OSi and Nb-OH groups can vary when $\text{Nb}_{4\text{R}}$ is generated. For the $\text{Nb}_{6\text{R}} + \text{Nb}_{4\text{R}}$, the stability of the structures is determined by the following factors: numbers of Nb-OSi groups; numbers of Nb-OH groups; H-bonds formation; distance between the two Nb atoms.

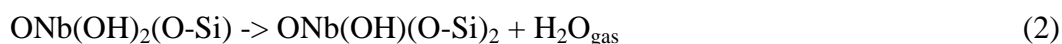
The most stable structures are those which have a large number of Nb-OSi and Nb-OH groups and implicitly H bonds formation. Between the Nb-OH groups and surface

silanol groups are formed H-bonds which stabilize the surface structure. The less stable structures are those which do not form H-bonds (the Nb–OH groups are pointed towards vacuum) and the distance between the Nb atoms is either too small or too large. An optimal Nb-Nb distance is found around 9 Å (See **figure 5**).

Thermodynamics and stability of the supported niobium oxide species

Considering the reaction energy (ΔE_{react} , **Table 1**) as calculated according to reaction (1) for the best grafting modes (mono-, di-, tri-, and tetra-grafting), we obtain -0.77, -0.51, +0.45 for $n = 1, 2$ and 3 , respectively, and -0.95 eV in the case of substitution ($n = 1$). These values suggest that the grafting of the niobium complex is favored compared to the initial situation (hydroxylated silica and $\text{O}=\text{Nb}(\text{OH})_3$ in the gas phase) for the mono- and di-grafted species, as well as for the insertion (structure E), while it is endothermic for the tri-grafted one. The endothermicity of the reaction leading to the tri-grafted species means that the gain in energy due to the formation of three water molecules is not sufficient to counter-balance the cost of breaking three Si-OH bonds. Indeed, the present calculations report electronic energies only, which are identical to the free energy at 0 K. Under given temperature T and pressure p , the contributions of entropy and chemical potentials have to be taken into account in the free energies.

In order to get a more precise picture of the respective stabilities of the mono-, di- and tri-grafted species at the silica surface, we performed calculations using the atomistic thermodynamics approach. To take into account deviations in surface composition and the presence of gas phases, one introduces appropriate chemical potentials to calculate an approximation of the Gibbs free energy. Assuming that the surface is in thermodynamic equilibrium with the gas phase, the chemical potentials are related to a given temperature T and pressure p . This procedure enables to extend the 0 K and zero pressure DFT results to experimentally relevant environments, thereby bridging the gap between ultra-high vacuum like conditions, and temperatures and gas phase pressures that are applied in realistic catalytic conditions. For this purpose the following equations are considered:



It is assumed that the niobia/silica system is in contact with a gaseous water reservoir. From the electronic energy, the free energy of the water/niobia/silica interface under known thermodynamic conditions may be estimated following the approximations used in our earlier studies.³⁶ They neglect the variation of the chemical potentials of the surfaces with the adsorption of water and the gas phase is modeled as a perfect gas. In the proposed scheme, the free energy of water (including the ZPE correction) in the gas phase is:

$$\Delta G(\text{H}_2\text{O}) = E(\text{H}_2\text{O}) - ((\Delta H_G - T\Delta S_G(T)) + RT \ln (p/p^\circ)) \quad (4)$$

where $E(\text{H}_2\text{O})$ is the electronic energy of water calculated at 0 K, ΔH_G and $\Delta S_G(T)$ are the enthalpy and entropy of gaseous water, calculated with the Gaussian03 code⁵⁷ as a function of the temperature, p is the partial pressure of water vapor and p° is the standard pressure (1 bar).

Using the above mentioned formalism, the free energies of reactions (2) and (3) for the formation of the di- and tri-grafted niobium complexes from the mono-grafted one at equilibrium conditions, are then expressed as:

$$\Delta G_2 = E(\text{ONb}(\text{OH})(\text{O-Si})_2) + \Delta G(\text{H}_2\text{O}) - E(\text{ONb}(\text{OH})_2(\text{Si-O})) \quad (5)$$

$$\Delta G_3 = E(\text{ONb}(\text{O-Si})_3) + \Delta G(\text{H}_2\text{O}) - E(\text{ONb}(\text{OH})(\text{Si-O})_2) \quad (6)$$

In this approach, it is assumed that the energies of the mono- to di- and di- to tri-grafted transitions are independent of the degree of hydration of the silica surface. It is known from experiments that silanols are stable at silica surfaces up to 673 K. Above this temperature, silanols begin to condensate into siloxane bridges.⁵⁸ Thus, our model with a density of 5.8 OH/nm² corresponding to conditions of a hydroxylated surface, holds up to the temperature of 673 K.

Figure 7 shows the surface free energy ΔF , defined as the free energy per surface area, of the grafted and the inserted (substituted) Nb-complexes on/in the silica surface as a function of temperature (T) for a water partial pressure (p) equivalent to the ambient air water partial pressure ($p_w = 1500$ Pa)⁵⁹. At these conditions, the mono-grafted model is the most stable up to $T = 230$ K, followed by the digrafted structure in the range of 230 - 660 K domain, finally at $T > 660$ K the trigrafted complex is found as the most stable configuration. These results are fully consistent with the experimental procedure used in

the synthesis of niobia-supported catalysts by grafting methods. From the thermodynamic approach one may conclude that the most stable species at low temperature is the mono-grafted $\text{ONb}(\text{OH})_2(\text{O-Si})$ complex. This configuration corresponds to the fully hydrated niobia catalyst on the silica support. An increase in temperature or a decrease in hydration stabilizes the di-grafted $\text{ONb}(\text{OH})(\text{O-Si})_2$ complex. High temperatures favor the formation of the pyramidal $\text{ONb}(\text{O-Si})_3$ arrangement. Nevertheless it should be noted that the penta-coordinated tri-grafted Nb species (C) with two hydroxyl groups is only 0.13 eV less stable than the di-grafted species. In the same diagram (**Fig. 7**) we have plotted the curve for the inserted species E, which is stable until 350 K (at ambient air water pressure) after which one might expect possible segregation to the surface and transformation into a di-grafted species such as structure B. This hypothesis is of course only valid without the presence of any unbridgeable kinetic barrier.

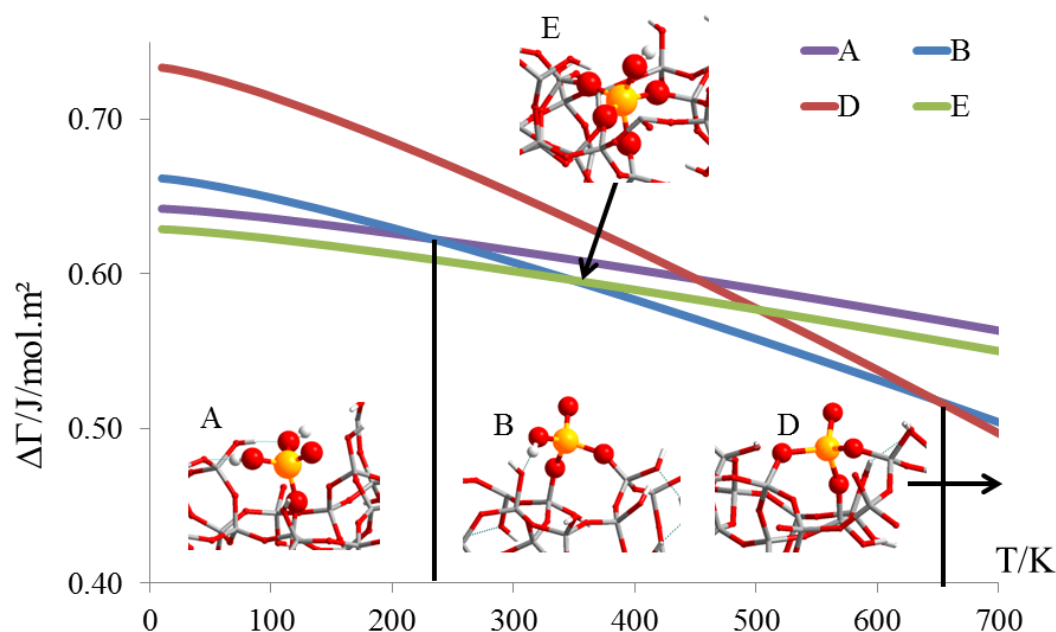


Figure 7. Phase diagram (surface energy vs. temperature) showing the stability ranges for the different grafting geometries.

Vibrational frequency analysis and Infrared Analysis

Table S1 and **Table S2** (See Supplementary Material) show the vibrations calculated for the models A to E and the double Nb/double unit cell models, together with the

assignment of the normal modes. The calculated values were not scaled but can easily be recalculated using the scaling factor to the experimental data using the value reported in ref.⁶⁰. It should be noted that experimentally, the SiO₂ spectra will be predominant over every individual Nb-O vibration mode that is why theoretical calculations are indispensable to interpret the spectra.

Only calculated frequencies in the range 600 - 4000 cm⁻¹ are reported. The region of 3000 - 4000 cm⁻¹ is dominated by the hydroxyl vibrations. Values higher than 3660 cm⁻¹ are assigned to free NbO-H stretching modes. Experimentally a narrow band is present at 3660 cm⁻¹ upon hydroxylation of dehydrated samples which is assigned to the stretching of free NbO-H groups²⁴. Lower values indicate hydrogen bonding: 3567 cm⁻¹ (distance O...H = 2.133 Å) and 3267 cm⁻¹ (distance O...H = 1.720 Å). One particular case is observed for Nb2-Structure 4, **Table S2**, for which the NbO-H stretching mode is assigned to a value of 2263 cm⁻¹.

The region between 1000 cm⁻¹ and 800 cm⁻¹ is dominated by Nb=O, Nb-O-Si, Si-O-Si stretching modes. From the calculated Nb=O distances, the vibration frequencies can be obtained in first approximation using the relationship of Hardcastle et al.³³ Nb=O bonds, with distances of 1.76 – 1.79 Å for one Nb atom grafted to silica surface, are predicted to vibrate between (906 – 941 ± 30) cm⁻¹ in first approximation. From our calculations we have found that the vibration frequencies for Nb=O bonds (one Nb atom grafted to silica surface) are between 1009 – 964 cm⁻¹ which is in very good agreement with experimental work.^{13, 14} Experimentally, the Nb=O vibration is observed at 982 cm⁻¹ on dehydrated 10% Nb₂O₅/Silica^{13, 14} and on dehydrated Nb-MCM-41 at ~985 cm⁻¹.¹³

The difference between the experimental and theoretical data is that, the calculated vibrations are slightly overestimated with respect to the bands observed due to the use of the DFT method and the harmonic approximation for the calculation of the vibrational frequencies.

For the case when two Nb atoms are grafted on a double silica unit cell model surface, the vibrational frequencies for Nb=O are between 835 - 905 cm⁻¹. The region between 700 – 600 cm⁻¹ is dominated by Nb-OH stretching of the O atom or bending of the H atom. The region below 600 cm⁻¹ corresponds again to the Nb-O-Si, Si-O-Si stretching modes. For a better understanding of the normal modes we compared them with the

experimental infrared spectrum for the case when two Nb are grafted on silica (see **Figure S1 and 9**).

From **Fig. S1** (See Supplementary Material), it can be observed that the surface infrared vibrational spectra can be divided into three regions: the high frequency domain between $4000 - 3000 \text{ cm}^{-1}$, the middle frequency domain between $2000 - 1500 \text{ cm}^{-1}$, the small frequency domain between $1500 - 500 \text{ cm}^{-1}$. The interval $3000 - 2000 \text{ cm}^{-1}$ is void of any frequency in our case.

In the high frequency domain ($4000 - 3000 \text{ cm}^{-1}$) we find a strong infrared active OH stretching vibration at 3770 cm^{-1} . The vibration of the OH stretching bond found at 3770 cm^{-1} is in the range of a free OH stretching vibration $3600 - 3800 \text{ cm}^{-1}$. In the middle frequency domain we do not observe any frequency in our case.

Unlike the previous regions, in the $1500 - 500 \text{ cm}^{-1}$ region, one can no longer assign a frequency to a single bond formation. Rather, in the so-called *fingerprint region*, several bonds are found to vibrate at nearly the same frequency giving rise to complex normal modes not easy to describe. With this in mind, we point however, towards the large presence of the Nb=O, Nb-O-Si, Si-O-Si stretching modes.

Comparing these data with experiment (See Figure 9) the following can be concluded: In the experimental spectrum no absorbance is observed, between $2000 - 4000 \text{ cm}^{-1}$, besides the 3740 cm^{-1} “large” band. This finding is in agreement with the theoretical models. No Nb=O nor Nb-O-Si vibration is visible in IR (only visible with RAMAN). It was found earlier that the $\nu\text{Si-OH} \approx \nu\text{Nb-OH}^{30}$, with Nb in oxidation state +5. The Nb-OH bands being less intense than the Si-OH band(s) due to the low concentration of Nb, their signal is drawn in the spectrum, however, a tiny shoulder was observed in zeolitic materials³⁰ at 3728 cm^{-1} . Due to the equilibrium conditions between the dehydrated and hydrated Nb(V) Lewis acid and base pairs in/on silica (See Structure 1), show Nb-OH IR bands which are even weaker because their presence is related to the degree of ambient humidity.

The structures corresponding to the experimental spectrum are the structures A and E as concerns the monomers and structures 1, 3, 6, and 8 for the Lewis acid base pair structures. However, taking the degree of hydration into account on structure E is

expected to be dominant in the monomer structures and structure 1 (most stable of the series) for the Lewis acid base pair structures. Both model structures show a Nb-OH vibration of 3770 cm^{-1} . The Nb-OH bands are thus expected to overlap with the terminal silanol bands.

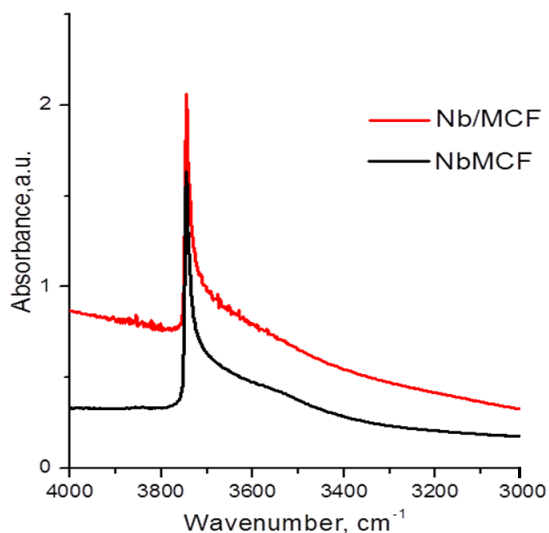


Figure 9. FTIR spectra of niobium containing mesoporous cellular foams (Nb/MCF prepared by the impregnation and NbMCF synthesized with the use of Nb source in one pot) activated under vacuum at 623 K.

Conclusion

In this study the insertion and grafting of Nb oxide species as a function of the degree of hydration has been characterized at the molecular level using periodic DFT confronted with experimental results.

The comparison of calculations with the experimental results (UV-Vis) indicates that two Nb structures are favored when niobium is inserted into silica, structure A (mono coordinated) and E (penta coordinated). The method of insertion (grafting or isomorphous substitution during the synthesis) does not determine the species formed but the structure of silica is the important parameter. On amorphous silica penta coordinated species is not formed whereas in mesoporous cellular foams (MCF) containing cavities such a form is easily generated. Therefore the catalytic activity of

niobium species located on amorphous silica and on ordered mesoporous molecular silica sieves is different.

The characterization of the earlier observed Nb Lewis acid base pairs in silica was achieved by constructing a large amorphous silica slab, which enabled the determination of the optimal distance between the Lewis acid and the Lewis base, and found to be about 9 Å. Its stability is probably depending on the local flexibility of the silica material. Moreover, the IR investigation provided a complete calculated spectrum between 500 and 4000 cm⁻¹, from which the overlap between the bands of the Nb-OH groups and the terminal silanol groups was confirmed, as was found for Nb substituted zeolites.

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Nb Oxide on Hydroxylated Silica: A Molecular Picture

