

Kr environment in feldspathic glass and melt: A high pressure, high temperature X-ray absorption study

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Jessica Hudspeth, Angelika D Rosa, Olivier Mathon, Tetsuo Irifune

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1	Kr environment in feldspathic glass and melt: a high pressure, high
2	temperature X-ray absorption study
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4	Céline Crépisson ^{a*} , Chrystèle Sanloup ^a , Laurent Cormier ^b , Marc Blanchard ^c , Jessica
5	Hudspeth ^a , Angelika D. Rosa ^d , Olivier Mathon ^d , Tetsuo Irifune ^e
6	
7	^a Sorbonne Université, CNRS, UMR 7193 - Institut des Sciences de la Terre de Paris (ISTeP),
8	4 place Jussieu, 75005, Paris, France
9	
10	^b Sorbonne Université, CNRS UMR 7590, MNHM, IRD, Institut de Minéralogie de Physique
11	des Matériaux et de Cosmochimie (IMPMC), 75005 Paris, France
12	
13	^c Géosciences Environnement Toulouse (GET), Université de Toulouse, CNRS, IRD, UPS, 14
14	avenue Edouard Belin, 31400 Toulouse, France
15	
16	^d European Synchrotron Radiation Facility, ESRF, 71 Avenue des Martyrs, 38000 Grenoble, 6
17	France.
18	
19	^e Ehime University, Geodynamic Research Center, Matsuyama, Japan
20	
21	*Corresponding author. Address : Céline Crépisson, Institut des Sciences de la Terre de Paris
22	(ISTeP), Sorbonne Université, Case 110, 4 place Jussieu, 75005 Paris, France. Tel.: +33 1 44
23	27 60 64. Email : <u>celine.crepisson@upmc.fr</u>
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26 Abstract

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28 Noble gases are used to trace the evolution of the Earth's atmosphere and magmatic 29 processes. However little is known about their solubility mechanisms in silicate melts or 30 glasses, whereas it could be fundamental to interpret these data. We report here the first *in situ* 31 X-ray absorption spectroscopy data at the Kr K-edge on a feldspathic Kr-bearing glass and 32 melt (0.5 wt % Kr) up to 2.7 GPa and 1086°C. Kr fitting of the EXAFS signal shows that Kr 33 is surrounded by oxygen atoms. Two Kr-O distances are identified: 2.49±0.1 Å and 3.32±0.1 34 Å. Results can be interpreted as two Kr populations. Kr surrounded by O atoms at 3.32±0.1 Å 35 suggests a filling of the large cavities available in the silicate network, in agreement with previous studies. On the contrary, Kr surrounded by O atoms at 2.49±0.1 Å, and the observed 36 37 + 0.6 eV shift of the edge position at high pressure, suggest Kr bonding to O atoms and Kr oxidation inside cages formed by the largest aluminosilicate rings (i.e. ~ 12-membered-rings). 38 39 Present results show that heavy noble gases incorporation in silicate melts can no longer be 40 considered as a passive filling of the voids.

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42 Keywords

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44 Krypton, XAS, silicate glass, silicate melt, high pressure

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46 <u>**1. Introduction**</u>

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48 Due to their supposed chemical inertia, high volatility, and high dilution, noble gases have 49 been intensively studied by geochemists to understand the formation of the Earth's 50 atmosphere, and to study mantle dynamics. None of Kr stable isotopes (^{78,80,82,83,84,86}Kr) is 51 purely radiogenic, and Pu and U fission yields are very low for Kr. Thus Kr cannot be used to 52 constrain chronology of the atmosphere evolution, and as such has been less studied than He, 53 Ne, Ar, and Xe. However, the small amount of radiogenic Kr makes it suitable to trace air 54 component in fluid inclusions (Böhlke and Irwin, 1992). Earth atmospheric Kr isotope 55 composition, unchanged since at least 3.3 Ga (Avice et al., 2017), falls in-between solar and 56 chondritic compositions (Pepin, 1991). Kr isotopic ratios in CO₂ magmatic well gases that probed the continental lithosphere, showed a significant enrichment in ⁸⁴Kr and ⁸⁶Kr 57 58 compared to the atmosphere (Holland et al., 2009). The discovery of Kr isotopic fractionation 59 in these well gases, taken as a proxy of deep mantle gases, was interpreted as the signature of 60 a primordial Kr chondritic component (Holland et al., 2009). Nevertheless Holland et al. 61 (2009) argue that early degassing of an accreted primordial chondritic composition, followed by mass fractionation during atmosphere loss, fails to explain the composition of the modern 62 63 atmosphere. Instead, late accretion of cometary material is assumed to explain Kr isotope 64 signature of the atmosphere. Although very few samples have been analyzed for all Kr 65 isotopes, mantle-derived basalts (MORBs, OIBs) have an atmospheric Kr isotopic 66 composition (Moreira et al., 1998; Trieloff et al., 2000; Graham, 2002). This homogeneous 67 composition could be due to atmospheric Kr recycling at subduction zones (Holland and Ballentine, 2006). 68

To explain highly precise Kr isotopic data, geochemists need an understanding of Kr
behavior in silicate melts, including Kr solubility value and mechanism, as well as Kr degree
of compatibility.

Fractionated fission Kr has also been observed in the Okhlo natural nuclear reactor. In nuclear plants, ⁸⁵Kr is a common fission product. Nuclear waste confinement by vitrification in SiO₂-rich glasses has been used for many years (Ojovan and Lee, 2011). Accumulation of Kr leads to the formation of nanobubbles where pressure may reach 2.6 GPa (Martin et al., 2015) under high temperature generated by irradiation. Understanding of Kr behavior in
silicate glasses is thus crucial for long nuclear waste storage.

The majority of studies dedicated to noble gases solubility in melts are based on 78 79 analyses of recovered glassy samples, and did not directly investigate the noble gas structural 80 environment in the silicate network, as noble gases can escape from the silicate structure at 81 ambient conditions. These studies showed that noble gas solubility decreases with increasing 82 radius (from He to Xe) (Caroll and Stolper, 1993, Lux, 1987, Shibata et al., 1998), and that 83 this solubility strongly depends on the chemical composition of the melt. Noble gas solubility 84 was first related to ionic porosity, i.e. the absolute volume of voids (Caroll and Stolper, 1993), 85 whereas Shibata et al. (1998) showed that it was better inversely correlated to the number of 86 non-bridging oxygen per silicon, even if the presence of Al may modify this trend (Bouhifd 87 and Jephcoat, 2006). Based on these findings and noble gas relative chemical inertia, noble 88 gases were supposed to passively fill the voids of the silicate network.

In situ structural data on the environment of heavy noble gases in silicate melts and 89 90 glasses are scarce. Wulf et al. (1999) performed an X-ray absorption study at the Kr K-edge in 91 a SiO₂ glass (1.97 wt % Kr). They identified a Kr-O distance of 3.45±0.1 Å, interpreted as 92 corresponding to a local rearrangement of the silicate network into a local clathrasil-type 93 environment around the Kr atom. The supposed chemical inertia of Kr and Xe, the most 94 reactive of the noble gases, has been questioned throughout the past 15 years, with the 95 synthesis of numerous Kr and Xe compounds (see Grochala, 2007 for a review). Bonding of 96 Xe to O in compressed haplogranitic magmas had also been evidenced by in situ X-ray 97 diffraction (Leroy et al. 2018).

In the present study, we perform *in situ* X-ray absorption spectroscopy (XAS) at the
Kr K-edge, to specifically access Kr structural environment in a feldspathic (sanidine) glass.

100 Starting $(Na,K)AlSi_3O_8$ glass composition was chosen as representative of the 101 chemistry of a major mineral of the continental crust, and proxy of silica-rich melts that can 102 be generated at pressure reaching 2 GPa in orogenic contexts (Jimenez-Munt et al., 2008), and 103 in subduction contexts from melting of hydrous basalt (Prouteau et al., 2001), or silica-rich 104 sediments (Turner et al., 2012). Experiments were conducted at pressures (P) and 105 temperatures (T) up to 2.7 GPa and 1086°C, relevant for both continental crust melting and 106 subduction zone contexts.

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108 **2.Material and Methods**

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110 **2.1 Sample: synthesis and characterization**

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112 The starting glass of Kr/Xe-bearing sanidine was synthesized in a piston cylinder press 113 at 3.5 GPa and 1400°C for 24 hours before quenching. For this purpose, Xe and Kr were 114 loaded as a 1:1 gaseous mix along with sanidine powder in a Pt capsule (Boettcher et al., 115 1989). Sanidine is a high-temperature feldspar, of atomic composition (K_xNa_{1-x})AlSi₃O₈ with 116 x higher than 0.32 (Bendel and Schmidt, 2008). In our experiments, sanidine powder was obtained by grinding a natural sanidine mineral (K_{0.52}Na_{0.43}Al_{1.05}Si_{2.96}O₈, formula unit 117 118 retrieved from electron microprobe analysis) from the Sorbonne Université mineralogical 119 collection.

120 Starting glass and recovered sample were mounted in epoxy, polished and C-coated. 121 They were observed by Scanning Electron Microscope (SEM) with a Zeiss Supra 55 122 (equipped with a field emission gun). Detection of Xe and Kr was optimized with an 123 acceleration voltage of 20 keV and a working distance of 6.6 mm in backscatter mode. Electron microprobe analyses (EMPA) were done on a CAMECA SX5, at CAMPARIS centre
at Sorbonne Université. Operating conditions were 15 kV accelerating voltage, 10 nA beam
current, and a focused beam for point analysis. We use the method from Montana et al. (1993)
for Kr, which was calibrated by measuring the counts on neighboring elements (i.e. Rb). Xe
was calibrated using a known standard analyzed by Particle-Induced X-ray Emission (data
published in Leroy et al., 2018).

130 The synthesized starting glass is chemically homogeneous (based on SEM and EMP 131 analyses) with presence of Kr/Xe gas bubbles indicating that saturation was reached. Contents 132 of 2.2 wt % Xe (0.36 at %) and 0.5 wt % Kr (0.13 at %) were measured by electron 133 microprobe analyses from an average of seven points of analysis in areas free of micrometric 134 gas bubbles. Average composition in wt % of the starting glass is given in Table 1. Recovered 135 sample was guenched from 1040°C and 2.7 GPa, where it reequilibrated as partial melt and feldspar crystals (Figure 1), following cooling from fully molten state (1415°C and 3 GPa) 136 137 which was reached with the initial purpose to collect XAS data at the Xe-edge. Kr and Xe are 138 only observed in the glass, with striking alignments of Kr/Xe gas nanobubbles (Figure 1). 139 Average analyses of the recovered sample are given in Table 1, although degassing upon 140 quenching prevents Xe and Kr contents to be determined by EMP analysis.

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142 2.2 Experimental set-up

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In-situ X-ray absorption spectroscopy (XAS) was performed on beamline BM23 at the ESRF, dedicated to general purpose XAS (Mathon et al., 2015). High pressure and temperature conditions up to 2.7 GPa and 1086°C (Table 2) were generated in a Paris Edinburgh Press (Besson et al., 1992). We used a cell assembly developed by Kono et al. (2014) and adapted from Yamada et al. (2011). More details on the cell-assembly and experimental set-up can be found in Rosa et al. (2016). The sample (0.75 mm diameter and 150 1.5 mm height) was encapsulated in a synthetic nanopolycrystalline diamond (NPD) cylinder 151 caped with Pt-Rh caps. The use of NPD prevents glitches from a single or polycrystalline 152 diamond capsule, which deteriorate the XAS signal. Temperature was determined using 153 calibration from Kono et al. (2014), and pressure using unit cell volume of hBN from the cell 154 assembly (based on X-ray diffraction measurements) and equation of state from Le Godec et 155 al. (2000). Error bars are $\pm 50^{\circ}$ C for temperature and ± 0.2 GPa for pressure.

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- 157 2.3 XRD and XAS measurements
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Angle-resolved X-ray diffraction data were recorded on a MAR165 CCD at 15 keV.
Image plates were integrated using Fit 2D software (Hammersley, 1997). Cell parameters of
hBN were determined using POWDER CELL software (Kraus and Nolze, 1996).

162 XAS transmission measurements were performed first at the Kr K-edge (14326 eV) up to 2.7 163 GPa, upon increase of temperature up to 1086°C, and upon decrease of temperature down to 164 505°C, before quenching to room temperature (Table 2). Xe K-edge spectra were recorded 165 following measurement at the Kr K-edge upon increasing temperature up to 1415°C and 3 166 GPa, and upon cooling to 1040°C at 2.7 GPa followed by quenching. Xe K-edge spectra are 167 however not of sufficient quality to allow fitting, given the very small absorption jump (0.09 168 compared to 0.25 at the Kr K-edge), and are thus not shown here. Energy calibration of the 169 edge position was made by frequent analyses of Kr gaseous standards. The beam size on the 170 sample was 300 x 80 µm². Transmitted X-ray intensity was recorded by ionization chambers 171 filled with gaseous Ar-He mixed gas. Signal was collected from 14244 eV to 14971 eV at the 172 Kr K-edge (451 points analysis). Counting times, number of scans, and P/T conditions are

given in Table 2. Energy steps were adapted as a function of the edge position, each scantakes about 1 hour.

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176 2.4 Processing of XAS data

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The edge of the standard was fixed to 14326 eV, and the edge of each sample was recalibrated using the averaged edge value of two gaseous Kr standards taken before and after the sample measurement. For Kr standard, no significant shift of the Kr K-edge was noted (0.4 eV at the largest). After correction of edge position, data were cut at 14800 eV for better normalization.

183 Data processing was performed using ATHENA and ARTEMIS packages, based on the 184 IFEFFIT programs (Ravel and Newville, 2005). Each scan was corrected with a pre-edge and 185 a post-edge baseline. Position of the edge was determined by the position of the first derivative of the absorption. The intensity of the jump at the edge was then normalized to one. 186 187 After merging of the scans collected at a given P/T condition, data were corrected using a 188 background function. We used an automated background subtraction method (AUTOBKG) 189 developed by Newville (Newville, 2001) and implemented in the ATHENA software. This correction minimizes the features below the first atom-atom distance in the R-space. R_{bkg} was 190 191 fixed to 1.2 Å.

To fit the data at the Kr K-edge, we used a k-range of ~ 2.2-6.5 Å⁻¹ (Table 3), due to low signal to noise ratio at higher k values. A small k-range ensures that all the different contributions to the Fourier transform are not artifacts but really arise from neighboring atoms. Real part of backward Fourier transform was exported to ARTEMIS software to be fitted using R-range ~ 1.1-3.8 Å (Table 3). Fitting made on real part of backward Fourier transform ensures that all XANES and noise contributions are removed from the EXAFS

198 signal. Fit parameters include: amplitude reduction factor (S0), average distances of the 199 neighboring atom to the central Kr atom, Debye Waller factor (σ^2), and difference of edge 200 position compared to standard value (ΔE_0). ΔE_0 values remain smaller than 7 eV, attesting 201 consistency of the fits. It was necessary to add a third order anharmonic cumulant expansion 202 fitting parameter (C3) for high pressure or high temperature measurements (Stern et al, 1991; 203 Farges et al., 1994). C3 is a measure of the asymmetry of the signal disorder, which arises in 204 glasses from configurational disorder associated to thermal vibration at high pressure and/or high temperature conditions. The absolute value of C3 remains smaller than 0.009 $Å^3$, similar 205 206 to values found by Cochain et al. (2015) on silicate glasses and melts at high pressure and temperature conditions. As no Kr standard was analyzed in the course of the experiment 207 208 besides Kr gas, it is not possible to determine coordination numbers (CN). We therefore fixed 209 S0 at a common value of 0.79, and fixed the value of CN. In total, seven parameters must be fitted (first interatomic distance, σ^2 and C3 for the two contributions and ΔE_0) whereas N, the 210 211 number of variables to fit, cannot exceed (Teo, 1986):

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$$N = \frac{2\Delta R \Delta k}{\pi}$$

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215 In our case $\Delta k \approx 4.3$ Å⁻¹ and $\Delta R \approx 2.8$ Å. Therefore maximum number of parameters *N*, 216 which can be fitted is ~ 7.7, indicating that we can theoretically fit our data as exposed above. 217

(1)

218 **<u>3.Results and Discussion</u>**

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220 **3.1. XAS data at the Kr K-edge**

222 XAS data were recorded at ambient conditions, at high pressure and high temperature, and 223 after quenching (Table 2). Potassic sanidine (KAlSi₃O₈) is expected to melt around 1500°C at 224 2.6-2.7 GPa (Boettcher et al., 1984). Reports on the KAlSi₃O₈-NaAlSi₃O₈ join under 225 anhydrous conditions at 1 atm showed that melting temperature of KAlSi₃O₈ drops by 100°C 226 for our starting composition (Na_{0.43}K_{0.52}Al_{1.05}Si_{2.96}O₈, atomic formula retrieved from EMPA) 227 (Schairer, 1950). Glass transition temperature (Tg) of felspathic glass at 1 atm was found in-228 between 907-948°C based on viscosity measurements (Petermann et al., 2008; Richet and 229 Bottinga, 1984). Moreover, viscosity is shown to decrease from 0 GPa to 2.5 GPa for both 230 sanidine melt (White and Montana, 1990) and albite melt (Kushiro, 1978). Thus Tg is 231 expected to be even lower than 907-948°C at our pressure conditions. Therefore our 232 measurements at 1086°C (and potentially at 920°C) were recorded above Tg and before 233 recrystallization, i.e. on the supercooled liquid, behaving like a melt. Well-defined oscillations 234 are visible in the X-ray absorption signal, and clearly differ from the signature of Kr gas and 235 Kr liquid under pressure (Di Cicco et al., 1996) (Figures 2 and 3).

236 Rapid damping of the EXAFS signal indicates disorder beyond the first shell. We compare in Figure 3 our data with the k²-weighted EXAFS function recorded at 27°C on UO₂ 237 238 (8 at % Kr) previously implanted with Kr ions (Martin et al., 2015). Oscillations observed after 4 Å⁻¹ clearly differ from our data sets. The magnitude of the Fourier transform (k-range 239 240 used is displayed in Table 3) clearly shows two peaks at all measurement conditions (Figure 4), indicating that two distances contribute to the signal, with the main distance at ~ 1.9 Å, 241 242 while Martin et al. (2015) reports a main distance at 3.44 Å. Kr environment in sanidine glass is therefore different from that observed by Martin et al. (2015) in UO₂, i.e. Kr nanoclusters. 243 244 There is no noticeable change in the room temperature glass with pressure: data at ambient 245 pressure and after quenching at 1.4 GPa being similar (Figures 2, 3 and 4). The relative 246 intensity of the two peaks however varies with temperature, the magnitude of the Fourier

transform (Figure 4) being different at ambient temperature, at 505°C and 706°C, and
especially in the melt at 920°C and 1086°C.

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250 **3.2 Resolving Kr environment in sanidine glass and melt**

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252 Variations of fitted Kr-O distances over our experimental P/T range are within error (i.e. ±0.1 Å). We thus evaluate averaged Kr-O distances for the whole P/T range, two Kr-O 253 distances are identified at 2.49±0.1 Å and 3.32±0.1 Å (Table 3, Fig. 5). No other neighboring 254 255 atom than oxygen could fit the observed XAS signal, although we considered other atomic 256 pairs such as Kr-Kr, Kr-Si, Kr-Na, or Kr-K. Identified distances confirm the absence of Kr 257 neighbors. Indeed Kr-Kr distance in solid cubic Kr is 3.68 Å at 2 GPa and room temperature 258 (Di Cicco et al., 1996). We can interpret these results as two distinct populations of Kr located 259 in two different sites of the silicate network. Results clearly differ from Wulf et al. (1999), 260 who only reported a Kr-O distance at 3.45 Å, as shown in Figure 5.

261 At ambient condition, there is no non-bridging oxygen in sanidine glass (KAlSi₃O₈), 262 and Al coordination number is four for all Al atoms (Thompson et al., 2011). Study of an 263 albite glass (NaAlSi₃O₈) quenched from the melt at 2.6 GPa showed that less than 3% of Al 264 atoms present a coordination number of five, whereas all remaining Al atoms have a 265 coordination number of 4 (Gaudio et al., 2015). Sanidine glass is thus fully polymerized, with all Na^+/K^+ compensating the charge deficit created by the presence of Al^{3+} in tetrahedral 266 position. Molecular dynamic simulations showed that in soda aluminosilicate glass, for 267 268 Na/Si=1, distribution of the aluminosilicate rings at ambient pressure varies from 2 to 13 269 membered-rings, with 6 or 7 membered-rings being the most present ring population (Xiang 270 et al., 2013). Similarly, at ambient pressure, a ring size distribution from 2 to 12-membered 271 rings is predicted for silica glass with 6 to 7 membered-rings being the dominant ring 272 population (Kohara et al. 2011; Guerette et al., 2015). To our knowledge, no pore size

distribution data is available for our glass composition. Based on the similarity of ring size 273 274 distribution between aluminosilicate and silica glass, we assume that size of cages within aluminosilicate rings range from 0.9 Å to 5.9 Å in diameter, like in silica glass (Guerette et 275 al., 2015). This size distribution indicates the presence of 5-5.9 Å-diameter voids, inside the 276 largest (i.e. ~ 12-membered) aluminosilicate rings. The Kr-O distance of 2.49±0.1 Å is 277 278 therefore compatible with a Kr atom located inside the largest observed rings of the silicate 279 network. This Kr-O distance falls in-between the sum of covalent radii of Kr and O (1.16 Å and 0.66 Å respectively (Cordero et al., 2008)) and the sum of ionic radii of Kr^0 and O^{2-} 280 (respectively 1.78 Å for a coordination number of 6 (Zhang et al. 1995) and 1.4 Å (Shannon, 281 1976)). This result indicates that Kr is interacting closely with the neighboring oxygen atoms, 282 283 suggesting an oxidation of Kr. Under pressure, the edge position shifts by + 0.6 eV (Figure 6). 284 Shift of the edge can be related to change in the oxidation state as observed for Cu, Co, Fe, 285 Mn (e.g. Klysubun et al., 2015). This observation further supports a change in Kr oxidation 286 due to Kr-O bonding. Kr is in the oxidation state II in most of Kr compounds synthesized, such as KrF_2 , KrF^+ or $Kr_2F_3^+$ (Grosse et al., 1963, Burbank et al., 1972, Lehmann et al., 2001). 287 288 Kr oxidation number is expected to increase compared to gaseous Kr, if Kr is surrounded by 289 several O atoms.

The Kr-O distance of 3.32±0.1 Å cannot correspond to the presence of Kr atoms inside 290 291 aluminosilicate rings, as the size distribution is limited to 5.9 Å in diameter. This larger Kr-O distance is similar to the 3.45 ± 0.1 Å distance reported in SiO₂ glass at ambient pressure and 5 292 293 K (Wulf et al., 1999), and close to predictions from first principle molecular dynamics. Zhang 294 et al. (2009) indeed found Kr-O = 3.5 Å in liquid silica, with Kr coordination number varying 295 from 10 to 15 depending on pressure. Similar result was found by Guillot and Sator (2012) using molecular dynamics with Kr-O = 3.45 Å in silica melt and 3.31 Å in rhyolite melt. This 296 large distance can be explained by the formation of large cavities in the melt where noble 297

298 gases are accommodated, in agreement with a passive filling of the void space, or with a 299 clathrasil-like environment (Wulf et al., 1999). Contribution of Kr fluid to this longer distance 300 cannot be ruled out if the few Kr bubbles observed on starting and quenched samples are not 301 fully homogenized with the melt at high pressure and temperature.

302 Disappearance of the larger Kr-O distance at higher temperatures (Figure 4), indicates 303 that smaller Kr-O distance is more favorable, possibly with the increase of Kr reactivity. 304 Moreover, the Debye Waller factor associated with the smallest Kr-O distance (σ^2 1), is twice 305 larger at room temperature than at high temperature (Table 3), suggesting that ordering of the 306 smaller Kr-O environment is higher. Measurements at high pressures and temperatures are all 307 above Kr melting point, whereas measurement after quenching at 1.4 GPa is below Kr 308 melting point (Ferreira and Lobo, 2008). Kr reactivity therefore seems enhanced above Kr 309 melting curve, suggesting that it is pressure-induced, i.e. it favors smaller volume, as observed 310 for Xe incorporation in olivine (Sanloup et al., 2011). Besides, the larger Kr-O distance 311 vanishes in the melt, i.e. at 920°C and 1086°C (Figure 4), indicating structural changes 312 between the melt and the glass. A complementary explanation could be a contribution of Kr 313 fluid to the longer Kr-O distance, which would diminish with temperature due to the increase 314 of Kr solubility in the melt.

Current knowledge of glass and silicate melt structures therefore supports the hypothesis of two Kr populations, with one population of Kr, oxidized and possibly located in the largest aluminosilicate rings, and one population filling the large cavities available in the silicate network.

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³²⁰ **4.Conclusion**

322 The present work provides evidence for the reactivity of Kr with silicate that adds up to the 323 previously reported evidences of Kr reactivity (Grochala, 2007). Kr oxidizes in feldspathic 324 glass and melt at high P/T conditions as shown by creation of short Kr-O bonds (2.49±0.1 Å) 325 and shift of the edge position. Kr behavior in compressed magmas thus appears to be 326 intermediate between that of an inert noble gas (He, Ne, Ar) and that of the heavier Xe. 327 Indeed Xe is shown to fully oxidize in compressed haplogranitic melts with Xe in the most 328 abundant 6-membered rings containing 12 nearest O, Xe-O distance of 2.05±0.05 Å is 329 similar to distance reported for covalent Xe-O bond in crystalline silicates (Probert, 2010; 330 Kalinowski et al., 2014; Crépisson et al., 2018). Both results question the supposed passive 331 filling of the void spaces by heavy noble gases (Caroll and Stolper, 1993; Shibata et al., 332 1998), and the assumption that all noble gases can be considered as zero-charge cation (e.g. 333 Brooker et al., 2003).

Creation of short Kr-O bonds in sanidine melt at depth, supports the role of Kr recycling at subduction zones, proposed by Holland and Ballentine (2006), as Kr will efficiently dissolve in the melt. Hydrated melts will now need to be studied to test the influence of water, which is thought to dramatically increase noble gases solubility (Nuccio and Paonita, 2000).

338 The Earth's atmosphere is depleted in heavy Kr isotopes compared to chondrites (Holland et 339 al., 2009). Observation of short Kr-O bonds in silicate melt, implying Kr reactivity, suggests 340 potential isotopic fractionation during partial-melting and melt percolation, as shown for 341 instance for Cu (Huang et al., 2017), and / or during fluid / melt interactions either upon fluid 342 injection at depth in subduction zones or degassing during magma ascent. This phenomenon 343 could have contributed to enrichment of well gases from the continental lithosphere in heavy 344 Kr isotopes, while most of the samples from the Earth's mantle (OIBs and MORBs) have an 345 atmospheric Kr isotopic composition due to intense recycling.

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Table 1: Average composition of the starting glass (7 data points) and recovered sample (20
data points both in the glassy and crystalline parts) in oxide wt %, in brackets: standard
deviation.

	Na ₂ O	SiO ₂	Al ₂ O ₃	K ₂ O	CaO	Xe	Kr	Total
Starting material	4.80 (0.2)	63.6 (0.40)	19.2 (0.14)	8.70 (0.24)	0.78 (0.05)	2.18 (0.04)	0.52 (0.05)	99.7 (0.47)
Recovered sample	2.9 (0.6)	65 (1)	19.8 (0.6)	11(1)	0.3 (0.2)	[0-1.3]*	[0-0.5]*	100(1)
564 * du	e to degassi	ng upon que	enching Kr a	nd Xe coni	ents canno	ot be prope	erly evaluat	ted by
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- 582 Table 2: *P/T* conditions and state of the sample for each XAS measurement. *n.s* = number of
- 583 scans, count. time = counting time per point analysis (451 point analysis).

Kr K-edge							
P (GPa)	T (°C)	state	n.s	count . time			
0	20	glass	7	2s			
2.6	920	melt	4	2s			
2.7	1086	melt	5	28			
2	706	glass	5	2s			
2	505	glass	5	2s			
1.4	20	quench	4	2s			

600 Table 3: Pressure-Temperature conditions and EXAFS-derived structural parameters. 601 Conservative error bars are ± 0.05 eV for ΔE_0 , ± 0.1 Å for Kr-O distances and ± 0.05 Å² for 602 σ^2 .

P (GPa)	Τ (°C)	k-range (Å ⁻¹)	R-range (Å)	$\Delta E_0 (eV)$
0	20	2.2 - 6.5	1.1 - 3.5	3.32
2.6	920	2.3 - 6.5	1.2 - 4.0	6.09
2.7	1086	2.3 - 6.5	1.1 - 3.8	7.00
2	706	2.3 - 6.5	1.1 - 3.5	6.45
2	505	2.3 - 6.5	1.1 - 3.8	5.84
1.4	20	2.2 - 6.5	1.1 - 4.0	3.50
P (GPa)	Τ (°C)	Kr-O1 (Å)	$\sigma^2 1 (\text{\AA}^2)$	$\mathbf{C3}_{1}(\text{\AA}^{3})$
0	20	2.42	0.025	-
2.6	920	2.45	0.009	0.0021
2.7	1086	2.51	0.014	0.0078
2	706	2.53	0.009	0.0085
2	505	2.53	0.012	0.0076
1.4	20	2.49	0.021	0.0035
P (GPa)	T (°C)	Kr-O2 (Å)	$\sigma^2 2 (\mathring{A}^2)$	$\mathbf{C3}_{2}(\text{\AA}^{3})$
0	20	3.38	0.048	-
2.6	920	3.27	0.040	-0.0090
2.7	1086	3.37	0.042	0.0090

	2	706	3.34	0.064	-0.0034
	2	505	3.29	0.045	-0.0090
	1.4	20	3.30	0.049	-0.0057
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- 623 Figure 1: SEM images in backscatter mode of recovered sample after XAS measurements.
- 624 Feldspars recrystallized upon cooling to 1040°C at 2.7 GPa, coexisting with partial melt
- 625 recovered as a glass upon quenching. Kr/Xe-nanobubbles (bright dots) are visible.



- 641 Figure 2: Normalized X-ray absorption spectra for sanidine glass and supercooled liquid,
- 642 gaseous Kr at ambient conditions, and liquid Kr at 0.75 GPa (Di Cicco et al., 1996).



Figure 3: k^2 -weighted EXAFS function for sanidine glass and supercooled liquid (color coding: same as Figure 2), and for UO₂ at room conditions (Martin et al., 2015).



Figure 4: Magnitude of the Fourier transform of the k^2 -weighted Kr EXAFS function without phase correction (same samples as Figure 3). The peak observed at ~1.9 Å corresponds to Kr-O = 2.49 Å, and the peak at ~2.9 Å corresponds to Kr-O = 3.32 Å. A general decrease of the amplitude is observed with increasing temperature (as noticed by Stern et al., 1991).



- 684 Figure 5: Real-part of backward Fourier transform (in dashed lines) and fit to the data (plain
- 685 black line) for sanidine glass and supercooled liquid, and Kr-bearing SiO₂ glass (Wulf et al.,
- *1999)*.



698 Figure 6: Shift of the Kr K-edge from 1.4 to 2.7 GPa (cf Table 2) compared to the standard

699 reference value (14326 eV).

