Unambiguous localization of titanium and iron cations in doped manganese hollandite nanowires

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New insights into the chemical and structural features of iron or titanium-doped K_xMnO_2 hollandites are reported. Neutron diffraction and atomically resolved transmission electron microscopy elucidate the localization of the dopant cations that could be one of the key factors governing the functional activity of these nanomaterials.

Hollandite oxides, α-MnO_2, belong to an intriguing class of microporous manganese oxides that have attracted special attention due to their remarkable properties for energy storage, such as pseudocapacity in supercapacitors and electrocatalysis in rechargeable batteries[1, 2]. These compounds also exhibit ion exchange[3] and magnetic properties[4, 5]. They are efficient materials for radioactive waste immobilization[6] and for catalysis in different reactions[7-10]. This multifunctionality arises from the one-dimensional tunnel structure of hollandites, formed by dimeric chains of edge-sharing MnO_6 octahedra oriented along the c crystallographic axis. Each dimeric chain is connected to two others in the ab plane by the octahedra corners, forming 2x2 and 1x1 tunnels along the c axis (Figure 1). The 1x1 tunnels are usually empty while the 2x2 ones are partially occupied by large cations such as K^+ or Ba^{2+}, which promote the partial reduction of Mn(IV) to Mn(III), and then the formation of a mixed Mn valence compound. The 2x2 channels are also usually partially filled with water molecules accompanying the cations. To tune and improve the numerous properties of hollandites, a lot of research has been devoted to design new pathways for controlling the particle size, morphology and for incorporating dopant cations[10-12]. This last approach has received special attention in catalysis[9, 13-18]. Nevertheless, in spite of the large number of studies that prove properties improvement due to doping, the position of the foreign cation is only inferred by indirect methods[13, 19, 20]. The present work focuses on tuning two synthetic pathways to allow easy and reproducible incorporation of Ti and Fe cations in hollandite nanowires, and then to thoroughly assess the presence and localization of dopant cations. Fe has been selected because this cation is known to enhance catalytic properties of hollandites[18, 20, 21].

Titanium, on the other side, has been selected because of its ability to form hollandite structures[22]. Titanium doping in manganese hollandites has not been reported yet.

Herein we have designed hollandite-type manganese oxides following two synthetic pathways (details in SI and Table SI 1). The first approach consists in a one-pot synthesis by the reaction between $Mn(SO_4)_2$ and $KMnO_4$ in acidic aqueous medium[23, 24]. Doped hollandite oxides were obtained by adding to the initial $Mn(SO_4)_2$ solution the foreign metal precursor, $(Fe(NO_3)_3 \cdot 9H_2O$ or titanium(IV) bis(ammonium lactato) dihydroxide. The second approach is a two-steps route where layered birnessite manganese oxide is first formed by precipitation in basic aqueous medium at room temperature, and then transformed into hollandite by acidic hydrothermal treatment[25]. Doped birnessites are obtained again by addition of Fe and Ti precursors to the reaction medium.
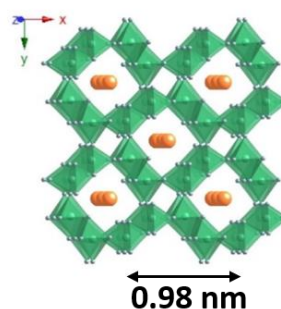

0.98 nm

Figure 1. Crystal structure of hollandite oxide (K_xMnO_2). Color code: K orange, Mn green and O grey.

The incorporation of Fe and Ti in the materials was confirmed by cationic elemental analysis using electron microprobe analysis (EMPA) (Table SI 1). The formation of doped hollandite-type (S.G. I4/m) single phases was supported by Le Bail analysis of the X-ray diffraction patterns (Figure 2a, Figure SI 1, Table SI 2). An increase in the cell parameters is observed upon doping (Table SI 4), in good agreement with the larger cationic radii of the dopants Ti^{4+} (0.61 Å) and Fe^{3+} (0.65 Å) compared to Mn^{4+} (0.53 Å)[26]. Both EMPA and diffraction data show unambiguously that the foreign cations are hosted in the oxide's frameworks. For all compositions, a strong anisotropy is observed, as the (002) peak ($\approx$ 65°) is more narrow than other peaks, which suggests preferential growth of nanowires along the [001] direction[27], as confirmed by TEM (Figure 2c-e). Diffraction peaks are also broadening (Table SI 3) with an increase of the dopant concentration, suggesting smaller particles or incorporation of larger strain. TEM (Figure 2c-e)

a. Departamento de Química Inorgánica, Facultad de Químicas, Universidad Complutense, 28040-Madrid, Spain.
b. Sorbonne Université, CNRS, Collège de France, Laboratoire Chimie de la Matière Condensée de Paris (LCMCP), 4 Place Jussieu, F-75005 Paris, France.
c. Facultad de Ciencias, Universidad de Cádiz, Campus Rio San Pedro, Puerto Real, Spain.
d. Institut Laue-Langevin, 71 avenue des Martyrs, CS 20156, 38042 Grenoble cedex 9, France
e. Centro Nacional de Microscopía Electrónica, Universidad Complutense, 28040 Madrid, Spain.

confirms that the length of the nanowires decreases upon doping.

To get further insights into the presence and localization of eventual anionic vacancies and into the cationic distribution of Fe, Ti and Mn in the $K_xMn_{1-y}(Fe, Ti)_yO_{2-\delta}$ phases, the materials were analyzed by neutron diffraction (ND). Since materials obtained by both synthetic pathways are structurally and compositionally analog according to the results described above, ND was performed only on one-pot materials. In order to avoid incoherent scattering of adsorbed water molecules (Figure SI 2), the samples were partially dehydrated at 180 °C for 24 h[28] (Figures SI 2 and SI 3). The ND data collected at room temperature were refined in the I4/m space group[29]. The atomic coordinates, thermal parameters and occupancy factors of atoms were refined. The Rietveld refinement patterns and difference plots for $K_{0.11}MnO_{2-\delta}$, $K_{0.11}Mn_{0.95}Fe_{0.05}O_{2-\delta}$, $K_{0.09}Mn_{0.8}Fe_{0.2}O_{2-\delta}$, and $K_{0.1}Mn_{0.9}Ti_{0.1}O_{2-\delta}$ are shown in Figure 2b, Figure SI 3a, SI 3b and SI 3c, respectively. The refined structural parameters for all four hollandites are listed in SI-2 (Table SI 4). For the undoped sample, the value of the final isotropic atomic displacement parameter of potassium cation is much larger than those for manganese and oxygen. This yields instabilities during the refinement so that the potassium content had to be fixed according to the value obtained by elemental analysis (Table SI 1). This behavior can be attributed to the easy-mobility of K^+ in the tunnel and to residual water molecules in the channels (Figure SI 4). The divergence between cell parameters values obtained by XRD and ND could be caused by the presence of a greater amount of water in the former. The Mn-O distances (Table SI 5) show distorted MO_6 octahedra in all hollandites, with three short Mn-O1 distances and three long Mn-O2 ones (Figure SI 5a). This feature corresponds to the displacement of the manganese cation from the octahedron center toward the three oxygen of the unshared edges to minimize Mn-Mn repulsions between edge-sharing octahedra (Figure SI 5b). The average Mn-O distances are in agreement with those observed in other MnO_2 phases (pyrolusite at 1.89 Å[30] and ramsdellite at 1.90 Å)[31] The composition of the undoped phase is $K_{0.11}MnO_{1.958(1)}$ according to EMPA and ND. The occupancy factors of the oxygen atoms reveal anionic vacancies mostly on the O2 position (Table SI 4), which corresponds to O^{2-} anions located in the connection between adjacent dimers (Figure SI 5c). This conclusion is in agreement with the hypothesis formulated by Wu and co-workers[32]. Importantly, we report herein the first experimental localization of oxygen vacancies in manganese hollandites.

The presence of Ti and Fe in the octahedral sites of doped hollandites does not significantly modify the Mn-O distances
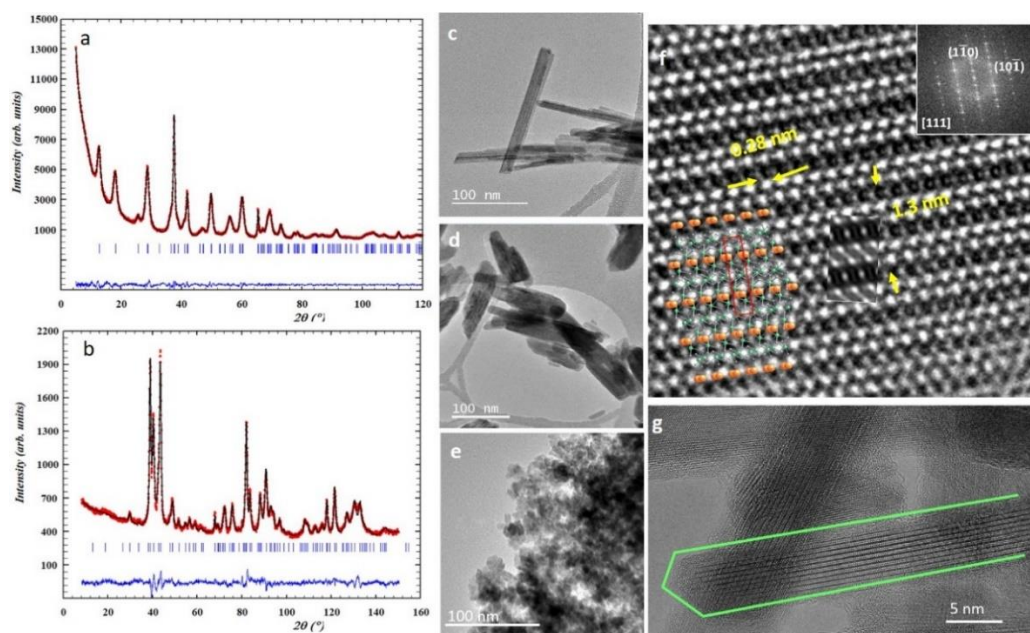

Figure 2. (a) Le Bail analysis of X-ray diffraction patterns and (b) Rietveld refined powder neutron diffraction pattern of the undoped hollandite obtained by one-pot synthesis. Low magnification TEM images of (c) $K_{0.12}MnO_\delta$; (d) $K_{0.12}Mn_{0.95}Fe_{0.05}O_{2-\delta}$, and (e) $K_{0.09}Mn_{0.8}Fe_{0.2}O_{2-\delta}$ (nominal composition). (f, g) HRTEM images at different magnifications for $K_{0.12}MnO_\delta$. FFT, calculated image (Δt= 8 nm, Δf= -20 nm) and schematic structural model are inserted in (f).

compared to the undoped $K_{0.11}MnO_{1.958(1)}$ (Table SI 5). For the Fe-doped hollandites, $K_xMn_{1-y}Fe_yO_{2-\delta}$, ($y = 0.05$ and 0.2) the octahedral site was constrained to be fully occupied but no constraint was imposed on the Mn:Fe ratio. The distribution of Mn and Fe among octahedra is summarized in Table SI 4. The refinement shows again the presence of anionic vacancies. The full compositions obtained were $K_{0.11}Mn_{0.946(2)}Fe_{0.054(2)}O_{1.899(14)}$ and $K_{0.09}Mn_{0.844(2)}Fe_{0.156(2)}O_{1.904(17)}$. For $K_{0.1}Mn_{0.9}Ti_{0.1}O_{2-\delta}$, the cationic composition could not be refined due to the very close scattering lengths of Mn (-3.73 fm) and Ti (-3.438 fm)[33]. Nevertheless, the refinement clearly shows that the Ti atoms cannot be located in the structure channels.

A representative HRTEM image (Table SI 6), its corresponding FFT and calculated image for the undoped sample (Figure 2f-g) are in good agreement with the hollandite unit cell. HRTEM images for other orientations and compositions are provided in Figure SI 6 and Figure SI 7.

In order to unambiguously solve the localization of the dopant cations and their electronic properties, an atomically resolved STEM spectrum-image study (SI) has been performed, despite the high instability of the samples under the focused electron beam. Characteristic spectrum-images studies of iron-doped samples are shown in Figure 3. Alternating atomic columns with large contrast are observed by High Angle Annular Dark Field (HAADF) imaging (Figure 3a, b), corresponding to Mn/dopant columns, in agreement with the expected structure (inserted schematic model) and structural refinements. K is not detected by HAADF due to its low content as well as to the presence of much heavier neighboring Mn cations. Nonetheless, the sum EELS spectrum (Figure 3c) shows characteristic edges of all the elements, $K-L_{2,3}$, $O-K$, $Mn-L_{2,3}$ and $Fe-L_{2,3}$, while the corresponding chemical maps confirm their distribution according to the crystal structure. Fe and Mn are collocated, so that Fe predominantly substitutes Mn, in agreement with structural refinement.

The low signal-to-noise ratio observed for Fe is due to its low concentration as well as to the low acquisition time (discussion SI 4) required in order to avoid sample degradation. Similar results are obtained for Ti doped samples (Figure SI 8). The mean oxidation states of Mn, Fe and Ti were evaluated from EELS spectra recorded along lines to have enough statistic and avoid sample damage (discussion SI 5). Both doped and undoped samples exhibit a $Mn-L_{2,3}$ edge typical of coexisting Mn^{3+} and Mn^{4+} oxidation states (Figure 3i and discussion SI 5) when compared with Mn^{2+}, Mn^{3+} and Mn^{4+} standards with a good agreement between the relative intensity ratio of the $Mn-L_2$ and $Mn-L_3$ white lines (discussion SI 5). Likewise, the oxidation states of Fe and Ti in the doped samples are +3 and +4 respectively. These values are in good agreement with the chemical composition obtained by chemical analysis and neutron diffraction.

Conclusions

In summary, nanoparticles of titanium and iron doped hollandites have been obtained according to two synthetic pathways. In these materials, dopant metal ions are hosted in the manganese lattice, as evidenced by neutron diffraction and atomically resolved transmission electron microscopy. We report herein i) the first titanium doping of manganese hollandite and ii) the first experimental atom-scale observation providing unambiguous information on dopant and oxygen vacancies localization in manganese hollandites. This knowledge could be a fundamental key to understand and design the functional behavior of these materials.
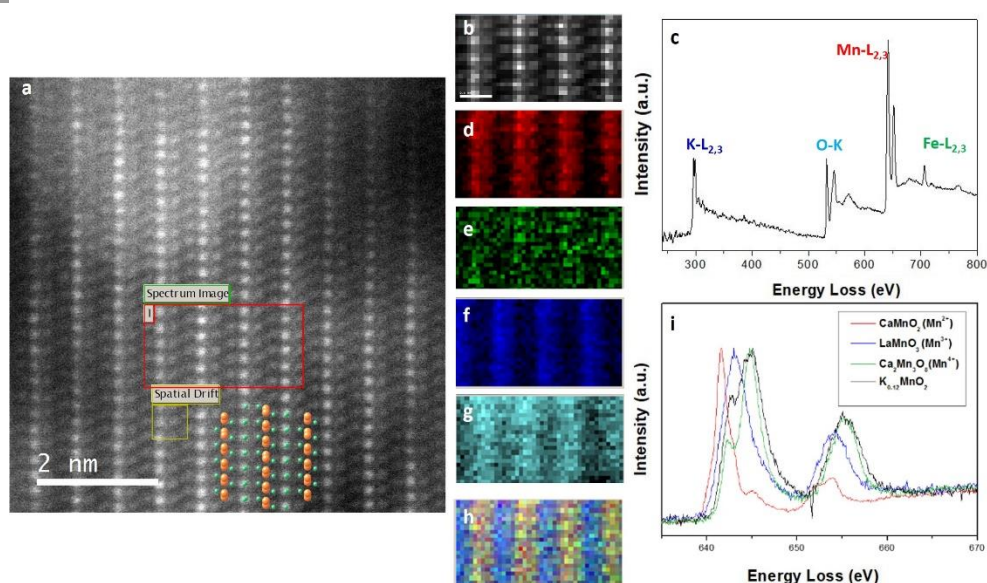

Figure 3. STEM-EELS study corresponding to Fe-doped hollandite. (a) HAADF images along the [111] zone axis, indicating the area where the spectrum image was performed; a schematic representation of the structure is also included. (b) HAADF image simultaneously recorded with the spectra. (c) Sum spectrum showing the edge of all the elements present. (d-g) Chemical maps obtained from the different signals (Mn red, K blue, Fe green, O soft blue). (h) Combined map showing all the elements. (i) Representative EELS spectra showing the $Mn-L_{2,3}$ signal for an undoped hollandite in comparison with references of Mn^{4+}, Mn^{3+} and Mn^2.

Conflicts of interest

There are no conflicts to declare

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