



HAL
open science

Solid Fluoride Electrolytes and Their Composite with Carbon: Issues and Challenges for Rechargeable Solid State Fluoride-Ion Batteries

Antonin Grenier, Ana-Gabriela Porras Gutierrez, Monique Body, Christophe Legein, Fabien Chretien, Encarnación Raymundo-Piñero, Mickaël Dollé, Henri Groult, Damien Dambournet

► To cite this version:

Antonin Grenier, Ana-Gabriela Porras Gutierrez, Monique Body, Christophe Legein, Fabien Chretien, et al.. Solid Fluoride Electrolytes and Their Composite with Carbon: Issues and Challenges for Rechargeable Solid State Fluoride-Ion Batteries. *Journal of Physical Chemistry C*, 2017, 10.1021/acs.jpcc.7b07988 . hal-01628316

HAL Id: hal-01628316

<https://hal.sorbonne-universite.fr/hal-01628316>

Submitted on 3 Nov 2017

HAL is a multi-disciplinary open access archive for the deposit and dissemination of scientific research documents, whether they are published or not. The documents may come from teaching and research institutions in France or abroad, or from public or private research centers.

L'archive ouverte pluridisciplinaire **HAL**, est destinée au dépôt et à la diffusion de documents scientifiques de niveau recherche, publiés ou non, émanant des établissements d'enseignement et de recherche français ou étrangers, des laboratoires publics ou privés.

Article

Solid Fluoride Electrolytes and Their Composite with Carbon: Issues and Challenges for Rechargeable Solid State Fluoride-Ion Batteries

Antonin Grenier, Ana-Gabriela Porras Gutierrez, Monique Body, Christophe Legein, Fabien Chretien, Encarnacion Raymundo-Piñero, Mickael Dollé, Henri Groult, and Damien Dambournet

J. Phys. Chem. C, **Just Accepted Manuscript** • DOI: 10.1021/acs.jpcc.7b07988 • Publication Date (Web): 24 Oct 2017Downloaded from <http://pubs.acs.org> on October 25, 2017**Just Accepted**

“Just Accepted” manuscripts have been peer-reviewed and accepted for publication. They are posted online prior to technical editing, formatting for publication and author proofing. The American Chemical Society provides “Just Accepted” as a free service to the research community to expedite the dissemination of scientific material as soon as possible after acceptance. “Just Accepted” manuscripts appear in full in PDF format accompanied by an HTML abstract. “Just Accepted” manuscripts have been fully peer reviewed, but should not be considered the official version of record. They are accessible to all readers and citable by the Digital Object Identifier (DOI®). “Just Accepted” is an optional service offered to authors. Therefore, the “Just Accepted” Web site may not include all articles that will be published in the journal. After a manuscript is technically edited and formatted, it will be removed from the “Just Accepted” Web site and published as an ASAP article. Note that technical editing may introduce minor changes to the manuscript text and/or graphics which could affect content, and all legal disclaimers and ethical guidelines that apply to the journal pertain. ACS cannot be held responsible for errors or consequences arising from the use of information contained in these “Just Accepted” manuscripts.



Solid Fluoride Electrolytes and their Composite with Carbon: Issues and Challenges for Rechargeable Solid State Fluoride-Ion Batteries

Antonin Grenier,[†] Ana Gabriela Porras-Gutierrez,[†] Monique Body,[§] Christophe Legein,[§] Fabien Chrétien,[‡] Encarnación Raymundo-Piñero,^{‡,Π} Mickael Dollé,[⊥] Henri Groult,[†] and Damien Dambournet^{‡,Π,}*

[†] Sorbonne Universités, UPMC Univ. Paris 06, CNRS, UMR 8234, PHENIX, F-75005 Paris, France

[§] Université Bretagne Loire, Université du Maine, UMR CNRS 6283, Institut des Molécules et des Matériaux du Mans (IMMM), Avenue Olivier Messiaen, 72085 Le Mans Cedex 9, France

[‡] CNRS, CEMHTI UPR3079, Univ. Orléans, F-4071 Orléans, France

[⊥] Département de Chimie, Université de Montréal C.P. 6128, succursale Centre-Ville Montréal, QC, H3C 3J7 Canada

^Π Réseau sur le Stockage Electrochimique de l'Energie (RS2E), FR CNRS 3459, France

Abstract

Solid-state batteries relying on fluoride-ion shuttle are still at their early stage of development. Assessing the fluoride solid electrolyte's electrochemical stability and its conduction properties in a mixture with carbon, as well as the possible interaction of fluoride-ion with carbon both during the

1 electrode preparation and upon electrochemical reactions, are mandatory to enable future practical
2 applications. Here, we discuss these points using LaF_3 doped with BaF_2 ($\text{La}_{0.95}\text{Ba}_{0.05}\text{F}_{2.95}$, LBF) as a
3 benchmark solid fluoride electrolyte. We establish that lithium may be used as a pseudo-reference
4 electrode to assess the electrochemical stability window of LBF and support the experiment with
5 thermodynamic calculations. We demonstrate the chemical compatibility of LBF with carbon upon ball-
6 milling, and investigate the electrical conductivity of the formed LBF-C composite. We use a LBF|LBF-
7 C|LBF cell (in this configuration, LBF serves as electronically blocking electrode) to assess the ionic
8 conductivity of the LBF-C composite. The results confirm that both electronic and ionic percolations are
9 insured within the LBF-C composite despite a noticeable decrease of the ionic conductivity. Finally, we
10 use a Li|LBF|LBF-C cell to evaluate the electrochemical fluorination of the carbon in the LBF-C
11 composite. Our results suggest an electrochemical activity of carbon with fluoride ions. The possible
12 interactions of carbon with fluorides to form insulating carbon fluoride (CF_x) must be considered when
13 determining the operating voltage of FIBs.
14
15
16
17
18
19
20
21
22
23
24
25
26
27
28
29
30
31
32
33

34 Introduction

35
36 The electronegativity and low mass of fluorine make fluoride ion batteries (FIBs) promising candidates
37 within the scope of developing new battery chemistries providing increased energy densities compared
38 to conventional metal-ion batteries.¹ Anji Reddy and Fichtner were the first to provide a proof-of-
39 concept of a rechargeable solid-state cell², raising the awareness of a conversion-based chemistry first
40 proposed in the 1970s.³⁻¹² It was more recently demonstrated that fluoride ions could be reversibly
41 intercalated in LaSrMnO_4 ¹³, that CuF_2 was a reversible conversion cathode¹⁴, and that oxyfluorides are
42 possible conversion materials¹⁵, further increasing the interest of the community for this chemistry.
43
44
45
46
47
48
49
50

51 Tysonite type (trigonal, $P-3c1$) solid solutions like $\text{La}_{1-x}\text{Ba}_x\text{F}_{3-x}$ (LBF) have been the focus of some
52 research efforts with the purpose of understanding and increasing the fluoride conductivity to improve
53
54
55
56
57
58
59
60

1 the electrochemical performance of FIBs.^{16–22} Nanocrystalline fluorite-type (cubic, *Fm-3m*) solid
2 solutions have also been explored for their interesting fluoride conductivity properties.^{23–25} However,
3 the electrochemical stability of solid fluoride electrolytes has never been investigated so far. The
4 investigation of the electrochemical stability window is crucial to determine the potential range in which
5 the cell can operate without degradation of the electrolyte. This is perfectly illustrated by advances in
6 the field of solid-state lithium ion batteries. Despite the efforts made by the community to develop solid
7 electrolytes capable of ionic conductivities equivalent to those of liquids²⁶, the high-rate capability of
8 cells using such electrolytes remains inferior.²⁷ These efforts have recently been focused on interfacial
9 phenomena which are the cause of the slow electrochemical kinetics.^{28–30} It is now fair to say that the
10 principle obstacle for developing all-solid-state batteries capable of delivering relatively high current
11 densities relies in the achievement of low resistance interfaces.³¹ Electrochemical stability of the
12 electrolyte and electrochemical/chemical compatibility with the cell components is therefore crucial to
13 avoid the formation of interphases and layers that can significantly impede mass and/or charge
14 transport. Analogous observations can be made for solid fluoride electrolytes.

15
16
17
18
19
20
21
22
23
24
25
26
27
28
29
30
31
32
33 In this work, we discuss the fundamental challenges inherent to the use of fluoride-ion conductors in
34 rechargeable batteries. **Figure 1** shows a schematic of a typical solid state fluoride ion battery
35 illustrating the challenges we aim to discuss. To allow for reversible electrochemical reactions, suitable
36 electrochemical stability of the electrolyte must be ensured. Moreover, ionic and electronic conduction
37 must be achieved throughout the composite electrode. The ionic and electronic conduction in this
38 electrode is typically ensured by adding a solid electrolyte and carbon. This practice therefore raises the
39 question of the interaction of fluoride ions with carbon to yield insulating carbon fluorides CF_x , as
40 previously mentioned by Nowroozi *et al.*¹³

41
42
43
44
45
46
47
48
49
50
51
52 Addressing these challenges is significantly complexified by both the lack of reference electrode and
53 lack of inert electrode against fluorination. While electrochemical investigations in metal-ion batteries
54
55
56
57
58
59
60

are generally performed using the corresponding metal as a reference electrode, no such equivalent exists for FIBs, as the elemental form of fluoride is fluorine gas, F_2 . Concerning the study of the electrolyte electrochemical stability window, this is typically done using an electrode which is inert against the electrolyte oxidation. In Li-ion batteries, noble metals like platinum are used as inert electrodes as they are stable at high potentials, where electrolyte decomposition occurs. They are only limited by Li-alloying reactions at potential < 0.2 V vs. Li^+/Li .³² In FIBs, the electrolyte decomposition is accompanied by fluorine evolution, which is highly reactive and able to form compounds with all materials typically used as inert electrode.

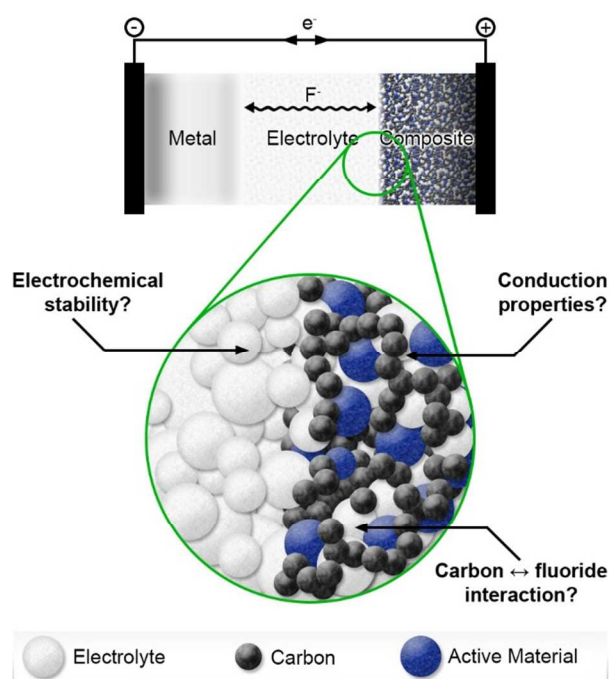


Figure 1. Schematic representation of a solid state fluoride ion battery with a zoom in on the interface between the electrolyte and the composite electrode. The arrows and questions illustrate the challenges that need to be addressed.

In an attempt to meet these challenges, we selected nanocrystalline LBF as model solid electrolyte since it exhibits a relatively high ionic conductivity (about 10^{-4} S.cm⁻¹ at 160 °C²⁰) and its conduction properties have been extensively studied.^{18,20,22} Moreover, LBF was successfully used in FIBs.^{2,13,14,33–35}

1 First, the suitability of Li as electrode for FIBs is assessed by time resolved electrochemical impedance
2 spectroscopy (EIS). Secondly, the electrochemical stability of LBF is studied using Li and Pt electrodes.
3
4 Thirdly, the electronic and ionic conductivity of a LBF-carbon (LBF-C) composite is investigated and
5
6 finally, the LBF-C is used to assess the interaction of carbon with fluorides.
7
8
9

10 11 12 **Experimental Section**

13
14 **Characterizations.** Powder X-ray diffraction (XRD) measurements were performed using a Rigaku
15
16 Ultima IV diffractometer with a Cu K α radiation ($\lambda = 1.5418 \text{ \AA}$), at a 2θ scan rate of $0.1^\circ/\text{min}$ and a step
17
18 of 0.017° , between 10° and 80° . Transmission electron microscopy (TEM) images were obtained with a
19
20 JEOL JEM 2011 on carbon coated copper grids impregnated by a suspension of the material in ethanol.
21
22

23
24 **Preparation of the materials.** The electrolyte, $\text{La}_{0.95}\text{Ba}_{0.05}\text{F}_{2.95}$ (LBF), was prepared by ball-milling
25
26 LaF_3 (Alfa Aesar, 99.99 %) and BaF_2 (Strem chemicals, 99.99 %) in N_2 atmosphere at 400 rpm for 12
27
28 hours with a Fritsch pulverisette 7 premium line planetary ball mill with ZrO_2 jars and milling balls (10
29
30 mm diameter). The detailed preparation of the electrolyte and the influence of the milling parameters on
31
32 its purity is detailed in the work of Chable *et al.*²² The XRD pattern of LBF can be found in **Figure S2**
33
34 and a presentation of its crystal structure in **Figure S3**. The LBF-C composite was obtained by an
35
36 additional milling step using the obtained LBF powder (90 wt %) and carbon black (Pure Black,
37
38 Superior Graphite, 10 wt %, see supporting information for details), denoted C. The mixture was ball-
39
40 milled in conditions similar as LBF preparation, for 4 hours. The carbon black was first outgassed for
41
42 12h at 150°C . Graphite (Cg) disks were cut from high purity graphite rods (Mersen, $\varnothing 10 \text{ mm}$), polished
43
44 (P400 SiC abrasive paper), sonicated 3 times in ultrapure water and dried overnight under vacuum at
45
46 150°C . Typical thickness of the graphite disks was 0.5 – 0.7 mm. Lithium foil (99.9%) was purchased
47
48 from Sigma Aldrich, and Pt foil (99.9 %) was polished (P2400 SiC abrasive paper) and rinsed before
49
50
51
52
53
54
55
56
57
58
59
60

1 use. Silver (Ag) conductive paint (RS Components) was painted on the LBF surface and dried at 150 °C
2 under vacuum for 12 h.
3

4 **Cell assembly.** Powders were pressed into 10 mm pellets at 5 t in a dry Ar glove box. The Li|LBF|Li
5 and Li|LBF|Pt cells were obtained by placing a pre-compacted LBF pellet (300 mg) between Li or Pt
6 disks. For the Ag|LBF|LBF-C|LBF|Ag cell, 30 mg of LBF-C and 300 mg of LBF powders were stacked
7 and pressed in a single step, and the pellet painted with Ag paint on each side. For the Li|LBF|LBF-C
8 and Li|LBF|C cells, a Cg disk was added to form LBF|LBF-C|Cg and LBF|C|Cg pellet stacks, using 15
9 mg of LBF-C and 5 mg of C, respectively. The stacks were then laid on a Li disk. All cell preparation
10 and assemblies were carried out in a dry Ar glove box. A summary of the cell configurations used in this
11 study is available **Figure S1** (supporting information). The stacks described above were mounted in
12 modified CR2032 coin-cells. A flat PTFE washer of the desired thickness was used to maintain the
13 stack at the center of the coin cell. A flat spring was used to maintain electrical between the stack and
14 the coin-cell case. Spacers (stainless steel disks) were used to adjust the thickness of the pellet stack to
15 the space available in the coin-cell. The crimped coin-cells were then tightened on a custom made
16 electrochemical test bench employing a screw system, so that the stacks could remain under constant
17 pressure for the duration of the electrochemical measurements. The test bench was placed in an oven set
18 to 30 °C. The cells were left at the designated temperature for at least 3 hours to allow reaching thermal
19 equilibrium before electrochemical measurements.
20
21
22
23
24
25
26
27
28
29
30
31
32
33
34
35
36
37
38
39
40
41

42 **Electrochemical measurements.** Electrochemical impedance spectroscopy (EIS) was recorded in the
43 13 MHz – 5 Hz frequency range, with a HP 4192A LF impedance analyzer, or in the 300 kHz – 3 mHz
44 frequency range, with a BioLogic VSP potentiostat equipped with a low current extension. The
45 amplitude of the sinusoidal potential excitation was adjusted in the 43 mV – 300 mV range depending
46 on the impedance of the cell, after checking the linearity of the measurement. Impedance spectra were
47 fitted using EC-Lab Zfit. Cyclic voltammetry (CV) was recorded at 30 °C with a BioLogic VSP
48
49
50
51
52
53
54
55
56
57
58
59
60

1 potentiostat channel equipped with a low current extension. All experiments were carried out at a scan
2 rate of $0.1 \text{ mV}\cdot\text{s}^{-1}$ between 7 V and -0.2 V, starting from OCV.

3
4 The ohmic drop was corrected using the resistance of the electrolyte determined from EIS
5 measurements. The current density, rather than the current, was used to allow comparison of the cells
6 during the investigation of the electrochemical compatibility of LBF with carbon. The contact area A of
7 the LBF|WE (working electrode) developed at the interface was estimated using the relation:
8
9
10
11
12

$$13 \quad A = \frac{L}{\sigma_i(\text{LBF}) \cdot R_{gb}}$$

14
15
16
17 With L the thickness of the LBF layer, σ_i the LBF conductivity at 30°C determined from conductivity
18 measurements on the Ag|LBF|Ag cell, and R_{gb} the resistance of the electrolyte determined from EIS
19 measurements on the relevant cell (see the supporting information for a detailed interpretation of the
20 EIS spectra obtained from the conductivity measurements). The thickness of the electrolyte was
21 measured with a micrometer and assumed to be the same for all experiments as the same mass of LBF
22 was used.
23
24
25
26
27
28
29
30

31 **^{19}F solid state Nuclear Magnetic Resonance (NMR).** ^{19}F solid-state Magic Angle Spinning (MAS)
32 NMR spectra were recorded on an Avance III Bruker spectrometer operating at 7 T (^{19}F Larmor
33 frequency of 282.2 MHz), using a 1.3 mm diameter probe head allowing spinning frequencies up to
34 70 kHz. One dimensional (1D) NMR spectra were recorded using a Hahn echo sequence for which
35 interpulse delay was synchronized with the rotor spinning frequency. A 90° pulse length of $1.25 \mu\text{s}$ was
36 used, recycle delays of 2 s were applied and 128 transients were accumulated. Due to air frictional
37 heating, the sample temperature varied by up to 32°C from 34 to 64 kHz (^{207}Pb isotropic chemical shift
38 of $\text{Pb}(\text{NO}_3)_2$ was used as a NMR thermometer^{36,37}). The maximum temperature gradient over the
39 dimension of the 1.3 mm rotor was estimated to be around 8°C . The ^{19}F chemical shifts are referenced
40 to CFCl_3 at 0 ppm.
41
42
43
44
45
46
47
48
49
50
51
52
53
54
55
56
57
58
59
60

Results & Discussion

1. Interface of LBF with lithium and electrochemical stability

Lithium metal was selected as counter electrode as it offers practical advantages such as a relatively low oxygen sensitivity (compared to lanthanides³³) and softness, providing good contact with the solid electrolyte. Due to the low potential of Li, reaction of LBF to yield LiF and metallic La (and Ba) is expected according to:



The calculated Gibbs free energies ($\Delta_r G^\circ = -120 \text{ kJ}\cdot\text{mol}^{-1}$ and $-15 \text{ kJ}\cdot\text{mol}^{-1}$ for LaF_3 and BaF_2 , respectively) of the corresponding reactions indicate that spontaneous reduction of LBF must occur. The formation of the interface should provide a convenient pseudo-reference redox couple as long as the interface is not entirely blocking. To investigate the interface properties of Li with LBF, a Li|LBF|Li cell was assembled and monitored by time-resolved electrochemical impedance spectroscopy (EIS) (**Figure 2**).

After 3 hours of contact, the impedance spectrum exhibits a complete semi-circle and an additional semi-circle, followed by a capacitive tail. The first semi-circle is characteristic of the electrolyte's ionic resistance and is denoted R_{gb} (see supporting information for more details). Over time, the resistance of the first semi-circle is stable, while the resistance of the second semi-circle increases. The fitting of all spectra was carried out with the equivalent circuit presented on **Figure 2 (c)**. The resistances of the two semi-circles, gathered from the fits, are reported on **Figure 2 (b)**.

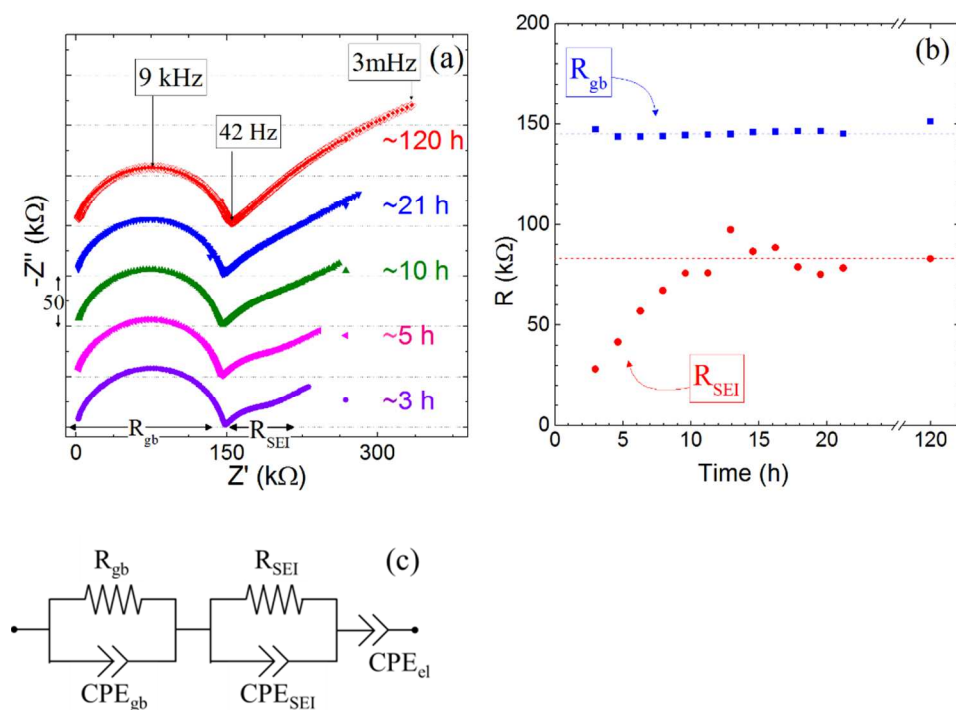


Figure 2. (a) Nyquist plot of the impedance spectra of a Li|LBF|Li cell in open circuit conditions at 30 °C. The frequency, in Hz, of a few points is indicated for comparison with **Figure S4** (b) Evolution of R_{gb} and R_{SEI} in function of time under open circuit conditions. (c) Equivalent circuit used to fit the data.

The appearance of a second semi-circle is attributed to the expected formation of a LiF|La-Ba interface layer resulting from the localized oxidation of Li to LiF and reduction of LBF to a La-Ba mixture. This layer can be described as a solid electrolyte interface (SEI) and its resistance is therefore denoted R_{SEI} . While the value of R_{gb} does not significantly vary over time, R_{SEI} gradually increases and stabilizes at around 82 kΩ after 15 h. Accordingly, the increase of R_{SEI} prior to its stabilization can be ascribed to the increase of the thickness of the SEI until it becomes sufficiently thick to impede significant local charge/mass transfer, and reach equilibrium. Nonetheless, R_{SEI} (≈ 41 kΩ for a single Li|LBF interface) is inferior to the electrolyte resistance R_{gb} (≈ 145 kΩ) so that the overall cell impedance remains sufficiently low to perform electrochemical characterizations. All these observations suggest that Li is a suitable electrode for the measurement of the electrochemical stability of fluoride solid electrolytes.

To evaluate the experimental electrochemical stability of LBF, a Li|LBF|Pt cell was assembled and cyclic voltammetry was performed at 0.1 mV/s scan rate, at 30 °C (**Figure 3**). The cell was polarized up to 7 V and down to -0.5 V, for two cycles. Two main features can be observed. One is the occurrence of redox peaks at around 0 V. The other is the oxidation current observed beyond 5 V. The redox peaks at around 0 V are typically observed with Li⁺ conducting electrolytes in analogous cells (Li|Li-electrolyte|Pt) and are generally attributed to the lithium plating/alloying and stripping (reduction and oxidation) at the Pt electrode and concomitant lithium oxidation and reduction at the Li electrode^{26,38-41}. Considering the structure of LBF, it is unlikely that Li⁺ can migrate through the electrolyte from the Li electrode to the Pt electrode. Therefore, the redox peaks around 0 V can be assigned to redox reactions involving LBF at both the Li and Pt electrode (*i.e.*, $La/Ba_{(s)} + 3F^- \leftrightarrow LBF_{(s)} + 3e^-$). These observations suggest that the SEI exhibits some reversibility, although further characterization would be needed to confirm this observation.

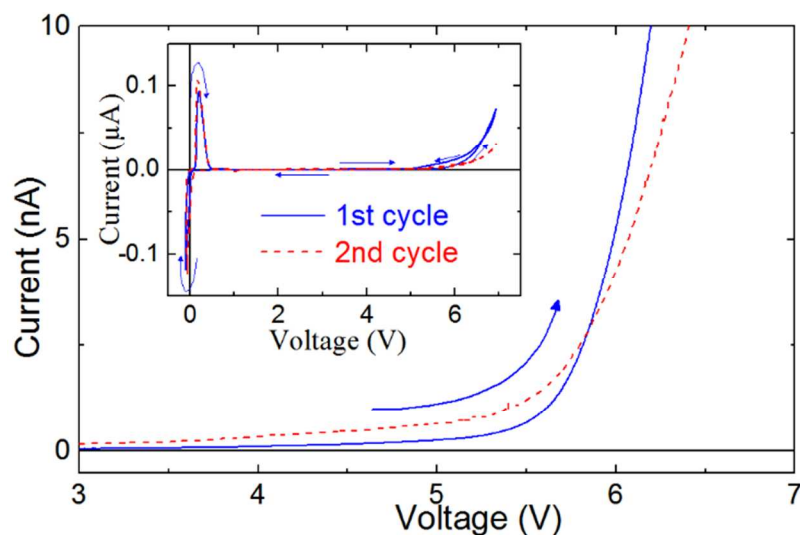


Figure 3. Cyclic voltammograms (forward scans from 3 to 7 V) of the Li|LBF|Pt cell performed at a 0.1 mV/s scan rate, at 30 °C. Inset: full scans between -0.05 and 7 V.

1 The steep increase of the oxidation current beyond 5 V can be ascribed to the electrochemical
2 decomposition of LBF. The decomposition potential can be calculated from thermodynamic data by
3 considering the following reaction:
4
5



6
7 The theoretical potential of such reaction is given by the following equations:
8
9

$$\Delta_r G^\circ = -nF\Delta E^\circ = \Delta_f H^\circ - T\Delta S^\circ \quad 3.$$

10 with n the number of moles of electrons exchanged upon decomposition, F the Faraday constant (96485
11 C.mol⁻¹), T the absolute temperature (K), and $\Delta_f H^\circ$ (kJ.mol⁻¹) and ΔS° (kJ.mol⁻¹.K⁻¹) are respectively the
12 standard enthalpy and entropy of the reaction. Calculations based on the aforementioned equations using
13 thermodynamic data from The Materials Project⁴² yield ΔE° of 5.73 V and 6.00 V for LaF₃ and BaF₂,
14 respectively, which is in good agreement with the value of 5.811 V calculated by Roos and Schoonman
15 for LaF₃.⁴³ The calculated decomposition potential of LaF₃ is in good agreement with our experimental
16 observations as the current dramatically increases beyond 5.8 V. Although, it is obvious that the current
17 significantly increases before the calculated thermodynamic value of 5.73 V. This current may be
18 attributed to the fluorination of the Pt electrode prior to F₂ evolution. The fluorination of Pt is further
19 suggested by the less steep increase of the decomposition current in the 2nd cycle (dashed line on **Figure**
20 **3**) which could be explained by the insulating effect of the PtF_x layer formed on the surface of Pt
21 electrode during the 1st cycle. For instance, the calculated potential of the reaction of Pt with LaF₃ (e.g.
22 3 Pt + 2 LaF₃ → 3 PtF₂ + 2 La) is ~4.17 V based on thermodynamic data of PtF₂ from Glassner⁴⁴.
23 Therefore, polarization of a Li|LBF|Pt cell beyond 4.17 V induces LBF reduction at the Li electrode,
24 and F₂ evolution and/or Pt fluorination at the Pt electrode. Experimental observations made by
25 Bezmelnitsyn et al.⁴⁵ are in agreement with this mechanism. They polarized a M|LBF|M cell (M = Ni,
26 Cu or graphite) by applying a galvanostatic current of 1.0 mA and observed F₂ evolution and the
27 formation of La/Ba dendrites at the negative electrode. XPS analysis of the dendrites showed that it was
28
29
30
31
32
33
34
35
36
37
38
39
40
41
42
43
44
45
46
47
48
49
50
51
52
53
54
55
56
57
58
59
60

1
2
3
4
5
6
7
8
9
10
11
12
13
14
15
16
17
18
19
20
21
22
23
24
25
26
27
28
29
30
31
32
33
34
35
36
37
38
39
40
41
42
43
44
45
46
47
48
49
50
51
52
53
54
55
56
57
58
59
60

a mixture of La and Ba metal with a Ba/La ratio corresponding to that in the LBF. Our experimental observations are summarized by the schematic illustration of the decomposition mechanism displayed on **Figure 4**.

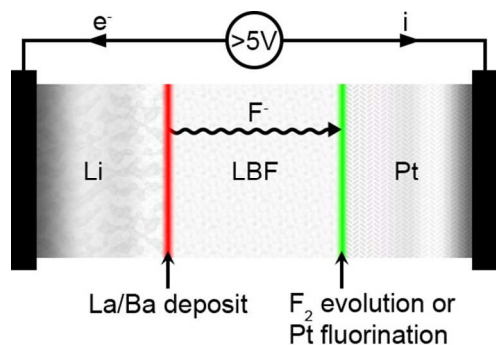


Figure 4. Schematic illustration of the decomposition mechanism of LBF.

In conclusion, we established that lithium is a suitable electrode for the measurement of the electrochemical stability of solid fluoride electrolytes. The SEI formed upon contacting Li with the fluoride solid electrolyte is not entirely blocking, and the presence of the LiF|La-Ba interface layer provides a redox couple that can be conveniently used to evaluate the electrochemical stability of the solid fluoride electrolyte. It must be noted that since the LiF would form in contact with the Li surface and the La-Ba mixture in contact with the LBF, the interface can be noted as Li|LiF|La-Ba|LBF. It can therefore be expected that the redox couple created at the Li|LiF|La-Ba|LBF interface is the LBF/La-Ba couple (i.e., La³⁺/La and Ba²⁺/Ba), the LiF|Li interphase acting as “current collector”. This result is promising as it is the first step in addressing a major challenge inherent to the study of FIBs, which is the lack of a reference electrode. Further work is required to verify whether this interphase provides a proper reference electrode, *i.e.*, the LBF/La-Ba couple at the interface is kinetically fast enough to provide a non-polarizable electrode. Nonetheless, the LBF/La-Ba couple provides a baseline from which thermodynamic calculations of the cell redox potentials can be performed. The potentials obtained from those calculations can be used to assign variations of the cell measured current to possible electrochemical reactions, bearing in mind the possible presence of overpotentials.

1 As for the electrochemical stability of LBF, we can conclude that its theoretical decomposition potential
2 (5.73 V), corresponding to F₂ evolution at the Pt electrode, is largely above the theoretical potential at
3 which the working electrode would be fluorinated. In our Li|LBF|Pt cell, we observed the apparition of
4 an oxidation current at potentials below that of the theoretical decomposition potential of LBF. This can
5 be explained by the fluorination of Pt prior to F₂ evolution at the Pt electrode or by the catalytic effect of
6 the Pt. Experimental observations of the actual decomposition potential will be very challenging
7 considering that unlike in metal-ion batteries, there is effectively no inert material against fluorine-ion.
8 Therefore, we propose that the best way to avoid reduction of the electrolyte is to calculate the redox
9 potential of electrochemical reactions involving both the FIBs' active materials and the electrolyte to
10 verify that the redox potential of reactions involving the active materials are below the redox potential
11 of reactions involving the electrolyte. In the case of solid solutions like LBF, we suggest doing
12 thermodynamic calculations using the electrolyte components (LaF₃ and BaF₂) as no thermodynamic
13 data are available for LBF. For instance, in a Bi|LBF|MgF₂ cell, the calculated redox potential of the
14 reaction $2 \text{ Bi} + 3 \text{ MgF}_2 \leftrightarrow 2 \text{ BiF}_3 + 3 \text{ Mg}$ is 2.65 V, below the values calculated for reactions involving
15 LaF₃ and BaF₂, i.e., $\text{Bi} + \text{LaF}_3 \leftrightarrow \text{BiF}_3 + \text{La}$ (2.83 V) and $2 \text{ Bi} + 3 \text{ BaF}_2 \leftrightarrow 2 \text{ BiF}_3 + 3 \text{ Ba}$ (3.11 V).
16 Reduction of MgF₂ should therefore occur before reduction of LBF.
17
18
19
20
21
22
23
24
25
26
27
28
29
30
31
32
33
34
35
36
37
38
39
40

41 2. Structural and electrical characterization of the LBF-C composite.

42 Metal fluorides are generally poor electronic conductors due to their large band gap, as predicted by the
43 band theory (e.g 4.681 eV for BiF₃⁴²). Their use as active material requires the development of
44 composites containing electronic conductive additives to ensure the electronic percolation in the
45 electrode. This is typically achieved by mixing carbon black, or other carbonaceous materials, with the
46 active material. Ball-milling is a common method to produce metal fluoride-carbon nanocomposites that
47
48
49
50
51
52
53
54
55
56
57
58
59
60

1 present good performance in conventional lithium-ion batteries.⁴⁶ The same approach can be used to
2 produce electrode composites for solid-state lithium and fluoride-ion batteries.
3

4 Additionally, solid electrolyte must be mixed with the active material and the carbonaceous additive to
5 achieve ionic conductivity within the electrode. Despite the fact that some metal fluorides available as
6 active material offer ionic conductivity properties (e.g. $\sigma_i \approx 1 \times 10^{-6}$ S.cm⁻¹ at RT for cubic BiF₃¹⁰), their
7 ionic conductivity is generally lower than that of LBF. Besides, upon operation of the FIB, the metal
8 fluoride will lose its ionic conductivity properties when reduced to its metallic state. Therefore, both
9 carbon and electrolyte must be present in the electrode composites to ensure electronic and ionic
10 conductivity through the bulk of the electrode. This raises the question of the interaction of
11 carbonaceous additives with the fluoride conducting electrolyte, both upon preparation of the electrode
12 materials and upon operation of the cell. Indeed, the fluorination of the carbonaceous material to yield
13 insulating graphite fluorite, CF_x, would be detrimental to the electrochemical performance of the
14 electrode.
15
16
17
18
19
20
21
22
23
24
25
26
27
28
29
30

31 To address this question, LBF and carbon black were mixed in a 9:1 weight ratio and ball-milled to
32 yield the LBF-C composite. The influence of ball-milling on the structure and composition of the
33 starting materials was investigated, the electrical properties of the LBF-C composite were measured by
34 means of electrochemical impedance spectroscopy, and the LBF-C was used as working electrode in a
35 Li|LBF|LBF-C to assess the fluorination of carbon.
36
37
38
39
40
41

42
43 a. Structural characterization of the LBF-C composite.
44

45 The XRD pattern of the LBF-C composite (**Figure S5**) indicates that no observable structural
46 modification of LBF takes place during the ball-milling process. The possible formation of CF_x during
47 ball-milling ($\text{LaF}_3 + \text{C} \rightarrow \text{LaF}_{3-x} + \text{CF}_x$)⁴⁷ was investigated using ¹⁹F NMR (**Figure S6**). ¹⁹F NMR shows
48 no sign of the formation of CF_x groups typically characterized by resonances in the -50 – -300 ppm
49 region.⁴⁸⁻⁵⁰ NMR further confirms that no changes occur in LBF as the ¹⁹F signal gives identical shifts
50
51
52
53
54
55
56
57
58
59
60

before and after ball milling. TEM images (**Figure S7**) show agglomerates of LBF of size ranging from a few dozen to several hundred nanometres, surrounded by large agglomerates of carbon black particles.

b. Electrical characterization of the LBF-C composite.

The electrochemical properties of the LBF-C composite was first investigated by EIS performed on a graphite|LBF-C|graphite cell denoted Cg|LBF-C|Cg thereafter. A Cg|Cg cell was used as a blank to deduct the electronic conductivity of the composite. Figure 5 shows the impedance spectra of the Cg|LBF-C|Cg and Cg|Cg cells recorded at 30 °C. The presented impedance spectra are characteristic of electronic conductors (single resistor), but present an inductive contribution at high frequencies, due to the wiring of the measurement setup. The Nyquist plot of a resistor ($Z_R = R$) is characterized by a single value on the x axis, yielding its resistance.

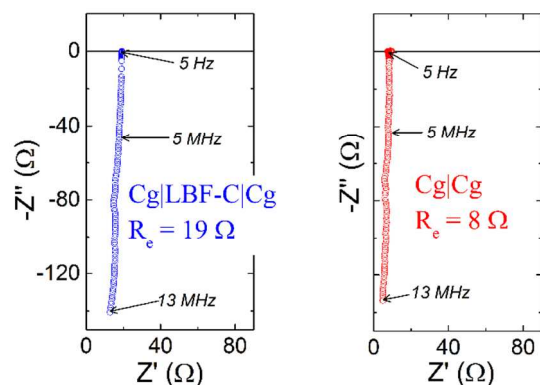
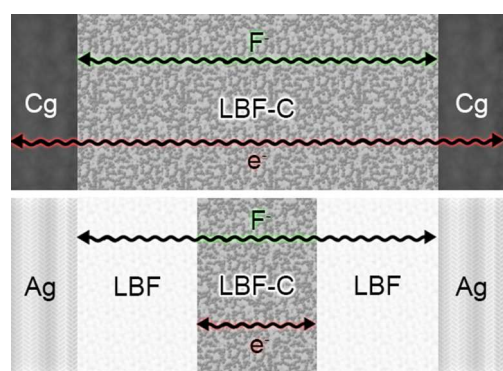


Figure 5. Impedance spectra of a Cg|LBF-C|Cg (a) and Cg|Cg (b) cell at 30 °C (Cg = carbon graphite disk).

The intersection of the inductive tail with the x axis ($-Z'' = 0$) gives the electronic resistance of the cell, R_e , yielding values of 19 Ω and 8 Ω, for Cg|LBF-C|Cg and Cg|Cg, respectively. An electronic resistance of 11 Ω is deduced for the LBF-C composite, yielding an electronic conductivity, σ_e , of about 10^{-2} S.cm⁻¹. This result suggests that the electronic percolation threshold is reached for 10 wt % C. Moreover, the electronic conductivity of LBF-C is several orders of magnitude higher than the ionic conductivity of

1 pure LBF (almost 5 orders of magnitude at 30 °C), indicating that the electronic conductivity of the
2 composite electrode is not the rate limiting phenomena.
3

4
5 The determination of the ionic conductivity of the LBF-C composite in the Cg|LBF-C|Cg setup is not
6 possible due to shorting created by the carbon present in the composite, as illustrated on **Figure 6**. To
7 counter this, LBF can be used as an electronically blocking electrode. The electrons are blocked at the
8 LBF|LBF-C interface so that the electronic conductivity of LBF-C does not short circuit the cell.
9
10 Consequently, the ionic conductivity of both LBF and LBF-C can be determined.
11
12
13
14
15
16
17
18
19



20
21
22
23
24
25
26
27
28
29
30
31
32 **Figure 6.** Schematic representation of the Cg|LBF-C|Cg and Ag|LBF|LBF-C|LBF|Ag cells used to
33 determine the electronic and ionic conductivity of LBF-C, respectively. Schematic electronic (e-) and
34 ionic (F-) migration routes are shown for the sake of clarity.
35
36
37
38
39

40 The impedance spectrum of the Ag|LBF|LBF-C|LBF|Ag cell recorded at 30 °C is shown on **Figure 7**.
41 The spectrum of a Ag|LBF|Ag cell recorded at the same temperature is also added for comparison. Both
42 spectra present a semi-circle, characteristic of the ionic conductivity, followed by a capacitive tail,
43 characteristic of blocking effects at the Ag electrodes. The frequency shift observed for the stack
44 configuration when compared to the sole LBF might be due to the slower migration of F⁻ in the LBF-C
45 layer, thus increasing the characteristic time of the process, thus decreasing the characteristic frequency.
46
47
48
49
50
51
52
53
54
55
56
57
58
59
60

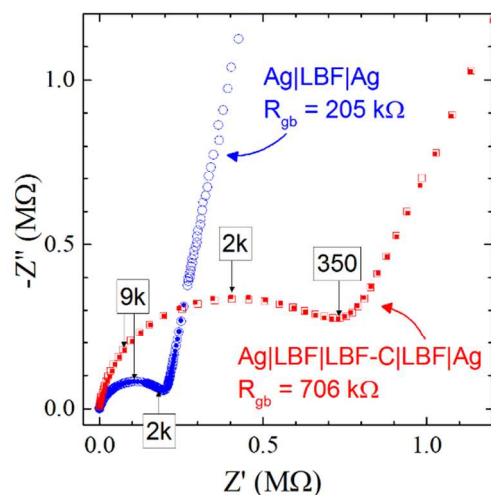


Figure 7. Nyquist plot of the impedance spectra of the \square Ag|LBF|LBF-C|LBF|Ag and \circ Ag|LBF|Ag cells recorded at 30 °C. The open symbols represent the data and the closed symbols represent the fits obtained.

The ionic conductivity of the LBF-C layer can be determined by subtracting the contribution of the LBF layers. The resistance of the LBF-C layer corresponds to the resistance of the Ag|LBF|LBF-C|LBF|Ag stack (706 k Ω) minus the resistance of the two LBF layers (2×205 k Ω). A resistance of 296 k Ω is obtained for the LBF-C layer yielding an ionic conductivity of 5×10^{-8} S.cm $^{-1}$ at 30 °C. This value is about one order of magnitude lower than the ionic conductivity of LBF alone (5.4×10^{-7} S.cm $^{-1}$, see supporting information) showing that the ionic percolation is maintained but also affected by the preparation of the LBF-C composite.

The fluoride ions mobility in the LBF-C composite was further investigated by ^{19}F MAS NMR (Figure 8). As previously reported for LBF, the F2 and F3 NMR resonances (having the largest isotropic chemical shifts) are no longer resolved and partially overlap with that of F1 (preventing reliable calculations of their relative intensities)¹⁸. These broadenings reflect the enhanced disorder induced by the ball milling technique. In the tysonite type solid solutions $\text{La}_{1-x}\text{M}_x\text{F}_{3-x}$, because F1–F1 exchanges are considerably faster than F1–F2,3 exchanges, the exchange between F1 and F2,3 results in a shift toward a larger frequency of the NMR resonance assigned to F1 and in a decrease of the relative intensity of the

F2,3 resonances.^{17,18} This is what is observed, when the temperature increases, for the LBF-C composite, to a similar extent as LBF,¹⁸ showing that the fluoride ions mobility is not impacted by the presence of carbon.

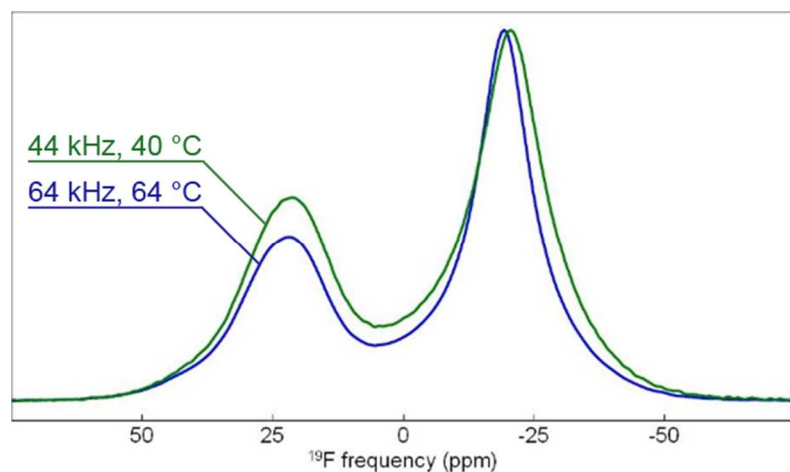


Figure 8. ¹⁹F MAS (64 kHz at 64°C, and 44 kHz at 40°C) NMR spectra of LBF-C.

Note that, the precise way the carbon interacts with LBF, *i.e.*, only due to morphology effects, or together with grain boundary effects, could be investigated in more details using dielectric spectroscopy⁵¹.

In summary, the investigation of the electrical properties of the LBF-C material revealed that both electronic ($\sigma_e \approx 10^{-2}$ S.cm⁻¹) and ionic percolations ($\sigma_i \approx 5 \times 10^{-8}$ S.cm⁻¹) are insured within the composite. The methodology adopted here provides a versatile tool to optimize the composite electrode by obtaining the best compromise between electronic and ionic conductivity.

3. Electrochemical activity of carbon

While in lithium batteries, it is well established that carbonaceous additives may contribute to the overall capacity due to the insertion of lithium ions at low potential, the possible electrochemical fluorination of carbon in FIBs is still an open question¹³. It is particularly important to address this question since the fluorination of carbon yields CF_x compounds which are electronic insulators. To investigate potential reactions with carbon, the electrochemical reactions occurring in Li|LBF|C and

Li|LBF|LBF-C cells were compared with a benchmark cell with no carbon, *i.e.*, Li|LBF|Pt cell. The cells were polarized up to 7 V. The complete CV scans on Li|LBF|C and Li|LBF|LBF-C cells can be found in supporting information (**Figures S7 and S8**, respectively). **Figure 9** shows the first cycle forward scans from 3 V up to 7 V and their corresponding derived curves for Li|LBF|Pt (from **Figure 3**), Li|LBF|C and Li|LBF|LBF-C cells.

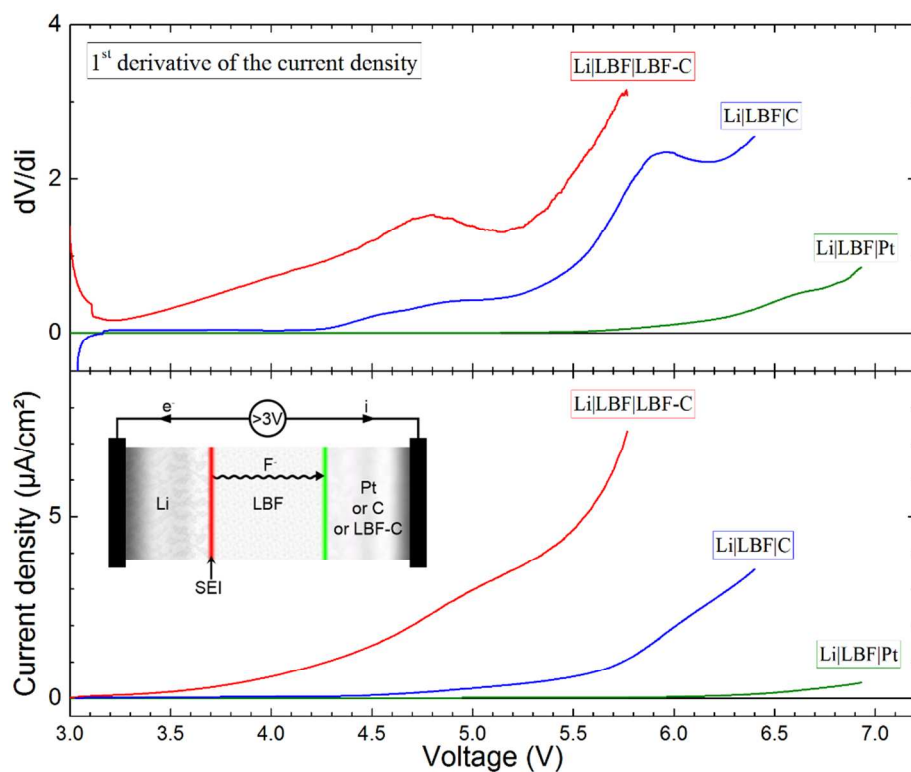


Figure 9. Overlaid forward CV scans and their 1st derivatives for the 3 Li based cells with different working electrodes: Pt, C and LBF-C. The curves were corrected for iR drop and normalized with the current density. Inset: schematic representation of the polarized cell illustrating the migration of fluoride ions towards the working electrode.

Upon polarization, fluoride ions are moving from the cathode side (SEI) towards the Pt or carbon-based working electrode anode, as represented on the schematic in inset of **Figure 9**. To ease the

1
2
3 discussion, the cells will be referred by the name of their respective electrode (*i.e.*, Pt cell, C cell and
4 LBF-C cell).
5
6

7
8
9 The voltammogram of the benchmark cell (Pt) was previously discussed (**Figure 3**) and is added for
10 comparison. The earlier onset of the current can be attributed to fluorination of Pt before F₂
11 evolution.
12
13
14

15
16
17 In the presence of carbon, *i.e.*, LBF-C and C cells, the current increases at lower voltages. The
18 derivative curves feature two main contributions located at 4.2-4.8 V, and beyond 5.2 V. The later
19 can be assigned to the electrochemical decomposition of the electrolyte. The early onset of the
20 decomposition of LBF in the LBF-C cell might be due to the increased contact area between the
21 solid electrolyte and the carbon that accelerates the kinetic of the reaction.⁵² The increase of the
22 current at lower voltages (4.2-4.8 V) suggests an electrochemical reaction of carbon. Carbon might
23 react with LaF₃ (and/or BaF₂) from LBF to yield CF_x as follow:
24
25
26
27
28
29
30
31
32



34
35
36 The standard potential of the electrochemical fluorination reaction of carbon was calculated using
37 the enthalpy of formation of CF_x obtained from the literature. According to Valerga *et al.*⁵³, the
38 entropy and enthalpy of formation of CF_x depends on the F/C ratio and can be calculated (at 298 K)
39 as follows:
40
41
42
43
44

$$45 \quad \Delta_f H_{CF_x}^0 = (4.53 - 178.01 x) \text{ kJ} \cdot \text{mol}^{-1} \quad 5.$$

$$46 \quad S_{CF_x}^0 = (4.62 - 16.91 x) \text{ J} \cdot \text{K}^{-1} \cdot \text{mol}^{-1} \quad 6.$$

47
48
49 The calculated potential for the formation of CF is about 4.19 V for a reaction involving LaF₃ and
50 4.47 V for a reaction involving BaF₂. The value of 4.19 V is close to the first contribution to the
51 current observed on the CV of the C and LBF-C cells, suggesting that carbon undergoes
52 electrochemical fluorination to form CF_x. This is particularly clear for the C electrode, as the
53
54
55
56
57
58
59
60

1
2
3 derivative shows a significant increase of the current observed at about 4.3 V, which is concomitant
4
5 with the calculated potential for the reaction of LaF_3 to form CF_x at the LBF|C interface (4.19 V).
6
7 The slight change of slope at about 4.5 V might be due to the onset of reactions involving the Ba^{2+} of
8
9 LBF to form CF (4.47 V). Finally, a frank change of slope is observed above 5.5 V, which
10
11 corresponds to the decomposition of LaF_3 from LBF (5.73 V) to yield elemental fluorine. In the case
12
13 of the LBF-C cell, it is difficult to accurately determine the onset of the current related to the carbon
14
15 reactivity as current increases as soon as the cell is polarized. This phenomenon may be attributed to
16
17 non-faradaic (i.e. capacitive) processes building up at the interface between LBF and C.
18
19 Nonetheless, the CV derivative of the LBF-C cell yields a similar shape than that of the C cell so that
20
21 similar conclusions can be made about the fluorination of carbon.
22
23
24
25
26

27 Our observations about the carbon's electrochemical activity with fluoride is in good agreement with
28
29 the work of Nowroozi *et al.*¹³ who showed that the electrochemical fluoride ion insertion into
30
31 LaSrMnO_4 is accompanied by an extra capacity that can be attributed to the fluorination of carbon to
32
33 form CF_x . Because CF_x compounds exhibit insulator properties due to the covalent nature of the C-F
34
35 bond inducing a sp^3 hybridization in these compounds⁵⁴, the progressive fluorination of the carbon
36
37 might eventually lead to a progressive loss of the operational electronic conduction pathways. One
38
39 might consider that the electrochemical window needs to be narrowed to avoid the formation of
40
41 carbon fluorides, as these processes are generally considered to be irreversible.
42
43
44
45

46 Another possible interaction of carbon with fluorides might be the formation of graphite
47
48 intercalation compounds (GICs) in which intercalation might be reversible, as it is the case in
49
50 lithium-ion batteries. To the best of our knowledge, no thermodynamic data are available on these
51
52 compounds so that no theoretical voltage could be calculated. Further experiments must be carried
53
54 out to explore this possibility.
55
56
57
58
59
60

Conclusion

The electrochemical stability and electrical properties of nanocrystalline fluoride solid electrolyte $\text{La}_{0.95}\text{Ba}_{0.05}\text{F}_{2.95}$, LBF, have been investigated both in its pure form and in a composite with carbon. Electrochemical characterizations have been performed using a lithium electrode. Upon contacting Li with the solid fluoride electrolyte, a solid electrolyte interface (SEI) of LiF and La-Ba mixture is formed, providing the LBF/La-Ba redox couple. This method has allowed us to establish the decomposition potential of LBF beyond 5.7 V, in good agreement with the calculated value. Subsequently, the electrical properties of the composite LBF-carbon was investigated using a conventional set up. It showed that only the electronic conductivity could be assessed, revealing that 10 wt % of carbon is sufficient to enable electronic percolation. The determination of the ionic conductivity within the composite was probed by a novel set up consisting of LBF|LBF-C|LBF stacks. The results showed that the ionic percolation is maintained within the LBF-C composite and that the interactions with carbon impacts on the ionic transport properties of the electrolyte. Finally, we demonstrated that carbon was electrochemically active in the LBF-C composite. The operating voltage of FIBs must be carefully considered as the fluorination of carbon to yield insulative carbon fluoride could be a serious limitation to the operation of FIBs.

Author Information

Corresponding Author

* E-mail: damien.dambournet@upmc.fr

Abbreviations

1
2
3 FIB, fluoride ion battery; SEI, solid electrolyte interface; EIS, electrochemical impedance
4 spectroscopy; LBF, $\text{La}_{1-x}\text{Ba}_x\text{F}_{3-x}$; C, carbon black; LBF-C, $\text{La}_{1-x}\text{Ba}_x\text{F}_{3-x}$ -carbon black mixture; Cg,
5
6 graphite disk, CF_x , carbon fluoride compounds.
7
8
9
10
11
12
13
14
15
16
17
18
19
20

21 **Supporting Information**

22
23
24 XRD pattern and ^{19}F NMR of LBF and LBF-C, TEM images of LBF-C, representation of tysonite-
25
26 type structure, ionic conductivity of LBF and complete cyclic voltammetry (-0.1 to 7V) of $\text{Li}|\text{LBF}|\text{C}$
27
28 and $\text{Li}|\text{LBF}|\text{LBF-C}$ cells are given in supporting information.
29
30
31

32 **Acknowledgement**

33
34
35
36 We gratefully acknowledge J. Chable and A. Gil-Martin for the preparation of the solid fluoride
37
38 electrolyte, S. Casale for the TEM images, D. Devilliers for fruitful discussions about
39
40 electrochemical measurements, and S. Leclerc for technical support. AG, HG and DD wish to thank
41
42 the French Fluorine network for their continuous support.
43
44
45

46 **Funding Sources**

47
48
49 This work was funded by the French National Research Agency (Project FLUOBAT-ANR-12-
50
51 PRGE-0009-01).
52
53
54

55 **References**

- 56
57 (1) Gschwind, F.; Rodriguez-Garcia, G.; Sandbeck, D. J. S.; Gross, A.; Weil, M.; Fichtner,
58
59
60

- 1
2
3 M.; Hörmann, N. Fluoride Ion Batteries: Theoretical Performance, Safety, Toxicity, and a
4 Combinatorial Screening of New Electrodes. *J. Fluor. Chem.* **2016**, *182*, 76–90.
5
6
7
8
9 (2) Anji Reddy, M.; Fichtner, M. Batteries Based on Fluoride Shuttle. *J. Mater. Chem* **2011**,
10 *21*, 17059–17062.
11
12
13
14 (3) Baukal, W. Electrochemical Secondary Cells Which Contains Only Solid Materials.
15 Patent CA 953780 A1, 1974.
16
17
18
19
20 (4) Lucat, C. Etude Des Propriétés de Conductivité Ionique de Quelques Fluorures de
21 Structure Fluorine, Ph.D. Dissertation, Université de Bordeaux I, 1976.
22
23
24
25
26 (5) Kennedy, J. H.; Miles, R. C. Ionic Conductivity of Doped Beta-Lead Fluoride. *J.*
27 *Electrochem. Soc.* **1976**, *123*, 47–57.
28
29
30
31
32 (6) Kennedy, J. H.; Hunter, J. C. Thin-Film Galvanic Cell Pb/PbF₂/PbF₂,CuF₂/Cu. *J.*
33 *Electrochem. Soc.* **1976**, *123*, 10–14.
34
35
36
37
38 (7) Borger, W.; Hullmeine, U.; Voss, E. Galvanic Cell with Solid Fluoride Ion-Conductive
39 Electrolyte. Patent US 3973990 A1, 1976.
40
41
42
43
44 (8) Schoonman, J. A Solid-State Galvanic Cell with Fluoride-Conducting Electrolytes. *J.*
45 *Electrochem. Soc.* **1976**, *123*, 1772–1775.
46
47
48
49 (9) Danto, Y.; Poujade, G.; Pistre, J. D.; Lucat, C.; Salardenne, J. A Pb|PbF₂|BiF₃|Bi Thin
50 Solid Film Reversible Galvanic Cell. *Thin Solid Films* **1978**, *55*, 347–354.
51
52
53
54
55 (10) Schoonman, J.; Wapenaar, K. E. D.; Oversluizen, G.; Dirksen, G. J. Fluoride-Conducting
56 Solid Electrolytes in Galvanic Cells. *J. Electrochem. Soc.* **1979**, *126*, 709–713.
57
58
59
60

- 1
2
3 (11) Schoonman, J.; Wolfert, A. Alloy-Anodes in Fluoride Solid-State Batteries. *J.*
4
5
6 *Electrochem. Soc.* **1981**, *128*, 1522–1523.
7
8
9 (12) Schoonman, J.; Wolfert, A. Solid-State Galvanic Cells with Fast Fluoride Conducting
10
11
12 Electrolytes. *Solid State Ionics* **1981**, *3–4*, 373–379.
13
14
15 (13) Nowroozi, M. A.; Wissel, K.; Rohrer, J.; Munnangi, A. R.; Clemens, O. LaSrMnO₄:
16
17 Reversible Electrochemical Intercalation of Fluoride Ions in the Context of Fluoride Ion
18
19 Batteries. *Chem. Mater.* **2017**, *29*, 3441–3453.
20
21
22
23 (14) Thieu, D. T.; Fawey, M. H.; Bhatia, H.; Diemant, T.; Chakravadhanula, V. S. K.; Behm,
24
25 R. J.; Kübel, C.; Fichtner, M. CuF₂ as Reversible Cathode for Fluoride Ion Batteries. *Adv.*
26
27 *Funct. Mater.* **2017**, *27*, 1701051.
28
29
30
31 (15) Grenier, A.; Porras-Gutierrez, A.-G.; Groult, H.; Beyer, K. A.; Borkiewicz, O. J.;
32
33 Chapman, K. W.; Dambournet, D. Electrochemical Reactions in Fluoride-Ion Batteries:
34
35 Mechanistic Insights from Pair Distribution Function Analysis. *J. Mater. Chem. A* **2017**, *5*,
36
37 15700–15705.
38
39
40
41 (16) Bhatia, H.; Thieu, D. T.; Pohl, A. H.; Chakravadhanula, V. S. K.; Fawey, M. H.; Kübel,
42
43 C.; Fichtner, M. Conductivity Optimization of Tysonite-Type La_{1-x}Ba_xF_{3-x} Solid
44
45 Electrolytes for Advanced Fluoride Ion Battery. *ACS Appl. Mater. Interfaces* **2017**, *9*,
46
47 23707–23715.
48
49
50
51
52 (17) Dieudonné, B.; Chable, J.; Mauvy, F.; Fourcade, S.; Durand, E.; Lebraud, E.; Leblanc, M.;
53
54 Legein, C.; Body, M.; Maisonneuve, V.; et al. Exploring the Sm_{1-x}Ca_xF_{3-x} Tysonite Solid
55
56 Solution as a Solid-State Electrolyte: Relationships between Structural Features and F –
57
58
59
60

- 1
2
3 Ionic Conductivity. *J. Phys. Chem. C* **2015**, *119*, 25170–25179.
4
5
6
7 (18) Chable, J.; Dieudonné, B.; Body, M.; Legein, C.; Crosnier-Lopez, M.-P.; Galven, C.;
8
9 Mauvy, F.; Durand, E.; Fourcade, S.; Sheptyakov, D.; et al. Fluoride Solid Electrolytes:
10
11 Investigation of the Tysonite-Type Solid Solutions $\text{La}_{1-x}\text{Ba}_x\text{F}_{3-x}$ ($X < 0.15$). *Dalt. Trans.*
12
13 **2015**, *44*, 19625–19635.
14
15
16
17 (19) Zhang, L.; Anji Reddy, M.; Fichtner, M. Development of Tysonite-Type Fluoride
18
19 Conducting Thin Film Electrolytes for Fluoride Ion Batteries. *Solid State Ionics* **2015**,
20
21 *272*, 39–44.
22
23
24
25 (20) Rongeat, C.; Anji Reddy, M.; Witter, R.; Fichtner, M. Solid Electrolytes for Fluoride Ion
26
27 Batteries: Ionic Conductivity in Polycrystalline Tysonite-Type Fluorides. *ACS Appl.*
28
29 *Mater. Interfaces* **2014**, *6*, 2103–2110.
30
31
32
33 (21) Dieudonné, B.; Chable, J.; Body, M.; Legein, C.; Durand, E.; Mauvy, F.; Fourcade, S.;
34
35 Leblanc, M.; Maisonneuve, V.; Demourgues, A. The Key Role of the Composition and
36
37 Structural Features in Fluoride Ion Conductivity in Tysonite $\text{Ce}_{1-x}\text{Sr}_x\text{F}_{3-x}$ Solid Solutions.
38
39 *Dalt. Trans.* **2017**, *46*, 3761–3769.
40
41
42
43 (22) Chable, J.; Martin, A. G.; Bourdin, A.; Body, M.; Legein, C.; Jouanneaux, A.; Crosnier-
44
45 Lopez, M.-P.; Galven, C.; Dieudonné, B.; Leblanc, M.; et al. Fluoride Solid Electrolytes:
46
47 From Microcrystalline to Nanostructured Tysonite-Type $\text{La}_{0.95}\text{Ba}_{0.05}\text{F}_{2.95}$. *J. Alloys*
48
49 *Compd.* **2017**, *692*, 980–988.
50
51
52
53 (23) Rongeat, C.; Reddy, M. A.; Witter, R.; Fichtner, M. Nanostructured Fluorite-Type
54
55 Fluorides As Electrolytes for Fluoride Ion Batteries. *J. Phys. Chem. C* **2013**, *117*, 4943–
56
57
58
59
60

- 1
2
3 4950.
4
5
6
7 (24) Krahl, T.; Scholz, G.; Kemnitz, E. Solid Solutions $\text{CaF}_2\text{-YF}_3$ with Fluorite Structure
8 Prepared on the Sol–Gel Route: Investigation by Multinuclear MAS NMR Spectroscopy.
9 *J. Phys. Chem. C* **2014**, *118*, 21066–21074.
10
11
12
13
14 (25) Düvel, A.; Bednarcik, J.; Šepelák, V.; Heitjans, P. Mechano-synthesis of the Fast Fluoride
15 Ion Conductor $\text{Ba}_{1-x}\text{La}_x\text{F}_{2+x}$: From the Fluorite to the Tysonite Structure. *J. Phys. Chem. C*
16 **2014**, *118*, 7117–7129.
17
18
19
20
21
22
23 (26) Kamaya, N.; Homma, K.; Yamakawa, Y.; Hirayama, M.; Kanno, R.; Yonemura, M.;
24 Kamiyama, T.; Kato, Y.; Hama, S.; Kawamoto, K.; et al. A Lithium Superionic
25 Conductor. *Nat. Mater.* **2011**, *10*, 682–686.
26
27
28
29
30
31 (27) Ohta, N.; Takada, K.; Zhang, L.; Ma, R.; Osada, M.; Sasaki, T. Enhancement of the High-
32 Rate Capability of Solid-State Lithium Batteries by Nanoscale Interfacial Modification.
33 *Adv. Mater.* **2006**, *18*, 2226–2229.
34
35
36
37
38
39 (28) Zhu, Y.; He, X.; Mo, Y. Origin of Outstanding Stability in the Lithium Solid Electrolyte
40 Materials: Insights from Thermodynamic Analyses Based on First-Principles Calculations.
41 *ACS Appl. Mater. Interfaces* **2015**, *7*, 23685–23693.
42
43
44
45
46
47 (29) Richards, W. D.; Miara, L. J.; Wang, Y.; Kim, J. C.; Ceder, G. Interface Stability in Solid-
48 State Batteries. *Chem. Mater.* **2016**, *28*, 266–273.
49
50
51
52
53 (30) Cheng, X.-B.; Zhang, R.; Zhao, C.-Z.; Wei, F.; Zhang, J.-G.; Zhang, Q. A Review of Solid
54 Electrolyte Interphases on Lithium Metal Anode. *Adv. Sci.* **2016**, *3*, 1500213.
55
56
57
58
59
60

- 1
2
3
4
5
6
7
8
9
10
11
12
13
14
15
16
17
18
19
20
21
22
23
24
25
26
27
28
29
30
31
32
33
34
35
36
37
38
39
40
41
42
43
44
45
46
47
48
49
50
51
52
53
54
55
56
57
58
59
60
- (31) Luntz, A. C.; Voss, J.; Reuter, K. Interfacial Challenges in Solid-State Li Ion Batteries. *J. Phys. Chem. Lett.* **2015**, *6*, 4599–4604.
- (32) Aurbach, D. *Nonaqueous Electrochemistry*; Marcel Dekker Inc: New York, 1999.
- (33) Rongeat, C.; Anji Reddy, M.; Diemant, T.; Behm, R. J.; Fichtner, M. Development of New Anode Composite Materials for Fluoride Ion Batteries. *J. Mater. Chem. A* **2014**, *2*, 20861–20872.
- (34) Hammad Fawey, M.; Chakravadhanula, V. S. K.; Reddy, M. A.; Rongeat, C.; Scherer, T.; Hahn, H.; Fichtner, M.; Kübel, C. In Situ TEM Studies of Micron-Sized All-Solid-State Fluoride Ion Batteries: Preparation, Prospects, and Challenges. *Microsc. Res. Tech.* **2016**, *79*, 615–624.
- (35) Grenier, A.; Porras-Gutierrez, A. G.; Groult, H.; Dambournet, D. Modified Coin Cells to Evaluate the Electrochemical Properties of Solid-State Fluoride-Ion Batteries at 150°C. *J. Fluor. Chem.* **2016**, *191*, 23–28.
- (36) Bielecki, A.; Burum, D. P. Temperature Dependence of ^{207}Pb MAS Spectra of Solid Lead Nitrate. An Accurate, Sensitive Thermometer for Variable-Temperature MAS. *J. Magn. Reson. Ser. A* **1995**, *116*, 215–220.
- (37) van Gorkom, L. C. M.; Hook, J. M.; Logan, M. B.; Hanna, J. V.; Wasylishen, R. E. Solid-State Lead-207 NMR of lead(II) Nitrate: Localized Heating Effects at High Magic Angle Spinning Speeds. *Magn. Reson. Chem.* **1995**, *33*, 791–795.
- (38) Boulineau, S.; Courty, M.; Tarascon, J.-M.; Viallet, V. Mechanochemical Synthesis of Li-

- 1
2
3 Argyrodite $\text{Li}_6\text{PS}_5\text{X}$ (X=Cl, Br, I) as Sulfur-Based Solid Electrolytes for All Solid State
4 Batteries Application. *Solid State Ionics* **2012**, *221*, 1–5.
5
6
7
8
9 (39) Kotobuki, M.; Kanamura, K.; Sato, Y.; Yoshida, T. Fabrication of All-Solid-State Lithium
10 Battery with Lithium Metal Anode Using Al_2O_3 -Added $\text{Li}_7\text{La}_3\text{Zr}_2\text{O}_{12}$ Solid Electrolyte. *J.*
11 *Power Sources* **2011**, *196*, 7750–7754.
12
13
14
15
16
17 (40) Feng, J. K.; Lu, L.; Lai, M. O. Lithium Storage Capability of Lithium Ion Conductor
18 $\text{Li}_{1.5}\text{Al}_{0.5}\text{Ge}_{1.5}(\text{PO}_4)_3$. *J. Alloys Compd.* **2010**, *501*, 255–258.
19
20
21
22
23 (41) Seino, Y.; Takada, K.; Kim, B.-C.; Zhang, L.; Ohta, N.; Wada, H.; Osada, M.; Sasaki, T.
24 Synthesis of Phosphorous Sulfide Solid Electrolyte and All-Solid-State Lithium Batteries
25 with Graphite Electrode. *Solid State Ionics* **2005**, *176*, 2389–2393.
26
27
28
29
30
31 (42) Jain, A.; Ong, S. P.; Hautier, G.; Chen, W.; Richards, W. D.; Dacek, S.; Cholia, S.;
32 Gunter, D.; Skinner, D.; Ceder, G.; et al. Commentary: The Materials Project: A Materials
33 Genome Approach to Accelerating Materials Innovation. *APL Mater.* **2013**, *1*, 11002.
34
35
36
37
38
39 (43) Roos, A.; Schoonman, J. Electronic Conductivity in $\text{La}_{1-x}\text{Ba}_x\text{F}_{3-x}$ Crystals. *Solid State*
40 *Ionics* **1984**, *13*, 205–211.
41
42
43
44
45 (44) Glassner, A. *The Thermochemical Properties of the Oxides, Fluorides, and Chlorides to*
46 *2500° K*; report ANL-5750, Argonne National Laboratory, Lemont, 1957.
47
48
49
50
51 (45) Bezmelnitsyn, V. N.; Bezmelnitsyn, A. V.; Kolmakov, A. A. New Solid-State
52 Electrochemical Source of Pure Fluorine. *J. Fluor. Chem.* **1996**, *77*, 9–12.
53
54
55
56
57 (46) Badway, F.; Cosandey, F.; Pereira, N.; Amatucci, G. G. Carbon Metal Fluoride
58
59
60

- 1
2
3 Nanocomposites. *J. Electrochem. Soc.* **2003**, *150*, A1318.
- 4
5
6
7 (47) Hua, X.; Robert, R.; Du, L. S.; Wiaderek, K. M.; Leskes, M.; Chapman, K. W.; Chupas, P.
8
9 J.; Grey, C. P. Comprehensive Study of the CuF_2 Conversion Reaction Mechanism in a
10
11 Lithium Ion Battery. *J. Phys. Chem. C* **2014**, *118*, 15169–15184.
- 12
13
14
15 (48) Giraudet, J.; Dubois, M.; Hamwi, A.; Stone, W. E. E.; Pirotte, P.; Masin, F. Solid-State
16
17 NMR (^{19}F and ^{13}C) Study of Graphite Monofluoride $(\text{CF})_n$: ^{19}F Spin–Lattice Magnetic
18
19 Relaxation and $^{19}\text{F}/^{13}\text{C}$ Distance Determination by Hartmann–Hahn Cross Polarization. *J.*
20
21 *Phys. Chem. B* **2005**, *109*, 175–181.
- 22
23
24
25 (49) Leifer, N. D.; Johnson, V. S.; Ben-Ari, R.; Gan, H.; Lehn, J. M.; Guo, R.; Lu, W.;
26
27 Muffoletto, B. C.; Reddy, T.; Stallworth, P. E.; et al. Solid-State NMR Studies of
28
29 Chemically Lithiated CF_x . *J. Electrochem. Soc.* **2010**, *157*, A148–A154.
- 30
31
32
33 (50) Vyalikh, A.; Bulusheva, L. G.; Chekhova, G. N.; Pinakov, D. V.; Okotrub, A. V.; Scheler,
34
35 U. Fluorine Patterning in Room-Temperature Fluorinated Graphite Determined by Solid-
36
37 State NMR and DFT. *J. Phys. Chem. C* **2013**, *117*, 7940–7948.
- 38
39
40
41 (51) Seid, K.-A.; Badot, J.-C.; Perca, C.; Dubrunfaut, O.; Soudan, P.; Guyomard, D.; Lestriez,
42
43 B. An In Situ Multiscale Study of Ion and Electron Motion in a Lithium-Ion Battery
44
45 Composite Electrode. *Adv. Energy Mater.* **2015**, *5*, 1400903.
- 46
47
48
49 (52) Han, F.; Zhu, Y.; He, X.; Mo, Y.; Wang, C. Electrochemical Stability of $\text{Li}_{10}\text{GeP}_2\text{S}_{12}$ and
50
51 $\text{Li}_7\text{La}_3\text{Zr}_2\text{O}_{12}$ Solid Electrolytes. *Adv. Energy Mater.* **2016**, *6*, 1501590.
- 52
53
54
55 (53) Valerga, A. J.; Badachhane, R. B.; Parks, G. D.; Karmachik, P.; Wood, J. L.; Margrave, J.
56
57
58
59
60

1
2
3 L. *Thermodynamic and Kinetic Data of Carbon-Fluorine Compounds*; Report Rice
4
5 University, Houston, 1974.
6
7

8
9 (54) Nakajima, T.; Watanabe, N. *Graphite Fluorides and Carbon-Fluorine Compounds*; CRC
10
11 Press: Boca Raton, 1991.
12
13
14
15
16
17
18
19
20
21
22
23
24

25 **Table of content:**
26
27

