

Supporting Information

1
2
3
4
5
6
7
8
9
10
11
12
13
14
15
16
17
18
19
20

pH-controlled self-assembled fibrillar network (SAFiN) hydrogels: evidence of a kinetic control of the mechanical properties

Ghazi Ben Messaoud,^{a,†} Patrick Le Griel,^a Daniel Hermida-Merino,^b Sophie L. K. W. Roelants,^{c,d} Wim Soetaert,^c Christian Victor Stevens,^e Niki Baccile^{a,*}

^a Sorbonne Université, Centre National de la Recherche Scientifique, Laboratoire de Chimie de la Matière Condensée de Paris, LCMCP, F-75005 Paris, France

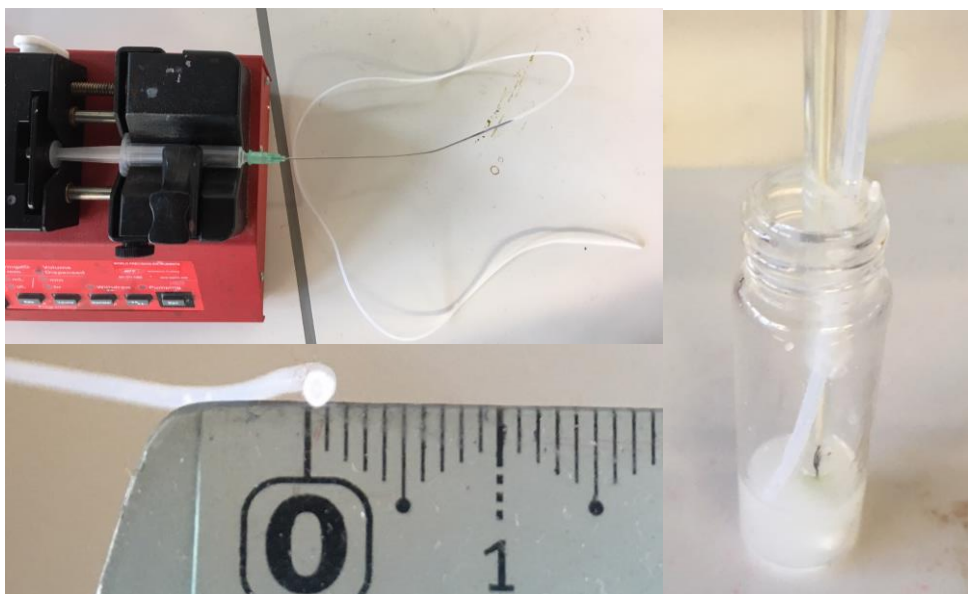
[†] Current address: DWI- Leibniz Institute for Interactive Materials, Forckenbeckstrasse 50, 52056 Aachen, Germany

^b Netherlands Organisation for Scientific Research (NWO), DUBBLE@ESRF BP CS40220, 38043 Grenoble, France

^c Ghent University, Centre for Industrial Biotechnology and Biocatalysis (InBio.be), Coupure Links 653, Ghent, Oost-Vlaanderen, BE 9000

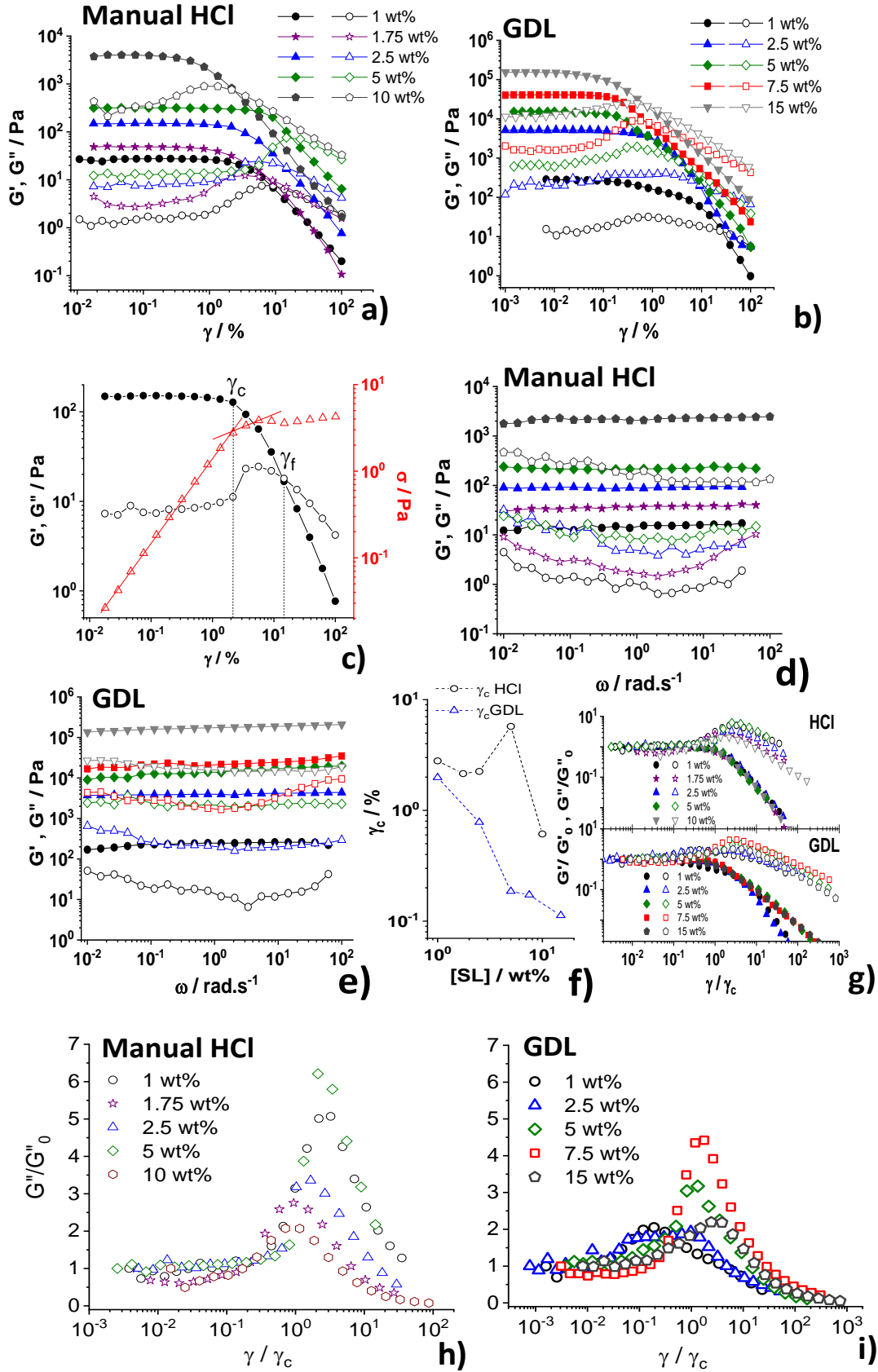
^d Bio Base Europe Pilot Plant, Rodenhuiszekaai 1, Ghent, Oost-Vlaanderen, BE 9000

^e SynBioC, Department of Green Chemistry and Technology, Ghent University, Coupure Links 653, 9000 Ghent, Belgium.



21

22 **Figure S 1 – Typical push-syringe apparatus used to acidify the SLC18:0 solution (1 mL, in the image). The**
23 **syringe is connected to a thin wall microbore PTFE tube, which is immersed in the solution. A pH**
24 **microelectrode monitors the pH at all time during acid injection.**



25

26 **Figure S 2 - Rheological properties of SLC18:0 hydrogels at pH-6 prepared either by manual acidification**
 27 **using HCl or upon GDL addition (molar ratio of SLC18:0:0.63 GDL) to basic solutions (initial pH 11).**

28 Concentration effects on the strain dependent storage (G') and loss (G'') moduli at a constant angular
29 frequency ($\omega = 6.28 \text{ rad}\cdot\text{s}^{-1}$) upon a) manual HCl acidification and b) GDL addition. c) Typical strain sweep
30 curve of SLC18:0 (C= 2.5 wt%) showing how γ_c and γ_f are determined. Concentration effect on the angular
31 frequency, ω , dependence of G' and G'' for hydrogels prepared using d) HCl or e) GDL. The legend for
32 panel a) apply to d) and the legend for panel b) apply to e). f) Concentration effect on the critical shear
33 strain (γ_c). g) Evolution of the reduced elastic (G'/G'_0) and viscous (G''/G''_0) reduced moduli as function of
34 the reduced shear strain (γ/γ_c). h-i) magnification of the viscous (G''/G''_0) reduced moduli as function of the
35 reduced shear strain (γ/γ_c) for SLC18:0 samples obtained upon h) manual HCl acidification or i) GDL
36 addition.

37

38 The rheological properties of SLC18:0 hydrogels prepared by manual acidification
39 using HCl or upon addition of GDL (molar ratio of SLC18:0:0.63 GDL) are shown in Figure S
40 2. As discussed in the main text, the shear strain dependency of storage (G') and loss (G'')
41 moduli demonstrate the same typical behavior (Figure S 2a, b). As shear strain increases, G'
42 decreases from a critical shear strain, γ_c (i.e, deviation from linearity), while G'' increases. The
43 estimation of the critical shear strain γ_c is obtained from the extent of the linear relationship
44 between the stress (σ) and the strain (γ) and where $\sigma = G^* \gamma$, with G^* being the complex
45 modulus calculated as $G^* = (G'^2 + G''^2)^{1/2}$ (Figure S 2c). The extent of the linear viscoelastic
46 regime from γ_c is related to structural changes and gel disruption. Finally, G' decreases
47 gradually with increasing shear strain amplitude while G'' exhibits a small strain hardening until
48 $G' = G''$, denoted as the fluidization strain (γ_f) from which both moduli decrease with strain, but
49 where $G'' > G'$ indicates a fluid-like behavior.

50 The evolution of $G'(\omega)$ and $G''(\omega)$ as function of angular frequency (ω) show that
51 comparable concentrations (1, 2.5 and 5 wt %), $G'(\gamma)$ of hydrogels prepared using GDL are at
52 least one order of magnitude higher than samples prepared using HCl, demonstrating the
53 influence of the acidification method on the final stiffness of the sample (Figure S 2d,e).

54 The evolution of the critical shear strain γ_c as a function of concentration and
55 acidification method (HCl or GDL) is shown in Figure S 2f. For hydrogels prepared using HCl,
56 the critical shear strain γ_c remains practically constant with concentration for $C_{\text{SLC18:0}} \leq 5 \text{ wt}\%$
57 before falling for $C_{\text{SLC18:0}} = 10 \text{ wt}\%$. On the contrary, γ_c decreases significantly with
58 concentration for the samples prepared using GDL, demonstrating that the sample becomes
59 more brittle with increasing concentration. This phenomenon indicates that if the sample
60 becomes stiffer due to the increase of G' with concentration, it is also more sensitive to
61 deformation. Such decrease of γ_c with concentration has been previously observed for other

62 SAFiNs systems¹⁻³ and it is in good agreement with the theoretical prediction for sterically
63 entangled solutions of biopolymers or cross-linked gels.⁴ It is in fact known that the decrease
64 of the theoretical γ_c with increasing concentration is attributed to the reduction of the mesh size
65 (the average spacing between fibers) and reduction of the entanglement length (distance
66 between entanglement points).⁴

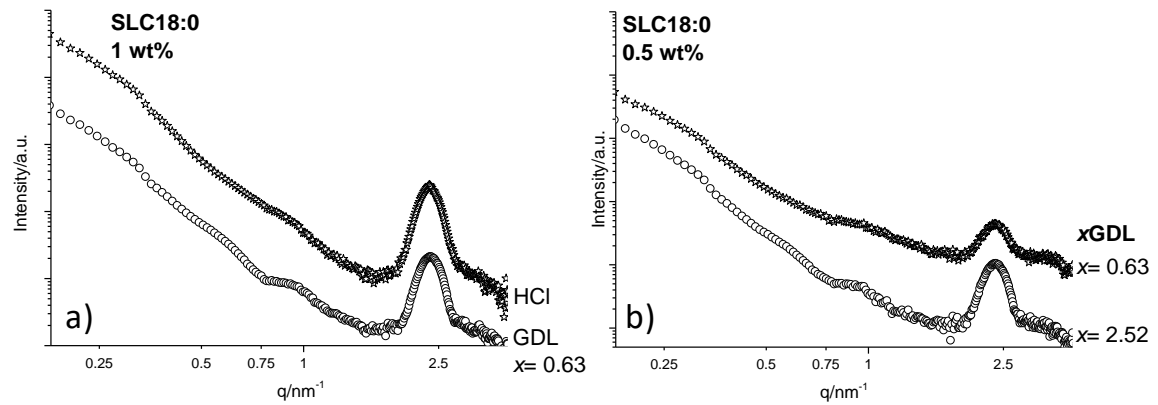
67 To investigate the yielding behavior of SLC18:0 samples under large amplitude
68 oscillatory shear (LAOS), G' and G'' moduli have been normalized by their respective values
69 in the LVER (G'/G'_0 ; G''/G''_0) and plotted as a function of the reduced shear strain amplitude
70 (γ/γ_c) (Figure S 2g). The reduced data show the decrease of G'/G'_0 and a local maximum in
71 G''/G''_0 curves. Based on the descriptive classification of the viscoelastic properties of complex
72 fluids as function of their LAOS behavior, SLC18:0 samples are considered as weak strain
73 overshoot fluids.⁵ All G'/G'_0 for HCl samples can be plotted on a master curve, while a small
74 deviation at high γ/γ_c is observed for samples prepared using GDL. Such strain hardening
75 overshoot was previously reported for a wide range of complex fluids like concentrated
76 emulsions,⁶ microgels suspensions,⁷ hydrophilic⁸ and hydrophobically-modified⁹ polymers
77 but also for SAFIN without discussing its origin.¹⁰ Depending on the complex fluid, the origin
78 of the strain overshoot can be attributed to an increase of the effective volume of temporal
79 structures,¹¹ to a variation of aggregate size in suspensions¹² or to a rearrangement of clusters¹³
80 during oscillatory shear deformation.⁵ However it's generally assumed that weak strain
81 overshoot is a result from the balance between the formation and the destruction of the network
82 junctions.⁵ In this work (refer to main text), we make the hypothesis, by combining rheology
83 and microscopy (optical and cryo-TEM), the overshoot can be explained by the presence of
84 spherulites and branched structures.

85 Figure S 2h and Figure S 2i bring out the intensity of the peak of the reduced viscous
86 modulus (G''/G''_0) as function of the reduced shear strain (γ/γ_c) for SLC18:0 samples obtained
87 upon manual HCl acidification or GDL addition. The intensity of the strain hardening of the
88 reduced loss modulus (G''/G''_0) varies in a considerable manner, however it does not show a
89 clear dependence on the SLC18:0 concentration. The global peak intensity of (G''/G''_0) was
90 relatively higher for samples prepared by manual acidification using HCl. One could suppose
91 that the higher presence of spherulites with an intermediate size in HCl samples compared to
92 GDL samples, may be responsible of the strain overshoot by analogy to flocks or clusters
93 formation in other weak strain overshoot fluids (emulsions and suspension). However the
94 presence also of weak strain overshoot for SLC180 samples prepared upon GDL addition may

95 contradicts this assumption. It's worth mentioning that hydrophilic polymers like xanthan gum
96 showed also a weak strain overshoot related to the formation of intermediate structural complex
97 by the association of the extended chains by hydrogen bonding.⁸ The formation of intermediate
98 structural complex between SLC18:0 ribbons by inter-ribbons interactions should not be
99 excluded. Under large deformations, these complex structures will first resist against the
100 imposed deformation resulting in an increase in G'' , before they break up from a certain
101 deformation limit beyond which the SLC180 fibers align with the flow field and resulting in a
102 decrease in G'' .

103

104

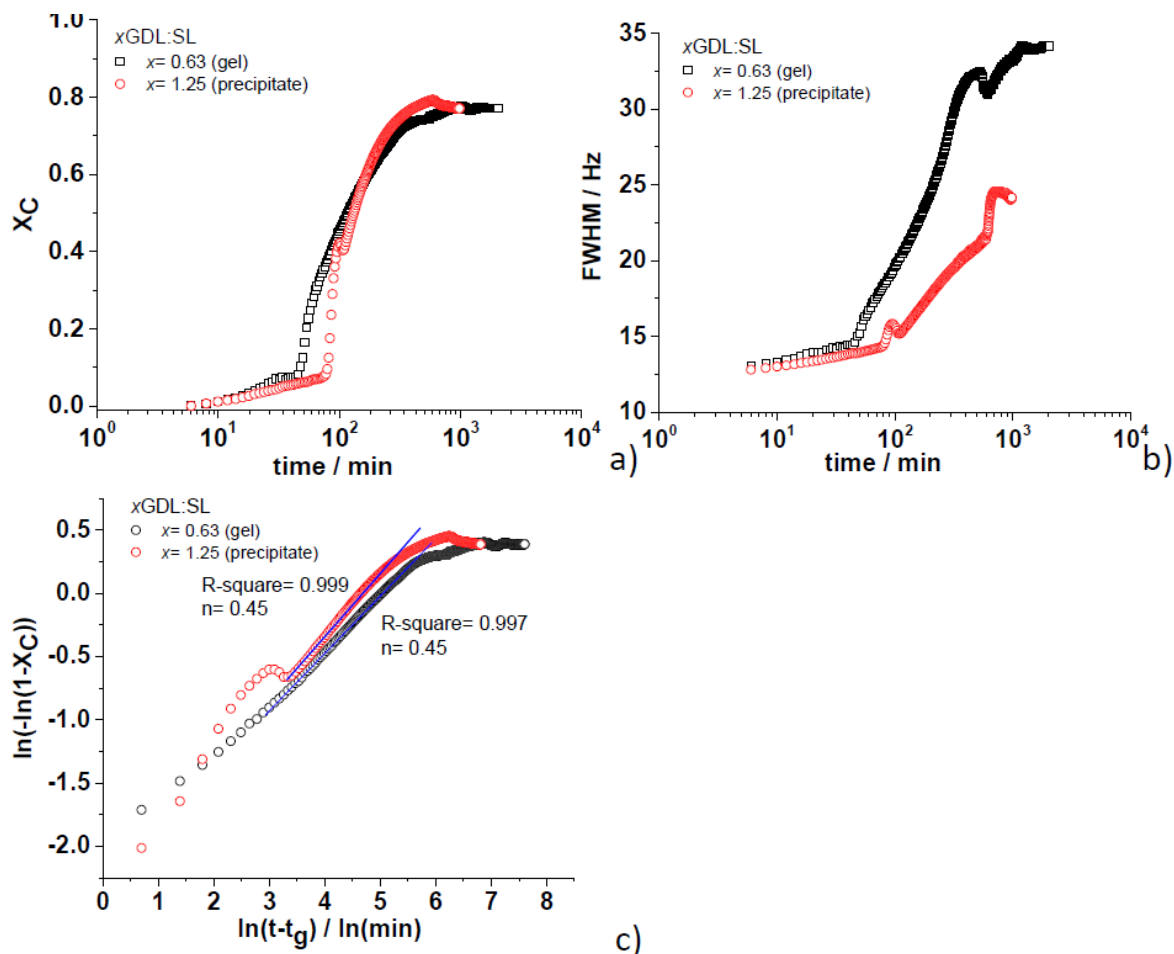


105

106 **Figure S 3 – SAXS patterns of a) SLC18:0 1 wt% solution acidified with HCl (manually) and GDL ($x=0.63$)**
 107 **and b) SLC18:0 0.5 wt% solution acidified with GDL at two different SLC18:0: x GDL molar ratios. The**
 108 **initial pH is 11 for all samples.**

109

110

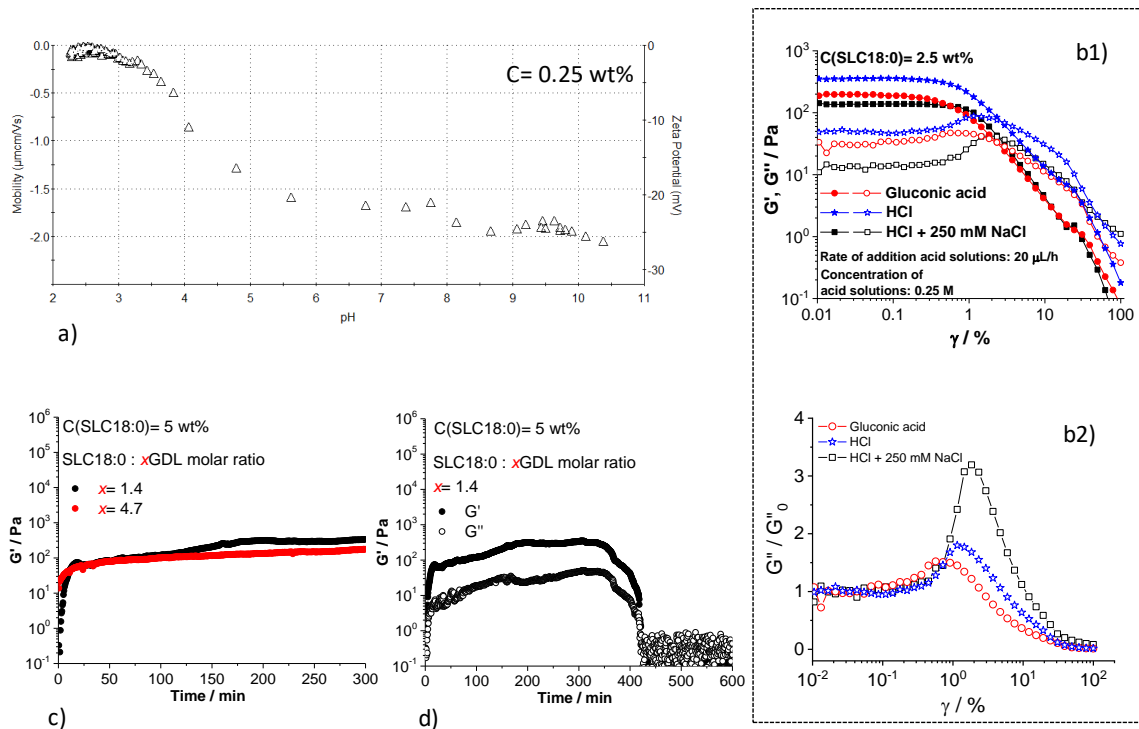


111

112 **Figure S 4 – a) Time evolution of the molar fraction of SLC18:0 (Initial concentration is 2.5 wt% and pH**
 113 **11) in the crystalline (fiber) phase upon addition of $xGDL$, x being the SLC18:0: $xGDL$ molar ratio; X_C from**
 114 **the solution 1H NMR spectra, as explained in the materials and method section. b) Time evolution of the full**
 115 **width at half maximum (FWHM); c) Avrami plot performed on X_C data in a).**

116

117



118

119 **Figure S 5 – a) Overall electrophoretic mobility and ζ -potential evolution measured as a function of pH from**
 120 **pH ~11 to pH ~2 on a SLC18:0 solution at $C = 0.25$ wt% measured at 25°C using a Malvern Zetasizer Nano**
 121 **ZS90 (Malvern Instruments). b1) Strain dependent storage (G') and loss (G'') moduli at a constant frequency**
 122 **($f = 1$ Hz) for a SLC18:0 solution ($V = 1$ mL) prepared at $C = 2.5$ wt% and initial pH ~10. The acid solutions**
 123 **are composed of either HCl or gluconic acid, both at 0.25 M. Rate of addition is $20\ \mu\text{L/h}$. The gluconic acid**
 124 **solution is prepared from a 2 M GDL solution left at rest overnight and with final pH of 1.5, whereas GDL**
 125 **spontaneously hydrolyzes into gluconic acid in water. The experiments referred to with filled and empty**
 126 **squares are recorded on a solution containing additional 250 mM NaCl and of which the acidification has**
 127 **been performed with HCl (0.25 M). b2) Normalized loss (G''/G''_0) moduli shown in b1) by their respective**
 128 **values in the LVER as function of the shear strain γ . c-d) Time-resolved evolution of storage (G') and loss**
 129 **(G'') moduli measured at ($f = 1$ Hz, $\gamma = 0.1\%$) on a SLC18:0 solution at $C = 5$ wt% and set at pH ~10.**
 130 **Acidification is performed with an excess of GDL, of which the molar ratio with respect to SLC18:0 are**
 131 **given in the figure.**

132

133

134 **References**

135 (1) Ozbas, B.; Kretsinger, J.; Rajagopal, K.; Schneider, J. P.; Pochan, D. J. Salt-Triggered Peptide
 136 Folding and Consequent Self-Assembly into Hydrogels with Tunable Modulus.
 137 *Macromolecules* **2004**, *37*, 7331–7337.

138 (2) Franceschini, A.; Filippidi, E.; Guazzelli, E.; Pine, D. J. Dynamics of Non-Brownian Fiber
 139 Suspensions under Periodic Shear. *Soft Matter* **2014**, *10*, 6722–6731.

140 (3) Breedveld, V.; Nowak, A. P.; Sato, J.; Deming, T. J.; Pine, D. Rheology of Block
 141 Copolyptide Solutions: Hydrogels with Tunable Properties. *Macromolecules* **2004**, *37*,

- 142 3943–3953.
- 143 (4) MacKintosh, F. C.; Käs, J.; Janmey, P. A. Elasticity of Semiflexible Biopolymer Networks.
144 *Phys. Rev. Lett.* **1995**, *75*, 4425–4428.
- 145 (5) Hyun, K.; Wilhelm, M.; Klein, C. O.; Cho, K. S.; Nam, J. G.; Ahn, K. H.; Lee, S. J.; Ewoldt, R.
146 H.; McKinley, G. H. A Review of Nonlinear Oscillatory Shear Tests: Analysis and Application
147 of Large Amplitude Oscillatory Shear (LAOS). *Prog. Polym. Sci.* **2011**, *36*, 1697–1753.
- 148 (6) Mason, T. G.; Bibette, J.; Weitz, D. A. Elasticity of Compressed Emulsions. *Phys. Rev. Lett.*
149 **1995**, *75*, 2051.
- 150 (7) Cloitre, M.; Borrega, R.; Leibler, L. Rheological Aging and Rejuvenation in Microgel Pastes.
151 *Phys. Rev. Lett.* **2000**, *85*, 4819.
- 152 (8) Hyun, K.; Kim, S. H.; Ahn, K. H.; Lee, S. J. Large Amplitude Oscillatory Shear as a Way to
153 Classify the Complex Fluids. *J. Nonnewton. Fluid Mech.* **2002**, *107*, 51–65.
- 154 (9) Pellens, L.; Gamez Corrales, R.; Mewis, J. General Nonlinear Rheological Behavior of
155 Associative Polymers. *J. Rheol. (N. Y. N. Y.)* **2004**, *48*, 379.
- 156 (10) O’Leary, L. E. R.; Fallas, J. A.; Bakota, E. L.; Kang, M. K.; Hartgerink, J. D. Multi-
157 Hierarchical Self-Assembly of a Collagen Mimetic Peptide from Triple Helix to Nanofibre and
158 Hydrogel. *Nat. Chem.* **2011**, *3*, 821–828.
- 159 (11) Tirtaatmadja, V.; Ta, K. C.; Jenkins, R. D. Rheological Properties of Model Alkali-Soluble
160 Associative (HASE) Polymers: Effect of Varying Hydrophobe Chain Length. *Macromolecules*
161 **1997**, *30*, 3271–3282.
- 162 (12) Raghavan, S. R.; Khan, S. A. Shear-induced Microstructural Changes in Flocculated
163 Suspensions of Fumed Silica. *J. Rheol. (N. Y. N. Y.)* **1995**, *39*, 1311.
- 164 (13) Parthasarathy, M.; Klingenberg, D. J. Large Amplitude Oscillatory Shear of ER Suspensions. *J.*
165 *Nonnewton. Fluid Mech.* **1999**, *81*, 83–104.
- 166