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Cold and Hot Gelling of Alginate-*graft*-PNIPAM: a Schizophrenic Behavior Induced by Potassium Salts

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ABSTRACT

Recently, alginates (ALG) characterized by high mannuronic content (M blocks) have been shown to undergo a reversible sol/gel transition during cooling in the presence of potassium salts. Cold gelling takes place at low temperatures, just below 0 °C for a KCl concentration of 0.3 mol/kg, but the aggregation process can be easily shifted to higher temperatures by increasing the salt concentration. In the present paper, we take advantage of this peculiar behavior to design a copolymer with schizophrenic gelling properties. For this purpose, side-chains of poly(*N*-isopropylacrylamide) (PNIPAM), characterized by a Lower Critical Solution Temperature (LCST) in water, were grafted on the alginate backbone. Working in semi-dilute solutions, we show by coupling DSC and viscoelastic measurements that ALG-*g*-PNIPAM solutions are able to form gels either by cooling or heating depending on the ionic environment. As the aggregation process of ALG and PNIPAM depends mainly and respectively on the nature of the cations and anions, the choice of the salt is then critical to control the self-assembly behavior and the gel properties. Moreover, as the gelation process of alginates driven by the aggregation of mannuronic sequences is characterized by a large hysteresis of 20-30 °C between gelling and melting, both ALG and ALG-*g*-PNIPAM polymers offer a large versatility not only in terms of salt (nature and concentration) but also in preparation history as different states (sol or gel) can be obtained at room temperature.

Keywords: alginate, poly(*N*-isopropylacrylamide), thermo-responsive, schizophrenic behavior, associating polymers, graft copolymers.

INTRODUCTION

Alginate is a water-soluble linear polysaccharide consisting of 1-4 linked α -L-guluronic (G) and β -D-mannuronic (M) acid residues arranged in a non-regular blockwise pattern of varying proportions of homopolymeric GG, MM and heteropolymeric alternating MG sequences.¹⁻³ Alginates are one of the main structural components in marine brown algae, and like pectins in plants, they play an important physiological role in the intercellular matrix forming hydrogels in the presence of divalent cations and bringing mechanical strength and flexibility.⁴ For alginates and pectins the gelation properties are induced by G-blocks (guluronic in alginates and galacturonic in pectins) which are able to bind cooperatively divalent cations forming the so-called egg-box model early described by Grant et al..⁵ This intrinsic ability to form ionotropic gels in the presence of multivalent cations, combined with their specific biological properties, biocompatibility and low toxicity, have lead to important developments in food, medicine and pharmacy. For instance, their gelling, viscosifying and stabilizing properties have been used for decades to improve and modify the texture of foods or to prepare wound dressing or dental impression materials.⁶⁻⁹ There is actually a growing literature dealing with applications of alginate in drug release or tissue engineering where they form very useful membranes and scaffolds.¹⁰⁻¹⁵

As many other polysaccharides like hyaluronic acid, dextran, carboxymethylcellulose or chitosan, alginates are also considered as an important polymeric platform for developing thermoresponsive assemblies that can be used on purpose for biomedical applications like drug delivery or tissue engineering.¹⁶⁻²⁰ For instance, alginates can be modified with polymer precursors characterized by a Lower Critical Solution Temperature (LCST) in water, like poly(ethylene oxide-co-propylene oxide),²¹⁻²² poly(oligo ethylene glycol methacrylate),²³ poly (*N*-vinyl caprolactam),²⁴ or poly(*N*-isopropylacrylamide)²⁵⁻²⁷ (PNIPAM) in order to design temperature responsive micelles or

vesicles for controlled drug delivery or copolymer solutions with sol/gel properties paving the way to injectable hydrogels and 3D cell culture. Although ionotropic gels formed by the complexation of alginates with multivalent cations are almost temperature independent, we have recently shown that alginates having a high mannuronic content (high-M) can form thermotropic gels upon cooling when they were studied at high ionic strength in the presence of monovalent cations. For instance, the cold gelation that was observed just below 0 °C for $[K_2CO_3]=0.2$ m can be shifted to higher temperatures by increasing the salt concentration: $T_{as}=23$ °C with $[K_2CO_3]=0.5$ m. The formation of these assemblies observed during cooling between mannuronic sequences was assumed to follow a UCST behavior (Upper Critical Solution Temperature) with a nucleation and growth mechanism responsible for the large hysteresis observed between gelling and melting temperatures.

Considering this specific property of high-M alginates, and concurrently to our initial strategy of developing injectable hydrogels, the goal of this work was to investigate the schizophrenic thermoresponsive behavior that can be developed by grafting LCST PNIPAM side-chains onto a high-M alginate backbone (see ALG-g-PNIPAM in **Figure 1**).

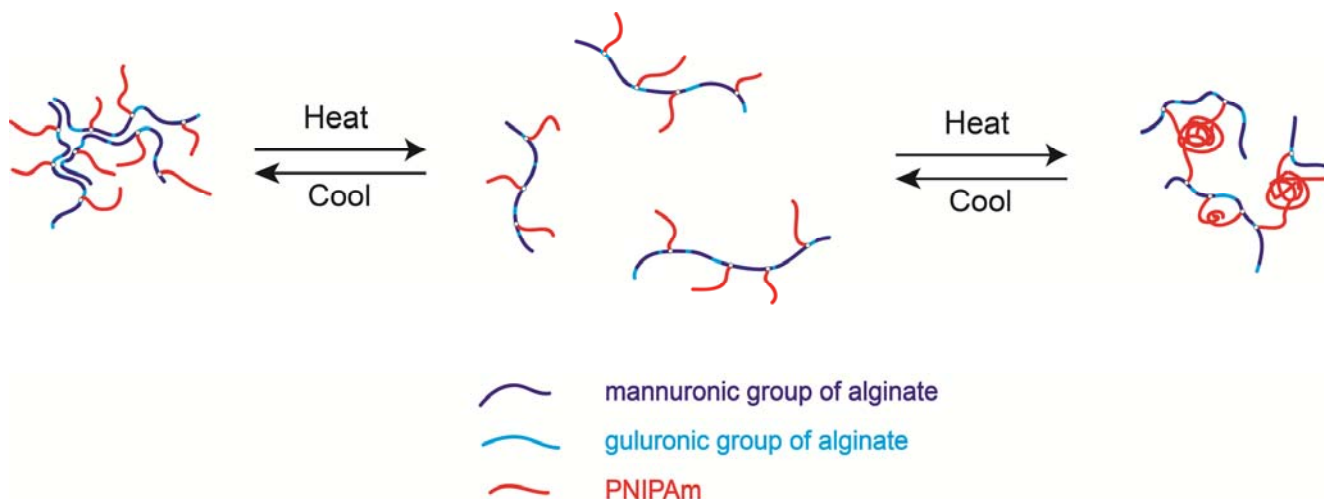


Figure 1. Schematic representation of the schizophrenic behavior of ALG-g-PNIPAM in aqueous solution of potassium salts as a function of temperature.

In order to deliver a clear and progressive message concerning these opposite sol/gel transitions, the following discussion is organized into three main parts considering: 1) the cold gelling process of high-M alginate (ALG) solutions with the impact of polymer and salt concentrations on the association temperature and hysteresis, 2) the hot gelation in pure water of alginates modified with PNIPAM side-chains (ALG-*g*-PNIPAM), in these conditions only PNIPAM grafts reversibly self-associate by changing the temperature, and 3) the superposition of the two mechanisms, cold and hot gelling, when various potassium salts are added into the ALG-*g*-PNIPAM solution.

EXPERIMENTAL SECTION

Materials

N-isopropylacrylamide (NIPAM, Aldrich), cysteamine hydrochloride (AET·HCl, Fluka), potassium peroxydisulfate (KPS, Aldrich), 1-Ethyl-3-(3-dimethylaminopropyl)carbodiimide (EDCI, Aldrich) and *N*-hydroxysuccinimide (NHS, Fluka) were used as received. Sodium alginate (ALG) was kindly provided by Brothier Laboratories (Fontevraud-l'Abbaye, France). All the inorganic salts, potassium carbonate (K₂CO₃ from Aldrich), potassium chloride (KCl from Fluka), sodium chloride (NaCl from Fluka), lithium chloride (LiCl from Aldrich) and potassium thiocyanate (KSCN from Aldrich) were used as received without further purification. All organic solvents were analytical grade and water was purified with a Millipore system combining inverse osmosis membrane (Milli RO) and ion exchange resins (Milli Q) for synthesis and purification.

Synthesis of ALG-*g*-PNIPAM

The synthesis of thermoresponsive alginates was performed according to a two-step process, originally developed with synthetic architectures²⁸⁻³⁰ and since widely extended to a wide variety of

polysaccharides.^{21, 31-33} Typically, PNIPAM telomer was first synthesized through free radical polymerization, as described elsewhere.³⁴ Then, the copolymer ALG-g-PNIPAM was obtained by coupling amino-terminated telomer with carboxylic group in alginate. The process is illustrated in **Figure 2** and the detailed procedure is described in supporting information.

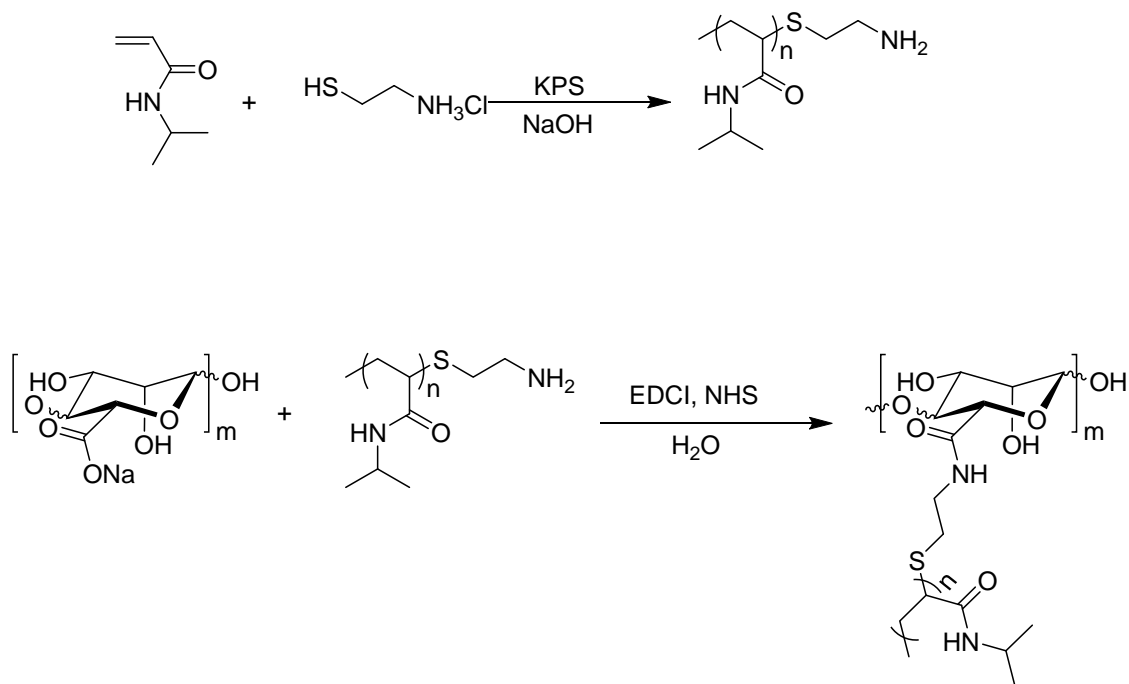


Figure 2. Two-step synthesis of graft copolymer ALG-g-PNIPAM.

The ¹H NMR spectra of ALG, PNIPAM as well as ALG-g-PNIPAM are given in supporting information (**Figure S1**) and their macromolecular characteristics are reported in **Table 1**.

Table 1 Characteristics of polymers

	M_n kg/mol	M_w kg/mol	D	M/G^a	Weight composition PNIPAM/ALG (wt/wt) ^b	Grafting density PNIPAM/ALG (mol/mol) ^b
ALG	60	134	2.2	1.6	-	-
PNIPAM telomer	14	22	1.6	-	-	-
ALG-g-PNIPAM	76	163	2.1		40/60	3/1

^{a,b} determined by ¹H NMR at 80 and 25 °C, respectively.

Size exclusion chromatography (SEC)

Experiments were carried out with two different chromatographic systems. The first one, which was applied to the absolute characterization of polymer samples, is a Viscotek triple detector (TDA 302) equipped with three columns (OH-pak). During the analysis carried out at 28 °C in aqueous media, the flow rate was controlled at 0.6 mL·min⁻¹ using 0.2 M NaNO₃ as mobile phase. The molar masses were derived from a universal calibration curve based on Pullulan standards from Sopares (OmniSEC software). The other chromatographic system from Waters is equipped with similar Shodex OH-pak columns thermostated at 10 °C and a differential refractometer (Shimadzu RID-10A). This system was mainly used to follow, without purification, the conversion of the reactions. The flow rate was controlled at 0.90 mL·min⁻¹ with 0.5 M LiNO₃ as mobile phase.

Preparation of polymer solutions

All the studies were performed with semi-dilute solutions, either in pure water or in salted media. For each solution, the polymer was allowed to dissolve in the salted medium on a shaking table (250 rpm) at room temperature for at least 24 h, until homogeneous solutions were obtained. The polymer concentrations are given in weight percentage (*wt%*) and the salt concentration is expressed in moles of salt per kg of solution (*m*).

Nuclear magnetic resonance (NMR)

The composition of the copolymer was characterized with a Bruker Avance III HD spectrometer operating at 400 MHz for ¹H, using a standard 5 mm broadband Smart probe at 25 °C. Complementary analyses dedicated to ALG-*g*-PNIPAM self-assembly were also carried out at different temperatures using a Bruker Avance III HD spectrometer operating at 600 MHz for ¹H, with 32 transients, 45 degree flip angle, 2.5 sec acquisition time, 2 sec relaxation delay. The temperature

control was achieved by a Bruker BCU II unit and a build in temperature control unit. The sample was allowed to equilibrate for 10 minutes at each temperature. The ^1H chemical shifts were referred to residual HOD peak at each temperature.

Rheology

The viscoelastic properties of ALG and ALG-g-PNIPAM were studied in aqueous solutions, in the semi-dilute regime, using a stress-controlled rheometer (Haake RS600) equipped with a cone/plate geometry (diameter 35 mm, angle 2, truncature 104 μm). The experiments were performed in the linear viscoelastic regime which was established for each sample by a stress sweep at 1 Hz. The temperature was controlled by a high power Peltier system that provides fast and precise adjustment of the temperature during heating and cooling stages. Typically, the experimental conditions were fixed at constant shear stress (2 Pa) and frequency (1 Hz). According to a previous study³⁵, a specific procedure was applied in order to remove the history of the sample (storage and preparation). For that purpose the sample was first heated to 60 °C (**step 1**) and then cooled from 60 to 2 °C (**step 2**) at a constant rate. Then, after 10 min at 2 °C, the sample was heated again from 2 to 60 °C (**step 3**) and immediately cooled from 60 to 2 °C (**step 4**). Heating and cooling scans were performed at 2 °C·min⁻¹ for all the samples, but slower rates (0.5 and 1 °C·min⁻¹) were also used to investigate the impact of heating and cooling procedures on the sol/gel transition. The reproducibility was checked for each experiment and a particular care was taken to avoid the drying of the sample by using a homemade cover which prevents from water evaporation during analysis. This was verified with the equivalence of steps 2 and 4. In the following, dynamic moduli (G' and G'') as well as complex viscosity (η^*) will be plotted between 2 and 60 °C using only steps 2 and 3.

Differential Scanning Calorimetry (DSC)

The phase transition of copolymers in aqueous solution was investigated by Differential Scanning Calorimetry using a DSC Q200 from TA instrument. Polymer solutions (ca. 80 mg), equilibrated with a reference filled with the same quantity of solvent, were submitted to temperature cycles between 2 and 60 °C. As for rheology experiments, the heating and cooling rates were fixed at 2 °C.min⁻¹. For each sample, the standard procedure was the following: (i) cooling from 20 to 2 °C; (ii) isotherm at 2 °C (10 min); (iii) **step 1**, heating from 2 to 60 °C; (iv) isotherm at 60 °C (10 min); (v) **step 2**, cooling from 60 to 2 °C; (vi) isotherm at 2 °C (10 min); (vii) **step 3**, heating from 2 to 60 °C. Only steps 2 and 3 will be used for the data analysis.

RESULTS AND DISCUSSION

Cold Gelation of Alginates

Sol/gel transition of ALG solutions

As discussed previously, alginates are widely known to viscosify and to form gels either by lowering the pH, or by adding multivalent cations that induce strong zip-complexes between the G-blocks.^{3, 36} Nevertheless these are not the only ways to prepare alginate gels in aqueous media and we have recently reported that alginates with a high mannuronic content (high-M) were also able to reversibly self-associate at low temperature in the presence of potassium salts.³⁵ This behavior, that has been initially highlighted with potassium carbonate, is displayed in **Figure 3** with an alginate solution ($C_p=5$ wt%) prepared in KCl 0.6 m. Starting at high temperature ($T=60$ °C), with a viscous solution ($G''>G'$), the dynamic moduli continuously increase during cooling down to 15 °C where a sharp sol/gel transition takes place. At $T=2$ °C the elastic modulus is more than 5 times higher than the loss

modulus and it is almost independent of the frequency in the range explored (10^{-1} - 10^1 Hz). The gelation process is reversible with an important hysteresis as the initial viscoelastic properties are recovered only above 40 °C. This viscoelastic loop, that can be repeated as many times as wanted, weakly depends on the rate of cooling/heating, at least in the range investigated (0.5 to 2 °C/min; see **Figure S2** in supporting information). The crossover between G' and G'' will be used in the following to define the sol/gel transition temperature as this simple criterion matches fairly well the critical conditions defined by Winter as the point where $\tan\delta=G''/G'$ becomes independent of frequency (see **Figure S2** in supporting information).³⁷ Using the experimental results reported in **Figure 3**, $T_{s-g}=12$ °C and $T_{g-s}=38$ °C can be determined for the sol/gel and the gel/sol transition temperatures, respectively.

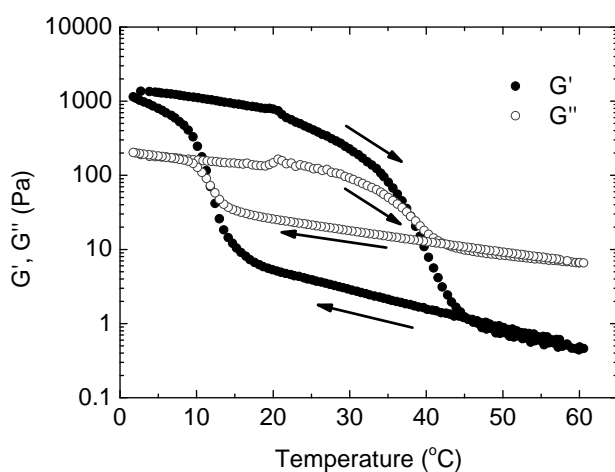


Figure 3

Temperature dependence of elastic (G' ; filled symbol) and loss (G'' ; hollow symbol) moduli of ALG solution ($C_p=5$ wt%, $[KCl]=0.6$ m). Heating and cooling rates = 2 °C.min⁻¹, $f=1$ Hz.

The association process between alginate chains, which is triggered at relatively high salt concentration and low temperature, results in the formation of an elastic gel if the polymer concentration is high enough. In the case of this alginate sample, this situation is observed for $C_p \geq 2$ wt%. At lower concentration ($C_p=1$ wt%), the solution retains a predominant viscous character at low temperature despite the large increase in dynamic moduli below 16 °C (see **Figure S3** in supporting information). In order to get a quick and quantitative understanding of the rheological behavior, the viscoelastic data can also be plotted by using the complex viscosity (η^*) as a single parameter (see

Figure 4). Indeed, as $\eta^* = \sqrt{(G'/\omega)^2 + (G''/\omega)^2}$, the complex viscosity can be identified to the Newtonian viscosity in the sol regime ($\eta^* = \eta_0$, for $G'' \gg G'$) while it scales with the plateau modulus (G_0) in the gel regime (for $G' \gg G''$, $\eta^* \cong G'/\omega \cong G_0/\omega$ with $\omega = 6.28$ rad/s in our conditions). As shown in **Figure 4**, sol/gel and gel/sol transitions give rise to abrupt changes of η^* versus temperature and this plot can be interestingly normalized by dividing η^* by the Newtonian viscosity (η_0) of the unassociated polymer solution: $\eta_{rel}(T) = (\eta^*(T)/\eta_0(T))$. For that purpose, the experimental data obtained in the sol regime have been fitted by the Andrade equation $\eta_0 = A \exp(E_a/RT)$ in order to extrapolate the Newtonian viscosity in the whole temperature range: between 2 and 60 °C (see **Figure 4**).

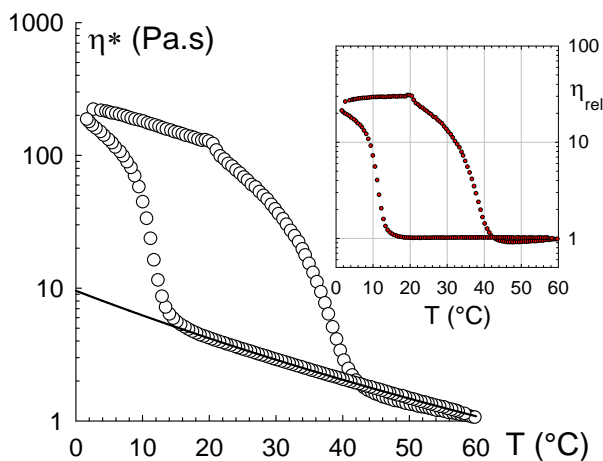


Figure 4

Temperature dependence of complex viscosity (η^* ; ○) and relative viscosity (η_{rel} ; ●) of ALG solution ($C_p = 5$ wt%, $[KCl] = 0.6$ m). Heating and cooling rates = 2 °C.min⁻¹, $f = 1$ Hz. Andrade equation (—): $\eta_0 = 5.5 \cdot 10^{-5} \exp(3295/T)$

In the Andrade equation mentioned above, A is a constant, R is the gas constant, T is the absolute temperature and E_a is the activation energy for viscous flow. All the alginate solutions investigated in this study display similar activation energies ($E_a = 26 \pm 2$ kJ/mol) which are typical of non associating polymer solution in this range of polymer concentrations.³⁸ As shown in **Figure 4** (inset), the plot of the relative viscosity (η_{rel}) clearly emphasizes the association process with the baseline corresponding to the “sol” domain (unassociating polymer chains) and large increase of the relative viscosity as soon as the association mechanism takes place leading to the physical gelation of the solution, at least for

concentrations higher than 1 wt%. This normalized curve will be calculated thereafter for all the samples in order to provide an efficient comparison of their associating behavior. Moreover, the relative viscosity will be also used to define systematically the beginning of the association process observed during cooling (T_{as}) as well as the end of the dissociation one observed upon heating (T_{dis}). For that purpose, the critical temperatures were determined at $\eta_{rel}=1.1$ (10 % deviation over more than 1000%); a criterion that can be accurately applied to all the formulations with less than 1 °C difference by comparison with lower deviation values. While viscoelastic measurements clearly highlight the macroscopic transition occurring for high-M alginates in the presence of high concentrations of potassium salt, the thermodynamic aspects of the phase transition can also be studied in a complementary manner by DSC. As previously reported, the cold gelation of high-M alginates is an exothermic process which slightly depends on the rate of cooling.³⁵ The enthalpogram given in **Figure 5**, obtained with the reference solution ([ALG]=5 wt% and [KCl]=0.6 m) using the same cooling and heating rates (2 °C/min), displays a well-defined exothermic peak ($\Delta H=5.5$ J/g_{ALG}) upon cooling that starts around $T_{as}=10.5$ °C with a maximum located at $T_{M-Cool}=8.8$ °C.

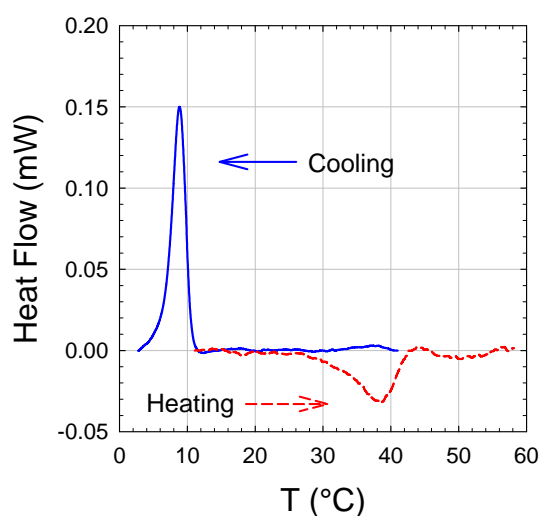


Figure 5
Thermograms of
ALG solution
($C_p=5$ wt%, [KCl]=0.6 m).
Cooling and heating
rates = 2 °C.min⁻¹.

By heating the solution from 2 °C back to 60 °C, a much broader endotherm is observed for the dissociation process that extends from about 30 °C up to $T_{dis}=44$ °C with a maximum located at

$T_{M-Heat}=37.6$ °C. Such large hysteresis and the transition enthalpy are in good agreement with our previous data reported with similar alginate samples studied in potassium carbonate solutions.³⁵ This phenomenon is attributed to the self-association of M-blocks into clusters following a nucleation and growth process. Even if the first events of self-assembling and self-dissociation taking place upon cooling and heating, respectively, are more easily detected by rheology rather than by calorimetry, the sol/gel transition temperatures are very close to the maxima of enthalpograms (see **Table 2**) pointing out the tight correlation between the formation of specific interactions and the dramatic increase of viscoelastic properties.

Influence of salt concentration

From complementary experiments performed with the former solution ($C_p=5$ wt% and $[KCl]=0.6$ m) by adding either a hydrogen bonding competitor ($[urea]=1$ m) or surfactant (sodium dodecyl sulfate: $[SDS]=(5-20)\cdot 10^{-3}$ m), we do not really observed significant modifications of the gelation process, neither on the transition temperatures nor on the absolute values of the dynamic moduli (see **Figure S4** in Supporting Information). These results highlight that hydrophobic interactions as well as hydrogen bonds are not the key interactions involved in the association mechanism of alginates and they emphasize the central role of ionic interactions. This is emphasized in **Figure 6**, where the relative viscosity of the alginate solution ($C_p=5$ wt%) has been studied with different KCl concentrations. Working in the temperature range 2 to 60 °C, the gelation process begins to be observed for KCl concentration of 0.4 m. In this case the association starts at very low temperature ($T_{as}=5.5$ °C) with a gelation threshold around $T_{s-g}=3$ °C and a dissociation taking place at higher temperature with a similar hysteresis ($T_{g-s}-T_{s-g}=25-30$ °C) as reported with the previous solution. By increasing the KCl concentration, the association temperature is shifted to higher temperatures, up to 19-20 °C for KCl 0.7 m, with a similar shift for the dissociation process.

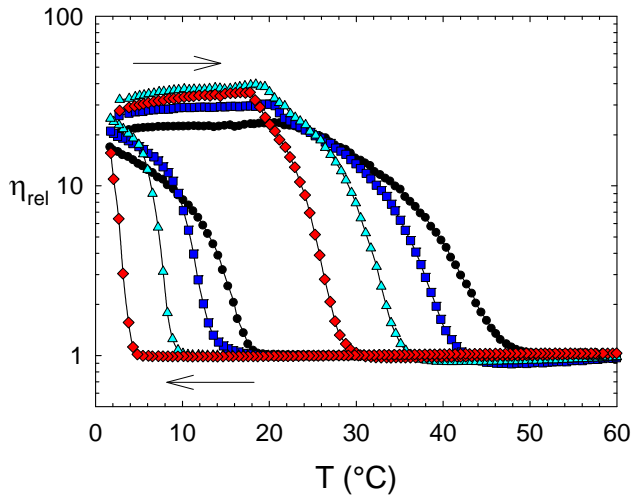


Figure 6

Temperature dependence of relative viscosity of alginate aqueous solution ($C_p=5$ wt%) with different KCl concentrations: 0.4 m (◆); 0.5 m (▲); 0.6 m (■) and 0.7 m (●) (heating and cooling rate = 2 °C.min⁻¹, $f=1$ Hz).

DSC experiments performed in similar conditions with the same solutions are in quantitative agreement with rheology for the transition temperatures and show only a slight decrease of the enthalpy with increasing salt concentrations (see **Table 2**). DSC also demonstrates that alginates can undergo transition at lower KCl concentration (0.3 m for instance) if the solution is cooled below 0 °C.

Table 2. Transition temperatures of alginate solutions ($C_p=5$ wt%) as a function of KCl concentration.

C_p wt%	[KCl] (m)	Rheology					DSC		
		Cooling		Heating		ΔT	Cooling		Heating
		$T_{s/g}$ °C	T_{as} °C	$T_{g/s}$ °C	T_{dis} °C	$T_{g/s} - T_{s/g}$ °C	T_{M-Cool} °C	ΔH_{gel} J/g _{ALG}	T_{M-Heat} °C
5	0	/	/	/	/	/	/	/	/
	0.2	/	/	/	/	/	/	/	/
	0.3	/	/	/	/	/	-3.2	6.7	20.2
	0.4	3.0	5.5	26.0	30	23.0	0.7	6.0	26.9
	0.5	7.5	10.5	32.5	37	25.0	5.4	6.2	33.2
	0.6	12.0	14.7	38.0	44	26.0	8.8	5.5	37.6
	0.7	15.0	19.3	41.7	50	26.7	14.9	5.5	44.7
1	0.6	/	16.4	/	51.1	/	10.1	6.8	44.5
2		13.5	15.0	38.0	46.8	24.5	10.1	7.0	43.9
3		13.0	15.3	40.5	44.9	27.5	9.8	6.2	41.5
4		11.3	13.0	39.5	44.2	28.2	8.9	5.9	39.2
5		12.0	14.7	38.0	44.0	26.0	8.8	5.5	37.6
6		10.5	12.9	37.0	40.9	26.5	8.7	5.4	37.6

Hot Gelation of Modified Alginates

The introduction of LCST polymer sequences (blocks or side-chains) within a hydrophilic macromolecular architecture has been a hot topic over the past 25 years as these systems provide an original response to the design of responsive aqueous formulations of paramount interest in many different fields such as oil recovery, cosmetics or biomedicine. Mostly prepared from synthetic polymers^{28, 30, 39-40}, their development has been extended to biopolymers, mainly polysaccharides, especially for biomedical applications.¹⁶⁻²⁰ This is the case of alginates that have been modified with various LCST polymers like poly(alkylene oxide) or poly(N-alkylacrylamide) in order to develop hot gelling properties.^{21-22, 25-27} Such behavior is highlighted in **Figure 7** with the ALG-g-PNIPAM sample studied at $C_p=5$ wt% in pure water where alginate itself does not self-associate. In these conditions, the solution which is originally viscous at low temperature undergoes a sol/gel transition by heating with a clear deviation of dynamic moduli, as well as the relative viscosity (see **Figure S5** in supporting information), at $T_{as}=34$ °C and a gelation threshold ($G' = G''$) at $T_{s/g}=44$ °C. Here, we can notice that the association temperature is in perfect agreement with the beginning of the phase transition ($T_{as}=34$ °C) that can be determined by DSC on an equivalent solution of PNIPAM precursor. Moreover, contrary to the thermogelation process of un-modified alginates in salted media, the self-assembling behavior induced by the micro-phase separation of PNIPAM side chains is reversible with practically no hysteresis as we can see from both viscoelasticity (**Figure 7**) and DSC experiments (see **Figure S6** in supporting information). By comparison with the transition enthalpy of the PNIPAM precursor ($\Delta H=40$ J/g_{PNIPAM}), the heat release occurring during the phase transition of the graft copolymer solution is quantitatively lower ($\Delta H=28$ J/g_{PNIPAM}). This variation, which has already been observed with alginates modified with other LCST stickers like poly(ethylene oxide-co-propylene oxide) (ALG-PEPO),²¹ as well as with other synthetic graft copolymers,⁴¹

mirrors a weaker dehydration process related to steric hindrance and/or electrostatic repulsions induced by the alginic backbone.

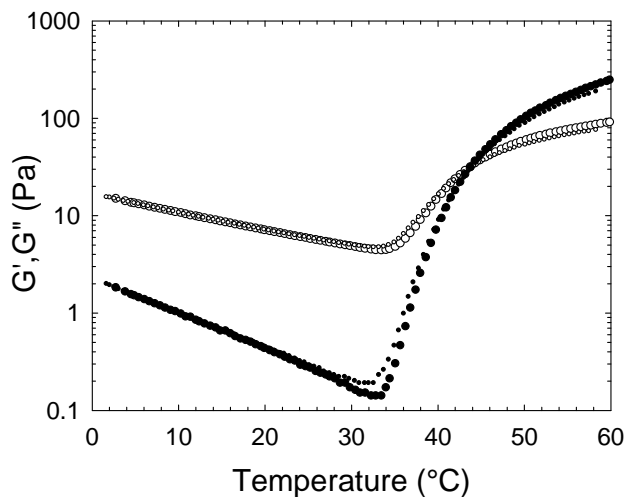


Figure 7

Temperature dependence of dynamic modulus G' (filled symbol) and G'' (hollow symbol) of ALG-g-PNIPAM aqueous solution ($C_p=5$ wt%, H_2O) upon heating (\bullet \circ) and cooling (\bullet \circ) (heating and cooling rate= 2 $^{\circ}C.min^{-1}$, $f=1$ Hz).

Complementary experiments have been carried out by 1H NMR as a function of temperature. From the series of NMR spectra obtained between 20 and 60 $^{\circ}C$ (**Figure 8**), we can see that the characteristic signals of NIPAM units, methyl (~ 1 ppm) and methine (3.8 ppm), sharply decrease within the interval 35-50 $^{\circ}C$ in good agreement with viscoelastic and DSC data.

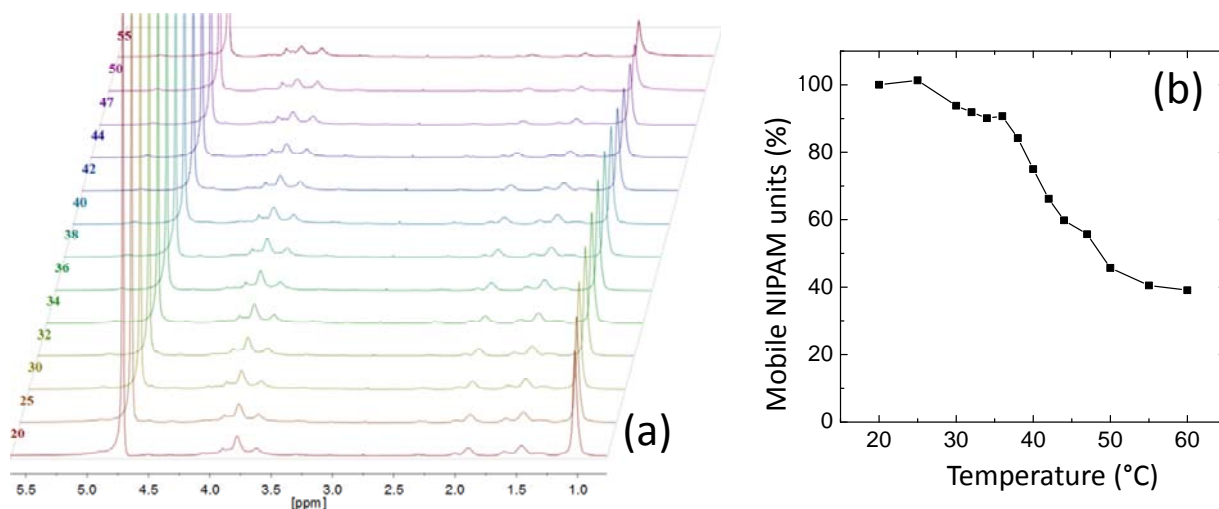


Figure 8. (a) Temperature variation of 1H NMR spectrum of ALG-g-PNIPAM in D_2O solution ($C_p=5$ wt%) and (b) temperature dependence of mobile NIPAM units within the graft copolymer.

This feature, which has been widely reported with PNIPAM copolymers or gels,^{34, 42-43} can be assigned to the formation of a solid-like rich-PNIPAM phase where the motions of the functional groups are strongly reduced. Quantitatively it appears that around 50 to 60% of NIPAM units have lost their mobility at 60 °C, within the timescale of the NMR experiment, and strongly self-interact inside rich-PNIPAM domains. This information can be interestingly compared with the 40% decrease of the transition enthalpy that has been observed by DSC between ungrafted and grafted PNIPAM chains if we assume that the remaining fraction of mobile NIPAM units (40 to 50%) remains hydrated in close contact with the aqueous phase due to steric hindrance imposed by the backbone.

Cold and Hot Gelation of Modified Alginates

Gel/Sol/Gel transitions of ALG-g-PNIPAM

Since high-M alginates and PNIPAM are characterized by opposite thermoresponsive behaviors in aqueous media, the main goal of this study was to couple these two precursors within the same macromolecular architecture in order to develop a versatile system able to develop antagonistic thermogelation properties. To that purpose, a first set of experiments was performed in [KCl]=0.6 m, where both high-M alginate and PNIPAM are known to exhibit thermoresponsive properties. The results of the dynamic analysis reported in **Figure 9** for a 5 wt% solution clearly emphasize such antagonistic behaviors, which can be discussed as follows. For this purpose we have also added in **Figure 9** a schematic representation of the associations taking place at low temperature between mannuronic sequences and at high temperature between PNIPAM side-chains. Of course for the sake of simplicity, this drawing does not consider the relative size of the associative sequences nor their number of aggregations which are much larger. At high temperature (T=60 °C), where ALG chains do not associate, the phase separation of PNIPAM side-chains into connecting micro-domains is

responsible for the gel formation ($G' \gg G''$) with an elastic modulus close to 1 kPa. By decreasing the temperature, the attractive interactions between PNIPAM stickers are progressively weakened, as well as the gel strength, and the association mechanism totally disappears below 26 °C when all PNIPAM side-chains recover their full solubility in the salt solution.

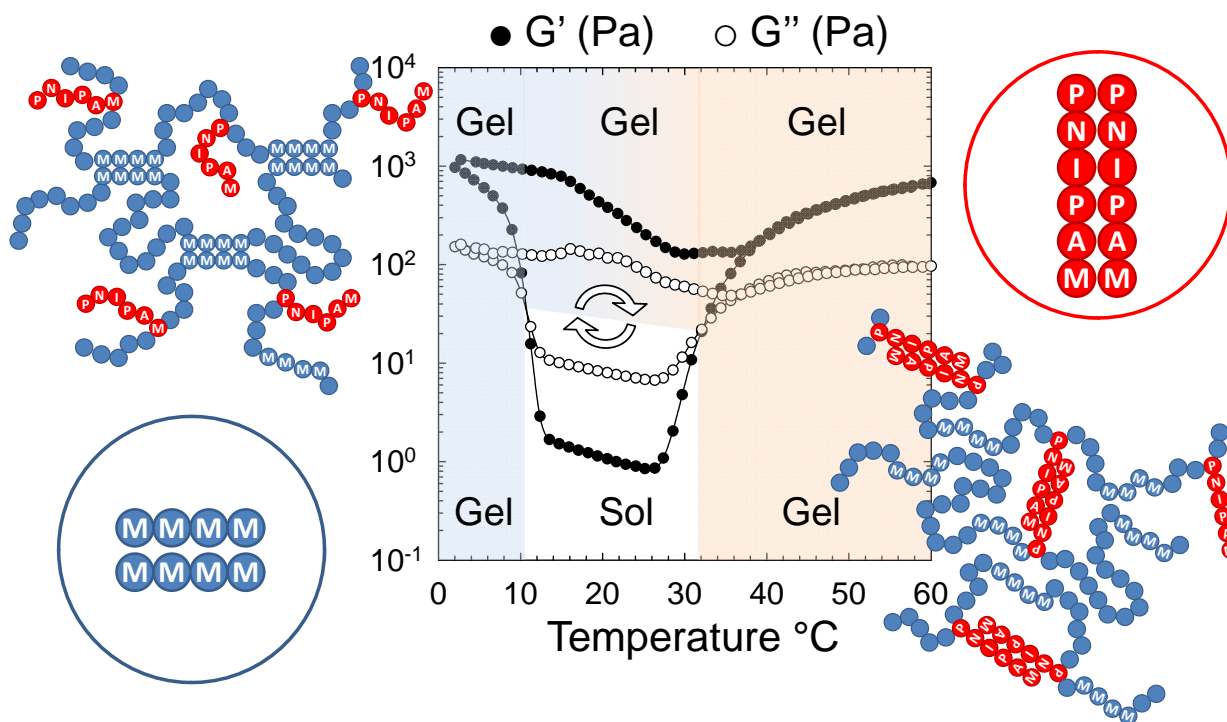


Figure 9. Temperature dependence of dynamic modulus G' (filled symbol) and G'' (hollow symbol) of ALG-g-PNIPAM aqueous solution ($C_p=5$ wt%, $[KCl]=0.6$ m) upon heating and cooling (heating and cooling rate= 2 °C.min $^{-1}$, $f=1$ Hz). Schematic representation of the schizotropic behavior.

Then, the solution keeps its liquid-like behavior between 26 and 14.5 °C where the association mechanism of high-M alginate chains takes over with cold gelling. As it will be discussed later, the transition temperatures measured with the graft copolymer are very close to those which can be determined separately with alginate or PNIPAM solutions. In other words, each polymer precursor works independently within the grafted structure without any strong interference with the other. Nevertheless, due to the great hysteresis of the association/dissociation process of alginates, once the

gel is formed at low temperature, the formulation retains its elastic behavior over the entire temperature range (2 to 60 °C) as the association of PNIPAM stickers overlaps the dissociation process of alginate chains. For instance, taking into account the viscoelastic data obtained with unmodified alginates, $T_{g/s}$ and T_{dis} are expected at about 38 and 44 °C, respectively (see **Table 2**), although the association of PNIPAM starts well below at $T_{as} \cong 26$ °C. As a consequence, by working at room temperature ($T \cong 20$ °C), it is possible to obtain two distinct behaviors from the same initial formulation: a viscous solution or an elastic gel, according to whether it has been prepared from high or low temperatures, respectively.

As shown in **Figure 10**, the dissociation process of PNIPAM stickers upon cooling, as well as the association of alginates chains, can be also easily distinguished by DSC as these transitions take place at different and well defined temperatures.

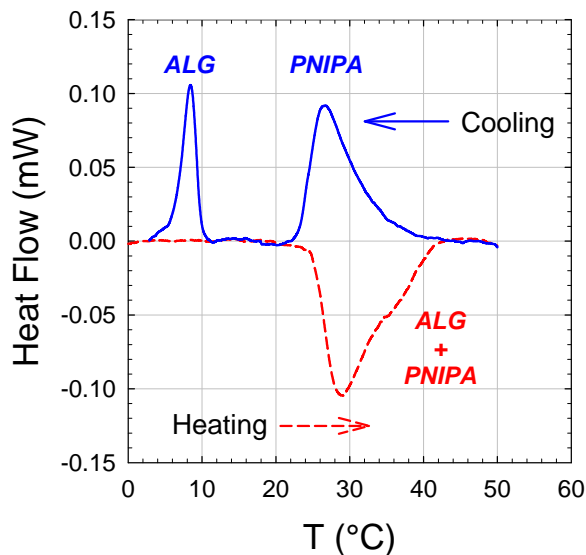


Figure 10

Thermograms of
ALG-g-PNIPAM solution
($C_p=5$ wt%, $[KCl]=0.6$ m).
Cooling and heating rates
are 2 °C.min⁻¹.

From the exothermic signatures recorded during cooling, the characteristic temperatures as well as the enthalpies were determined and reported in **Table 3**. The comparison of these values with those given in **Table 2** for alginate solutions clearly demonstrates that association and gelation process take place at similar temperature, whether alginate chains are grafted or not. The transition enthalpy also

remains comparable within the limit of accuracy of the measurements. Concerning the self-assembly induced by PNIPAM stickers, the hysteresis remains weak, less than 3 °C, and the transition temperatures are close to those determined from the solution of PNIPAM precursor investigated in the same conditions ($T_{M-Cool}=25.4$ °C; $T_{M-Heat}=27.3$ °C).

Table 3. DSC analysis: influence of the nature of salt and its concentration on the transition temperature and enthalpy of ALG-g-PNIPAM solutions ($C_p=5$ wt%).

Salt	(m)	ALG			PNIPAM		
		Cooling		Heating	Cooling		Heating
		T_{M-Cool} °C	$ \Delta H_{Cool} $ J/g _{ALG}	T_{M-Heat} °C	T_{M-Cool} °C	$ \Delta H_{Cool} $ J/g _{PNIPAM}	T_{M-Heat} °C
KCl	0	/	/	/	34.3	28	37.0
	0.2	/	/	/	32.6	21	34.5
	0.3	-4.4	8.3	15.6	31.3	25	32.9
	0.4	0.5	7.2	22.6	29.5	18	31.2
	0.5	5.4	5.1	/	28.3	25	30.0
	0.6	8.5	5.8	/	27.1	26.5	29.6
	0.7	12.1	5.5	/	25.4	26	27.4
LiCl	0.6	/	/	/	30.7	24	32.9
NaCl	0.6	/	/	/	27.6	25	29.4
KCl	0.6	8.5	5.8	/	27.1	26.5	29.6
KSCN	0.6	7.5	5.3	/	38.7	15	40.2
K ₂ CO ₃	0.3	3.6	6.3	/	23.5	22	26.1

Complementary experiments performed in the same salt conditions (KCl 0.6 m) by varying the polymer concentration display very similar transition temperatures and enthalpies for both ALG backbone and PNIPAM side-chains (see **Figure 11** and **Tables S1** and **S2** in supporting information). The main differences arise from the viscoelastic behavior that is strongly influenced by the polymer concentration with the level of chain overlapping and entanglements. At low polymer concentration (see $C_p=1$ wt% in **Figure 11**), the phase transition of alginates induced by cooling gives rise to a 10 times increase of the viscosity (here $\eta^* \sim G''$) without gelation as already reported for alginate solutions.

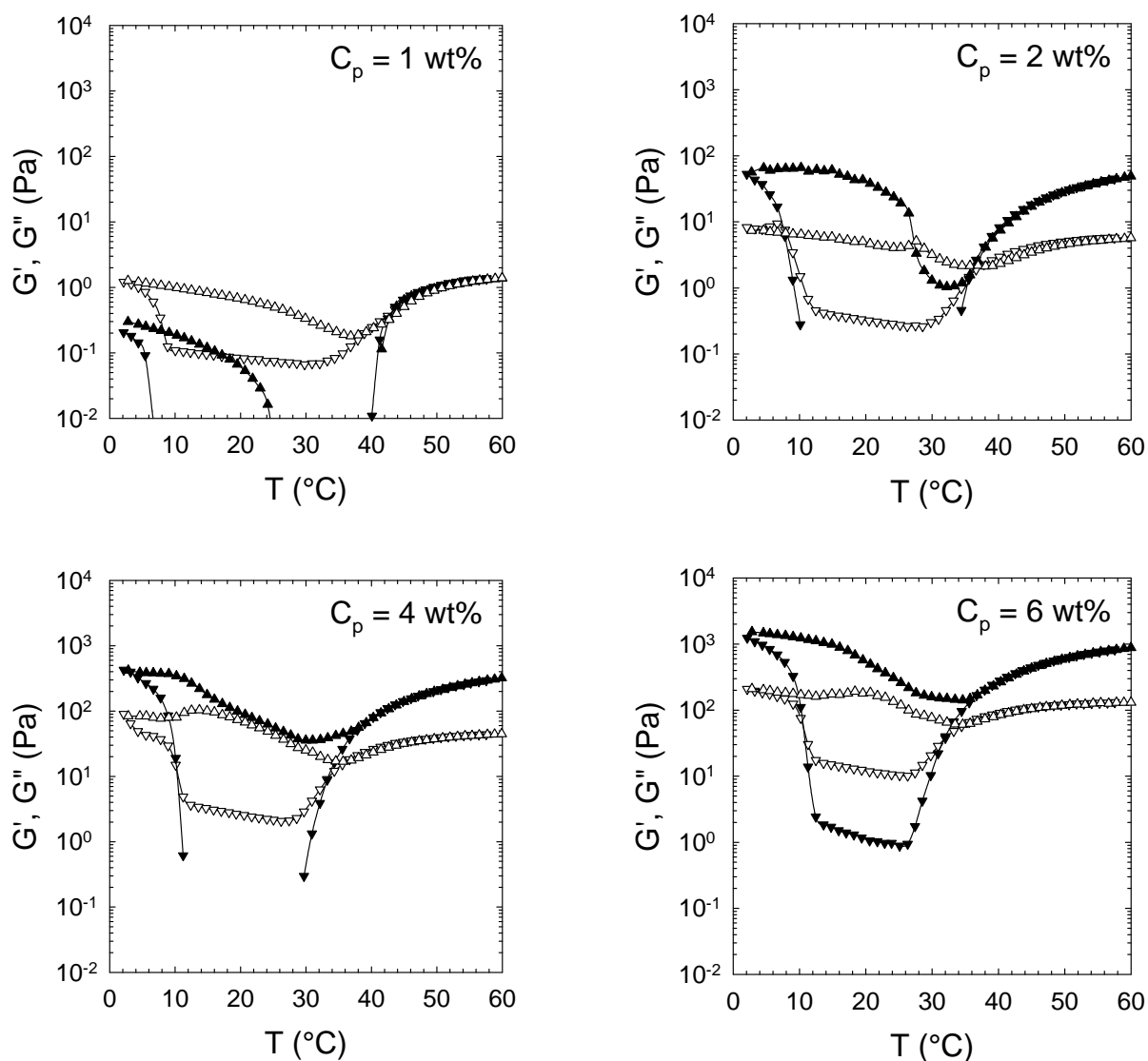


Figure 11. Temperature dependence of dynamic modulus G' (filled symbol) and G'' (hollow symbol) of ALG-g-PNIPAM aqueous solution ($[KCl]=0.6$ m) for different polymer concentrations upon cooling (\blacktriangledown) and heating (\blacktriangle).

In a different way, the association driven by PNIPAM stickers above 30 °C reaches the critical conditions of sol/gel transition at about 42 °C with $G' \cong G''$. For $C_p \geq 2$ wt%, gelation is effective at both low and high temperatures. The main difference is that in the low concentration regime (see $C_p=2$ wt% in **Figure 11** and also $C_p=3$ wt%; see **Table S1** and **S2** in supporting information) there is an intermediate sol state observed during heating taking place between the dissociation of ALG domains and the association of PNIPAM side-chains. Such intermediate state disappears at higher

concentration ($C \geq 4$ wt%), leading to a continuous gel state over the whole range of temperature (see also **Table S2** in supporting information). The main conclusion is that working in KCl 0.6 m, gelation is obtained above a critical concentration $C_p^c = 1-2$ wt % upon cooling and $C_p^c = 1$ wt% upon heating.

Tuning the Gel/Sol/Gel transitions with salts

Influence of salt concentration

As salt influences the thermodynamic behavior of alginates and PNIPAM in aqueous solutions, it is an interesting tool for controlling the transition temperatures and therefore the viscoelastic state of the formulations over the whole range of temperatures. Using KCl as a reference, we can see in **Figure 12** that the rheology of grafted alginates can be finely controlled by tuning the salt concentration.

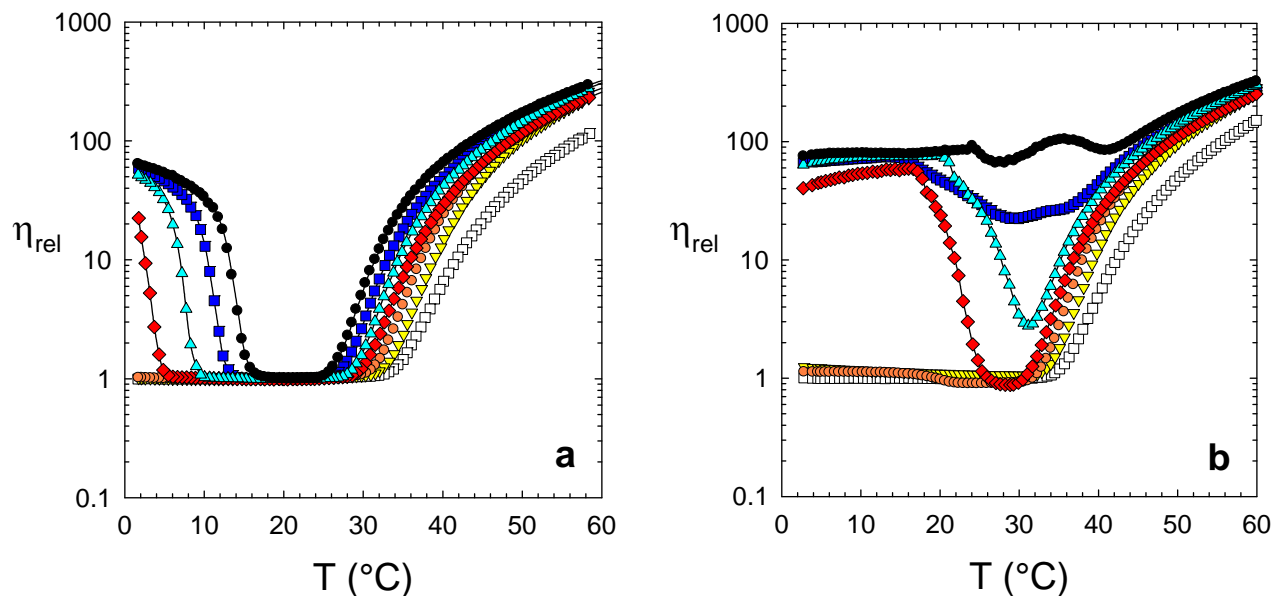


Figure 12. Temperature dependence (**a**: cooling; **b**: heating) of the relative viscosity of ALG-*g*-PNIPAM aqueous solutions ($C_p = 5$ wt%) prepared with KCl at different molar concentrations: 0 (\square); 0.2 (\blacktriangledown); 0.3 (\bullet); 0.4 (\blacklozenge); 0.5 (\blacktriangle); 0.6 (\blacksquare) and 0.7 (\bullet).

During cooling (**Figure 12a**), the properties at low temperature are those already described with the alginate sample, i.e. cold gelation for $[KCl] > 0.3$ m and a shift of the transition towards higher temperatures with increasing salt concentration. Concurrently, the transition temperature of PNIPAM is reduced by about 10 °C by salting out with increasing the KCl concentration from 0 to 0.6 m (see

also **Table S3** in supporting information). As KCl facilitates the two aggregation mechanisms (UCST-type of ALG and LCST-type of PNIPAM), it comes that the temperature range where the solution remains liquid is considerably reduced as the salt concentration increases. If we except the solutions prepared with $[KCl] < 0.4$ m, which undergo a single sol/gel transition at high temperature without any strong hysteresis, the solutions prepared at higher salt concentrations display a large set of behaviors during heating due to the overlay of the two aggregation processes (**Figure 12b**). For instance, with KCl 0.4 m, the solution fully recovers a non associating state only in a very narrow temperature range, around 28 °C. At higher salt concentrations, the copolymer chains never fully dissociate and highly viscous or elastic behaviors are observed in the intermediate temperature range for 0.5 and 0.6-0.7 m, respectively (see **Figure S7** in supporting information).

Influence of the nature of salt

While salt concentration is known to impact the phase transition of polymer in aqueous media, the nature of salt can also have a strong influence that can vary according to the mechanism involved. It is well known for instance that the transition temperature of PNIPAM in aqueous solution follows a recurring trend known as the Hoffmeister series that has been established from the ability of the salts to precipitate certain proteins from aqueous solutions.⁴⁴⁻⁴⁶ This behavior is more pronounced for anions than cations with typical orders: $CO_3^{2-} > F^- > Cl^- > Br^-$ for anions and $K^+ \cong Na^+ > Li^+$ for cations. While the above mentioned anions, referred as kosmotropes, decrease the LCST of PNIPAM (salting-out), some others like I^- and SCN^- , named chaotropes, are known to have the opposite trend (salting-in). As shown in **Figure 13a**, the effect of chloride salts on the sol/gel transition related to PNIPAM side-chains is in good agreement with the literature data with a weaker effect for lithium ($T_{g/s} \cong 36$ °C) compared to sodium and potassium ($T_{g/s} \cong 32$ °C). The impact of salt is totally different for the association of alginate chains as only potassium ions are able to trigger the aggregation

process at low temperature. Even if the salt concentration is quite high for high-M alginate, this case is somehow reminiscent with κ -carrageenan which form gel in aqueous media during cooling much more efficiently and with much higher modulus with KCl compared to NaCl or LiCl. One of the explanations given for carrageenan, which could also be applied to the case of alginates, is that K^+ , being less hydrated than Na^+ and Li^+ and having the smallest hydrated radius despite its larger non-hydrated size, interacts more strongly with sulfonate groups forming even ion pairs or bridges between negative charges.⁴⁷⁻⁴⁹ While comparable effects were also observed with gellan gum which contains carboxylic groups⁵⁰⁻⁵¹, these effects were not really described in the case of alginates even if some associations promoted by K^+ were suggested by Rees et al..⁵²⁻⁵³

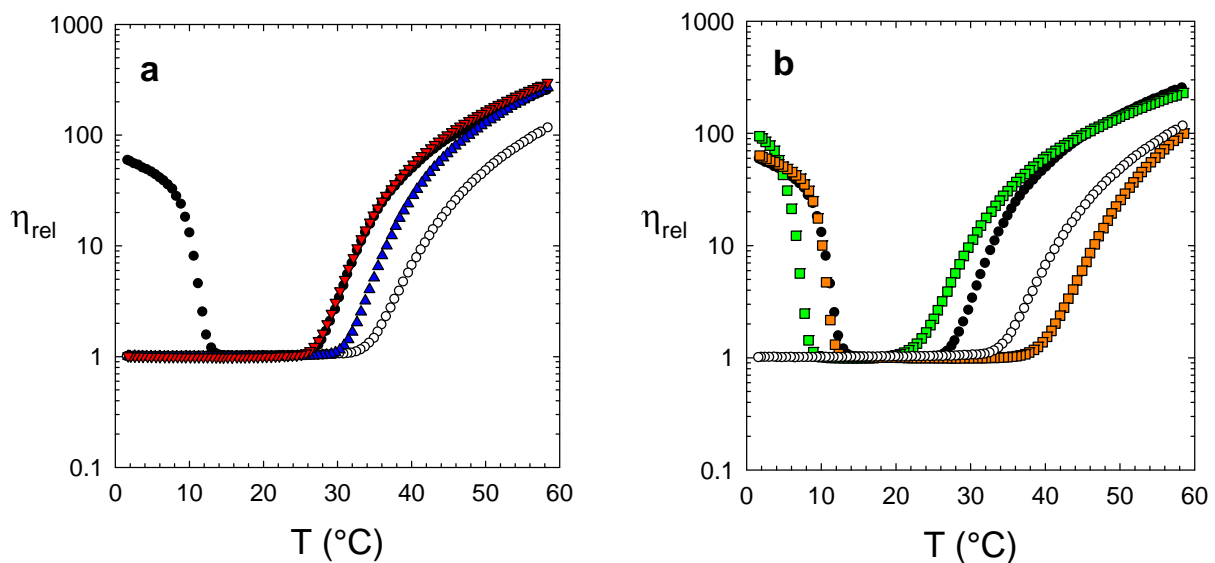


Figure 13. Temperature dependence (during cooling) of the relative viscosity of ALG-g-PNIPAM aqueous solution ($C_p=5$ wt%) in the presence of different (a) chloride salt (salt concentration=0.6 m) LiCl (\blacktriangle), NaCl (\blacktriangledown) and KCl (\bullet); (b) potassium salt ($[K^+]=0.6$ m): K_2CO_3 (\blacksquare), KSCN (\blacksquare) and KCl (\bullet). H_2O (\circ) is used as reference. (cooling rate= 2 $^{\circ}C \cdot min^{-1}$, $f=1$ Hz).

In **Figure 13b**, where the influence of anions was investigated with different potassium salts, we observe an opposite trend with a strong impact of anions on the hot gelation process and a weak influence on the cold one. For a fixed concentration of $[K^+]=0.6$ m, the sol/gel transition of alginate

chains only decreases from 11.5 °C for KCl to 10.2 °C for KSCN and 7.9 °C for K₂CO₃ while the hot gelation process spans over 20 °C: from 28.6 for K₂CO₃ and up to 48.6 °C for KSCN. The relative dissociation temperatures of PNIPAM domains clearly highlighted in **Figure 13b** as a function of salt are in very good agreement with the literature with a salting-in effect of KSCN which increases the transition temperature with respect to water, and salting-out with KCl and K₂CO₃, the later being much more efficient in decreasing the LCST.⁴⁶ From the alginate side, there is almost no difference between KCl and KSCN 0.6 m but lower transition temperatures are obtained for K₂CO₃ 0.3 m, as well as for K₂SO₄ 0.3 m and K₃PO₄ 0.2 m (results not shown here), meaning that multivalent anions interfere negatively on the gelation process of alginates.

General discussion

Using the macromolecular data obtained from SEC for ALG at relatively high ionic strength ($[\eta]=385$ mL/g in $[\text{NaNO}_3]=0.2$ m), the overlap concentration of alginate chains can be roughly estimated at $C^*\cong 1/[\eta]=2.5$ g/L $\cong 0.25$ wt %. Above this concentration, that separates the dilute and the semi-dilute unentangled regimes, the polymer chains start to overlap and increase their contact up to the concentration C_e where entanglements dominate the viscoelastic properties. For neutral polymer or polyelectrolyte solutions studied at high salt concentration, the ratio between these two concentrations is $C_e/C^*\cong 5-10$ but other criterions can be used to define C^* and C_e as the concentrations where the viscosity of the polymer solutions become equal to $\eta^*\cong 2\eta_s$ and $\eta_e\cong 50\eta_s$ respectively, with η_s the viscosity of the solvent.⁵⁴⁻⁵⁵ Moreover, in the semi-dilute unentangled regime ($C^*<C<C_e$), the variation of the specific viscosity ($\eta_{sp} = (\eta - \eta_s)/\eta_s$) versus the concentration follows a power law $\eta_{sp} \propto C^\alpha$ with moderate scaling coefficients ($\alpha=2$ or $5/4$ in Θ -conditions or good solvent, respectively), whereas in the semi-dilute entangled regime ($C>C_e$) these coefficients are much larger ($\alpha=14/3$ or $15/4$ in Θ -conditions or good solvent, respectively). As shown in **Figure**

14, the log-log plots of η_{sp} versus the concentration obtained for both ALG and ALG-g-PNIPAM in KCl 0.6 m at 20 °C upon cooling, i.e. in absence of specific associations, gives relatively high exponents ($\alpha=3.5-3.8$) in agreement with theoretical values expected for entangled macromolecules in good solvent. These scaling relations are determined for $C_p \geq 2$ wt % as the experimental values of η_{sp} obtained at 1 wt% depart from the above relations and the viscosities are clearly below the criterion $\eta_e \cong 50\eta_s$. If the viscosity of ALG and ALG-g-PNIPAM solutions follow the same trend, the main difference between the two samples originates from the concentration of ALG backbone that mainly contributes to entanglements and which is 40 wt% lower in the ALG-g-PNIPAM compared to ALG when prepared at the same total concentration. From this first analysis we can conclude that the entanglement regime is reached for concentrations above 1 wt% and this is typically the concentration regime where a true gel behavior with elastic properties was observed for alginates solutions.

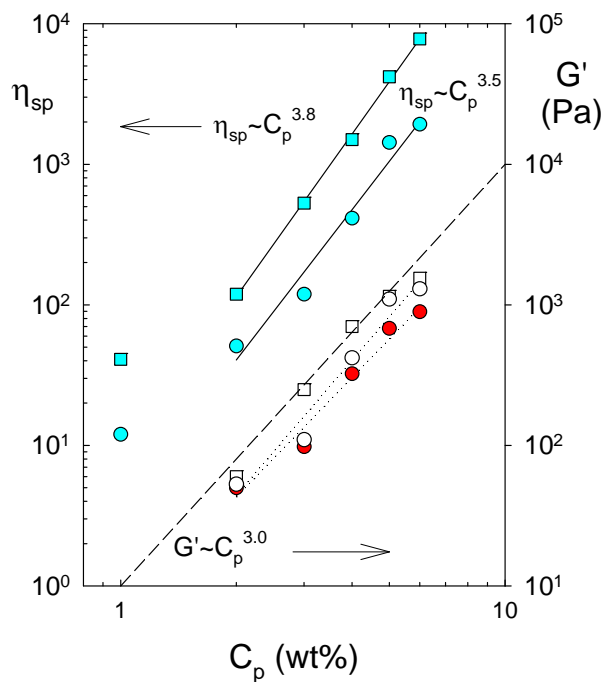


Figure 14

Concentration dependence of the specific viscosity (η_{sp}) for solutions prepared in KCl 0.6 m at 20 °C upon cooling.
 (■ ALG and ● ALG-g-PNIPAM)

Concentration dependence of the elastic modulus (G') of alginate gels prepared in KCl 0.6 m:

- ALG at T = 2 °C
- ALG-g-PNIPAM at T = 2 °C
- ALG-g-PNIPAM at T = 60 °C

For entangled networks made up of linear chains with temporary cross-links, the elastic modulus (G) has contributions from both cross-links (G_c) and entanglements (G_e) at times shorter than the lifetime

of a cross-link (τ) and longer than the Rouse time of an entanglement strand (τ_e). This is typically the situation for ALG and ALG-g-PNIPAM solutions as the physical interactions taking place at low (2 °C) and high (60 °C) temperatures are “strong” enough, *i.e.* the dissociation rate constant is slow at the time scale of the experiment ($t < \tau$). If we plot the elastic modulus versus the polymer concentration (**Figure 14**), all the gels prepared by heating or cooling exhibit very similar concentration dependence with a scaling exponent ($G \propto C_p^\beta$) close to 3. By comparison, similar values were already reported in previous works for cold gelation of ALG sample in K_2CO_3 0.3 m³⁵ ($\beta=2.6$) as well as for nano-hybrid networks where adsorbing silica nano-particles are embedded into a chemical hydrogel of poly(*N,N*-dimethylacrylamide)⁵⁶ ($\beta=2.6$). With reference to the non-associated state, where only the entanglements contribute to the elastic modulus as $G_e \propto C_p^{9/4}$ (assuming good solvent conditions), the strong dependence of the modulus with the concentration in the gel states, $G \propto C_p^{9/4} C_p^{3/4}$, can be considered as a signature of the coupling between physical cross-links and entanglements. If the modulus of the gels formed at low and high temperature in the presence of 0.6 m KCl are very close and have a similar concentration dependence, it should be known that this similarity strongly rely on the composition of the copolymer and that the mechanisms of association are of very different natures.

At high temperature, the general picture that can be given for this graft copolymer is that of the formation of a micellar network where PNIPAM chains self-assemble into micro-domains physically cross-linking the water-soluble matrix; alginate in the present case. The nano-metric aggregates of PNIPAM, that retain only a small fraction of water (30 wt%), have a very low dynamics and these physical cross-links have a very long lifetime which make these physical gels very stable with time. While PNIPAM is well-known as a temperature-responsive polymer, it is also very sensitive to many molecules that can be added into the solution and inorganic salts in particular. The effect of salts on

the LCST of PNIPAM has been widely studied and the key points are that the solubility of PNIPAM chains is mainly affected by anions and that the impact of these anions depends strongly if they are kosmotropic or chaotropic.^{46, 57} In the first case, highly hydrated anions, like SO_4^{2-} , CO_3^{2-} , F^- ..., decrease the solubility of PNIPAM by polarizing the water molecules which form hydrogen bonds around the amide group of NIPAM units and by increasing the surface tension at the water/hydrophobic interface (salting-out). Less hydrated anions also increase the surface tension but they can also bind to macromolecules, like SCN^- , giving rise to salting-in.⁴⁶ From a general point of view we can see that the solubility of PNIPAM is a complex mechanism that involves very different interactions like hydrogen bonds, hydrophobic and even ion binding.

At low temperature, the mechanism of gelation of high-M alginates has hardly been studied but it can be emphasized from our series of experiments that neither the hydrophobic interactions nor the hydrogen bonds are the key interactions of the association process. The gelling mechanism induced at low temperature by the association between mannuronic sequences is mainly induced by the cations and in the series of alkali metals that has been tested Li^+ , Na^+ , K^+ , only potassium is able to drive the association process. We can also mention that larger alkali metals like Cs^+ (not shown here) behave similarly to potassium. The hypothesis is then that chaotropic cations, which are less hydrated than kosmotropic ones (Li^+ and Na^+), can more strongly interact with carboxylic groups of mannuronic units by forming ions pairs or inter-chain bridges as it was reported for the salt-induced gelation of carrageenans.⁴⁷⁻⁴⁹

Although the exact mechanism of cold gelation of high-M alginates remains an open question, the main result of this paper is that a copolymer with schizophrenic thermo-responsive properties, coupling UCST and LCST behaviors, can be readily obtained by grafting PNIPAM chains on a high-M alginate backbone. Since the association process of alginate and PNIPAM are strongly

modified, respectively by cations or by anions, this means that the choice of salt is critical and that one or the two mechanisms can be selectively regulated by choosing the appropriate salt. This selective activation/deactivation of the self-assemblies is exemplified in **Table 4** for three different temperatures, 5, 32 and 38 °C, on the basis of our experimental data and their extrapolation to other salts.

Table 4. Different states of ALG-g-PNIPAM solutions obtained during cooling in the presence of various monovalent salts (0.6 m) at three given temperatures: “- -”: sol state; “—”: gel state.						
	No salt	NaCl	KCl	KSCN	NaSCN ^a	KI ^a
5 °C	- -	- -	—	—	- -	—
32 °C	- -	—	—	- -	- -	- -
38 °C	—	—	—	- -	- -	—
^a Predicted values						

CONCLUSIONS

The aim of this work was to take advantage of the specific behavior of high-M alginates, which show reversible gelation upon cooling in certain salt conditions, to design an original copolymer with schizophrenic gelling properties. This has been demonstrated by grafting PNIPAM side-chains onto the alginate backbone. Based on this grafted architecture, we have shown that the two opposite association processes, driven by the LCST of PNIPAM side-chains and the UCST of mannuronic sequences, can be triggered independently from one another without interference between the two. Indeed, since the aggregation of PNIPAM which couples the hydrophobic interactions and the hydrogen bonds can be reinforced by anions, and the association of mannuronic sequences is clearly controlled by cation binding, the schizophrenic behavior of the graft copolymer can then be selectively controlled by the choice of the salt. In addition, since the association/dissociation behavior of high M alginates exhibits a high hysteresis, greater than 20 °C, their derivatives offer a great versatility not only in terms of salt (nature and concentration) but also in terms of preparation since

different states, sol or gel, can be obtained at the same temperature. While the exact mechanism of the aggregation process of high-M alginates remains under investigation, this polysaccharide forms a very interesting platform for developing responsive systems for biomedical applications. Such work is actually developed in our group in order to investigate how the nature, the size and/or the number of LCST side-chains can modify the viscoelastic properties under small and large deformations.

ASSOCIATED CONTENT

Supporting information

Detailed synthesis of ALG-g-PNIPAM, sol/gel transition of ALG as a function of polymer concentration and added SDS or urea, viscoelastic and calorimetric analysis of ALG-g-PNIPAM as a function of KCl concentration, transition temperatures of ALG-g-PNIPAM solutions for different salts, salt concentrations and polymer concentrations.

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