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1 **New constraints on Xe incorporation mechanisms in olivine from First-**  
2 **principles calculations**

3  
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26           **Abstract**

27

28           Storage of Xe at depth in silicate minerals has recently been proposed to explain the  
29 low Xe abundance in the Earth's and Mars' atmospheres compared to other noble gases (the  
30 so-called 'Missing Xenon' issue). Evidences for incorporation, and thus reactivity of Xe in  
31 olivine at high pressure and high temperature are based on variations in cell parameters and  
32 the appearance of a new Raman band. To constrain the, so far only hypothetical, Xe  
33 incorporation mechanism in olivine, we theoretically investigated models of Xe-bearing  
34 olivine using density functional theory. Three types of incorporation mechanisms are tested:  
35 Xe for Si and Xe for Mg substitutions, and interstitial Xe. Xe for Si substitution, implying an  
36 oxidation of Xe, is found to be the only mechanism consistent with experimental  
37 observations, leading to an increase of cell parameter  $a$  and the appearance of a new Raman  
38 band around  $720\text{-}750\text{ cm}^{-1}$  associated with Xe-O stretching vibrations. Raman spectroscopy  
39 makes it possible to identify Xe incorporation site, even at low Xe content, due to high Xe  
40 polarizability. An estimation of Xe content in olivine, based on present work and previous *in*  
41 *situ* experimental results, shows that up to 0.4 at % Xe could be stored in olivine at depth.

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## 51 1.INTRODUCTION

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53 Noble gases, highly volatile and poorly reactive elements, are of particular interest in  
54 Earth and planetary sciences to study and date the formation of the atmosphere (Avice and  
55 Marty, 2014) or to trace magmatic sources and degassing processes. Xenon, the heaviest of  
56 the noble gases, with nine stable isotopes ( $^{124,126,128,129,130,131,132,134,136}\text{Xe}$ ), is at the heart of a  
57 long-standing question, the so-called ‘Missing Xenon’ issue. Elemental Xe, compared to  
58 other noble gases, is missing in the atmospheres of the Earth and Mars relative to chondritic  
59 abundances (Anders and Owen, 1977). According to Ozima and Podosek (1999), around 90%  
60 of Earth’s primordial Xe is missing from the atmosphere. In addition, the atmospheres of  
61 Earth and Mars present a strong deficit in Xe light isotopes ( $^{124,126,128}\text{Xe}$ ) (Krumennacher,  
62 1962). Some Archean rocks present a Xe isotopic composition falling in between primitive  
63 chondritic and present-Earth atmospheric values, with an enrichment of the light non-  
64 radiogenic Xe isotopes relative to the present-Earth atmosphere of 2.1 ( $\pm 0.3$ ) % amu<sup>-1</sup> for a  
65 3.5 Gy barite (Pujol et al., 2009), and 1 ( $\pm 0.5$ ) % amu<sup>-1</sup> for a 3.0 ( $\pm 0.2$ ) Gy quartz (Pujol et al.,  
66 2011). These findings indicate a storage and fractionation of Xe throughout the Archean  
67 (Avice and Marty, 2014) which could continue nowadays through subduction processes  
68 (Holland and Ballentine, 2006). Recently, a number of studies have supported storage of Xe  
69 in silicates at depth. Xe oxides have been synthesized: XeO<sub>2</sub> at ambient pressure and  
70 temperature (Brock and Schrobilgen, 2010), and Xe<sub>3</sub>O<sub>2</sub> and Xe<sub>2</sub>O<sub>5</sub> at respectively 97 GPa and  
71 83 GPa, and at around 2000 K (Dewaele et al., 2016); as well as a Xe-bearing perovskite  
72 (Britvin et al., 2015) at ambient pressure. Furthermore, experiments brought evidence of Xe-  
73 incorporation in SiO<sub>2</sub> phases (Sanloup et al., 2002, 2005) and olivine (Sanloup et al., 2011) at  
74 crustal and upper mantle P-T conditions. If the presence of pure Xe oxides in the Earth is  
75 unlikely, storage in silicates as a trace element may be a significant phenomenon in Xe

76 retention at depth. Furthermore, a granite from the Sudbury impact structure (Kuroda et al.,  
77 1977) as well as siliceous sediments (Matsubara et al., 1988; Matsuda and Matsubara, 1989)  
78 have been found to be enriched in Xe by 2 to 3 orders of magnitude.

79 A detailed understanding of incorporation mechanisms of Xe, and reliable values of Xe  
80 solubility in minerals are presently needed to quantify the role of Xe incorporation in minerals  
81 in the context of the 'Missing Xenon' issue at the Earth's scale. Reliable partition coefficients  
82 are also needed to understand the fate of Xe in melting and crystallizing processes (from the  
83 magma ocean era to the current day magmatic processes). All available data on Xe-solubility  
84 in minerals are based on *ex situ* analyses of Xe-saturated samples, synthesized at high  
85 temperatures and, in some cases, at high pressures. Measured Xe solubility in olivine varies  
86 from 6.5 ppb down to 0.67 ppt (Broadhurst et al., 1992; Heber et al., 2007; Hiyagon and  
87 Ozima, 1986). Similarly, olivine melt partition coefficient varies from 90 to  $6 \cdot 10^{-4}$   
88 (Broadhurst et al., 1992; Hiyagon and Ozima, 1986; Heber et al., 2007) and the compatibility  
89 (or not) of Xe in minerals remains controversial. In samples recovered from high pressure and  
90 temperature experiments, Xe is at least partly located in gas bubbles, and inconsistencies in  
91 Xe solubility and partition coefficient mainly stem from whether or not bubbles are analyzed,  
92 keeping in mind that their sizes range from the nm to the mm scale, and that part or all of  
93 these bubbles may form upon quenching to room conditions. Consequently, the evaluation of  
94 Xe content should be preferentially based on *in situ* measurements as proposed in the present  
95 study.

96         Recent experiments have focused on the incorporation mechanism of Xe in olivine,  
97 one of the most widespread terrestrial minerals. Sanloup et al. (2011) studied Xe-bearing San  
98 Carlos olivine up to 7 GPa and 2000 K by *in situ* x-ray diffraction and Raman spectroscopy  
99 reporting shifts of lattice parameters in presence of Xe at high pressure and for temperatures  
100 above Xe melting point. Cell parameter *a* increases, *c* decreases (although *c* is strongly

101 temperature dependent) while  $b$  is not affected. This lattice distortion was interpreted as  
102 potentially due to Xe incorporation on Si sites, as Si release from olivine was observed. A  
103 new Raman mode, in presence of Xe, (disappearing within few minutes after quench) was  
104 also observed at  $786\text{ cm}^{-1}$  in presence of Xe at 0.9 GPa after heating at 1800 K. This new  
105 Raman band is close to the stretching of  $\text{XeO}_4$  at  $776\text{ cm}^{-1}$  (Gerken and Schrobilgen, 2002),  
106 and is in the Si-O stretching region of olivine (Noël et al., 2006). As for other heavy rare  
107 gases, Ar was not observed to substitute to Si in olivine up to 7 GPa-2000 K (Sanloup et al.,  
108 2011). In the absence of studies devoted to the incorporation of Kr in minerals at depth, we  
109 can temporarily conclude that this mechanism is specific to Xe and of major relevance to the  
110 ‘Missing Xe’ issue.

111 First-principles calculations represent a powerful complementary approach to identify  
112 incorporation sites of trace elements in minerals. Regarding Xe in silicates only two  
113 theoretical studies are so far available. Probert (2010) investigated different incorporation  
114 mechanisms of Xe in  $\alpha$ -quartz, but the lack of related measurements prevents definitive  
115 conclusion. Kalinowski et al. (2014) investigated Xe incorporation in fibrous silica, which can  
116 be found on Earth in moganite or chalcédoine (Hopkinson et al., 1999). Computed Raman  
117 spectra of Xe-bearing fibrous silica possibly reproduce observations for Xe-rich quartz  
118 (Sanloup et al., 2002) and  $\text{XeO}_2$  (Brock and Schrobilgen, 2011).

119 In the present work, we study the structure, stability and vibrational properties of  
120 different incorporation sites of Xe in olivine, by means of first-principles calculations based  
121 on density functional theory (DFT). These theoretical results are then compared to the  
122 experimental data of Sanloup et al. (2011) to infer potential mechanisms of Xe storage in  
123 Earth’s interior.

124

## 125 **2.METHODS**

126

127           Various systems are investigated: two reference compounds ( $\text{XeO}_4$  molecule,  $\text{Xe}_3\text{O}_2$   
128 condensed phase); pure forsterite ( $\text{Mg}_2\text{SiO}_4$ , the Mg-rich end member of the olivine series,);  
129 Xe-bearing forsterite in which one Xe atom substitutes one Si (or Mg) atom or in which one  
130 Xe atom is added at an interstitial position. In all cases, the charge of the simulation cell is  
131 neutral. Xe-bearing forsterite is simulated using supercells of various sizes:  $2 \times 1 \times 2$ ,  $4 \times 1 \times 2$ ,  
132  $3 \times 1 \times 3$ ,  $4 \times 2 \times 4$  (with respect to the forsterite unit cell containing 28 atoms) corresponding to a  
133 concentration of approximately 0.89, 0.45, 0.40, 0.11 at % Xe (or 5.5, 2.8, 2.5, 0.72 wt %  
134 Xe), respectively.

135 All calculations are done within density functional theory (DFT) and the generalized gradient  
136 approximation (GGA-PBE) of Perdew et al. (1996), by using the PWscf code of the Quantum  
137 ESPRESSO package <http://www.quantum-espresso.org/>; Gianozzi et al., 2009). Norm-  
138 conserving pseudopotentials (Troullier and Martins, 1991) are used. Electronic wave-  
139 functions (charge-density) are expanded up to an energy cut-off of 80 (480) Ry. Equilibrium  
140 structures are obtained after full relaxation of atomic positions and cell parameters (total  
141 energy is converged within 1 mRy/atom and forces to less than  $10^{-4}$  Ry/au). Electronic k-point  
142 sampling of  $\text{Xe}_3\text{O}_2$  and pure forsterite uses  $6 \times 16 \times 10$  and  $2 \times 1 \times 2$  shifted grids, while that of the  
143 Xe-bearing larger cells uses one shifted k-point. Spin polarized calculations were tested,  
144 obtaining zero magnetic moments for all the systems. Vibrational properties (dynamical  
145 matrices, dielectric constants, Born effective charges, and Raman tensor) are obtained within  
146 the approach of Baroni et al. (2001) and Lazzeri and Mauri (2003). For Raman tensors and  
147 effective charges calculations a finer k-point sampling is used ( $4 \times 2 \times 4$  grid for pure forsterite  
148 and a  $2 \times 2 \times 2$  one for Xe-bearing forsterites).

149 For the Xe-bearing forsterites, actual DFT calculations of the dynamical matrices (and, thus,  
150 of the interatomic force constants, IFCs) have been done only for the  $2 \times 1 \times 2$  cell. To simulate

151 larger cells, we have used a force-matching approach similar to that of Lazzeri and  
152 Thibaudeau (2006). One considers a forsterite cell (larger than the 2x1x2 cell) containing one  
153 Xe atom: the IFCs among the atoms in the neighborhood of the Xe atom (a sphere centred on  
154 the Xe atom with a 6.35 Å radius i.e. containing ~ 80 atoms) are taken from the DFT  
155 calculations done on the 2x1x2 cell, the other IFCs (for atoms located further than 6.35 Å  
156 from the Xe atom) are taken from pure forsterite bulk calculations.

157 In the following, calculated Raman spectra are done in a back-scattering geometry and, unless  
158 otherwise stated, the spectra are obtained after averaging polarizations and sample orientation.  
159 We note that for the kind of systems presently studied, vibrational frequencies are generally  
160 underestimated by DFT-GGA calculations. To make the comparison with measurements  
161 easier, all theoretical frequencies are multiplied by a factor 1.05 (in order to match the most  
162 intense theoretical Raman bands of pure forsterite with the measured ones, Table S1) as in  
163 Lazzeri and Mauri (2003).

164

## 165 **3.RESULTS**

166

### 167 **3.1 Reference compounds and pure forsterite**

168

169 As a test, we calculated the structure of the XeO<sub>4</sub> tetrahedral molecule and Xe<sub>3</sub>O<sub>2</sub>  
170 condensed phase (Table 1). For XeO<sub>4</sub>, Xe-O bond length is 2.0% larger than the experimental  
171 value (Gundersen et al., 1970) whereas the one from Probert (2010) is 4.8% larger (Table 1).  
172 Xe<sub>3</sub>O<sub>2</sub> was simulated at 97 GPa using the orthorhombic *Immm* space group as predicted by  
173 Hermann and Schwerdtfeger (2014) and synthesized by Dewaele et al. (2016). Bond lengths  
174 are well reproduced (within 0.5%) whereas cell parameters are slightly different (within  
175 1.5%) (Table 1).

176

177 Forsterite ( $\text{Mg}_2\text{SiO}_4$ ) has an orthorhombic structure with three distinct oxygen sites  
178 (O1, O2 and O3), two distinct magnesium sites (Mg1 and Mg2) and one silicon site. Our  
179 calculated cell parameters for pure forsterite ( $a = 4.78 \text{ \AA}$ ,  $b = 10.25 \text{ \AA}$ ,  $c = 5.99 \text{ \AA}$ ) are in good  
180 agreement with single crystal x-ray diffraction measurements recorded on pure forsterite at  
181 ambient conditions, i.e.  $a = 4.752(3) \text{ \AA}$ ,  $b = 10.193(8) \text{ \AA}$ ,  $c = 5.977(5) \text{ \AA}$  (Hazen et al., 1976).  
182 Similar values were obtained by previous studies using also GGA (which tends to increase  
183 cell parameters) and similar computational parameters (Balan et al., 2011). Born effective  
184 charges (Table S2) and calculated Raman frequencies of pure forsterite (Table S1) are in good  
185 agreement with the ones calculated by Noël et al. (2006). Computed polarized Raman spectra  
186 are in agreement with McKeown et al. (2011) (Fig. S1). The average Raman spectrum  
187 compares well with the experimental spectrum of San Carlos olivine (Sanloup et al., 2011).  
188 The presence of Fe in San Carlos olivine ( $(\text{Mg}_{0.9}\text{Fe}_{0.1})_2\text{SiO}_4$ ) does not significantly change the  
189 average Raman spectrum compared to pure forsterite (Fig. 5).

190

### 191 **3.2 Xe incorporation models**

192

193 Three main types of Xe incorporation mechanisms are investigated: Xe for Si  
194 substitutions ( $\text{Xe}_{\text{Si}}$ ), Xe for Mg substitutions (in Mg1 and Mg2 sites) ( $\text{Xe}_{\text{Mg1}}$ ,  $\text{Xe}_{\text{Mg2}}$ ), and Xe  
195 in interstitial site ( $\text{Xe}_i$ ). We relaxed atomic positions at constant volume for these four  
196 mechanisms for 0.89 at % Xe. No significant change was observed in the configurations of  
197 Xe incorporation sites for more diluted defects (0.45 at % Xe or 0.40 at % Xe, depending on  
198 the supercell) or for simultaneous relaxation of atomic positions and cell parameters. The  
199 configurations and selected parameters of the four main Xe incorporation sites ( $\text{Xe}_{\text{Si}}$ ,  $\text{Xe}_{\text{Mg1}}$ ,  
200  $\text{Xe}_{\text{Mg2}}$  and  $\text{Xe}_i$ ) are reported (Figs. 1, 2, 3) following simultaneous relaxation of atomic  
201 positions and cell parameters at 0 GPa for 0.89 at % Xe. *CIF* files for all configurations can

202 be found in supplementary materials. One third of the trace of the Born effective charge  
203 tensor ( $Z^{\text{eff}}$ ) is used to evaluate atomic charges, as discussed by Ghosez et al. (1998).

204 Xe incorporation sites can be described in terms of the closest surrounding oxygens.  
205 Indeed only oxygens are located within the sum of covalent radii that Xe forms with other  
206 elements (i.e.  $\sim 2 \text{ \AA}$  for oxygen). These oxygens appear to be strongly affected by the  
207 presence of Xe in term of variations of  $Z^{\text{eff}}$  (Figs. 1, 2, 3) compared to pure forsterite (Table  
208 S2). For the atoms located beyond the sum of covalent radii no significant variations of  $Z^{\text{eff}}$   
209 are observed.

210

### 211 3.2.1 Most stable configurations of Xe incorporation sites

212

213 For the most stable Xe for Si substitution, Xe is located on the face of the oxygen  
214 tetrahedron (Fig. 1). Xe is surrounded by one O1 and two O3 at a distance of  $\sim 2 \text{ \AA}$  forming a  
215 plane including Xe, with dihedral angles inferior to  $2^\circ$ .  $Z^{\text{eff}}$  for Xe is slightly more positive  
216 than that for Si in pure forsterite (+ 2.85; Table S2). These charges significantly differ from  
217 the formal ones (+ 4). This indicates a covalent character of Xe-O bonds involving the closest  
218 oxygens (O1 and O3).

219

220 Regarding Xe for Mg substitutions,  $\text{Xe}_{\text{Mg}2}$  incorporation site was found more stable  
221 than  $\text{Xe}_{\text{Mg}1}$  by 1.05 eV. For both configurations, Xe is located in the former oxygen  
222 octahedron. Nevertheless, two oxygen atoms are located close to the sum of covalent radii of  
223 Xe and O with Xe-O bond lengths  $\sim 2 \text{ \AA}$  whereas others O atoms are located significantly  
224 further (Fig. 2). The geometry is nearly linear with angle O3\*-Xe-O3\* close to  $180^\circ$  for  
225  $\text{Xe}_{\text{Mg}2}$  and O2-Xe-O2 equal to  $180^\circ$  for  $\text{Xe}_{\text{Mg}1}$ .  $Z^{\text{eff}}$  for Xe in  $\text{Xe}_{\text{Mg}1}$  and  $\text{Xe}_{\text{Mg}2}$  are slightly  
226 higher than that for Mg1 (+ 1.95) and Mg2 (+ 2.10) in pure forsterite (Table S2). The small  
227 excess of  $Z^{\text{eff}}$  of Xe in these configurations, compared to charge of Mg in pure forsterite, is

228 counterbalanced by more negative  $Z^{\text{eff}}$  for the two closest oxygens (Fig. 2) compared to pure  
229 forsterite (Table S2). These results suggest a covalent character of Xe-O bonds involving  
230 these oxygens.

231

232 For  $\text{Xe}_i$ , no oxygen atoms are located within the sum of covalent radii of Xe and O  
233 (Fig. 3).  $Z^{\text{eff}}$  for Xe is close to 0 indicating that Xe is nearly neutral in this configuration.  
234 Small positive  $Z^{\text{eff}}$  for Xe, is counterbalanced by slightly more negative  $Z^{\text{eff}}$  for surrounding  
235 oxygens (Fig. 3) compared to pure forsterite (Table S2). However, large Xe-O distances  
236 discard the possibility of covalency for Xe-O bond.

237

### 238 3.2.2 Less stable configurations of Xe incorporation sites

239

240 For Xe for Si substitution (which was found to best reproduce the experimental  
241 observations as further detailed), we investigated possible configurations by slightly moving  
242 Xe within the oxygen tetrahedron visible in figure 1. We identified another configuration:  
243  $\text{Xe}_{\text{Si}}(\text{b})$ , less stable than  $\text{Xe}_{\text{Si}}$  by 0.54 eV, where Xe is quasi aligned with O1 and O2 (O1-Xe-  
244 O2 = 157 °) though Xe remains inside the tetrahedron. For both  $\text{Xe}_{\text{Si}}$  and  $\text{Xe}_{\text{Si}}(\text{b})$  there exist  
245 *non symmetric variant* (respectively  $\text{Xe}_{\text{Si}}'$  and  $\text{Xe}_{\text{Si}}(\text{b})'$ ) with Xe slightly shifted toward one of  
246 the two O3 of the oxygen tetrahedron so that symmetry of the Xe incorporation sites is  
247 broken. These *non symmetric variants* are slightly more stable than the corresponding  
248 symmetric variants by 55 meV (for  $\text{Xe}_{\text{Si}}$  and  $\text{Xe}_{\text{Si}}'$ ) and 54 meV (for  $\text{Xe}_{\text{Si}}(\text{b})$  and  $\text{Xe}_{\text{Si}}(\text{b})'$ ).  
249 We checked that these results are not due to incomplete convergence or artefacts of  
250 calculations. A major concern was Van der Waals interactions, which may be crucial for  
251 calculations involving noble gases. We therefore repeated our calculations after adding a  
252 correction for Van der Waals interactions via a pairwise atomic potential proportional to  $R^{-6}$   
253 (with R the distance between the atoms of a pair) as implemented by Barone et al. (2009) for

254 DFT, based on results from Grimme (2006). We obtained similar geometries for the four  
255 configurations and similar relative stabilities as previously found. We only observe a small  
256 variation in differences of energies between configurations: from 0.54 eV to 0.68 eV between  
257  $\text{Xe}_{\text{Si}}$  and  $\text{Xe}_{\text{Si}}(\text{b})$  and from 55 meV to 35 meV and from 54 meV to 46 meV, between  
258 respectively  $\text{Xe}_{\text{Si}}$  and  $\text{Xe}_{\text{Si}}'$  and  $\text{Xe}_{\text{Si}}(\text{b})$  and  $\text{Xe}_{\text{Si}}(\text{b})'$ .

259

### 260 3.2.3 Comparison with other systems

261

262 In our calculation, the most stable Xe for Si substitutions (i.e.  $\text{Xe}_{\text{Si}}$  and  $\text{Xe}_{\text{Si}}'$ ) present a  
263 planar geometry with Xe located on the face of the oxygen tetrahedron, surrounded by one O1  
264 and two O3 (Fig. 1). As emphasized by Kalinowski et al. (2014), a planar structure is  
265 common for Xe compounds: in particular under the form of  $\text{XeO}_4$  units or chains present in  
266  $\text{XeO}_2$  (Brock and Schrobilgen, 2011)  $\text{Xe}_3\text{O}_2$  and  $\text{Xe}_2\text{O}_5$  (Dewaele et al., 2016). This planar  
267 geometry was also proposed for Xe incorporation in fibrous silica (Kalinowski et al., 2014)  
268 and in  $\alpha$ -quartz via Xe for Si substitution (Probert, 2011). Other examples of incorporations  
269 associated with planar geometries have been reported on boron incorporation in olivine  
270 (Ingrin et al., 2014) and carbonate incorporation in fluor-apatite (Yi et al., 2013). In both  
271 cases, the carbon and boron atoms share significantly covalent bonds with surrounding  
272 oxygens.

273 Less stable configurations for Xe for Si substitution (i.e.  $\text{Xe}_{\text{Si}}(\text{b})$  and  $\text{Xe}_{\text{Si}}(\text{b})'$ ), as well  
274 as Xe for Mg substitutions, present nearly linear geometries as in common compounds such  
275 as  $\text{XeF}_2$  or noble gas hydrides.

276 For Xe for Si and Xe for Mg substitutions the nearest oxygen atoms have Xe-O bond  
277 lengths within the sum of the covalent radii of Xe and O ( $\sim 2 \text{ \AA}$ ) (Figs. 1, 2). Also, the values  
278 of  $Z^{\text{eff}}$  indicate a certain degree of covalency of Xe-O bonds. Xe-O bonds are thus likely to be

279 at least partially covalent for the closest oxygen atoms. This result implies that Xe can react  
280 with olivine by oxidation processes.

281

### 282 3.2.4 Energetics of the Xe-bearing models

283

284 Let us define:

$$285 \Delta E(\text{Xe}_{\text{Si}}) = E(\text{Xe}_{\text{Si}}) - E_{\text{Xe}} - E_{\text{Fs-Si}}$$

$$286 \Delta E(\text{Xe}_{\text{Mg}}) = E(\text{Xe}_{\text{Mg}}) - E_{\text{Xe}} - E_{\text{Fs-Mg}}$$

$$287 \Delta E(\text{Xe}_i) = E(\text{Xe}_i) - E_{\text{Xe}} - E_{\text{Fs}}$$

288

289 where  $E(\text{Xe}_{\text{Si}})$ ,  $E(\text{Xe}_{\text{Mg}})$ ,  $E(\text{Xe}_i)$  are the calculated energies for the cell used to simulate the  
290 different type of incorporation mechanisms ( $\text{Xe}_{\text{Si}}$ ,  $\text{Xe}_{\text{Mg}}$  and  $\text{Xe}_i$ ).  $E_{\text{Xe}}$  is the energy of an  
291 isolated Xe atom.  $E_{\text{Fs-Si}}$  and  $E_{\text{Fs-Mg}}$  are the energies of the forsterite cell with one neutral Si or  
292 Mg vacancy (the atomic structure was fully relaxed) and  $E_{\text{Fs}}$  is the energy of a pure forsterite  
293 cell. The calculated  $\Delta E$  values are reported in table 2 and represent the energy of the various  
294 incorporations in a forsterite presenting Si and Mg vacancies. Occurrence of Si and Mg  
295 vacancies in synthetic forsterite samples is suggested by the observation of related  
296 hydroxylated point defects (e.g. Lemaire et al. 2004, Balan et al. 2011, Ingrin et al. 2013,  
297 Balan et al. 2017, Blanchard et al. 2017). How the present  $\Delta E$  values can be compared to  
298 measurements is questionable. Indeed, besides the fact that the vacancies in the calculations  
299 are neutral, temperature effects are not included. Both effects are expected to be relevant  
300 (Verna and Karki, 2009). That said, according to table 2, Xe incorporation in the Si vacancy  
301 site ( $\text{Xe}_{\text{Si}}$ ) is more stable than Mg-vacancy ( $\text{Xe}_{\text{Mg}}$ ) and interstitial ( $\text{Xe}_i$ ) incorporations. A  
302 different conclusion would be reached if considering forsterite without vacancies. In this case  
303 the relative stability would be dominated by the formation energy of the vacancy itself, which

304 is higher for Si than for Mg (Béjina et al., 2000, Stashans and Flores, 2013), as can be seen in  
305 Table 2 by comparing:

$$306 \Delta E_v(\text{Si}) = E_{\text{Fs-Si}} - E_{\text{Fs}} + E_{\text{Si}}$$

$$307 \Delta E_v(\text{Mg}) = E_{\text{Fs-Mg}} - E_{\text{Fs}} + E_{\text{Mg}},$$

308 where  $E_{\text{Si}}$  and  $E_{\text{Mg}}$  are the energies of one atom of Si and Mg in the equilibrium diamond and  
309 *hcp* bulk, respectively.

310 The relative stability of the three different incorporation mechanisms (Table 2),  
311  $\Delta E(\text{Xe}_{\text{Si}}) < \Delta E(\text{Xe}_{\text{Mg}}) < \Delta E(\text{Xe}_i)$ , results from an interplay of oxidation state and strain  
312 induced on the host structure. Although we cannot unambiguously disentangle these effects  
313 and determine which one dominates, we note that there is a correlation between the value of  
314  $\Delta E$  and the Xe oxidation estimated in Figs. 1-4: the more stable structures having the highest  
315 Xe oxidation (as it would be expected if oxidation was the only stabilization mechanism at  
316 play). On the other hand, the  $\Delta E$  ordering can simply reflect the fact that different inclusions  
317 correspond to different volume increase of the host forsterite: the more stable structures  
318 corresponding to the smaller volume increase (bottom panel of Fig.4), as will be discussed in  
319 the next section.

320

### 321 **3.3 Cell parameters vs. Xe concentration**

322

323 For each configuration, relaxation was done at 0, 2, 5 and 5.7 GPa for 0.89 at % Xe,  
324 0.45 at % Xe and 0.40 at % Xe. Cell parameters are only shown at 0 GPa and 5.7 GPa (Fig.  
325 4), but the overall trend is similar at 2 GPa and 5 GPa (Table S3). We observe an increase of  
326 volume with increasing Xe content (+ 1.6% for Xe for Si substitutions, of + 2.3% for Xe for  
327 Mg substitutions, and + 4% for Xe in interstitial site at 0 GPa from 0 at % Xe to 0.89 at %

328 Xe). However, variations of each cell parameters are strongly dependent on the Xe  
329 incorporation site (Fig. 4), and this can help to determine the likeliest incorporation site.

330 Variations of cell parameters for  $\text{Xe}_{\text{Si}}$  incorporation site reproduce the experimentally  
331 observed increase of cell parameter  $a$  associated with very small variations of cell parameter  $b$   
332 (Sanloup et al., 2011) compared to the thermal equation of state of olivine (Liu and Li, 2006).  
333 Other incorporation mechanisms do not reproduce this concomitant evolution of  $a$  and  $b$ .  
334 Experimental variations of cell parameter  $c$  are strongly temperature dependent (Sanloup et  
335 al., 2011), therefore they are not used for comparison with calculations, as they are unlikely to  
336 be reproduced quantitatively by our calculations at 0 K. In contrast, the relative variations of  $a$   
337 and  $b$ , between the pure and the Xe-bearing systems can be used to assess the concentration of  
338 Xe in olivine. For  $\text{Xe}_{\text{Si}}$ , calculated increase of cell parameter  $a$  from 0 at % Xe to 0.40 at %  
339 Xe, is about 0.4% at all investigated pressures. This increase is close to the experimentally  
340 observed one (i.e. + 0.2% at 2 GPa, + 0.4% at 5 GPa and + 0.2% at 5.7 GPa). Consequently  
341 the Xe content in olivine at high pressure and temperature, studied by Sanloup et al. (2011),  
342 can be estimated to be equal or slightly less than 0.40 at % Xe. We note that the Xe content  
343 thus determined does not significantly vary with pressure between 2 GPa and 5.7 GPa.

344

### 345 **3.4 Raman spectra**

346

347 Raman spectra were computed for 0.89 at % Xe (Fig. 5). The two most intense Raman  
348 bands of olivine at  $822\text{ cm}^{-1}$  and  $855\text{ cm}^{-1}$  are clearly visible for  $\text{Xe}_{\text{Si}}$  and  $\text{Xe}_{\text{Mg2}}$  incorporation  
349 sites. For  $\text{Xe}_{\text{i}}$  and  $\text{Xe}_{\text{Mg1}}$  these two olivine bands are mixed with other contributions linked to  
350 the presence of Xe. New well-resolved and intense Raman bands appear in the  $650\text{-}800\text{ cm}^{-1}$   
351 region for  $\text{Xe}_{\text{Si}}$  and, although less intense, for  $\text{Xe}_{\text{Mg1}}$  (Fig. 5). All these new Xe-related Raman  
352 bands involve the nearest oxygens around Xe (described and indexed in Figs. 1, 2, 3). For

353  $\text{Xe}_{\text{Si}}$ , the  $720 \text{ cm}^{-1}$  band is linked to strong Xe-O1 stretching along the  $a$  axis (and weak  
354 stretching of a  $\text{SiO}_4$  tetrahedron close to Xe also along  $a$  axis), whereas the weak  $685 \text{ cm}^{-1}$   
355 band is linked to Xe-O2 bending. For  $\text{Xe}_{\text{Mg1}}$  the  $739 \text{ cm}^{-1}$  band is linked to the two Xe-O2  
356 twisting. A new well-resolved intense band is also visible for  $\text{Xe}_{\text{Mg2}}$  at  $900 \text{ cm}^{-1}$  linked to the  
357 two Xe-O3\* symmetrical stretching with a wagging component. The bands are visible in  
358 parallel polarization along axis or planes corresponding to the maximal direction of the  
359 associated vibrational modes (Fig. 5). For  $\text{Xe}_i$  (which closest oxygen is only located at  $2.66$   
360  $\text{Å}$ ), no new intense band is observed.

361 For  $\text{Xe}_{\text{Si}}$  and  $\text{Xe}_{\text{Mg}}$ , lots of poorly resolved bands are visible below  $650 \text{ cm}^{-1}$ , and  
362 above  $800 \text{ cm}^{-1}$ . These weak bands are linked to the high Xe polarizability, which implies that  
363 Xe presence will be felt by all atoms of the cell. In order to test this hypothesis, we calculate  
364 Raman spectra for lower Xe contents. Results for  $0.11 \text{ at } \% \text{ Xe}$  are shown in figure 6. Weak  
365 non-resolved bands below  $650 \text{ cm}^{-1}$  and above  $800 \text{ cm}^{-1}$  visible for  $0.89 \text{ at } \% \text{ Xe}$  (Fig. 5)  
366 disappear with decreasing Xe content. Furthermore, for all Xe incorporation sites, pure  
367 forsterite bands are well identified for  $0.11 \text{ at } \% \text{ Xe}$  not only at  $822 \text{ cm}^{-1}$  and  $855 \text{ cm}^{-1}$  but  
368 also at  $340 \text{ cm}^{-1}$ ,  $611 \text{ cm}^{-1}$ ,  $882 \text{ cm}^{-1}$ ,  $923 \text{ cm}^{-1}$ ,  $964 \text{ cm}^{-1}$  (Fig. 6) implying that the presence of  
369 Xe is just felt locally as expected for a diluted defect.

370 For  $0.11 \text{ at } \% \text{ Xe}$ , Raman bands induced by Xe are still visible for  $\text{Xe}_{\text{Si}}$  (at  $720 \text{ cm}^{-1}$ ),  
371  $\text{Xe}_{\text{Mg2}}$  (at  $900 \text{ cm}^{-1}$ ), and  $\text{Xe}_{\text{Mg1}}$  (at  $710 \text{ cm}^{-1}$ ). In contrast, spectra calculated for  $\text{Xe}_i$  only  
372 present minor variations compared to pure forsterite. Presence of Xe in interstitial site can  
373 therefore not be evidenced by Raman spectroscopy.

374 We also computed Raman spectra for  $\text{Xe}_{\text{Si}}'$ ,  $\text{Xe}_{\text{Si}}(\text{b})$  and  $\text{Xe}_{\text{Si}}(\text{b})'$  for  $0.89 \text{ at } \% \text{ Xe}$ .  
375 Spectra show well-defined Xe-related bands at  $745 \text{ cm}^{-1}$  for  $\text{Xe}_{\text{Si}}'$ ,  $748 \text{ cm}^{-1}$  for  $\text{Xe}_{\text{Si}}(\text{b})$  and  
376  $750 \text{ cm}^{-1}$  for  $\text{Xe}_{\text{Si}}(\text{b})'$  (Fig. S2) similar to the Xe-related band at  $720 \text{ cm}^{-1}$  in Raman spectrum  
377 computed for  $\text{Xe}_{\text{Si}}$  (Fig. 5).

378           The bands observed between  $720\text{ cm}^{-1}$  and  $750\text{ cm}^{-1}$  for Xe for Si substitutions and at  
379  $710\text{ cm}^{-1}$  for  $\text{Xe}_{\text{MgI}}$  are good candidates to explain the new Raman band observed by Sanloup  
380 et al. (2011) (Fig. 6), keeping in mind that only  $\text{Xe}_{\text{Si}}$  reproduces observed variations of cell  
381 parameters. Broadness of the experimental Raman band may be caused firstly by the  
382 coexistence of  $\text{Xe}_{\text{Si}}$ ,  $\text{Xe}_{\text{Si}}'$ ,  $\text{Xe}_{\text{Si}}(\text{b})$  and  $\text{Xe}_{\text{Si}}(\text{b})'$  at high temperature. It may also be caused by  
383 the chemical complexity of San Carlos olivine (see for instance trace and major element  
384 analyses of San Carlos olivine in Spandler and O'Neill, 2010). Indeed there exist a variety of  
385 Si vacancies, which slightly differs in terms of structural configuration and electronic  
386 properties due to Fe and trace elements. Distribution of Xe in these Si vacancies will lead to a  
387 range of slightly different vibration modes and consequently to a broad Raman signal as  
388 compared to the present ideal model. Eventually, the high temperature of measurement (1800  
389 K) induces broadening of the bands (Kolesov and Geiger, 2004). In cases of band ascribed to  
390 chemical defects, this broadening can highly depends on the interaction of the trace element  
391 with its crystalline host (e.g. Balan et al., 2017). High temperature may also induce a shift in  
392 Raman frequency, as observed for characteristic bands of forsterite (decrease of  $15\text{ cm}^{-1}$  from  
393 273 K to 1473 K, Kolesov and Geiger, 2004).

394

#### 395           3.4.1 Xe concentration from Raman spectra

396

397           Our calculated Raman spectra can be used to approximate the experimental Xe content  
398 in olivine. Raman spectrum was computed for 0.45 at % Xe for  $\text{Xe}_{\text{Si}}$  (Fig. 7). The intensity of  
399 the Raman spectrum calculated for  $\text{Xe}_{\text{Si}}$  was integrated within the four energy windows  
400 shown in figure 7. The intensity was, first, normalized to the integrated intensity in window  
401 "3" (corresponding to the characteristic olivine bands) and, then, the contribution from pure  
402 forsterite was subtracted. The normalized integrated intensity of Xe-related bands is linearly

403 related to the Xe content (Fig. 8). The corresponding value extracted from the measured  
404 Raman spectrum of Xe-rich olivine (Sanloup et al., 2011) is 2.3, which corresponds to  $\sim 0.2$   
405 at % Xe in figure 8. This evaluation must be considered as a first approximation and only  
406 represents a lower limit due to the significant signal to noise ratio in the experimental  
407 spectrum. Comparing Raman spectroscopy and X-ray diffraction, the evaluation of Xe  
408 content based on XRD-determined cell-parameters variation (Sec. 3.3) should be more  
409 reliable.

410

#### 411 **4.DISCUSSION**

412

413 This work brings new insights on Xe storage at depth, attesting the possibility of Xe  
414 incorporation in olivine. We tested here three types of incorporation mechanisms: Xe for Mg  
415 and Xe for Si substitutions, and interstitial Xe. Only Xe for Si substitutions can reproduce the  
416 observed evolution of the cell parameters induced by Xe at different pressures (Sanloup et al.,  
417 2011): the increase of  $a$  is substantially more important than the increase of  $b$  (variations of  
418  $c$ , which are strongly temperature dependent, are unlikely to be reproduced by zero  
419 temperature calculations). Furthermore, Xe for Si substitutions induce new Raman bands  
420 between  $720\text{ cm}^{-1}$  and  $750\text{ cm}^{-1}$  comparable with the broad band centred at  $786\text{ cm}^{-1}$  observed  
421 on Xe-rich San Carlos olivine (Sanloup et al., 2011). Besides these two main results,  
422 calculations reported in Table 2 also indicate that Xe for Si substitution might be the most  
423 stable from the energetic point of view. The results of table 2 are, however, to be taken only  
424 as a tentative estimation of the energetics, since calculations do not include temperature  
425 effects and consider vacancies as neutrally charged. Consequently, reactivity of Xe with the  
426 olivine is explained by substitution of Xe with silicon inducing Xe oxidation via formation of  
427 Xe-O bonds with an important degree of covalency. Presence of Fe and other redox sensitive

428 trace elements may modulate the reactivity of Xe. Incorporation sites may also slightly  
429 change their geometries compared to pure forsterite as studied here.

430 Raman spectroscopy turns out to be a valuable tool for the study of Xe incorporation  
431 in minerals. According to our findings, the high Xe polarizability makes it possible to retrieve  
432 a Xe signature for 0.11 at % Xe, and likely below. Moreover, the strong polarization of the  
433 Xe-related bands could, in principle, allow to discriminate experimentally among different  
434 incorporation mechanisms.

435 As detailed in the introduction, evaluation of Xe content based on *in situ*  
436 measurements is strongly needed. The present calculations can be used to estimate the Xe  
437 content in olivine from the *in situ* measurements (Sanloup et al., 2011). This has been done in  
438 two independent ways: i) by comparing the calculated and the X-ray diffraction measured  
439 values of cell parameters; ii) by comparing the Xe-related Raman-band intensities. According  
440 to i) or ii), the Xe concentration is  $\sim 0.40$  at % Xe or  $\sim 0.20$  at % Xe, respectively. We judge  
441  $0.40$  at % Xe to be more reliable and  $0.20$  at % Xe should be simply used as a lower  
442 boundary. However, the qualitative agreement between the two values is very encouraging,  
443 suggesting that future improvements of the present simplified structural model (based on a  
444 simple model of iron-free olivine) could provide a theoretical calibration to determine Xe-  
445 content from Raman measurements.

446 Using previously determined Xe solubility value of  $0.2 - 0.4$  at % for olivine, at  $0.9$   
447 GPa and  $1800$  K, and a value of  $0.08$  mol% Xe for a tholeitic melt at  $1.5$  GPa and  $1773$  K  
448 (Schmidt and Keppler, 2002), we found a partition coefficient for olivine/tholeitic melt  
449 ( $^{ol./melt}D_{Xe}$ ) of  $2.5 - 5$ . A value of  $^{ol./melt}D_{Xe}$  close to, or higher than  $1$ , in agreement with  
450 Hiyagon and Ozima (1986) and Broadhurst et al. (1992), challenges the supposedly  
451 incompatible behaviour of Xe in olivine. However, our value of  $^{ol./melt}D_{Xe}$  significantly differs  
452 from Heber et al. (2007) ( $^{ol./melt}D_{Xe} = 6.10^{-4}$ ). Heber et al. (2007) evaluated the Xe content of

453 olivine exclusively on regions free of bubbles and melt inclusions, on the basis that they  
454 might have formed upon quenching at ambient conditions, and if so, did not reflect the  
455 equilibrium at high pressure and temperature. In contrast Hiyagon and Ozima (1986) and  
456 Broadhurst et al. (1992) extracted Xe by heating the whole sample. Our result thus questions  
457 the actual quest for free-bubble analysis of recovered samples as a guaranty of reliable  
458 estimates of Xe content at high pressure and temperature in corresponding melt or minerals.

459 Sanloup et al. (2011) observed Xe incorporation in olivine approximately throughout  
460 the entire range of lithospheric mantle P/T conditions. Assuming that olivine represents 60%  
461 of the mantle volume (Bina, 1998), and that 90% of the Earth's primordial Xe is missing from  
462 the atmosphere (Ozima and Podosek, 1999), storing the entire amount of the 'Missing Xe' in  
463 the olivine of the upper mantle requires a Xe content in olivine of 0.012 ppb. This estimate is  
464 much lower than the solubility value of 0.4 at % (i.e. 2.5 wt % Xe) found here, hence the  
465 entire amount of the 'Missing Xe' could be stored in the upper mantle.

466 Although more data are needed on the dependence of Xe solubility in olivine on  
467 olivine chemical composition (e.g. Fe and other trivalent cations), pressures and temperatures,  
468 but also in other minerals, these preliminary results prove that storage of Xe at depth can no  
469 longer be discarded in the 'Missing Xenon' issue.

470 To our knowledge, there is currently no data available on the potential isotopic  
471 fractionation of Xe during incorporation in minerals. However, it could be a key to explain  
472 particular terrestrial Xe isotopic compositions, such as the observed  $^{136}\text{Xe}$  enrichment in  
473 mantle xenoliths of Mt Quincan (North Queensland, Australia), Victoria (Southeastern  
474 Australia) and Reunion Island, which cannot be explained by mass dependent fractionation  
475 (Czuppon et al., 2009).

476

477

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483

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676 Table 1: Bond lengths and cell parameters of Xe reference compounds: XeO<sub>4</sub> molecule and  
 677 Xe<sub>3</sub>O<sub>2</sub> at 97 GPa.

<b>XeO<sub>4</sub></b>	<b>This study</b>	<b>Calc.<sup>a</sup></b>	<b>Exp.<sup>b</sup></b>
Xe - O (Å)	1.77	1.82	1.736
<b>Xe<sub>3</sub>O<sub>2</sub></b>	<b>This study</b>	<b>Calc.<sup>c</sup></b>	<b>Exp.<sup>c</sup></b>
a (Å)	8.377	8.388	8.457
b (Å)	3.206	3.195	3.166
c (Å)	4.881	4.881	4.904
Xe <sup>4+</sup> - O (Å)	2.01	2.01	1.99
Xe <sup>0</sup> -Xe <sup>0</sup> (Å)	3.05	3.05	3.06
Xe <sup>4+</sup> -Xe <sup>0</sup> (Å)	2.95-3.00	2.95-3.00	2.96-3.01
Xe <sup>4+</sup> - Xe <sup>4+</sup> (Å)	3.21	3.19	3.17

678 <sup>a</sup> Probert, 2010, <sup>b</sup> Gundersen et al. (1970), and <sup>c</sup> Dewaele et al. (2016).

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695 Table 2: Formation energies for the different Xe incorporation mechanisms.  $\Delta E$  and  $\Delta E_v$  are  
696 defined in the text.

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	$\Delta E$ (eV)		$\Delta E_v$ (eV)
Xe <sub>Si</sub>	-0.12	Si	15.2
Xe <sub>Mg1</sub>	2.75	Mg <sub>1</sub>	8.9
Xe <sub>Mg2</sub>	1.48	Mg <sub>2</sub>	10.0
Xe <sub>i</sub>	8.21		

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721 **Figure captions**

722

723 Figure 1: Calculated structure of the  $\text{Xe}_{\text{Si}}$  incorporation site in forsterite. Note the localization  
724 of the Xe on the face of the oxygen tetrahedron. Relevant O-Xe-O angles, Xe-O bond length,  
725 and one third of the trace of the Born effective charge tensor ( $Z^{\text{eff}}$ ) are indicated.  $a, b,$  and  $c$   
726 forsterite lattice parameters are along x, y, and z axes, respectively.

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728 Figure 2: Calculated structure of the  $\text{Xe}_{\text{Mg1}}$  and  $\text{Xe}_{\text{Mg2}}$  incorporation sites (see also Fig.1  
729 caption). O3 and O3\* atoms are equivalent in pure forsterite.

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731 Figure 3: Calculated structure of the  $\text{Xe}_i$  incorporation site (see also Fig.1 caption).  
732 O3a/O3b/O3c/O3d atoms (as O2/O2\* and O1/O1\*) are equivalent in pure forsterite. For  
733 clarity, only details for O within 3.05 Å around Xe are given.

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735 Figure 4: Calculated evolution of cell parameters  $a, b, c$  and volume as a function of Xe  
736 content at 0 GPa (black symbols) and 5.7 GPa (orange symbols). Dotted lines are guides for  
737 the eyes.

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739 Figure 5: Calculated Raman spectra of the different stable configurations of Xe incorporation  
740 in forsterite (0.89 at % Xe) and pure forsterite vs. measurements (exp.) from Sanloup et al.  
741 (2011). The spectrum on Xe-rich olivine was recorded at 0.9 GPa after quenching T from  
742 1800 K. The spectrum on Xe-free olivine was recorded at 0.9 GPa and 300 K before heating  
743 (i.e. before reaction of Xe with olivine). The labels associated with some calculated Xe-  
744 related bands indicate strong polarization: the bands are visible only in parallel polarization  
745 along specific axes (corresponding to the  $a, b, c$  lattice parameters).

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747 Figure 6: Calculated Raman spectra of the different stable configurations of Xe incorporation  
748 in forsterite (0.11 at % Xe) and pure forsterite vs. measurements (exp.) from Sanloup et al.  
749 (2011).

750

751 Figure 7: Calculated Raman spectra for  $\text{Xe}_{\text{Si}}$  for various Xe content. Measurements (exp.)  
752 from Sanloup et al. (2011). The four energy windows (1, 2, 3 and 4) used to integrate the  
753 intensity of Raman bands (see text) are shown.

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755 Figure 8: Normalized integrated intensity of Xe-related Raman bands for  $\text{Xe}_{\text{Si}}$  incorporation  
756 site as a function of Xe content from the present calculations. The dotted line is a linear fit.  
757 The horizontal dashed line indicates value for experimental Raman spectrum of Xe-rich  
758 olivine from Sanloup et al. (2011).

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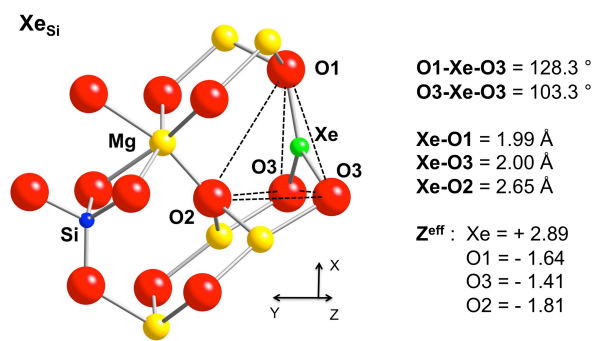
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**Figure 1**

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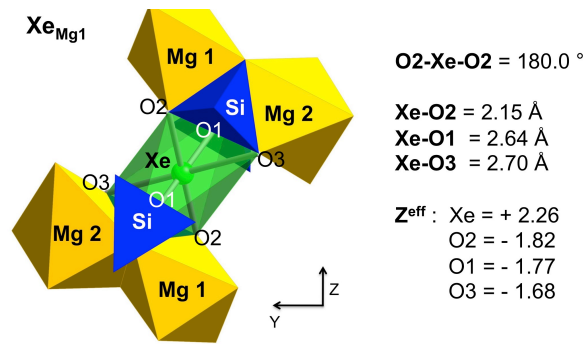
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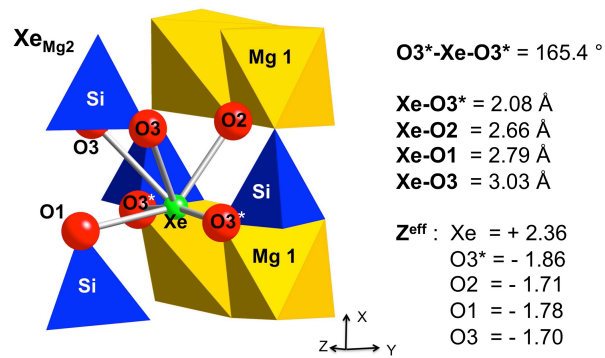
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Figure 2

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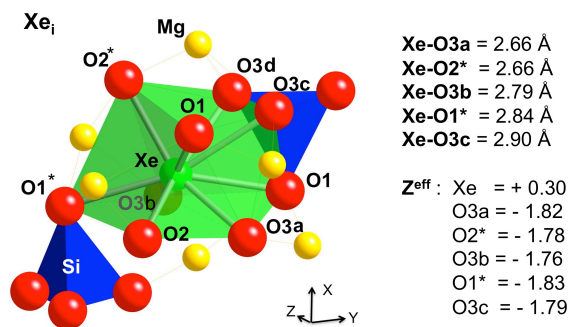
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Figure 3

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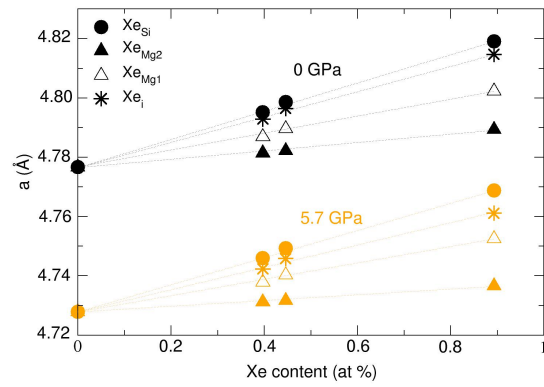
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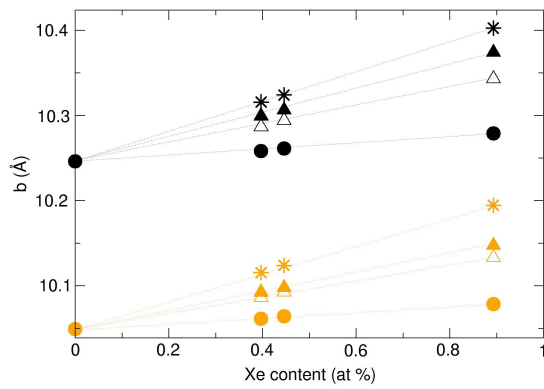
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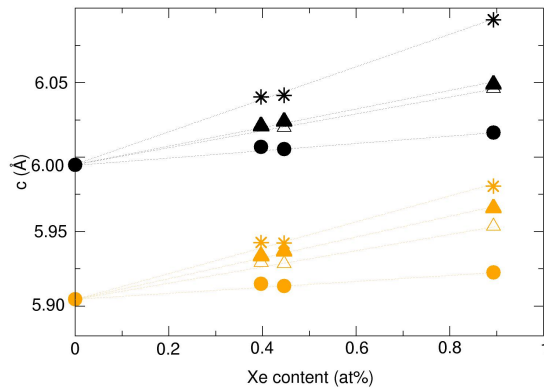
Figure 4



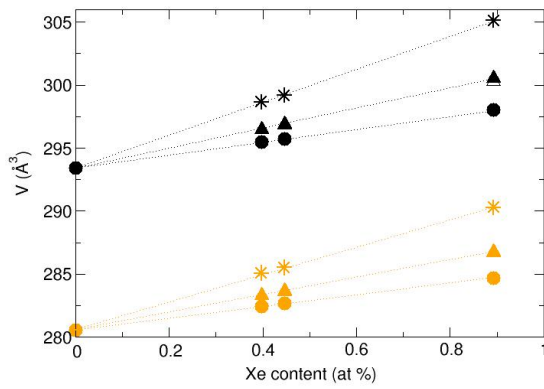
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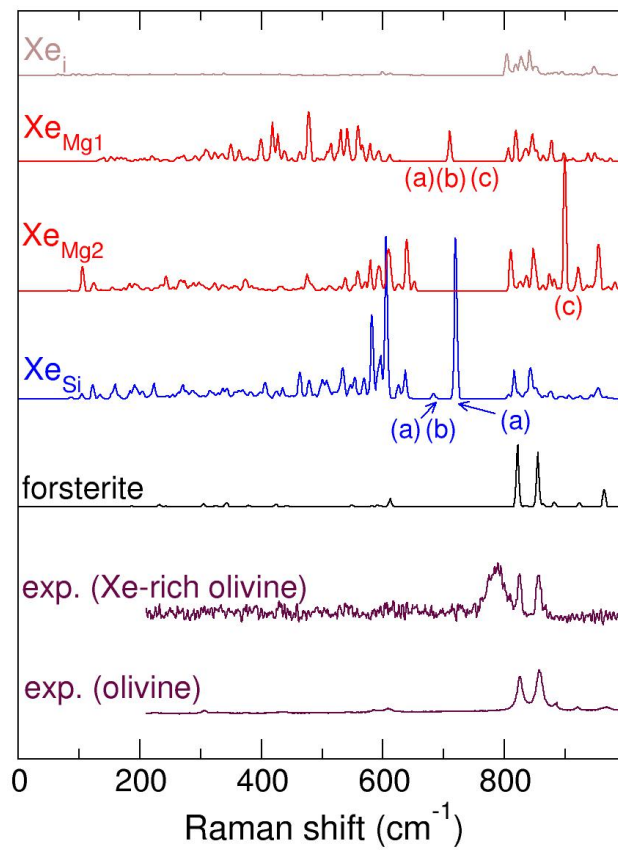


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Figure 5

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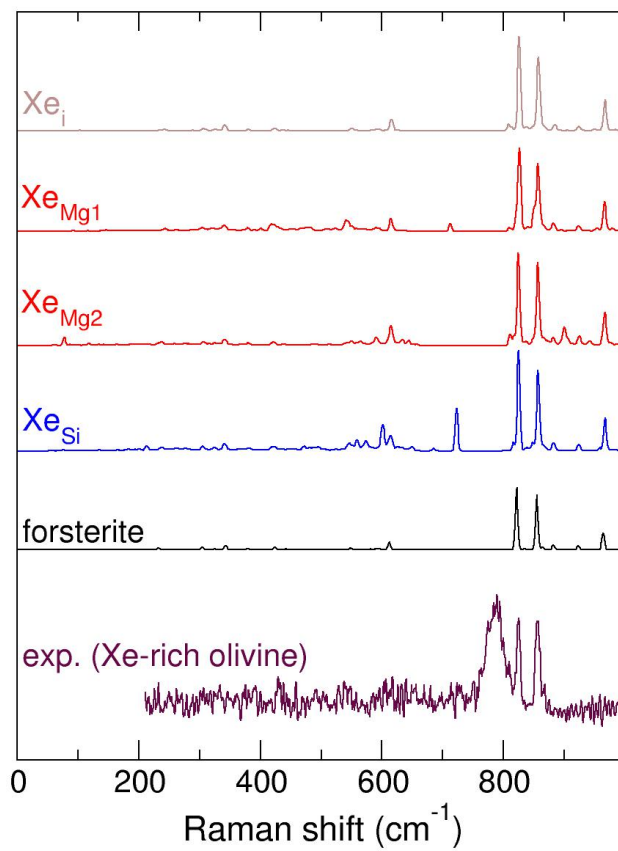
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**Figure 6**

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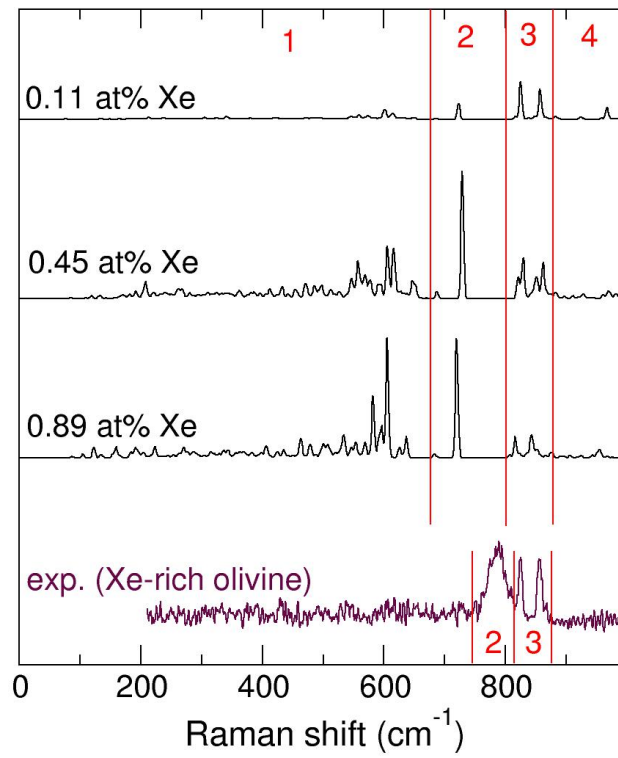
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Figure 7

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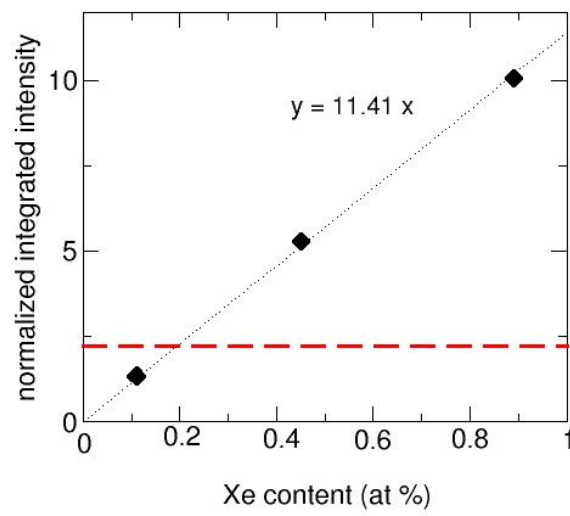
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**Figure 8**

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