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Dielectric properties of a single nanochannel investigated by high frequency impedance spectroscopy

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Abstract

The high frequency electric impedance spectroscopy is used to characterize the dielectric properties of a single nanochannel in a micro/nanofluidic device. The simulated electric impedance results lead to the determination of two conductance regimes, a non ideal capacitance and its surface charge.

Keywords: nanochannel; impedance, high frequency, conductance, constant phase element, surface charge.

1. Introduction

The understanding of the physics of ionic transport in biologic membranes, solid-state nanofluidic devices or porous material offers new perspectives from biology to energy storage domains. [1],[2],[3],[4],[5],[6],[7],[8],[9] In such systems characterized by at least one nanometer dimension, the role of the surface charge is exacerbated,[10],[11] and new phenomena such as surface-charge mediated transport and permselectivity occur. The unbalanced ionic transport between anionic and cationic species through a nanofluidic structures initiates a polarization effect related to the concentration domain, which can be applied for focusing and locally preconcentrate ionic species. [12], [13] Compared to the classical microfluidic field, the nanometer dimension of the channel requires specific theoretical predictions and experimental discussions.[14],[15],[16],[17]. The conductance law through nanochannels depends on the ionic strength of the background electrolyte.[18],[19],[20]. At high concentration for which the Debye length is much smaller than the channel height, conductance through the nanochannel follows the solution conductivity; at low concentration corresponding to a large Debye length with respect to the channel height, the nanochannel conductance becomes concentration independent and is fully governed by surface charge. Recently, Reed et. al. [20] have investigated the conductance change in an intermediate regime of ionic strength and have speculated about a surface charge inversion in divalent systems where ion-ion correlations exist. In this context, development of direct real-time measurement of conductance in nanofluidic channel is of great interest and necessitates new strategies for more accurate characterization.

In this letter, a new methodology to determine accurately the nanochannel electric properties in a micro/nano/microfluidic device using the high frequency (HF) electric impedance spectroscopy is presented. Indeed, the main limitation encountered with the use of traditional commercially available EIS instrument is the HF measurement accuracy. To overcome this limitation, we combined a HF EIS Frequency Analyzer with a dielectric interface apparatus dedicated to electric impedance measurements for high resistive materials.[21],[22],[23]

2. Experimental

2.1. Chemicals and reagents

KCl and NaOH (from Sigma-Aldrich) were used as background electrolyte solution and as washing solution for the glass surface regeneration, respectively.

2.2. Nanochannel and microchannel fabrication

The hybrid micro/nanofluidic device displayed in Fig. 1 was fabricated in borosilicate glass chip with a 1.47 \pm 0.01 µm height microchannel and a 170 \pm 10 nm height nanochannel. Nanochannel was made using the following process. 1) First, 300 nm of amorphous silicon (α -Si) was deposited by plasma-enhanced chemical vapor deposition on a 500 µm glass wafer. 2) Then AZ-5214 reversible photoresist was spin-coated (5000 rpm, 30 s) on the α -Si glass substrate. 3) The microchannels were designed by UV lithography on a SUSS MicroTec MJB4 mask aligner. 4) In the regions that were not protected by the resin, the α -Si was etched by reactive-ion etching (RIE) with SF₆. 5) The microchannels were then etched, on α -Si not protected areas of the glass, using ammonium fluoride – hydrofluoric acid mixture (AF 875-125, etching mixture, Aldrich). 6-9) The nanochannel was made by the same method after a second spin-coating, lithography and RIE (SF₆) procedure. 10) The reservoirs, where solutions were injected, were drilled by sandblasting. 11) The remaining α -Si was removed using RIE (SF₆). 12) The glass chip and another 300 µm glass wafer were then carefully washed using, successively, trichloroethylene, acetone and isopropyl alcohol, the etching mixture for a few seconds, and deionized water.. 13) The two wafers were bonded without any intermediate layer using a wafer bonder SB6 (SUSS MicroTec). During thermal bonding, wafers were put in contact for 2h under vacuum at 550°C with an applied pressure of 3-5 bars. 14) A 5 mm layer of polydimethylsiloxane (PDMS) with holes aligned with the reservoirs was bonded on the chip using air plasma for 70 s.

- Fig. 1 -

2.3. Instrumentation and measurement

Electric impedance measurements were performed with a frequency response analyser (FRA 1255B) coupled with a dielectric interface (DI 1296A) which extends frequency range up to 20 MHz. The originality derives from the use of this latter instrument which contains a very high input impedance allowing measurement of ultralow currents. [24],[25] Impedance was measured between two AglAgCl junctions located in the microchannel reservoirs (Fig. 1). The frequency range was varied from 1 MHz to 0.1 Hz. The sinusoidal ac signal amplitude was set to 100 mV, and no dc bias was imposed in order to avoid any faradic reactions.

3. Results and discussion

3.1. Electric impedance spectroscopy on micro nanodevice

To determine the nanochannel dielectric properties, the procedure described below was followed for various KCl concentrations. Solutions were introduced by capillarity in the device and measurements started only after a stabilization time of 45 minutes in order to reach the steady state conditions for complete solution uptake through the nanochannel. The results are displayed according the Nyquist or Bode (modified as phase and log (imaginary) vs. log (f)) plot representations. As displayed in Fig. 2A-B for 3×10^{-4} M KCl concentration, characteristic frequencies are indicated on the plot to visualize the time constants of phenomena involved. Indeed, the semi-circle shape suggests a traditional R//Celectrical circuit with a characteristic frequency $f_{\rm RC}$ that can be also identified on the $Z_{\rm j}$ and phase Bode diagrams. This $f_{\rm RC}$ value, equal to 1168 Hz, corresponds to the maximum of Z_{i} , equal to R/2 (R, semi-circle diameter). Furthermore, the nanochannel resistance, R, can be estimated as the difference on the real-axis between the asymptotic resistance limit at zero-frequency, $R_{\rm LF}$, and that as frequency tends towards ∞ , $R_{\rm HF}$, (Fig. 2A-B). As one can observe on real-axis, $R_{\rm HF}$, is negligible, then R identifies itself with R_{LF} . It is observed that the nanochannel capacitance, C, obtained in the HF domain is not ideal ($\phi \neq 90^{\circ}$) because the impedance phase tends towards a constant value equals to - 88° (Fig. 2B). This experimental result is also confirmed since the α -slope value for Z_j with frequency on loglog scale indicated in Fig. 2B is equal to - 0.98 (α = - 88°/90° \approx - 0.98 thus different from the unit). In brief, it had been assumed that nanochannel capacity was a non ideal capacity because it is not independent of frequency. Indeed, this analysis of the data showed that it decreased with increasing frequency. Consequently, the glass/ nanochannel interface behaves as a RC filter where a non ideal capacitor can be theoretically represented by a constant phase element (CPE). The changes on the effective CPE coefficient, Q, can be calculated from the obtained raw data of the imaginary part in the HF domain, as proposed by Orazem *et. al.*[26] with equation 1 as follows

$$Q = \sin\left(\frac{\alpha\pi}{2}\right) \frac{-1}{Z_j(f)(2\pi f)^{\alpha}}$$
(1)

with Q and α being the CPE element and the CPE exponent, respectively.

In equation 1, when $\alpha = 1$, the effective CPE coefficient, Q_{eff} , becomes an effective capacitance, C_{eff} . The change of Q was plotted by using α -slope numerical value previously determined ($\alpha = -0.98$) and the Z_j data at each frequency. As expected an asymptotic value $Q_{\text{lim}} = 1.08 \times 10^{-12} \text{ F s}^{(\alpha-1)}$ was deduced by plotting the Q_{eff} variation that behaves as a CPE in the HF domain (Fig. 2C). At this stage, it can be assumed that the observed CPE behavior originates probably from the nanochannel geometry which induces a non uniform current and potential distribution at the nanochannel entrance with a CPE exponent, α , taking into account the role of concentration polarization.

For exploring changes on conductance and capacitance, several KCl solutions were successively introduced from low to high concentrations $(3 \times 10^{-6} \text{ M} - 3 \text{ M})$. The data analysis was performed with a homemade software (Simad) using the electrical equivalent circuit displayed in Fig. 3A that corresponds to an analytical expression taking into account a CPE coefficient *Q* instead of pure capacitor *C*, as follows

$$Z(\omega) = R_{HF} + \frac{R}{1 + (j\omega)^{\alpha} Q \cdot R}$$
(2)

where ω is the angular frequency (rad s⁻¹), defined by $\omega = 2 \pi f$, and *f* is the frequency (Hz). As shown in Fig. 3B-C, the nanochannel resistance, *R*, decreases when the nanochannel ionic concentration increases from 3×10^{-6} to 3 M. The corresponding conductance $R^{-1} (\Omega^{-1})$ or *G* (S) *vs* ionic concentration is plotted in the log-log scale (see Fig. 3D). Two conductance regimes are highlighted: a plateau conductance appears for concentrations lower than 3×10^{-2} mol m⁻³, while a linear conductance variation starts from concentrations higher than 30 mol m⁻³. These conductance regimes in a nanochannel have been also observed in the literature, using other methods such as dc conductance,[18] or by conventional electrochemical impedance spectroscopy measurements.[27] Indeed, as proposed in literature [18],[27], the conductance of a nanochannel filled with a 1:1 background electrolyte, in the steady-state conditions, neglecting electroosmotic effects, can be expressed as the sum of the bulk conductance (nanochannel opened), first term in equation 3, and the conductance due to the excess of counterions equilibrating surface charges (nanochannel closed), second term in equation 3, as follows

$$R^{-1} = G = 2\mu_{\pm}c_i \cdot N_A \cdot e\frac{wh}{l} + 2\mu_{\pm}\frac{\sigma_0 w}{l}$$
⁽³⁾

where μ_{\pm} and μ_{+} correspond to the average value of the ionic mobility and the positive charge mobility, respectively. c_i represents the ionic concentration, N_A the Avogadro number, e the electron charge. w, h and l are the nanochannel width, height and length, respectively. σ_0 is the effective surface charge density in the diffuse layer assuming that the species are immobile in the inner Helmholtz plane (IHP).

In the configuration here described the nanochannel width and length are identical leading to a simplified equation 3 as follows

$$G = 2\mu_{\pm}c_i \cdot F \cdot h + 2\mu_{\pm}\sigma_0 \tag{4}$$

where *F* is the Faraday constant.

A transition concentration, c_t , delimiting these two regimes is obtained after equating the first and the second term, as follows

$$c_{t} = \frac{\mu_{+}\sigma_{0}}{\mu_{\pm}\cdot F\cdot h} = t_{+}\cdot \frac{\sigma_{0}}{F\cdot h}$$
(5)

where the role of mobility ratio t_+ (cationic transport number) is highlighted.

For the simulated conductance curve, we obtained an effective surface charge density value, $\sigma_0 = -2.5 \times 10^{-2}$ C m⁻² that is closely similar to those measured in the literature.[18],[28] In addition, c_t graphically determined around 2.6 mol m⁻³ is in good agreement with the one estimated *i.e.* 3.1 mol m⁻³ (t_{+} =0.98 in equation 5).

As displayed in Fig. 3D, it is obvious that the deviation to the electrical neutrality in the double layer (second term in equation 4) becomes qualitatively predominant compared to the average ionic strength at low concentrations. For the two lower concentrations, the Debye length is of the order of magnitude of the nanochannel height.[29] We also observe that the non ideal capacitance obtained does not depend on KCl concentration, confirming that it is an intrinsic parameter (a geometric capacity : $C = (\varepsilon_R \varepsilon_0 wl)/h$) would be more modulable by increasing the length of the nanochannel, *l*, and not its section *h*. In future work, we will investigate on the transition zone where a well-established phenomenon such as pre-concentration depends on the ionic strength of the background electrolyte. [20]

4. Conclusions

We demonstrated that HF electric impedance spectroscopy can be used to determine nanochannel dielectric properties in a nanofluidic device. Modelling the obtained electric impedance data highlighted the existence of a linear variation of the nanochannel conductance for higher ionic concentrations, and of a plateau conductance for lower concentrations. A nanochannel conductance, surface charge, and non ideal capacitance were determined equal to 8.33 nS, -25 mC m⁻² and 1.08 pF s^{α -1}, respectively.

Conflict of interest

The authors declare that they have no conflicts of interests.

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Figure captions

Fig. 1: Left : Global view of a microfluidic chip. At each corner, a hole is visible, where electrolyte is injected. Conductance changes between the two AglAgCl junctions are measured either between holes 1 and 3 or between 2 and 4. The nanochannel is in the middle, framed with blue solid line. Right : A schematic view of the micro/nano/micrometric geometry of the device in the middle. The microchannel is reduced to nanometric size on a short distance.

Fig. 2: A. Nyquist representation of electric impedance data of 170-nm high nanochannel measured in a 3×10^{-4} M KCl electrolyte solution for frequencies from 1 MHz to 0.1 Hz. The diameter loop allows a quick estimation of $R \approx 134.9$ M Ω for the nanochannel resistance. B. Bode representation of the electric impedance imaginary part (right-axis) superimposed with the Bode representation of the electric impedance phase (left-axis), underlining the inflexion point (-45° phase) when the maximum frequency ($f_{RC} = 1168$ Hz) is reached. C. Bode representation of the effective CPE coefficient defined in equation 1, underlining a plateau obtention in the high-frequency asymptote, Q_{lim} , where an accurate value for Q is estimated when $\alpha \neq 1$.

Fig. 3: Nyquist plots for experimental $Z_{g,exp}(\omega)$ and modeled $Z_{g,sim}(\omega)$ electric impedance from 1 MHz to 0.1 Hz with a voltage of 0. 1 Vac, for different KCl concentrations in the hybrid micro nanochannel device. A. Electrical equivalent circuit proposed for the hybrid micro nanochannel device, where Q, R, and R_{HF} are attributed to the nanochannel non ideal capacitance, nanochannel resistance and microchannel resistance, respectively. B. Nanochannel filled with low KCl concentrations: 3×10^{-6} M (a), 3×10^{-5} M (b), 3×10^{-4} M (c), 3×10^{-3} M (d), 3×10^{-2} M (e). C. Nanochannel with high KCl concentrations: 3×10^{-1} M (f), 3M (g). D. Measured and simulated conductances (left-axis: blue circles) across the nanochannel against the KCl ionic concentration. Two regimes appear in the log-log scale delimited by the transition zone concentration ($C_t = 2.6$ mol m⁻³ KCl): a plateau conductance for concentrations lower than 3×10^{-2} mol m⁻³. The theoretical conductance curve change is calculated with equation 4 and the following parameters were found: effective surface charge density, $\sigma_0 = -2.5 \times 10^{-2}$ C m⁻², the nanochannel thickness, h = 170 nm, the average value of the ionic mobility (KCl) and the positive (K⁺) charge mobility $\mu_{\pm} = 7.765 \times 10^{-8}$ m² V⁻¹ s⁻¹ and $\mu_{\pm} = 7.62 \times 10^{-8}$ m² V⁻¹ s⁻¹. Simulated capacitances (right-axis : green squares) across the nanochannel against the KCl ionic concentration.

Figures

Figure 1



Figure 2



