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Alkylthiol self-assembled monolayers on Au(111) with tailored tail groups for attaching gold nanoparticles

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Abstract

Self-Assembled Monolayers (SAM) on Au(111) are able to control the functionality of a gold surface. We use scanning tunnelling microscopy (STM) in air and contact angle measurements to compare the morphology and the chemistry of three alkylthiol SAMs differing by their tail groups: 1,9-nonanedithiol (NDT) and 1,4-butanedithiol (BDT) and 11-mercaptoundecanol (MUOH). STM reveals very different morphologies: hexagonal lattice for MUOH and parallel rows for NDT and BDT. In the case of NDT, we find that the thiol tail groups may form disulfide bridges for long immersion times. The availability of –SH group for chemical reactions is checked by attaching gold nanoparticles (AuNPs). When the thiol tail group is available, AuNPs readily attach as shown with atomic force microscopy (AFM). When disulfide bridges are formed the gold surface is not able to bind nanoparticles.

KEYWORDS: disulfide, dithiol self-assembled monolayer, Au(111) surface,

contact angle, STM, gold nanoparticles.

PACS: 68.08.Bc, 68.08.De

Introduction

Alkanethiol molecules have attracted much attention because they form self-assembled monolayers (SAMs) on gold surfaces [25] and other noble metal surfaces [40, 45]. Applications are considered in nanoelectronics [1, 2], sensing [23, 35], catalysis by gold nanoparticles [26, 27] nanoparticle immobilisation [7, 11], adhesion control [14, 46] and in fabrication of nanohybrid architectures [27, 32, 48]. Within this well-organized organic monolayer, the alkanethiol molecules are all parallel, with the $-SH$ group that forms a covalent bond with the gold surface ($\sim 50 \text{ kcal}\cdot\text{mol}^{-1}$) [45]. This side is termed the head group, whereas the outward moiety is the tail group. The quality of self-organization processes not only depends on the gold-thiol bond but also on the interaction between the neighbouring alkyls due to lateral van der Waals interaction ($\sim 1 \text{ kcal}\cdot\text{mol}^{-1}$ per $-CH_2-$ group) [8, 42]. The increase of alkyl chain length leads to its tilting within a SAM instead of standing upright [21, 25, 36, 41] or in some cases to laying phase [8] even when concentration is sufficient to form dense packing. This tilting appears since molecules optimize the intermolecular interaction and balance the head group-substrate interaction, thus minimizing the surface energy. Typical alkanethiol molecules within SAMs may have an orientation with an average tilt angle of $\sim 27^\circ$ - 35° with respect to the surface normal [10, 16, 30]. Such structures are achieved when concentration of 10-1000 mM are used [45]. For lower concentrations, alkanethiol may adopt a flat-laying orientation arranged along rows on the surface forming the so-called “stripe phase” [45].

The tail group that points outward the surface can have different chemical compositions which define the chemical and physical properties of the functionalized surface. For example, $-CH_3$ and $-CF_3$ as tail groups make the SAM surface hydrophobic, metallophobic and highly anti-adherent,[36], while $-COOH$, $-NH_2$ or $-OH$ tail groups yield hydrophilic surfaces with good metal ion and protein binding properties [3, 25, 45]. Among tail groups a very popular one is the $-SH$ (thiol) end-group since it allows attaching various nanoobjects (noble metal

nanoparticles, nanorods, etc) to a gold substrate or cross-link them between each other [5, 6, 18, 27, 32, 34, 48]. To that end, dithiol molecules are of particular interest because they can be used for attaching gold nanoparticles (AuNPs). However the sulfur-metal interfacial chemistry remains a controversial topic[45]. It was found that for a given dithiol, changes of the concentration and deposition conditions may lead to lying phase [22], loops or standing phase on the gold surface, the latter being called “brush-like” structure [45]. Elaborated methods of Au(111) functionalization were proposed, for AuNPs immobilization to gold via an hexanedithiol linker for example [28].

In the present paper, we compare three SAMs on Au(111) with different tail groups ($-SH$ and $-OH$) and different chain lengths. The capacity of forming ordered monolayers is probed by STM and correlated to their chemical properties by contact angle measurements. In the case of dithiol-based SAMs, we compare how the nature of the tail groups leads to the immobilization of AuNPs. We show that in certain conditions long alkyl chains may lead to the formation of disulfide groups that cannot retain AuNPs. We also establish a procedure for avoiding or minimizing this disulfide formation.

Methods

Sample preparation. The Au(111) substrates are thin gold layers evaporated on mica purchased from Phasis (Switzerland). Before deposition, atomically-flat terraces were prepared by annealing in a propane-butane flame ($\sim 50^\circ\text{C} - 150^\circ\text{C}$ for 90 sec, then at $\sim 350^\circ\text{C}$, 2 – 3 sec). STM images of the resulting substrate reveal the well-known herringbone reconstruction pattern of Au(111) (figure 1-a) as an evidence of the chemical purity of the surface before molecules deposition [32]. As it is well known the herringbone structure is made of alternative hcp and fcc domains [17].

Figure 1. (a) STM images ($300 \times 300 \text{ nm}^2$) of clean Au(111) surface after flaming. The inset shows the zigzag stripes of the reconstructed herringbone structure of gold. (b) Chemical structures of butanedithiol (BDT), nonanedithiol (NDT) and mercaptoundecanol (MUOH)

Three different SAMs were prepared with butanedithiol (BDT), nonanedithiol (NDT) and mercaptoundecanol (MUOH) dissolved in ethanol with a concentration $1.5 \times 10^{-3} \text{ mol.L}^{-1}$ (figure 1-b). The freshly annealed gold surfaces were immersed in these ethanolic solutions for either 30 s (**process #1**) or 40 min (**process #2**) at ambient conditions. In order to remove the excess of physisorbed molecules, the substrates were rinsed in pure ethanol (HPLC-grade) and dried in nitrogen flow. All chemicals were purchased from Sigma-Aldrich and used as received.

The synthesis of gold nanoparticles was achieved following the Turkevich-Frens method [13, 20]. An aqueous solution of HAuCl_4 ($2.5 \times 10^{-4} \text{ mol.L}^{-1}$) was heated up to boiling in an Erlenmeyer flask. Then 1 mL of an aqueous sodium citrate solution ($1.7 \times 10^{-2} \text{ mol.L}^{-1}$) was added under vigorous stirring. When AuNPs are formed, the solution has a clear ruby-red colour. It was stored at 4°C to avoid nanoparticle aggregation.

In the case of the dithiol solutions of NDT and BDT, the gold substrates were immersed in the aqueous AuNPs suspension for 1 hour. We checked visually that no aggregation occurred

upon deposition: the few colloidal solutions that turned blue were discarded. The resulting substrates with AuNPs were finally rinsed several times with ultrapure water and dried in a nitrogen flow.

The structure of monolayers on Au(111) surface was investigated using a commercial STM (NT-MDT, Russia). STM tips were mechanically cut from a 0.25 mm Pt/Ir (80:20) wire. The calibration of STM piezoelements was checked by imaging the structure of SAM of MUOH on Au(111) and distances are measured with an accuracy of 0.05 nm. All STM-images were recorded in the constant-current mode. The STM tunnelling parameters typically used in our experiments were 0.1-1.0 V and 0.2-0.5 nA.

The AFM characterizations of AuNPs attachment was performed by a Bruker Multimode apparatus operating in the PeakForce® mode. STM and AFM measurements were carried out at ambient conditions.

The measurements of contact angle at solid-liquid-gas interface were performed immediately after substrate functionalization and its drying. The volume of the deposited droplets was equivalent to 3 μ l. The contact angle measurements were performed by custom-made apparatus (Institute of Physics NAS of Ukraine) with software *Micro-Measure version 1.2*. (license GNU). The results were processed using *ImageJ*. Each time, a minimum of five measurements were carried out and the error was below $\pm 2^\circ$ (details in the Supplementary data). The calibration was achieved by measurements of contact angles at water-vapor interface for bare graphite and Au(111) surfaces in comparison with the values obtained by commercially manufactured equipment [42].

Results

The STM images of the three different SAMs prepared following process #2 reveal periodic structures with either bright spots in the case of MUOH (figure 2-a) or rows in the case of BDT and NDT (figure 3). All these structures exhibit randomly distributed large dark spots (pits)

which are known features for such surfaces [44]. The depth of these pits is a multiple of the height of a monoatomic gold step ($2.4 \text{ \AA}$), and they are a consequence of the strong interaction between $-SH$ groups and gold atoms at the Au(111) surface: alkythiol molecules strongly modify the initial surface reconstruction and pull out gold atoms during self-assembling [44]. Such behaviour occurs with every SAM studied in this work. On a smaller scale, the STM contrasts depends on the nature of the tail groups ($-OH$ and $-SH$) and the length of the alkyl chains ($-[CH_2]_n-$, $n = 11, 9, 4$), as shown in the insets of figures 2 and 3. For MUOH, the bright spots are arranged into a hexagonal structure $(\sqrt{3} \times \sqrt{3})R30^\circ$ and the lateral distance between the neighbouring spots is equal to $5 \pm 0.3 \text{ \AA}$.

Figure 2. STM images ($45 \times 45 \text{ nm}^2$) of (a) hexagonally packed SAM of MUOH. Each bright spot of the inset ($4.3 \times 4.3 \text{ nm}^2$) corresponds to terminal $-OH$ tail groups. The distance between the spots is $5 \text{ \AA}$. (b) Schematic models for the packing of thiols on a gold surface: “brush-like structure” (side view); (c) hexagonal $(\sqrt{3} \times \sqrt{3})R30^\circ$ structure (top view). The red disks indicate the positions of the sulfur atom bound in the three-fold hollow sites of the gold.

For NDT, the SAM is organized into parallel rows separated from each other by $15 \text{ \AA}$. When several domains are observed, an angle of 60° is found between the orientation of rows which

suggests an epitaxial relationship between SAM and the underlying Au(111) substrate. The variation of scanning directions, as well as STM-tunnelling parameters did not influence the angle between molecular rows and single bright spots were never observed with NDT SAM. Interestingly, when the NDT-SAM is prepared with a short immersion time following **process#1**, the image displays irregular domains and is hardly observable with STM (see SI, Fig. S1). STM images of BDT SAM prepared according to **process #2** reveals domains with a similar row-like structure as for NDT. The lateral distance between rows is also equal to 15 Å (figure 3). However, the ordered domains cover just 70% of the surface with BDT whereas they represent 95% with NDT.

Figure 3. STM images ($90 \times 90 \text{ nm}^2$) of the SAMs formed on Au(111) with (a) NDT and (b) BDT. Row-like structures appear due to the formation of disulfide between neighbour $-SH$ tail groups (see text). The immersion time of Au(111) in the corresponding solutions of NDT and BDT was 40 min.

Wetting properties of the surface. The contact angles measurements at the droplet-gold interface gives a value of 66° in the case of a freshly annealed gold surface (figure 4-a) which is

indicative of gold atoms in a reduced state [38]. For MUOH, BDT and NDT, the contact angles were found to be equal to 22° (not presented), 47° and 68° respectively (figure-4 b, c). This is consistent with a hydrophilic surface in the case of MUOH and a more hydrophobic surface with BDT and NDT. One would expect that these two SAMs have the same contact angles, since both molecules have the same $-SH$ terminal groups pointing outwards the surface. This will be addressed below. Moreover if the gold substrate is immersed in NDT for 40 min (instead of 30 sec) the contact angle rises from $68 \pm 0.4^\circ$ to $74 \pm 0.5^\circ$. For BDT the surface also becomes more hydrophobic with an increase of the contact angle by at least 14° . A complete set of contact angles values of BDT, NDT and for different samples is given in the Supplementary Data (see Table S1).

Figure 4. Optical microscope images of an ultrapure water droplet on various SAMs along with the values of the corresponding contact angles for (a) annealed gold, (b) BDT for 30 sec, (c) NDT for 30 sec and (d) NDT for 40 min.

AFM measurements were performed after deposition of AuNPs on NDT SAM in two cases: on Au(111) exposed to NDT for 30 sec and on Au(111) exposed to NDT for 40 min. AFM revealed drastic differences. The substrate exposed to NDT for 30 sec (**process #1**) was covered by a large number of spots: $\sim 350 \text{ spots}/\mu\text{m}^2$ (540 and 600 $\text{spots}/\mu\text{m}^2$ for two other experiments, not shown). The profile analysis in figure 5-a, shows that the height of each spot is

17±1 nm which matches with the expected size of AuNPs from the colloidal solution [39]. These spots can be unambiguously assigned to the AuNPs, which are known to be spherical. The lateral size of the spots is 30 nm (figure 5-a, inset) is slightly larger than the actual particle diameter due to the convolution with the AFM tip. Particles were never displaced on the surface during AFM measurements. This suggests that AuNPs are strongly anchored to the surface by NDT. The analysis of different areas also revealed a homogeneous coating of the surface by AuNPs in spite of the presence of flat Au(111) terraces and gold steps edges. It means that Au(111) surface is covered entirely by NDT molecules. No preferred adsorption positions of AuNPs were found. The surface density of AuNPs is relatively low (350 AuNPs/μm²). They cover about 15% of accessible gold surface. A longer immersion time of NDT-covered substrates into AuNPs colloidal solution leads to an increase of AuNPs coverage. In the case of the longer functionalization time in NDT (**process #2**, 40 min) and same immersion time in colloidal AuNPs (1 h) the AFM images display consistently less than 10 AuNPs/μm². In figure 5-b a surface density of 8 AuNPs/μm² is counted. The AuNPs are mainly trapped by defects like step edges, while the probability of trapping by atomically flat terraces seems much lower.

a)

b)

Figure 5. AFM images of AuNPs after immobilization by the $-SH$ tail groups of the NDT SAM. (a) $0.7 \times 0.7 \mu\text{m}^2$ area with high coverage of AuNPs in the case of short immersion into NDT solution (30 sec); the inset is a cross section along line AB; (b) $3 \times 3 \mu\text{m}^2$ area after 40 min immersion into NDT showing very few AuNPs attachment.

Discussion

STM shows that MUOH SAMs form a hexagonal lattice where the average distance between the bright spots is of $5 \pm 0.5 \text{ \AA}$. The average occupancy for one spot is 0.23 nm^2 (calculated from figure 2-a). In the case of adsorption of alkanethiol into a $(\sqrt{3} \times \sqrt{3})R30^\circ$ structure, each molecule is known to occupy an average area per molecule of 0.216 nm^2 which is consistent with our value [22] [24]. Each spot represents the position of one $-OH$ terminal group of the MUOH molecule (see figure 2-b) and form the “brush-like structure” [33]. Each sulfur atom is bound equivalently to the Au(111) lattice on a hollow site (Figure 2-c). The OH moiety points outwards the surface and explains the hydrophilic character of the surface (contact angle of 22°). This MUOH SAM serves as a reference when assessing the surface properties of NDT and BDT SAMs. For these two latter SAMs the adsorption configuration of the head group onto Au(111) surface remains the same, bonding in the three-fold hollow site, while the tail group may form a different structure.

In the case of NDT and BDT, as shown in figure 3, the SAM has a row-like structure with lateral distances of $\sim 15 \text{ \AA}$ when prepared with **process#2**. At that point, it is important to first discuss the possible adsorption geometries. Dithiol molecules can in principle binds to gold through both terminal sulfur atoms so that the molecule would anchor to two points on the surface. Such examples are reported for hexanedithiol [22], benzenedithiol [19] on Au(111) and octanedithiol on Ag(111)[4]. Given that sulfur adsorbs on the three-fold hollow site of Au(111), this double end adsorption results into a back-folded geometry where the molecule forms a bridge over the surface. If we consider the possibility of a back-folded geometry for NDT (total

length of a free molecule is 13.69 Å) and if we assume that this also leads to a self-organization into rows, we found that the inter-row distance would be 11.4 Å. This estimation is based on force field optimization with ChemSketch® software and is explained in the Supplementary Data (see Fig. S2). This value is significantly smaller than the 15 Å obtained from the STM measurement. For BDT, the free molecule has a length of 7.13 Å and the back-folded geometry would result in even narrower stripes. This tends to rule out the back-folded geometry. If now we assume another possibility where NDT and BDT interact with the surface through the lateral alkyl groups with Van der Waals interaction, the molecules would be completely stretched and lie flat. Under these conditions the stripes would reach a period of 13 Å for BDT and 20 Å for NDT. This is not compatible with STM observation since in both cases the period is identical. Moreover, if the molecule was in a flat-lying phase, no thiol bond would be available for attaching AuNPs as well as in a case of back-folding phase, and the situation depicted in Fig. 5-a would not have been observed. Therefore, this is very unlikely that NDT and BDT adopt this flat lying geometry.

In a detailed study, Daza Millone and co-authors have shown by XPS that NDT prepared in ethanol with 24h immersing forms preferentially a standing-up phase.[9] They reach this conclusion by measuring the ratio of the S2p core level shift at 163.5 eV related to free S-H bonds or disulfides and 162 eV related to thiolates. Interestingly, by monitoring the 163.5 eV peak, they conclude that the signal comes from terminal sulfur atoms that have formed disulfide bridges. Their experimental conditions correspond to our **process#2**. Therefore it indicates that BDT and NDT adopt a standing-up geometry.

The explanation of this striped-phase can be found by comparing **process#1** and **process#2**. When the NDT SAM is prepared according to **process#1**, the domains with parallel rows appear to be much smaller (self-organized area of 10×10 nm²) and the surface is poorly organized as shown in Figure S1 (Supplementary Data). The formation of the parallel lines is a relatively slow process that occurs inside the NDT solution. Moreover the value of 15 Å matches well

with the distance between paired rows and can be explained by a pairing of molecules. Actually at ambient conditions the oxidation of two $-SH$ groups leads to the formation of disulfide as shown in Figure 6 [29, 43]. The observation of straight rows over large distances (~ 50 nm) as shown in Fig. 3-a suggests a self-organization process which starts from the moment of association of the first two molecules (discussed below) and supposes a strong surface reorganization. It was not possible to observe with STM the *in-situ* transition of a SAM from single $-SH$ groups, to double row. It suggests that the NDT SAM undergoes this transition in solution. The transition was only measured with contact angle technique.

In the case of BDT, the smaller proportion of striped domains (70% of the area) observed after 40 min dipping time indicates that the disulfide formation is not as likely to be formed and that the surface still contains unoxidized $-SH$ groups. In a previous study, it has been shown that BDT SAM formation leads to a mix of standing-up and flat-lying phases, contrary to NDT which leads to disulfides formation.[9] And thus, the disulfides formation is lower for BDT compare to NDT. This weaker tendency of BDT for forming rows can be explained by the shorter chain length and by the weaker van der Waals interactions between the alkyl chains in neighbouring molecules [33]. Such difference appeared due to (i) a stronger lateral interaction between $-[CH_2]-$ units of alkyl chains of NDT (~ 5 k.cal. mol.⁻¹) [42] and (ii) an increase of the steric effects confining the accessible range of tilt angles, for NDT during the process of the self-organization [42]. Notice that the concentration of the molecules during SAM formation was equivalent for both molecules. Therefore, concentration should probably have no influence on the ordering of the SAMs [25]. At the same time the more rigid structure of BDT will oppose the disulfide formation.

Figure 6. (a) Schematic model of the double row structure of dithiol molecules on Au(111) surface and (b) scheme of the oxidation of thiol ($-SH$) tail groups of alkyldithiols in presence of oxygen that result into disulfide.

This row-like surface of NDT turns out to be rather hydrophobic (contact angle of $68 \pm 0.4^\circ$) which is consistent with a $-SH$ terminated surface. Interestingly when the reaction time of NDT with Au(111) is increased from 30 sec to 40 min, the surface becomes more hydrophobic (contact angle of $74 \pm 0.5^\circ$, see Table S1 in the Supplementary Data). This slight contact angle increase is below the error bars and is consistent with the progression of the disulfide formation since the $-S-S-$ moiety is expected to be hydrophobic (non-polar bond). A short reaction time of NDT leads to a less disulfide and more available $-SH$ tail groups.

The NDT SAM is made of vertically oriented molecules which are densely packed. This monolayer has few defects. Therefore, the resulted contact angle is defined by the interaction of water and the thiol tail groups of NDT. When the alkyl chain length is shorter such as the case of BDT, the self-organization of the molecules into parallel vertical units is less favoured due to the weakening of van der Waals interaction between the short alkyl chains. Therefore the surface is only partially covered by domains with vertically oriented molecules. The residual surface is made of disordered domains, where the molecules could even adopt a low-coverage molecular phase with a horizontal orientation [33]. Therefore, the contact angle is not only defined by the interaction of water with the thiol tail groups but also with the hydrophobic

methylene groups of the alkyl chain. A similar impact of the methylene groups on the contact angle was observed for mixed SAMs formed by long alkyl chains which are partially folding on a vertically oriented short alkyl chains [12].

This formation of the disulfide can be used for controlling the attachment of AuNPs on the NDT SAM. The activation of such chemical reaction occurs in the presence of oxygen [47] and plays a critical role in biochemistry because it prevents the folding of some proteins and their oxidation [37]. The reaction of two terminal $-SH$ groups of neighbouring rows causes the formation of disulfide $-S-S-$ covalent bonds. AuNPs are known to bind to the SAM through the terminal SH groups. Disulfide can also bind to gold [15, 25, 31] but with a lower affinity. Navarro *et al.* studied the binding of disulfide on gold nanoparticles and concluded that the bond was weaker than with planar gold. The AFM results support this assumption. When **process #1** is used for preparing the NDT SAM, AuNPs can bind to the SAM and a surface coverage of $350 \text{ AuNPs}/\mu\text{m}^2$ is observed after 1 h immersion time. However, when **process #2** is used, disulfide is formed over the entire surface and AuNPs can hardly bind. A coverage of only $8 \text{ AuNPs}/\mu\text{m}^2$ is measured. For intermediate immersion time, an intermediate coverage is obtained (see Fig. S3). This low affinity of disulfide for attaching AuNPs can be used as a practical way for controlling the surface coverage of AuNPs on a SAM.

Conclusions

In summary, we have studied the formation of three SAMs on Au(111) with MUOH, BDT and NDT. MUOH is found to form a $(\sqrt{3} \times \sqrt{3})R30^\circ$ structure which is highly hydrophilic. For NDT and BDT, the terminal $-SH$ groups oxidized and create disulfide by pairing two neighbouring molecules and form rows spaced with $\sim 15 \text{ \AA}$. With short-chain BDT this process is limited to small domains whereas with NDT the formation of large row-like domains is enabled. A kinetic mechanism is also unravelled for NDT so that a short immersion (30 sec) in NDT will limit the formation of disulfide bridges and preserve $-SH$ tail groups. This property is

used for controlling the attachment of gold nanoparticles. When $-SH$ tail groups are available AuNPs can be attached whereas this attachment process is blocked when these groups react into disulfide. In the case of a NDT SAM prepared with a short immersion time (30 sec) we could deposit an amount of 500 AuNPs/ μm^2 .

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