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Development of a thermodynamic approach to assist the control of the precipitation of hydroxyapatites and associated calcium phosphates in open systems

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#### **ABSTRACT**

The understanding of the precipitation mechanisms of calcium phosphates (CaPs) remains extremely challenging despite its fundamental interest for material sciences and its practical implications in medicine, paleosciences, and chemistry, especially for heterogeneous catalysis. The competitive precipitation of CaPs, such as hydroxyapatite, octacalcium phosphate and dicalcium phosphate, has been shown experimentally to be strongly impacted by the pH of the precipitating medium. However, the thermodynamic validation of this observation, based on the solubility isotherms reported in the literature, could not be comprehensively substantiated. In the present study, we developed a thermodynamic prediction model adapted to the synthesis of CaPs in open systems (dropwise synthesis). By overcoming the theoretical limits of the solubility isotherms to explain their precipitation features, this new thermodynamic model makes it possible

to shed light on the influence of the common ion effect on the stability of these materials in aqueous solution. Finally, this work paves the way to a methodological approach that could be of broad interest in the field of the precipitation of inorganic materials when a continuous flow addition is used.

#### INTRODUCTION

Calcium phosphates (CaPs) are materials of wide interest in many fields (medicine, 1,2 geosciences,<sup>3,4</sup> paleosciences,<sup>5–7</sup> chemistry<sup>8–10</sup>...) due, among other things, to their biocompatibility, and mechanical and surface properties. They present distinct Ca/P ratios ranging from 1.00 to 1.67 and are likely to precipitate in aqueous solution under soft chemistry conditions. This is the case of hydroxyapatite (HAp, Ca/P = 1.67) of generic formula  $Ca_{10}(PO_4)_6(OH)_2$ , amorphous calcium phosphate (ACP, Ca/P = 1.50) that consists of  $Ca_9(PO_4)_6$  units generally referred to as Posner's clusters, 11 octacalcium phosphate (OCP, Ca/P = 1.33) of generic formula Ca<sub>8</sub>(HPO<sub>4</sub>)<sub>2</sub>(PO<sub>4</sub>)<sub>4</sub> · 5H<sub>2</sub>O, as well as dicalcium phosphate dihydrate (DCPD, CaHPO<sub>4</sub> · 2H<sub>2</sub>O, Ca/P = 1.00) and dicalcium phosphate anhydrous (DCPA, CaHPO<sub>4</sub>, Ca/P = 1.00). Hydroxyapatite is the main mineral constituent of bones and teeth, while ACP and OCP are proposed to be its *in vivo* precursors in the biomineralization mechanisms. <sup>12,13</sup> Interestingly, owing to its great flexibility, the HAp structure exhibits peculiar affinity for both cationic and anionic substitutions. Together with its non-stoichiometric properties, often expressed as the modulation of the Ca/P ratio, such a tunable composition impacts its bioactive, mechanical and surface properties, making this system attractive for dental treatments, 14,15 osteointegration, 16 heavy-metal waste water or soil remediation<sup>17,18</sup> and heterogeneous catalysis applications.<sup>8–10,19</sup> As these different minerals are likely to precipitate competitively in aqueous solution, the understanding of the precipitation mechanisms could make it possible to design highly selective syntheses in terms of crystalline phase composition, crystallographic defects incorporation and textural properties. Such insights into these *in vitro* mechanisms could also provide keys for the appreciation of *in vivo* mineralization processes, such as osteogenesis.<sup>20,21</sup> In addition, HAp materials have emerged as eco-friendly heterogeneous catalysts,<sup>17</sup> both as a bulk tunable active phase and as a suitable support for the controlled surface deposition of highly-dispersed catalytic metallic centers,<sup>9</sup> being of interest for both liquid solvent-free medium,<sup>10,22,23</sup> and gas phase reactions.<sup>9,24</sup> Given the significant impact of the preparation conditions on the texture and the composition (non-stoichiometric properties) of the apatitic compounds,<sup>8</sup> there is a crucial need in the rationalization of the key synthesis parameters to be tuned to control the surface reactivity.

Hydroxyapatite is extensively prepared by co-precipitation routes, <sup>8,25</sup> which are highly sensitive to the pH of the reaction medium that strongly influences the nature of the precipitated calcium phosphate(s).<sup>26</sup> In particular, gel and NH<sub>3</sub> vapor diffusion techniques were developed as biomimetic protocols to study, *inter alia*, the effects of matrix molecules on CaPs mineralization.<sup>27–30</sup> Besides, numerous protocols reported to date for its preparation involve the dropwise addition of a phosphate source in a calcium one (or *vice versa*). As a result, the thermodynamic modelling of CaPs dissolution in closed systems (solubility isotherms), widely cited to explain the formation of HAp in alkaline and neutral media, and that of DCPA in an acidic one, appears to be inadequate to account for the thermodynamic precipitation predictions in such current open systems.<sup>31,32</sup>

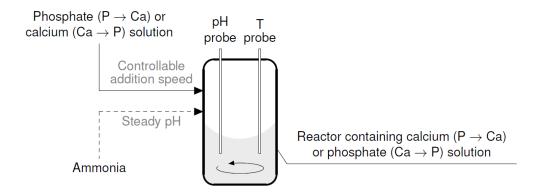
The present study aims at investigating how thermodynamics influences the precipitation reactions. To decorrelate the thermodynamic and kinetic aspects of CaPs precipitation, the present study focuses on their dropwise syntheses, under quite soft chemistry conditions (water as a

solvent, moderate temperatures, long reaction times, and high concentration of precursors). The combination of a theoretical and an experimental approach made it possible to analyze the influence of various experimental parameters such as the pH, the temperature, the final concentration ratio  $[Ca^{2+}]/[H_xPO_4^{(3-x)-}] = [Ca^{2+}]/[P]$  introduced into the reactor as well as the order of introduction of the precursor ions on materials synthesized under thermodynamic control. It is rationalized how setting these experimental parameters can help control the HAp morphology in the absence of organic solvent or additive. 33-35 From an industrial point of view, the establishment of such interconnection between the experimental parameters and the properties of the synthesized materials is of the utmost interest. 36-38 The purpose of this study lies in the indepth analysis of the thermodynamic aspects of the competitive precipitation of CaPs, whereas the complementary kinetic aspects will be investigated in a forthcoming article. Synthesized CaP powders have been characterized by X-Ray diffraction (XRD), Raman spectroscopy, X-Ray fluorescence spectroscopy (XRF), scanning electron microscopy (SEM) and N2 sorption. The synthesis routes implemented at stationary pH to validate the thermodynamic model have made it possible to greatly modulate the Ca/P ratio of the HAp material as well as its morphology and specific surface area (SSA), which is of particular interest in the context of applications in heterogeneous catalysis in order to establish structure-activity relationships.<sup>39</sup>

#### MATERIALS AND METHODS

**Material synthesis.** Calcium phosphates were prepared via a co-precipitation procedure using an automated reactor (Optimax 1001 synthesis workstation, Mettler Toledo) that allows controlling, *inter alia*, both pH and temperature profiles as well as the addition rate of the precursor solutions (Figure 1).<sup>40</sup> Calcium and phosphate solutions were prepared by dissolving 12.81 mg of

 $Ca(NO_3)_2 \cdot 4H_2O$  (Sigma Aldrich, purity  $\geq 99.0$  %) and 3.74 mg of  $(NH_4)_2HPO_4$  (Acros Organics, purity  $\geq 99.0 \%$ ) in 250 mL of ultrapure water (resistivity of 18.2 M $\Omega \times$  cm at 25 °C), respectively. Two synthesis routes,  $Ca \rightarrow P$  and  $P \rightarrow Ca$ , were considered depending on the order of introduction of the precursor ions into the reactor. Both syntheses were carried out at 80 °C (unless stated otherwise) and at constant pH in the reactor of 9.0, 6.5 and 4.2. The typical procedure is detailed in the case of the Ca  $\rightarrow$  P route performed at 80 °C and operating at steady pH of 9.0. The phosphate solution ( $[P]_0 = 0.13 \text{ mol/L}$ , 200 mL) was first adjusted to pH 10.0 by the addition of a small volume of concentrated ammonia (Sigma Aldrich, 28 %) before being poured into the reactor and maintained under an inert atmosphere (N<sub>2</sub>). This phosphate solution was then gradually heated up to 80 °C (5 °C/min) under mechanical stirring (400 rpm). The temperature of the reaction medium is known to influence the acid-base equilibrium governing the pH. The pH of the phosphate solution was found to decrease as the temperature increased (S1 and Figure S1A) and was adjusted to a value of 9.0 by adding concentrated ammonia drop by drop (Figure S1B). Depending on the targeted final ratio [Ca<sup>2+</sup>]/[P] of 1.67 or 1.50, an appropriate volume of the calcium solution ( $[Ca^{2+}]_0 = 0.22 \text{ mol/L}$ ), beforehand degassed by bubbling  $N_2$  for 15 minutes and adjusted to pH 10.0 by the addition of concentrated ammonia, was poured at a controlled speed (2.2 mL/min) into the reactor. During this step, the pH was kept constant with the automated addition of suitable volumes of concentrated ammonia. Note that the pH of the medium in the reactor is easier to maintain constant by the system during a  $Ca \rightarrow P$  rather than a  $P \rightarrow Ca$  synthesis because of the higher buffering capacity of the phosphate solution compared to that of the calcium one (Figure S1C). In some cases, once the addition step was completed, a 2 h maturation step at 80 °C and pH 9.0 was eventually performed. The reactor temperature was then cooled down to 20 °C within a few minutes. In most cases, the white precipitate was recovered by centrifugation (8500 rpm) and washed three times with distilled water to mainly remove ammonia, and ammonium and nitrate ions. The wet sample was then recovered on a watch glass and dried at room temperature for 12 hours before being finely ground in an agate mortar. In the case of syntheses at pH 4.2, as discussed in section II - 1.2., the precipitates were rapidly recovered by Büchner filtration and washed promptly with distilled water.



**Figure 1.** Schematic representation of the automated reactor used for the precipitation of calcium phosphates.

**X-ray diffraction.** Laboratory X-ray diffraction patterns were recorded with a Brucker D8 Advance diffractometer equipped with a copper source ( $\lambda_{\text{Cu}-K_{\alpha 1}} = 1.54056 \text{ Å}$  and  $\lambda_{\text{Cu}-K_{\alpha 2}} = 1.54439 \text{ Å}$ ) and a LynxEye detector. XRD patterns were recorded by 0.02 ° steps in the  $2\theta$  range of 8–90 ° for high-angle diffraction. For low-angle XRD, a blade was added to the experimental set-up to prevent the detector from directly receiving the incident X-ray beam and patterns were recorded by 0.02 ° steps in the  $2\theta$  range of 1–6 °. The indexation of the XRD lines of the CaP phases is reported in S2 (Table S2A).

**Raman spectroscopy.** Raman spectra were collected with a Kaiser Optical system equipped with a charge couple detector (CCD) and a laser with  $\lambda = 785$  nm (P = 12 or 25 mW, resolution = 4 cm<sup>-1</sup>, accumulation time = 30 s, 30 scans per spectrum). Poorly crystalline powders were

analyzed with a laser power of 25 mW instead of 12 mW to improve the signal-to-noise ratio, without any noticeable observation of material degradation. Assignment of the Raman bands of the CaP phases is provided in S2 (Table S2B).

Scanning electron microscopy. Scanning Electron Microscopy with Field Emission Gun (SEM-FEG) was used for morphological investigation of the materials. High resolution images were obtained with a Hitachi SU-70 microscope. The samples were fixed on an alumina SEM support with a carbon adhesive tape and were observed without any metal coating. Secondary electron detector (SE-Lower) was used to observe the samples. The low accelerating voltage was 1 kV to prevent charging effects of the insulating CaPs and the working distance was around 10 to 15 mm.

N<sub>2</sub> sorption. Specific surface area measurements were carried out on a BELSORP-max instrument (BEL Japan) at 77 K after evacuation of the samples at 140 °C for 12 h under vacuum. This parameter was estimated by using the BET method in the  $0.05 < p/p^0 < 0.30$  domain.

**X-ray fluorescence spectroscopy.** Ca/P bulk ratio of the prepared CaPs was determined following a quantitative XRF protocol. In order to avoid matrix and grain size effects, all samples were melted into "beads" and analyzed with a X-ray fluorimeter Epsilon 3xl (Panalytical) equipped with an Ag X-ray tube operating under He atmosphere, with two conditions:  $12 \text{ kV}-25 \text{ }\mu\text{A}$  during 120 s for the analysis of calcium and  $10 \text{ kV}-30 \text{ }\mu\text{A}$  during 120 s for that of phosphorus. The bead was prepared by mixing approximately 0.114 g of sample, 1.231 g of fluxing agent (LiBO<sub>2</sub>/Li<sub>2</sub>B<sub>4</sub>O<sub>7</sub>) and 0.019 g of non-wetting agent (LiBr) in a platinum crucible. The mixture was heated to 1050 °C for 25 minutes in a fusion instrument (LeNeo fluxer, Claisse). Calibration curves were obtained from beads of mechanical mixtures of the precursors  $\text{Ca}(\text{NO}_3)_2 \cdot 4\text{H}_2\text{O}$  and

(NH<sub>4</sub>)<sub>2</sub>HPO<sub>4</sub> used for the material synthesis (Figure S3). Their relative proportions were chosen in order to prepare reference samples with a Ca/P ratio framing those of calcium phosphates, although with this methodology, the response is perfectly linear over the entire concentration range.

#### **RESULTS AND DISCUSSIONS**

The solubility of CaPs has been found to be strongly dependent on the pH of the aqueous medium. Being very weakly soluble in basic media, these minerals solubilize to a significant extent in acidic ones (S4). As an increase in the solubility of CaPs at low pH is accounted for by the speciation diagram of the phosphate species in solution (Figures S4A and S4B). As the pH decreases, the protonation of the PO<sub>4</sub> and HPO<sub>4</sub> groups of CaPs, into HPO<sub>4</sub>, H<sub>2</sub>PO<sub>4</sub> and H<sub>3</sub>PO<sub>4</sub> species, leads to a shift of the dissolution-precipitation equilibria towards the dissolution of the CaPs (Table 1, S4). These observations, formalized by the solubility isotherms, are often reported to explain the formation of DCPA and HAp under acidic and neutral/basic conditions, respectively. However, due to the different Ca/P ratios exhibited by CaPs, the comparison of their solubility isotherms cannot be used as a basis for the comprehensive understanding of their competitive precipitation.

**Table 1.** Ca/P molar ratio of calcium phosphates, dissolution-precipitation equilibria and associated solubility product  $K_{s,app}^{CaP}$  at 37 and 80 °C. Acid-base equilibria and associated acidity constant  $K_{ai}$  and  $K_w$  at 37 and 80 °C.

Ca/P	Dissolution-precipitation equilibrium(a)	$K_{s,app}^{CaP}$ (37 °C)	<i>K</i> <sup>cap</sup> <sub>s, app</sub> (80 °C)	
1.67	$HAp_{(s)} = 5 Ca_{(aq)}^{2+} + 3 PO_{4(aq)}^{3-} + HO_{(aq)}^{-}$	$10^{-58.6}$	$10^{-59.6}$	
1.50	$ACP_{(s)} = 3 Ca_{(aq)}^{2+} + 2 PO_{4(aq)}^{3-}$	$10^{-29.5}$	$10^{-31.0}$	
1.33	$OCP_{(s)} = 4 Ca_{(aq)}^{2+} + 3 PO_{4(aq)}^{3-} + H_3O_{(aq)}^{+} + \frac{3}{2}H_2O_{(l)}$	$10^{-48.7}$	$10^{-49.5} - 10^{-52.5}$	
1.00	$DCPD_{(s)} = Ca_{(aq)}^{2+} + HPO_{4(aq)}^{2-} + 2 H_2O_{(l)}$	$10^{-6.56}$	$10^{-7.00}$	
1.00	$DCPA_{(s)} = Ca_{(aq)}^{2+} + HPO_{4(aq)}^{2-}$	$10^{-7.03}$	$10^{-7.40}$	
	Acid-base equilibrium(b)	$K_a (37  ^{\circ}\text{C})$ $K_a (80  ^{\circ}\text{C})$		
	$H_3PO_{4(aq)} + H_2O_{(l)} = H_2PO_{4(aq)}^- + H_3O_{(aq)}^+$	$K_{a1} = 10^{-2.20}$	$10^{-2.37}$	
	$H_2PO_{4(aq)}^- + H_2O_{(l)} = HPO_{4(aq)}^{2-} + H_3O_{(aq)}^+$	$K_{a2} = 10^{-7.17}$	$10^{-7.09}$	
	$HPO_{4(aq)}^{2-} + H_2O_{(l)} = PO_{4(aq)}^{3-} + H_3O_{(aq)}^{+}$	$K_{a3} = 10^{-12.3}$	$10^{-11.9}$	
	$2 H_2 O_{(l)} = H_3 O_{(aq)}^+ + HO_{(aq)}^-$	$K_w = 10^{-13.6}$	10 <sup>-12.6</sup>	

<sup>&</sup>lt;sup>(a)</sup> Data on CaPs dissolution-precipitation equilibria were taken and/or extrapolated from the literature. <sup>44–47</sup> Note that, for ACP, the thermodynamic data used in this study are those associated with the β–Ca<sub>3</sub>(PO<sub>4</sub>)<sub>2</sub> mineral having a close structure. <sup>48</sup>

#### I. Thermodynamic modelling of the precipitation of calcium phosphates in open systems

In the present study, CaPs were synthesized using an automated reactor (Figure 1) allowing for the differentiation of two synthesis routes (Ca  $\rightarrow$  P and P  $\rightarrow$  Ca) with the gradual addition of a solution (containing Ca<sup>2+</sup> or PO<sub>4</sub><sup>3-</sup> ions) into the reactor containing that of the other precursor (see *Materials and Methods*). During the addition step, the two synthesis routes involved different common ion effect<sup>51</sup> and successive disruptions of equilibrium whose thermodynamic influence is

 $<sup>^{(</sup>b)}$  Data on the acid-base equilibria involving phosphate species at 80  $^{\circ}$ C were extrapolated from the study of Rard and Wolery,  $^{49}$  using the Ellingham approximation.  $^{50}$ 

reported in this paper. This experimental system has the major advantage of allowing the precipitation to proceed at constant pH during the addition step, which is not possible with those developed for vapor and gel diffusion syntheses involving a pH gradient out of thermodynamic equilibrium. This study thus describes a methodology based on aqueous solution chemistry principles to provide an in-depth understanding of the competitive precipitation of CaPs in open systems (dropwise synthesis). On the basis of thermodynamic modelling, the influence of various parameters such as the pH and the order of introduction of the precursor ions could be investigated.

#### I - 1. Thermodynamic differentiation in the Ca $\rightarrow$ P and P $\rightarrow$ Ca routes: a common ion effect

The impact of the order of introduction of the precursors on the precipitation processes and the structural properties of the final product was investigated. The present methodology consists first in predicting which CaP material is the thermodynamic product at the beginning of the addition step (Figure 2).

A precipitate forms spontaneously in a solution at a given temperature T if this solution becomes rich enough in precursor ions to satisfy the supersaturation criterion  $S \ge 1$ , also expressed as  $Q \ge K_s$  (thermodynamic driving force, S5), where Q is the quotient of the precipitation reaction characteristic of the considered solution and  $K_s$  is the solubility constant of the phase prone to precipitate at T. In that respect, at the beginning of the addition stage in the  $Ca \to P$  synthesis, the minimum concentration of calcium ions  $[Ca^{2+}]_{min}$ , to be introduced into the reactor containing the phosphate solution, to precipitate the HAp mineral should verify the equation:

$$Q = a(\text{Ca}^{2+})_{min}^{5} a(\text{PO}_{4}^{3-})_{r}^{3} a(\text{HO}^{-})_{r} = \gamma_{2\pm}^{5} \gamma_{3\pm}^{3} \gamma_{\pm} \frac{[\text{Ca}^{2+}]_{min}^{5} [\text{PO}_{4}^{3-}]_{r}^{3} [\text{HO}^{-}]_{r}}{(c^{\circ})^{9}} = K_{s}^{HAp} \quad \text{(Eq. 1)}$$

with  $\gamma_{\pm}$ ,  $\gamma_{2\pm}$  and  $\gamma_{3\pm}$  the activity coefficients associated with HO<sup>-</sup>, Ca<sup>2+</sup> and PO<sub>4</sub><sup>3-</sup> ions calculated from the Davies equation deriving from the Debye-Hückel theory,  $[HO^-]_r$  and  $[PO_4^{3-}]_r$  the initial concentrations of hydroxide and phosphate ions present in the reactor and  $c^{\circ}$  the standard concentration of 1 mol/L. The HAp material spontaneously precipitates if the concentration  $[Ca^{2+}]_{min}$  introduced into the reactor is:

$$[Ca^{2+}]_{min} = \frac{1}{\gamma_{2\pm}} \left( \frac{K_s^{HAp}(c^{\circ})^9}{\gamma_{3+}^3 [PO_4^{3-}]_r^3 \gamma_{\pm} [HO^{-}]_r} \right)^{\frac{1}{5}}$$
(Eq. 2)

The  $[HO^-]_r$ ,  $[PO_4^{3-}]_r$ ,  $\gamma_{\pm}$ ,  $\gamma_{2\pm}$  and  $\gamma_{3\pm}$  parameters depend on the pH of the solution, which is maintained constant in the reactor in the present study. Eq. 2 indicates that the more alkaline the medium, the lower the minimum concentration of calcium ions to be introduced for precipitating the HAp material, as also illustrated in Figure 2 A.

Similarly, at the beginning of the P  $\rightarrow$  Ca synthesis, the minimum concentration of phosphate species  $[P]_{min} = [H_3PO_4]_{min} + [H_2PO_4^-]_{min} + [HPO_4^{2-}]_{min} + [PO_4^{3-}]_{min}$  to be introduced into the reactor to precipitate HAp has to verify:

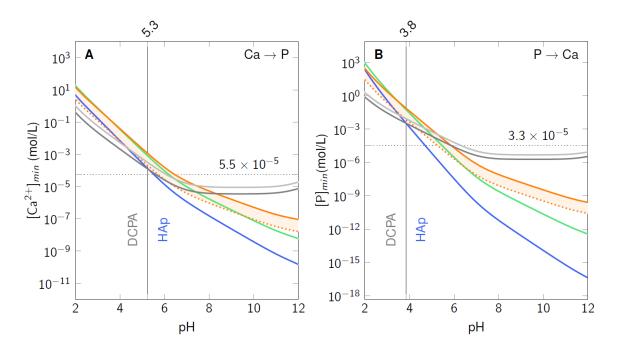
$$Q = a(\text{Ca}^{2+})_r^5 a(\text{PO}_4^{3-})_{min}^3 a(\text{OH}^-)_r = \gamma_{2\pm}^5 \gamma_{3\pm}^3 \gamma_{\pm} \frac{[\text{Ca}^{2+}]_r^5 [\text{PO}_4^{3-}]_{min}^3 [\text{OH}^-]_r}{(c^{\circ})^9} = K_s^{HAp} \quad (\text{Eq. 3})$$

Hence, the HAp mineral precipitates spontaneously if the concentration in phosphate species introduced into the reactor is at least:

$$[P]_{min} = \frac{1}{n\gamma_{3\pm}} \left( \frac{K_s^{HAp}(c^{\circ})^9}{\gamma_{2\pm}^5 [Ca^{2+}]_r^5 \gamma_{\pm} [OH^{-}]_r} \right)^{\frac{1}{3}}$$
 (Eq. 4)

with n the molar fraction in phosphate ions.

The same thermodynamic approach was applied to predict the conditions for the precipitation of all of the CaPs materials (S5). The pH dependence of the corresponding calculated  $[Ca^{2+}]_{min}$  and  $[P]_{min}$  concentrations is plotted in Figure 2 for a reaction temperature of 80 °C. Under given conditions, the most stable CaP material is the one requiring the lowest precursor concentration to spontaneously precipitate. The HAp crystalline phase, constituted of phosphate groups, appears to be the thermodynamically-predicted material to be formed at the beginning of the  $Ca \rightarrow P$  (Figure 2 A) and  $P \rightarrow Ca$  (Figure 2 B) synthesis routes carried out in basic, neutral or slightly acidic media. During the addition step proceeding in a more acidic media, the DCPA material, consisting of hydrogen phosphate groups, is predicted to be the most stable material initially formed in aqueous solution at 80 °C. Interestingly, the modelling of the competitive CaPs precipitation predicts thermodynamic differentiation at the beginning of the  $Ca \rightarrow P$  and  $Ca \rightarrow P$  and  $Ca \rightarrow P$  ca synthesis routes as the DCPA/HAp stability frontier is shifted by 1.5 units of pH, decreasing from 5.3 to 3.8, respectively (Figure 2 A, B). Such an influence of the order of introduction of the precursors under acidic conditions has never been reported in earlier literature studies to the best of our knowledge.

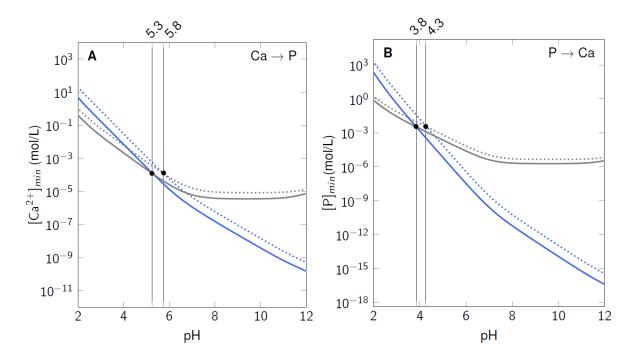


**Figure 2.** Thermodynamic traces predicting the minimum concentration in precursor to be initially introduced into the reactor for both Ca  $\rightarrow$  P (**A**) and P  $\rightarrow$  Ca (**B**) routes in order to precipitate the different calcium phosphates: HAp (blue), ACP (green), OCP (orange), DCPD (light grey) and DCPA (dark grey) at 80 °C (modelling parameters: [P]<sub>r</sub> = 0.13 mol/L, and [Ca<sup>2+</sup>]<sub>r</sub> = 0.22 mol/L). For OCP, due to the important uncertainty on its solubility product at 80 °C (Table 1), the corresponding domain is highlighted in orange. Specified concentrations of [Ca<sup>2+</sup>] = 5.5 × 10<sup>-5</sup> mol/L and [P] = 3.3 × 10<sup>-5</sup> mol/L are those estimated in the reactor after one drop of the corresponding precursor is added (see *Materials and Methods*). [Ca<sup>2+</sup>]<sub>min</sub> or [P]<sub>min</sub> are the minimum concentrations in calcium ions or phosphate species to be introduced into the reactor containing the phosphate or the calcium solution to precipitate the CaP minerals, respectively.

#### I - 2. Influence of the temperature

The influence of the temperature on the formation of CaPs in aqueous solution was studied by plotting the thermodynamic traces  $[Ca^{2+}]_{min} = f(pH)$  and  $[P]_{min} = f(pH)$  calculated at 37 °C (Figure S5A) and 80 °C (Figure 2 A, B). It appears that the reaction temperature has very little influence on the stability of the calcium phosphates, and the HAp and DCPA materials remain the thermodynamic products as a function of the pH at physiological temperature (Figure S5A). A decrease in temperature from 80 to 37 °C leads to a slight increase in the  $K_s^{CaP}$  values (Table 1),

which translates in the need for the addition of slightly higher amounts of precursor ions to spontaneously precipitate both HAp and DCPA minerals (Eq. 2, Eq. 4 and Figure 3). In line with the conclusions of Vereecke and Lemaître,<sup>32</sup> a lower reaction temperature is found to be favourable to the thermodynamic stability of the DCPA material over a moderately wider range of pH with an increase in this parameter by about 0.5 unit for both synthesis routes.



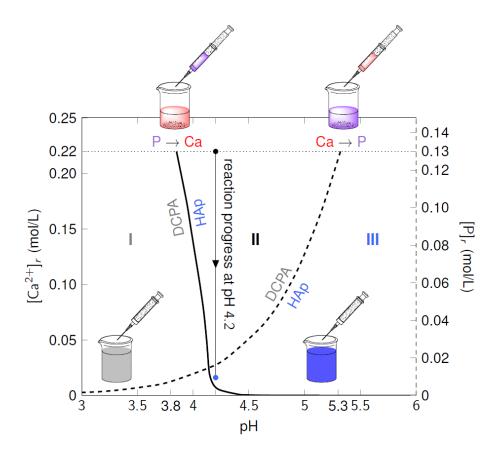
**Figure 3.** Influence of temperature on the competitive precipitation of HAp (blue) and DCPA (dark grey) at the beginning of the Ca  $\rightarrow$  P (**A**) and P  $\rightarrow$  Ca (**B**) synthesis routes. The DCPA/HAp stability frontiers at 80 (solid lines) and 37 °C (dotted lines) are indicated by vertical lines and black dots.  $[Ca^{2+}]_{min}$  or  $[P]_{min}$  are the minimum concentrations in calcium ions or phosphate species to be introduced into the reactor containing the phosphate or the calcium solution to precipitate the HAp (blue) and DCPA (dark grey) minerals, respectively.

### I - 3. Successive disruptions of equilibrium during the addition step: influence of the $\lceil Ca^{2+} \rceil / \lceil P \rceil$ ratio

At a given pH, the competitive precipitation of CaPs in P-rich (Ca  $\rightarrow$  P route) or Ca-rich (P  $\rightarrow$  Ca route) solutions has to deal with different CaPs stabilities (Figure 2). In particular, the HAp mineral is thermodynamically stable over a wider range of pH when  $\ddagger$  precipitated in a Ca-rich solution (Figure 2 **B**) rather than in a P-rich one (Figure 2 **A**) in agreement with its high intrinsic Ca/P ratio making it the most calcium enriched CaP mineral (Table 1).

During the addition step, the  $[Ca^{2+}]_r$  and  $[P]_r$  concentrations in the reactor decrease due to both the precipitation of CaPs and the dilution of the reaction medium. The pH stability frontier between HAp and DCPA appears to be more or less shifted during that step, depending on the selected synthesis route (Figure 4). Such a pH shift can be calculated using the same methodology as that used for the prediction of the DCPA/HAp stability frontier at the beginning of the reaction by considering the corresponding decrease in the  $[Ca^{2+}]_r$  and  $[P]_r$  concentrations in the reactor, in other words, by taking into account the progress of the precipitation reaction. Considering the Ca → P route at 80 °C, the pH stability frontier between the HAp and DCPA materials, initially at pH 5.3 (Figure 2 A), progressively shifts to lower pH as the addition step progresses and the [P]<sub>r</sub> concentration decreases in the reactor (Figure 4, dashed line). Within this synthesis route, the stability domains of HAp and DCPA are on the right-hand side (Figure 4, region III) and the lefthand side (Figure 4, regions I and II) of the dashed trace, respectively. A reverse tendency is observed for the  $P \rightarrow Ca$  route regarding the impact of the addition step on the position of the stability frontier moving to higher pH during the progress of the precipitation reaction (Figure 4, solid line). Regions I and III in Figure 4 define experimental conditions boundaries for which the

order of introduction of the precursor ions has no influence on the nature of the predicted thermodynamic product, *i.e.* DCPA (region I) or HAp (region III). In contrast, a thermodynamic differentiation should occur in region II for which HAp is predicted to thermodynamically precipitate first in the  $P \rightarrow Ca$  route (right-hand side of the solid line), whereas DCPA is expected as the thermodynamic product at the beginning of the  $Ca \rightarrow P$  route (left-hand side of the dashed line). The present identification of such a thermodynamic differentiation represents an important breakthrough compared to the earlier thermodynamic modelling reported in the literature to date that was traditionally based upon the solubility isotherms. <sup>31,32</sup> The thermodynamic differentiation between both synthesis routes is observed from pH 3.8 to 5.3 at the beginning of the reaction and gradually diminishes with the precipitation progress until disappearing at the end of the addition step, when the targeted final  $[Ca^{2+}]/[P]$  ratio is achieved in the reactor (Figure 2 A, B and Figure 4). When the precipitation is performed in region II of Figure 4, the nature of the precipitated nuclei is expected to transform from DCPA to HAp (hydrolysis, region II  $\rightarrow$  region III) or from HAp to DCPA (dehydration, region II  $\rightarrow$  region I) for the  $Ca \rightarrow P$  and  $P \rightarrow Ca$  routes, respectively.



**Figure 4.** Evolution of the DCPA/HAp stability frontier during the addition stage for the  $P \rightarrow Ca$  (solid line) and  $Ca \rightarrow P$  (dashed line) synthesis routes at 80 °C. Regions I and III correspond to the stability regions of DCPA and HAp for both synthesis routes, respectively. Region II is the thermodynamic differentiation region between the  $Ca \rightarrow P$  and the  $P \rightarrow Ca$  synthesis routes for which DCPA and HAp are respectively the most stable crystalline phases at the beginning of the addition step. Note the presence of a fourth region that is thermodynamically not accessible under the operating conditions used in the present study. The black and blue dots indicate respectively the beginning and the end of the addition step of syntheses carried out at constant pH 4.2 (vertical arrow), assuming that the thermodynamic equilibria are achieved as soon as the droplet is added. The horizontal dashed line highlights the calcium and phosphate species concentrations of the starting solutions in the reactor at the beginning of the syntheses. Note that the  $[Ca^{2+}]_r$  and  $[P]_r$  concentrations are different from 0 at the end of the addition stage at pH 4.2 because of the important solubility of hydroxyapatite under those conditions (Figure S5B).

According to the model proposed in the present study, in the case of thermodynamically-controlled experiments, OCP and DCPD materials are not expected to precipitate in aqueous solution by varying the order of introduction of the precursors, *i.e.* by playing with the common

ion effect. The formation of these minerals could thus result from non-thermodynamically-controlled conditions, as detailed in a forthcoming paper, or from the modification of the experimental conditions. As the dissociative power of the solvent greatly affects the solubility of ionic solids such as DCPD,<sup>52</sup> the use of hydroalcoholic or organic media appears, for instance, as an alternative solution to make the OCP and DCPD materials stable in solution. However, the lack of thermodynamic data on the solubility of CaPs in such environments makes the understanding of the CaPs competitive precipitation difficult.

## II. Experimental validation of the predictive thermodynamic model of the competitive precipitation of CaPs

CaPs were precipitated at constant pH and under thermodynamic control (Table 2 and S1 *Synthesis pH profile*) to assess of the reliability of the predictive capabilities of the thermodynamic model adapted for open systems (Section I). Based on this model, three stationary pH conditions were considered regarding both synthesis routes. At pH 6.5 and 9.0, the HAp material is expected to be the thermodynamic product regardless of the synthesis route (Figure 4, region III). In region II of Figure 4, a thermodynamic differentiation is anticipated with the precipitation of the DCPA and HAp minerals in the  $Ca \rightarrow P$  and  $P \rightarrow Ca$  routes, respectively. To promote a thermodynamic control, the precipitation reactions were carried out at a moderate temperature of 80 °C and over a relatively long period of time (addition rate: 2.2 mL/min, addition stage  $\approx 1$  h 30, optional additional maturation step of 2 h). As the present study may also have an impact in the field of biomineralization, two syntheses were performed at the physiological temperature of 37 °C. The influence of the final  $[Ca^{2+}]/[P]$  ratio of precursor ions introduced into the reactor was

investigated at pH 6.5 in the Ca  $\rightarrow$  P route as the formation of HAp exhibiting a Ca/P ratio lower than 1.67 is expected under acidic conditions.<sup>53,54</sup>

**Table 2.** Experimental conditions applied in the synthesis of calcium phosphates, crystalline phases identified by XRD, Ca/P molar ratios determined by XRF and specific surface areas (SSA) of synthesized materials. All of the experiments were performed with an addition rate of the corresponding precursor solution of 2.2 mL/min.

Sample	nII	Synthesis	Final	Maturation	Crystalline	Ca/P	SSA
reference	pН	route	$[Ca^{2+}]/[P]$	time (h)	phase		$(m^2/g)$
1	9.0	$P \rightarrow Ca$	1.67	2	HAp	1.72	39
2			1.67	_	HAp	1.71	38
3		Ca → P	1.67	_	HAp	1.66	102
4 (37 °C)			1.67	2	HAp	1.65	195
5	<i>(</i> <b>.</b>	P → Ca	1.67	_	HAp	1.63	15
6 (37 °C)			1.67	_	HAp	1.52	122
7	6.5	Ca → P	_ 1.67	_	HAp	1.53	44
8			1.50	_	HAp	1.53	43
9	4.2	P → Ca	1.50	2 at pH 7.0	НАр	1.57	53
10			1.50	2	HAp/OCP	1.44	45
11			1.67	_	HAp/OCP	1.43	55
12		Ca → P	1.67	_	DCPA	1.04	< 2
13			1.67	15	HAp/DCPA		

#### II - 1. Nature and properties of the CaPs synthesized under thermodynamic control

#### II - 1.1. Samples prepared at pH 6. 5 and 9. 0 (Figure 4, region III)

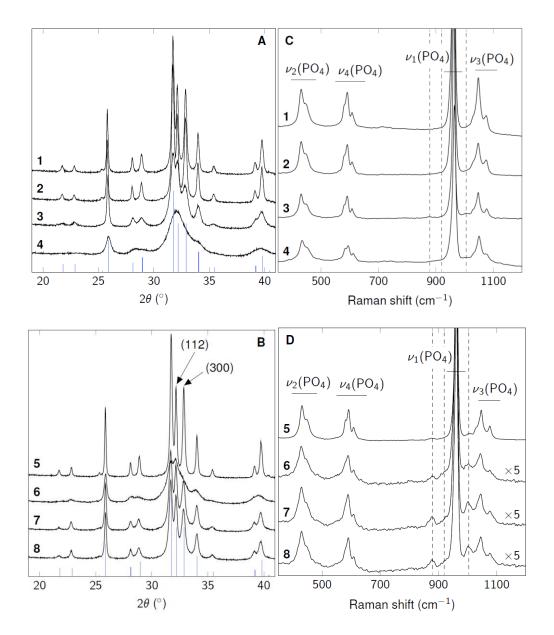
**Hydroxyapatite precipitation.** The predictive model accounts for HAp precipitation at stationary pH 6.5 and 9.0 without any thermodynamic differentiation (Figure 4, region III). In accordance with this prediction, all of the samples prepared at such pH (Table 2, samples 1 to 8) exhibit XRD patterns that perfectly match that of the pure HAp crystalline phase (Figure 5 **A**, **B**, S6 and Table S2A). However, the crystallinity of the HAp phase appears to be surprisingly impacted by the order of introduction of the precursor ions. A close examination of the full width

at half maximum of the diffraction lines provides evidences for a slight loss in HAp crystallinity for the samples obtained following the  $Ca \rightarrow P$  route compared to those prepared by the  $P \rightarrow Ca$ pathway (Figure 5 A, B, sample 7 vs sample 5, and sample 3 vs sample 2). This loss can be inferred to a lowering in the size of the crystallites, as supported by the significant changes in the specific surface area (Table 2). Indeed, by simply switching the order of introduction of the ion precursors from P  $\rightarrow$  Ca to Ca  $\rightarrow$  P, the SSA increases from 38 (sample 2) to 102 m<sup>2</sup>/g (sample 3) for materials prepared at pH 9.0 and from 15 (sample 5) to 44 m<sup>2</sup>/g (sample 7) for those synthesized at pH 6.5 (Table 2). Similarly, the particularly wide diffraction lines observed for samples 4 and 6, formed at physiological temperature (Figure 5 A, B), is attributed to barely crystallized particles. Consequently, an increase in the precipitation temperature from 37 up to 80 °C results in a decrease in the SSA of the materials ranging from 195 to 102 m<sup>2</sup>/g for those obtained following the Ca  $\rightarrow$  P route at pH 9.0 (Table 2, samples 4 and 3) and from 122 to 15 m<sup>2</sup>/g for those prepared with the  $P \rightarrow Ca$  route at pH 6.5 (Table 2, samples 6 and 5). Thus, for the same synthesis duration, the use of a higher reaction temperature promotes a thermodynamic control of the precipitation process that logically tends to bring the system to a stable state more rapidly, i.e. more crystallized in the case of HAp precipitation.<sup>55</sup>

**Hydroxyapatite stoichiometry.** The HAp materials prepared at pH 6.5 and 9.0 exhibit different Ca/P ratios ranging from 1.52 to 1.72 (Table 2, samples 1 to 8). Both the pH and the order of introduction of the precursor ions are shown to influence this ratio with the formation of substoichiometric materials (Ca/P < 1.67) at pH 6.5. A simplified formula  $Ca_{10-x}(HPO_4)_x(PO_4)_{6-x}(OH)_{2-x} \cdot nH_2O$  was reported for such minerals, also referred to as calcium-deficient hydroxyapatite (CDHAp, x = 0–1 and Ca/P = 1.67–1.50). The presence of hydrogen phosphate groups  $HPO_4^{2-}$  as crystallographic defects is responsible for the Raman bands

at 879, 920 and 1011 cm<sup>-1</sup> assigned to [P—OH] and  $\nu_1$  (HPO<sub>4</sub>) stretching modes (Table S2B). The related bands are clearly observed for the HAp materials prepared at pH 6.5 (Table 2 and Figure 5 **D**), under which the CaHPO<sub>4(aq)</sub> species in solution, is the predominant ion pair in the reactor compared to  $CaH_2PO_{4(aq)}^+$  and  $CaPO_{4(aq)}^-$ . 57,58 As expected, a lowering in the Ca/P ratio is correlated with an increase in the intensity of these Raman contributions and makes it possible to identify the formation of CDHAp at pH 6.5. On the contrary, such defects are extremely limited in HAp precipitated at pH 9.0 (Figure 5 C), as such an alkaline pH favours the stability of CaPO<sub>4(aq)</sub> species in solution compared to that of CaHPO<sub>4(aq)</sub> and CaH<sub>2</sub>PO<sup>+</sup><sub>4(aq)</sub>. <sup>58</sup> Interestingly, at pH 6.5, the  $P \rightarrow Ca$  route (Table 2, sample 5) appears to be less favourable to the incorporation of such HPO<sub>4</sub><sup>2-</sup> defects (Ca/P ratio closer to 1.67) in the HAp structure than the Ca  $\rightarrow$  P route (Table 2, sample 7), suggesting differences in the corresponding precipitation pathways. Given the substoichiometry observed for the samples prepared at pH 6.5, the influence of the final [Ca<sup>2+</sup>]/[P] ratio of precursors ions introduced into the reactor during the addition step was investigated by decreasing its value from 1.67 (sample 7) to 1.50 (sample 8). Neither the XRD patterns nor the Raman spectra and the XRF quantification show any distinguishable structural modification between these two HAp samples (Figure 5 B, D and Table S2), which demonstrates that the final [Ca<sup>2+</sup>]/[P] ratio in the reactor does not drive the stoichiometry of the formed HAp materials. In contrast, the relative intensity between the diffraction lines at 32.20 and 32.90 °, associated with the (112) and (300) diffraction planes, of samples 7 and 8 is different from that observed for the sample prepared by the  $P \rightarrow Ca$  route at the same pH (Figure 5 B and Table 2, sample 5). The existence of distinct microstructures, leading to different preferred orientations when preparing samples for XRD analysis, could be at the origin of this observation.<sup>59</sup> The higher

SSA of samples 7 and 8 compared to that of sample 5 (Table 2), also suggests differences in textural properties between these samples as shown in section II - 2.



**Figure 5.** Structural characterizations of the samples synthesized at pH 9.0 and 6.5 (Figure 4, region III). Diffraction patterns of samples 1, 2, 3 and 4 prepared at pH 9.0 (**A**) and samples 5, 6, 7 and 8 prepared at pH 6.5 (**B**). The corresponding positions of the diffraction lines expected for HAp are reported in blue (ICDD reference card n° 00-009-0432). Raman spectra of samples 1, 2, 3 and 4 synthesized at pH 9.0 (**C**) and samples 5, 6, 7 and 8 precipitated at pH 6.5 (**D**). In **C** and **D**, Raman contributions at 879, 920 and 1011 cm<sup>-1</sup>, characteristic of the presence of hydrogen phosphate groups in the samples (Table S2B), are indicated by vertical dashed lines.

#### II - 1.2. Samples prepared at pH 4. 2 (Figure 4, region II)

The proposed model predicts a DCPA/HAp thermodynamic differentiation region in between pH 3.8 and 5.3 (Figure 4, region II). In this region, the order of introduction of the precursor ions is expected to impact the thermodynamic nature of the formed nuclei depending on the progress of the precipitation reaction. To validate the existence of such a differentiation region, samples 11 and 12 were synthesized at pH 4.2 (Table 2). The recovered precipitates were rapidly filtered on Büchner and washed promptly with distilled water to avoid denaturing the isolated CaPs materials (S7). The crystalline phase composition of these samples is found to be different, as shown by the comparison of their XRD patterns and Raman spectra (Figure 6), demonstrating the importance of a different common ion effect taking place during the addition steps of the  $Ca \rightarrow P$  (phosphate type) and  $P \rightarrow Ca$  (calcium type) synthesis routes. In agreement with the thermodynamic predictions (Figure 4), the  $Ca \rightarrow P$  route leads to the formation of DCPA, as supported by the corresponding diffraction pattern (Figure 6 A, sample 12) and the presence of the characteristic vibrational bands at 903 and 988 cm<sup>-1</sup> attributed to  $\nu_1(HPO_4)$  and  $\nu(P-OH)$  vibration modes of hydrogen phosphate groups present in this mineral, respectively (Figure 6 C and Table S2B). Under such experimental conditions, DCPA is the thermodynamic product during a large part of the addition step as the phosphate type common ion effect is favourable to the stabilization of this material having the lowest Ca/P ratio of the CaPs (Figure 4, region II). At the end of the addition step, when the common ion effect switches to calcium-type, HAp becomes the thermodynamic product (Figure 4, region III). Nevertheless, in sample 12, no trace of HAp could be observed experimentally under the present conditions due to the unfavourable kinetics of DCPA hydrolysis into HAp under acidic conditions<sup>60</sup>, as indicated in Eq. 5:

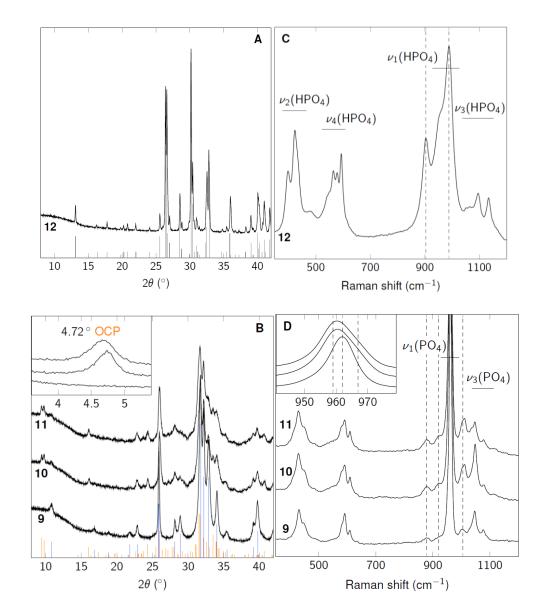
$$Ca_6(HPO_4)_{6(s)} + (4-x) Ca_{(aq)}^{2+} + (8-2x) HO_{(aq)}^{-} + n H_2O_{(l)} =$$

$$Ca_{10-x}(HPO_4)_x(PO_4)_{6-x}(OH)_{2-x} \cdot nH_2O_{(s)}$$
 (Eq. 5)

Such a conclusion is supported by the experiment for which the precipitate was matured during 15 h (Table 2, sample 13). This maturation time allowed the system to approach its state of equilibrium (Figure 4, blue dot) by partial hydrolysis of the DCPA to HAp (S8). The presence of DCPA as a unique material in sample 12 indicates that the stable DCPA crystals formed at the beginning of the addition step could be preserved in a metastable way at the end of the addition step, as the establishment of the final equilibrium state is not reached rapidly due to hydrolysis kinetic limitations (dissolution-reprecipitation mechanism). This shows that thermodynamic control is not achieved at the end of the addition step at pH 4.2 in the Ca  $\rightarrow$  P route. In contrast, during the  $P \rightarrow Ca$  route at pH 4.2, HAp is predicted to be formed over the whole experiment (Figure 4, region II). The XRD pattern of sample 11 did not show the presence of DCPA but mainly that of HAp (Figure 6 B). In addition, low-angle diffraction lines are observed at 4.72, 9.44 and 9.77 ° (Figure 6 **B**) that are not indexed to the HAp crystalline structure (ICDD reference card n° 00-009-0432). Despite close crystallographic structures between the HAp and OCP minerals, these three low-angle diffraction lines are characteristic of the (001), (020), and (110) diffraction planes of OCP, respectively, whereas that at 10.8 ° corresponds to the (100) plane of the HAp structure (Figure 6 B and Table S1A). The presence of OCP is also confirmed by Raman spectroscopy as shown in the magnification of the  $v_1(PO_4)$  spectral region reported in the inset of Figure 6 **D**. Compared to the Raman spectrum of sample 9, composed of the pure CDHAp phase, sample 11 exhibits a slight shift of the maximum of the  $v_1(PO_4)$  vibrational band to lower wavenumbers and a shoulder at higher wavenumbers, which are both characteristic of the presence of OCP in mixture with CDHAp.<sup>21</sup> In line with such a composition, sample 11 exhibits a Ca/P ratio of 1.43 (Table 2), lying between those of OCP (1.33) and CDHAp (1.67–1.50). If the HAp

mineral was predicted as the thermodynamic product precipitating in the P  $\rightarrow$  Ca route at pH 4.2 (Figure 4, region II), this is not the case for OCP, which has been reported to form in non-basic media in earlier studies.<sup>26,61</sup>

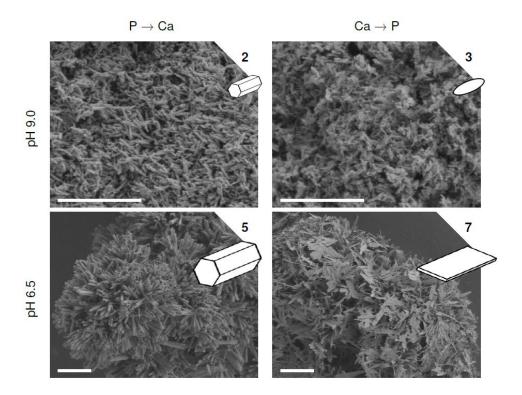
The present results put particular emphasis on the importance of the introduction order of the precursor ions on the nature of the precipitated materials at pH 4.2. To a large extent, the observed differences can be explained reliably from the thermodynamic methodology developed in this study apart from the formation of the OCP phase which is discussed in the section II - 2. The proposed thermodynamic model appears to be more appropriate to account for the competitive precipitation of CaPs under realistic experimental conditions (dropwise synthesis) compared to the earlier approach based on the solubility isotherms.<sup>31,32</sup>



**Figure 6.** Structural characterization of the samples synthesized at pH 4.2 (Figure 4, region II). Diffraction patterns of sample 12 obtained via the Ca  $\rightarrow$  P route (**A**) and samples 9, 10 and 11 precipitated following the P  $\rightarrow$  Ca route with corresponding low-angle pattern in the inset (**B**). The position of the diffraction lines expected for DCPA (ICDD reference card n° 00-009-0080), HAp (ICDD reference card n° 00-009-0432) and OCP (ICDD reference card n° 00-026-1056) are reported in grey, blue and orange, respectively. Raman spectra of sample 12 (**C**) and samples 9, 10 and 11 with magnification between 940 and 980 cm<sup>-1</sup> in the inset (**D**). Vertical dashed lines highlight the Raman shifts at 903 and 988 cm<sup>-1</sup> ( $\nu_1$ (HPO<sub>4</sub>) and  $\nu$ (P—OH) vibration modes in hydrogen phosphate groups of DCPA), at 879, 920 and 1011 cm<sup>-1</sup> (characteristic of the presence of hydrogen phosphate groups in both HAp and OCP), at 959 and 967 cm<sup>-1</sup> (typical of OCP) and at 962 cm<sup>-1</sup> (typical of HAp) (Table S2B).

### II - 2. Morphologies, Oswald ripening control and incorporation of defects as a function of pH: towards the limits of the predictive precipitation thermodynamic model

The quite soft chemistry conditions used in the present study result in HAp materials with very different particle sizes and surface areas ranging from 15 to 195 m<sup>2</sup>/g (Table 2). This may be of particular interest in the field of heterogeneous catalysis, which is known to be surface dependent.<sup>8,62</sup> In addition to these differences, SEM characterization of the prepared samples showed the existence of two distinct morphologies associated with rod-like and platelet/ribbonlike crystallites (Figure 7 and Figure S9). Given the hexagonal symmetry of HAp, the typical rodlike morphology with a hexagonal section (S10), observed for the materials prepared with the P  $\rightarrow$ Ca route at pH 6.5 (Figure 7, sample 5) and 9.0 (Figure 7, sample 2), was expected, even though the pH of the precipitation medium still strongly influences the particle size distribution and the associated SSA of the prepared samples (Table 2). The order of introduction of the precursor ions into the reactor is found to play a key role in controlling the HAp morphology. At pH 6.5, the P  $\rightarrow$ Ca route led to rod-like crystallites (Figure 7, sample 5) and the Ca → P route to platelet/ribbonlike ones (Figure 7, sample 7). The platelet/ribbon morphology obtained in a slightly acidic medium has already been reported in earlier studies using (i) additives that modified both the ionic precursors speciation and the growth processes, (ii) electrochemical (energy consuming) protocols or (iii) biomimetic synthesis operated at physiological pH and temperature. 63-65 As another common feature with the biomimetic materials, the HAp platelet-like materials were found to be calcium-deficient with a Ca/P ratio of 1.53 (Table 2, samples 7 and 8) due to the presence of hydrogen phosphate groups (Figure 5 **D**). Samples 7, 8, 9 and 10, showing platelet/ribbon-like particles (Figures 7 and 8), exhibit specific surface areas (43–53 m<sup>2</sup>/g, Table 2) comparable to those measured for the rod-like crystallites of samples 1 and 2 (Table 2, Figure 7). The origin of such differences in shape and particle size of the precipitated CaPs likely result from different reaction pathways related to their kinetics of nucleation/growth/solid-solid phase transformation, which will be detailed in a forthcoming paper. It remains possible to play on the thermodynamic processes to tune the HAp morphology by setting up a maturation step at stationary pH (see Materials and Methods). Due to the dropwise synthesis mode, the precipitates formed during the addition step do not remain in the reactor for the same duration, leading to a heterogeneous crystallite size distribution. The maturation step should favour the Oswald ripening process, which accounts for the change of an inhomogeneous structure over time, i.e. the dissolution of small crystallites and the growth of larger ones due to the redeposition of dissolved species.<sup>66</sup> Yet this phenomenon is disfavoured in an alkaline medium as CaP crystallites were shown to be poorly soluble in such an environment (Figures S4A and S4B). At pH 9.0, the addition of a 2-hour maturation step in the  $P \rightarrow Ca$  route (sample 1) did not lead to significant changes in the HAp crystallite size as attested by the similar SSA of samples 1 and 2 (38–39 m<sup>2</sup>/g, Table 2). In contrast, the addition of such a maturation step at the end of the  $P \rightarrow Ca$  protocol for the syntheses carried out at pH 4.2 was found to bring an increase in the size of the crystallites and a decrease in the SSA of the prepared materials by about 18 % (Figure 6 **B** and Table 2, sample 11: 55 m<sup>2</sup>/g, sample 10: 45 m<sup>2</sup>/g). Such a finding is in accordance with the high apparent solubility of both HAp and OCP in these acidic conditions (Figure S4A).



**Figure 7.** Impact of the P  $\rightarrow$  Ca or Ca  $\rightarrow$  P route on the morphology (rod-like or platelet/ribbon-like, respectively) of the HAp crystallites prepared at pH 9.0 (samples 2 and 3) or 6.5 (samples 5 and 7) as investigated by scanning electron microscopy. The numbers indicated on the top right corners of the images refer to as the sample references listed in Table 2. Scale bars: 2  $\mu$ m.

Based on these results, a  $P \rightarrow Ca$  synthesis was carried out at pH 4.2 and the resulting particles were matured in a neutral medium (Table 2, sample 9). Such a maturation step is expected to be favourable to the hydrolysis of OCP to CDHAp:

$$\operatorname{Ca}_{8}(\operatorname{HPO}_{4})_{2}(\operatorname{PO}_{4})_{4} \cdot \operatorname{5H}_{2}\operatorname{O}_{(s)} + (2-x)\operatorname{Ca}_{(aq)}^{2+} + (4-2x)\operatorname{HO}_{(aq)}^{-} =$$

$$\operatorname{Ca}_{10-x}(\operatorname{HPO}_{4})_{x}(\operatorname{PO}_{4})_{6-x}(\operatorname{OH})_{2-x} \cdot n\operatorname{H}_{2}\operatorname{O}_{(s)} + (7-x-n)\operatorname{H}_{2}\operatorname{O}_{(l)}$$
(Eq. 6)

but to be unfavourable to the Oswald ripening process.<sup>67,68</sup> This protocol allowed the preparation of a pure CDHAp material (Figure 6 **B**, **D**) for which the characteristic SSA of about 53 m<sup>2</sup>/g and the platelet/ribbon-like morphology of the OCP/HAp mixture obtained at pH 4.2 and 80 °C without maturation step (Table 2, sample 11 and Figure 8) were preserved. The conservation of

the OCP platelet/ribbon-like morphology during the OCP hydrolysis into CDHAp indicates the involvement of a solid-solid phase transformation, controlled by diffusion, as proposed by Tseng *et al.*<sup>67</sup>

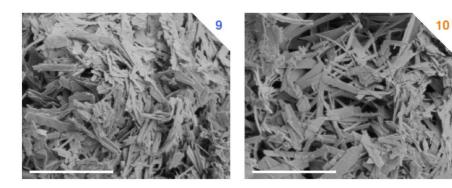


Figure 8. Preservation of the morphology of OCP crystallites during their hydrolysis to HAp. Scanning electron microscopy images of samples 9 (HAp) and 10 (partially hydrolysed OCP into CDHAp) presenting crystallites with a platelet/ribbon-like morphology obtained via the  $P \rightarrow Ca$  route at pH 4.2 and matured for 2 hours at pH 7.0 and 4.2, respectively. The numbers indicated on the top right corners of the images refer to as the sample references listed in Table 2. Scale bars: 2  $\mu m$ .

The present study therefore allowed the development of syntheses to tune (1) the morphology of the HAp material and (2) the incorporation of defects (hydrogen phosphate groups) into its crystalline structure. Surprisingly, not only the pH but also the order of introduction of the precursor ions into the reactor ( $P \rightarrow Ca$  or  $Ca \rightarrow P$ ) have a major impact on these two features. In particular, the order of introduction of the precursor ions exhibits the more pronounced influence on the final HAp morphology, while earlier literature studies in the field attributed the observed changes in the morphology of the CaP particles prepared under hydrothermal or physiological conditions to pH only.  $^{34,65,69}$  However, it must be stressed that the complete rationalization of the mineralisation processes of CaPs cannot be done exclusively on the basis of the proposed thermodynamic model. This model allows for the prediction of a final state of equilibrium from a

given initial state only and in no case of the reaction pathway leading to it indeed. Nevertheless, these pathways may also influence the final morphology of the thermodynamically-predicted CaPs and the incorporation of crystallographic defects. Finally, the present study also reveals that the formation of DCPA and HAp materials could involve reaction intermediates such as OCP, as supported by its identification in some of the experiments whose operating conditions did not allow for a complete thermodynamic control (Table 2, samples 10 and 11).

#### **CONCLUSION**

The results of the present study provide unique insights into the precipitation of CaPs at the thermodynamic level. The dropwise synthesis, involving successive equilibrium disruptions, allows to differentiate two synthesis routes (P  $\rightarrow$  Ca and Ca  $\rightarrow$  P), which led to the precipitation of materials with distinct structural properties. In particular, the synthesis procedures detailed in the present work constitute robust and simple protocols to obtain HAp particles with a wide variety of morphologies and quantity of hydrogen phosphate defects under soft chemistry conditions. The proposed thermodynamic model takes the influence of the pH, the temperature and the order of introduction of the precursors into account to understand the competitive precipitation of CaPs in open systems. DCPA was found to be obtained regardless of the route considered when operating at a pH lower than 3.8, whereas HAp is preferentially formed at a pH higher than 5.3. In between pH 3.8 and 5.3, a thermodynamic differentiation region was identified theoretically and validated experimentally. This differentiation phenomenon sheds light on the importance of the common ion effect on the comparative stability of the DCPA and HAp crystalline phases. In the HAp pHstability region, the use of the  $P \rightarrow Ca$  route and/or basic conditions favour the precipitation of weakly defective materials, while the Ca → P pathway and/or more acidic conditions favor the

incorporation of hydrogen phosphate defects. At a given pH, the order of introduction of the precursor ions appears from the present study to be a major parameter to control the morphology of the HAp from platelets/ribbons to rods. The present work can therefore be used as a basis to help researchers select the appropriate synthesis procedure to prepare the targeted HAp materials of proper surface area (Table 2) and morphology (Figures 7 and 8). The developed approach could be adapted to the study of the competitive precipitation of other materials of major interest such as clays,<sup>69</sup> and magnesium silicates,<sup>70</sup> which are widely used as purifying agents or in the cosmetic field.

Under particular conditions, the proposed thermodynamic model was not able, however, to account for (i) the formation of OCP and (ii) the formation of singular platelet/ribbon-like crystallites with various defectiveness. These apparent discrepancies, likely related to kinetic aspects, will be discussed in a forthcoming article. It is anticipated that the combination of thermodynamic and kinetic approaches will further allow to propose complementary detailed precipitation mechanisms accounting for the possibility to obtain numerous HAp morphologies by modulating the pH and above all, the synthesis route used.

#### ASSOCIATED CONTENT

Supporting information on the synthesis and characterization of materials as well as mathematical details on the developed thermodynamic model are available free of charge via the Internet at

- S1 pH control
- S2 Indexation of the XRD lines and Raman bands of CaPs
- S3 XRF calibration traces

S4 - Apparent solubility of CaPs

S5 - Precipitation conditions of CaPs for the Ca  $\rightarrow$  P and P  $\rightarrow$  Ca synthesis routes

S6 - Low-angle diffractograms of HAp samples prepared at pH 6.5

S7 - DCPA hydrolysis into CDHAp during the washing step

S8 - Sample 13

S9 - High magnification images of rod-like and platelet/ribbon-like crystallites

S10 - Hexagonal section of samples 2 and 5

#### **AUTHOR CONTRIBUTIONS**

Study conception and design: COSTENTIN, REYNAUD and THOMAS

Acquisition of data: CASALE, NOWAK and REYNAUD

Analysis and interpretation of data: COSTENTIN, REYNAUD and THOMAS

Drafting of manuscript: REYNAUD

Critical revision: COSTENTIN and THOMAS

#### Conflicts of interest

There are no conflicts to declare.

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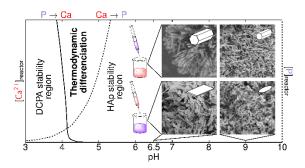
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Thermodynamics of the precipitation of calcium phosphates shows the importance of the pH medium and the order of introduction of the precursor ions on the textural (morphology, surface area) and structural (defects) properties of hydroxyapatites.

## **Supplementary Information for**

Development of a thermodynamic approach to assist the control of the precipitation of hydroxyapatites and associated calcium phosphates in open systems

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This document includes:

#### S1 - pH control 44

pH\_meter operation

Figure S1A

Synthesis pH profile

Figure S1B

Figure S1C

S2 - Indexation of the XRD lines and Raman bands of CaPs

Table S2A

Table S2B

**S3 - XRF calibration traces** 

Figure S3

**S4 - Apparent solubility of CaPs** 

Figure S4

#### S5 - Precipitation conditions of CaPs for the $Ca \rightarrow P$ and $P \rightarrow Ca$ synthesis routes

Figure S5A

Figure S5B

S6 - Low-angle diffractograms of HAp samples prepared at pH 6.5

Figure S6.

S7 - DCPA hydrolysis into CDHAp during the washing step

Figure S7

S8 - Sample 13 65

Figure S8

S9 - High magnification images of rod-like and platelet/ribbon-like crystallites

Figure S9

S10 - Hexagonal section of samples 2 and 5

Figure S10

### S1 - pH control

pH meter operation. The syntheses were carried out using an automated reactor controlling the pH during the addition step. In order to check the correct operation of the pH probe, the pH of the medium was modelled during the temperature rise step for the  $P \rightarrow Ca$  synthesis. The reactor initially at 20 °C contained, 200 mL of a calcium solution (0.22 mol/L) adjusted to about pH 10 by adding a few drops of concentrated ammonia ( $C_0 \approx 0.002$  mol/L in the reactor). This solution was heated up to 80 °C under stirring (400 rpm) and a significant decrease in pH of about 1.5 unit was observed (Figure S1A). The acid-base reaction which imposes the pH during this temperature rise step is:

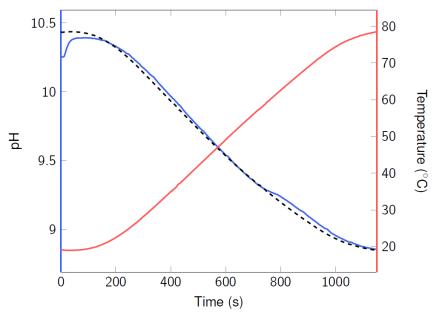
$$NH_{3(aq)} + H_2O_{(l)} = NH_{4(aq)}^+ + OH_{(aq)}^-$$
 (Eq. S1)

associated with the following equilibrium constant:

$$K^o = \frac{K_w}{K_a(\mathrm{NH_4^+/NH_3})}$$
 (Eq. S2)

with  $K_w$  the autoprotolysis equilibrium constant of water and  $K_a(NH_4^+/NH_3)$  the acidity constant associated with the  $NH_4^+/NH_3$  acid-base couple The pH of the medium can be expressed as a function of the equilibrium constants  $K_w$  and  $K_a(NH_4^+/NH_3)$  as well as the  $C_0$  ammonia

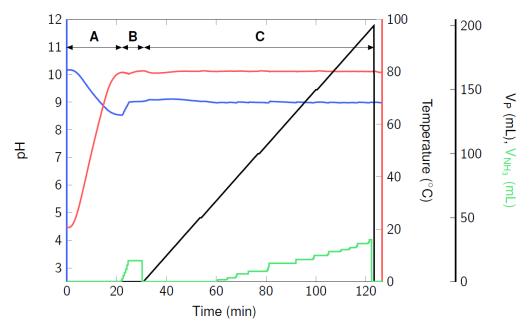
concentration in the reactor initially introduced at 20 °C. The change in pH during the temperature rise results from the temperature dependence of the equilibrium constants. <sup>1,2</sup> The theoretical model was found to fit reasonably well the experimental data, thus confirming the correct operation of the pH meter at 80 °C (Figure S1A).



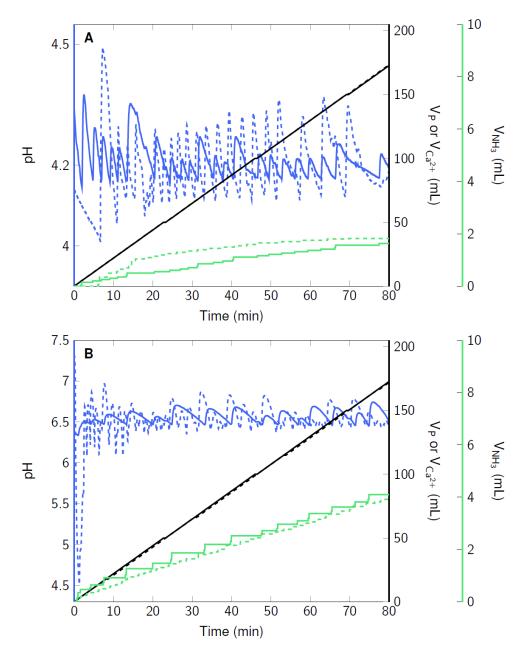
**Figure S1A.** Experimental (blue trace) and modelled (dashed black trace) pH evolution in the reactor during the temperature rise step from 20 to 80 °C (orange trace) (Table 2, sample 1).

Synthesis pH profile. The synthesis of calcium phosphates is particularly sensitive to the pH of the reaction medium. The automated reactor allows for the control of this parameter, as illustrated in Figure S1B showing a characteristic pH profile recorded for an experiment carried out at pH 9.0 and 80 °C. During the temperature rise (Figure S1B, step A), the pH was found to decrease because of the influence of the temperature on the acid base equilibrium (*pH meter operation* section). The pH was then rapidly readjusted to pH 9.0 by adding concentrated ammonia drop by drop (Figure S1B, step B). The calcium or phosphate solution is then gradually added into the reactor. The pH could be reasonably kept constant by the addition of concentrated ammonia controlled by the automated reactor (Figure S1B, step C and Figure S1C) in the course of the addition of the solutions of phosphate or calcium precursors. A white precipitate is instantaneously formed during that step as the first drops of solutions of precursors are added for all experiments performed at pH 9.0 and 6.5, whereas the formation of such a precipitate was slightly delayed for the syntheses performed at pH 4.2 (Table 2). These observations are in line with the minimal concentrations

[Ca<sup>2+</sup>]<sub>min</sub> and [P]<sub>min</sub> calculated from the methodology developed in this study implying that, at pH 4.2, the concentration of Ca<sup>2+</sup> (Ca  $\rightarrow$  P) or H<sub>x</sub>PO<sub>4</sub><sup>(3-x)-</sup> (P  $\rightarrow$  Ca) ions into the reactor by one drop, is not sufficient to induce the precipitation of any CaP contrary to what occurs for the syntheses performed at pH 6.5 and 9.0 (Figure 3 **A**, **B**). For the synthesis performed at pH 9.0 and 80 °C (Table 2, sample 1), once the precipitation step was completed, few amounts (approximately 1 mL) of ammonia were required to keep the pH constant due to the need in counterbalancing the gaseous NH<sub>3</sub> released during the maturation step.



**Figure S1B.** Temperature profile (red trace), pH profile (blue trace) and volumes added of phosphate precursor (black trace) and ammonia (green trace) solutions into the reactor in the case of a  $P \rightarrow Ca$  synthesis performed at 80 °C at pH 9.0 (Table 2, samples 2).



**Figure S1C.** Comparison of the pH stability (blue traces) in the reactor during the addition step at 80 °C and pH 4.2 (**A**) or 6.5 (**B**) between the P  $\rightarrow$  Ca (Table 2, sample 5, dashed line) and Ca  $\rightarrow$  P (Table 2, sample 7, solid line) synthesis routes. The pH control parameters are: mixing time = 12 s, minimum pH deviation = 0.02 and maximum pH rate = 0.3 pH unit/min. Note that the pH is easier to maintain constant when the calcium solution is added gradually into the reactor (Ca  $\rightarrow$  P route) compared to that of the phosphate solution (P  $\rightarrow$  Ca route).

### S2 - Indexation of the XRD lines and Raman bands of CaPs

Both XRD and Raman spectroscopy techniques were complementary used to identify the nature of calcium phosphate(s) obtained after washing/drying steps as well as their crystallinity and the origins of the substoichiometry of HAp (Tables S2A and S2B).

**Table S2A.** XRD assignment of the diffraction lines below 40  $^{\circ}$  for crystalline calcium phosphates: HAp (reference card  $n^{\circ}$  00-009-0432, ICDD), OCP (reference card  $n^{\circ}$  00-026-1056, ICDD), DCPD (reference card  $n^{\circ}$  00-009-0077) and DCPA (reference card  $n^{\circ}$  00-009-0080). The relative intensity (Int.) of each (h k l) diffraction line is indicated as a percentage of the most intense diffraction line.

	HAp			ОСР			DCPD			DCPA	
2 <i>θ</i> (°)	(hkl)	Int. (%)	2 <i>θ</i> (°)	(hkl)	Int. (%)	2 <i>θ</i> (°)	(hkl)	Int. (%)	2 <i>θ</i> (°)	(hkl)	Int. (%)
			4.72	(010)	100						
			9.44	(020)	15						
10.82	(100)	12	9.77	(110)	13	11.68	(020)	100			
			14.51	(1-20)	2				13.12	(010)	14
			16.04	(-101)	8						
			16.35	(021)	2				16.31	(100)	4
			17.00	(1-11)	1						
			17.37	(-111)	4	17.98	(-111)	2	17.76	(0-11)	4
			18.41	(1-31)	2						
18.78	(110)	4	18.84	(0-31)	2						
			18.99	(040)	1						
			19.65	(031)	3				19.80	(011)	2

			19.75	(-121)	3				20.26	(-101)	4
			20.67	(131)	2	20.93	(021)	100	20.79	(1-20)	4
21.82	(200)	10	21.60	(230)	2						
			22.67	(1-40)	5				22.04	(101)	4
22.90	(111)	10	22.91	(201)	4						
			23.01	(-201)	3						
			23.48	(041)	3	23.39	(040)	8			
			23.74	(221)	5	23.71	(130)	< 1			
			24.30	(-211)	10	24.50	(-131)	2	24.03	(1-21)	4
25.35	(201)	2	25.49	(231)	8				25.58	(-121)	14
25.88	(002)	40	25.87	(2-21)	17						
			26.00	(002)	20				26.43	(020)	70
			26.36	(-221)	6				26.59	(2-20)	75
			26.91	(-1-51)	8				26.75	(2-10)	16
			27.18	(1-50)	8				27.00	(002)	10
			27.78	(250)	3						
28.13	(102)	12	28.04	(241)	2						
			28.47	(-1-22)	5				28.49	(-1- 21)	20
28.97	(201)	18	28.61	(-112)	3				28.77	(0-12)	6
			29.21	(0-32)	5	29.26	(041)	75	29.90	(-221)	2

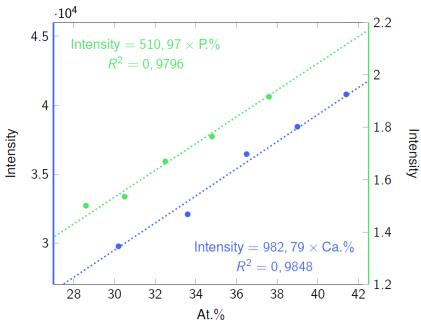
			29.60	(330)	3				30.19	(-112)	100
			30.31	(-122)	5				30.41	(-102)	35
			30.66	(-151)	4	30.51	(-221)	50	30.68	(2-11)	4
			31.10	(251)	10				31.02	(021)	8
			31.55	(260)	33	31.30	(-112)	10	31.17	(111)	4
31.77	(211)	100	31.70	(2-41)	32	31.97	(200)	2	31.44	(012)	2
32.20	(112)	60	32.18	(-1-42)	15				32.38	(1-30)	10
			32.59	(331)	12				32.48	(2-30)	20
32.90	(300)	60	33.06	(042)	8				32.89	(102)	35
			33.52	(070)	17	33.54	(150)	4			
34.05	(202)	25	33.97	(1-61)	12	33.82	(131)	4			
			34.24	(3-30)	7	34.15	(220)	50			
			34.38	(2-22)	7	34.43	(-202)	30			
			34.92	(-161)	5	35.11	(002)	4	34.73	(2-31)	4
35.48	(301)	6	35.25	(-1-71)	4	35.42	(060)	2	35.42	(-122)	4
			36.10	(-2-51)	2	35.60	(-132)	4	35.91	(0-22)	16
			36.27	(052)	3				36.06	(-1- 12)	2
			36.53	(1-70)	2	36.90	(-241)	14	36.76	(201)	2
			38.02	(180)	2	37.10	(022)	16	37.26	(-212)	2
			38.52	(271)	3				38.25	(2-22)	4

39.20	(212)	8	39.06	(-302)	2				39.04	(120)	10
			39.65	(361)	2				39.37	(2-12)	2
			39.76	(-1-81)	2	39.71	(061)	4			
39.82	(310)	20	39.89	(062)	2						

**Table S2B.** Raman shift (cm $^{-1}$ ) of PO $_4^{3-}$  and HPO $_4^{2-}$  groups of (CD)HAp, ACP, OCP, DCPD and DCPA reported in earlier works. $^{3-6}$ 

	(CD)HAp	ACP	ОСР	DCPD	DCPA	
$ u_3(PO_4) $ or		1118	1112 (HPO <sub>4</sub> <sup>2-</sup> hydrated layer)	1132 1119	1131 1094	
	1077 1064		$1079 (PO_4^{3-} \text{ and } HPO_4^{2-})$	1079 1061	1094	
$v_3(\text{HPO}_4)$ stretch	1057 1048 1041	1050	1052 (PO <sub>4</sub> <sup>3-</sup> ) 1048 (PO <sub>4</sub> <sup>3-</sup> )			
	1041 1034 1029		1036 (PO <sub>4</sub> <sup>3-</sup> ) 1027 (PO <sub>4</sub> <sup>3-</sup> )			
$ \nu_1(PO_4) $ or $ \nu_1(HPO_4) $ stretch	1011 (HPO <sub>4</sub> <sup>2-</sup> CDHAp)		1011 (HPO <sub>4</sub> <sup>2-</sup> ) 1005 (HPO <sub>4</sub> <sup>2-</sup> )		988	
	941–961	951	966 (PO <sub>4</sub> <sup>3-</sup> ) 959 (PO <sub>4</sub> <sup>3-</sup> )	986		
	920 (CDHAp)		916 (apatite layer)			
ν(P— OH) stretch	879 (CDHAp)		874 (hydrated layer)	878	900	
	614		619 (PO <sub>3</sub> <sup>-</sup> )			
$v_4(PO_4)$ or $v_4(HPO_4)$ stretch	607 591 580	594	$609 (PO_4^{3-})$ $591 (PO_4^{3-} \text{ and } HPO_4^{2-})$ $577 (PO_4^{3-} \text{ and } HPO_4^{2-})$ $556 (PO_4^{3-})$ $523 (PO_4^{3-})$	588 525	588 574 563	
$\nu_2(PO_4)$	448 433	451	451 (PO <sub>4</sub> <sup>3-</sup> ) 427 (PO <sub>4</sub> <sup>3-</sup> )			
or $\nu_2(\text{HPO}_4)$ stretch		419	409 (HPO <sub>4</sub> <sup>3-</sup> ) 353 (HPO <sub>4</sub> <sup>2-</sup> hydrated layer)	411 381	420 394	

### S3 - XRF calibration traces



**Figure S3.** XRF calibration traces for the quantification of calcium (blue) and phosphorous (green). The dots represent the results of XRF analyses on beads of mechanical reference mixtures (see *Materials and Methods*). The dotted lines are the linear regressions used in the study to determine the Ca/P ratios of the synthesized CaPs.

## S4 - Apparent solubility of CaPs

The phosphate and hydrogen phosphate groups constitutive of CaPs confer them a solubility in aqueous medium that is greatly influenced by the pH of the latter. The apparent solubility product  $K_{s,app}^{CaP}$  associated with each calcium phosphate mineral is considered in order to analyze the influence of the acid-base equilibria on their dissolution in water. This temperature-dependent parameter is expressed from the solubility constants  $K_s^{CaP}$  of the CaPs and the acidity constants  $K_{ai}$  associated with the various phosphate species (Table 1).

In the case of the HAp crystalline phase, the dissolution-precipitation equilibrium in a buffered medium can be defined as:

$$Ca_{5}(PO_{4})_{3}OH_{(s)} + (9x + 6y + 3z + q)H_{3}O_{(aq)}^{+} = 5 Ca_{(aq)}^{2+} + 3x H_{3}PO_{4(aq)} + 3y H_{2}PO_{4(aq)}^{-} + 3z HPO_{4(aq)}^{2-} + 3n PO_{4(aq)}^{3-} + (1 - q) HO_{(aq)}^{-} + (9x + 6y + 3z + 2q) H_{2}O_{(l)}$$
 (Eq. S3)

where x, y, z and n stand for the molar fractions associated with  $H_3PO_4$ ,  $H_2PO_4^-$ ,  $HPO_4^{2-}$  and  $PO_4^{3-}$  entities, respectively, and q and 1-q the molar fractions of oxonium and hydroxide ions,

respectively. The dissolution equilibrium is associated with an apparent equilibrium constant  $K_{s,app}^{HAp}$  which can be expressed according to:

$$K_{s, app}^{HAp} = \frac{K_s^{HAp}}{K_{a1}^{3x} K_{a2}^{3(x+y)} K_{a3}^{3(x+y+z)} K_w^q}$$
 (Eq. S4)

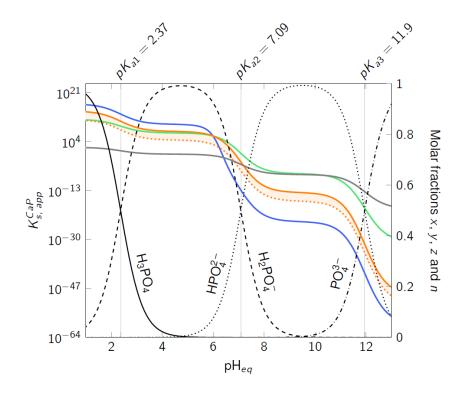
where  $K_w$  is the autoprotolysis constant of water. Detailed demonstration for obtention of Eq. S4 is provided below. Due to the dependence of the molar fractions x, y, z, n and q on the pH at the equilibrium (Figure S4A and Eqs. S16–S19), the apparent solubility product is also a function of the latter, *i.e.*  $K_{s,app}^{HAp} = f(pH_{eq})$ . Hence, as commonly written in a strongly basic medium where  $x \simeq y \simeq z \simeq q \simeq 0$  and  $n \simeq 1$  (Figure S4A and Figure S4B), the dissolution equilibrium comes down to:

$$Ca_5(PO_4)_3OH_{(s)} = 5 Ca_{(aq)}^{2+} + 3 PO_{4(aq)}^{3-} + HO_{(aq)}^{-}$$
 (Eq. S5)

with

$$K_{s,app}^{HAp} = K_s^{HAp} = a(\text{Ca}^{2+})_{eq}^5 a(\text{PO}_4^{3-})_{eq}^3 a(\text{HO}^{-})_{eq}$$
 (Eq. S6)

where  $a(i)_{eq}$  is the activity of the species i at the equilibrium and  $K_s^{HAp} = 10^{-58.6}$  at 37 °C.<sup>7</sup> Likewise, the pH dependence of the apparent solubility product can be expressed for each CaPs as developed below. For all of them, it appears that the lower the pH of the medium solution at the equilibrium, the higher the apparent solubility (Figure S4A). Such a significant increase in the solubility of CaPs in an acidic medium is accounted for by the speciation diagram of the phosphate species in solution. As the pH decreases, the protonation of the  $PO_4^{3-}$  and  $HPO_4^{2-}$  groups of the CaPs, into  $HPO_4^{2-}$ ,  $H_2PO_4^{-}$  and  $H_3PO_4$  species, leads to a shift of the dissolution-precipitation equilibria towards the dissolution of the CaPs (Figure S4A). Although the  $K_{s,app}^{HAp}$  parameter is dependent on the temperature of the medium (Table 1), its evolution as a function of the pH<sub>eq</sub>, is only slightly influenced in the 37–80 °C temperature range at atmospheric pressure (Figures S4A and S4B). This result reflects the weak influence of this parameter on the stability of CaPs in aqueous solution.



**Figure S4A.** Evolution of the apparent solubility product  $K_{s,app}^{CaP}$  of HAp, ACP, OCP and DCPA as a function of the pH at the equilibrium at 80 °C. As the DCPD trace is almost superimposed with that of DCPA, it is not plotted in the figure for a readability reason. For OCP, due to the important uncertainty on the value of its solubility product at 80 °C (Table 1), the corresponding domain is highlighted in orange. The speciation of the phosphate species in aqueous solution at 80 °C:  $H_3PO_4$  (x, solid line),  $H_2PO_4^-$  (y, dashed line),  $HPO_4^{2-}$  (z, dotted line) and  $PO_4^{3-}$  (n, dotdashed line).

In the acidic domain, the DCPA mineral is the CaP with the lowest apparent solubility product, whereas HAp has the highest one (Figure S4A). The reverse situation is observed in the alkaline domain, with DCPA and HAp becoming the most and the least soluble CaPs, respectively. These observations are fully consistent with the numerous protocols reported in the literature that report on the formation of DCPA and HAp under acidic and neutral/basic conditions, respectively.<sup>8,9</sup> However, the  $K_{s,app}^{CaP}$  parameter is not fully relevant to predict which CaP mineral will thermodynamically precipitate first under realistic operating synthesis conditions. In that respect, at a given pH in open systems (see *Materials and Methods*), the Ca/P ratio of these compounds (Table 1), varying from 1.00 (DCPD and DCPA) to 1.67 (HAp), needs to be taken into account.

The apparent solubility product  $K_{s,app}^{CaP}$  of a specified CaP mineral was determined based on its solubility constant  $K_s^{CaP}$  and the equilibrium constants involving phosphate species  $(K_{ai})$  and

water  $(K_w)$  defined on Table 1. The computational method used is detailed in the case of hydroxyapatite. This mineral solubilizes in water according to Eq. S5.

Phosphate ions  $PO_4^{3-}$  are involved in the acid-base equilibrium:

$$HPO_{4(aq)}^{2-} + H_2O_{(l)} = PO_{4(aq)}^{3-} + H_3O_{(aq)}^{+}$$
 (Eq. S7)

with

$$K_{a3} = \frac{a(PO_4^{3-})_{eq}a(H_3O^+)_{eq}}{a(HPO_4^{2-})_{eq}}$$
(Eq. S8)

The protonation of PO<sub>4</sub><sup>3-</sup> ions allows the involvement of two other acid-base equilibria:

$$H_2PO_{4(aq)}^- + H_2O_{(l)} = HPO_{4(aq)}^{2-} + H_3O_{(aq)}^+$$
 (Eq. S9)

with

$$K_{a2} = \frac{a(\text{HPO}_4^{2-})_{eq}a(\text{H}_3\text{O}^+)_{eq}}{a(\text{H}_2\text{PO}_4^-)_{eq}}$$
(Eq. S10)

and

$$H_3PO_{4(aq)} + H_2O_{(l)} = H_2PO_{4(aq)}^- + H_3O_{(aq)}^+$$
 (Eq. S11)

with

$$K_{a1} = \frac{a(H_2PO_4^-)_{eq}a(H_3O^+)_{eq}}{a(H_3PO_4)_{eq}}$$
 (Eq. S12)

Hydroxide ions are involved in the autoprotolysis equilibrium of water:

$$2 H_2 O_{(l)} = HO_{(aq)}^- + H_3 O_{(aq)}^+$$
 (Eq. S13)

with

$$K_w = a(H_3O^+)_{eq}a(HO^-)_{eq}$$
 (Eq. S14)

No acid-base/complexation equilibrium involving Ca<sup>2+</sup>cations was considered.

The apparent equilibrium constant  $K_{s,app}^{HAp}$  can be expressed based on Eq. S4 as a function of the activity of the different species:

$$K_{s,app}^{HAp} = \frac{a(\text{Ca}^{2+})_{eq}^{5}a(\text{PO}_{4}^{3-})_{eq}^{3}a(\text{HO}^{-})_{eq}}{K_{a1}^{3x}K_{a2}^{3(x+y)}\left(\frac{a(\text{PO}_{4}^{3-})_{eq}a(\text{H}_{3}\text{O}^{+})_{eq}}{a(\text{HPO}_{4}^{2-})_{eq}}\right)^{3(x+y+z)}\left(a(\text{H}_{3}\text{O}^{+})_{eq}a(\text{HO}^{-})_{eq}\right)^{q}} \\ K_{s,app}^{HAp} = \frac{a(\text{Ca}^{2+})_{eq}^{5}a(\text{PO}_{4}^{3-})_{eq}^{3}a(\text{HO}^{-})_{eq}^{1-q}}{K_{a1}^{3x}\left(\frac{a(\text{HPO}_{4}^{2-})_{eq}a(\text{H}_{3}\text{O}^{+})_{eq}}{a(\text{H}_{2}\text{PO}_{4}^{-})_{eq}}\right)^{3(x+y)}\left(\frac{a(\text{PO}_{4}^{3-})_{eq}}{a(\text{HPO}_{4}^{2-})_{eq}}\right)^{3(x+y+z)}}{a(\text{H}_{3}\text{O}^{+})_{eq}^{q+3(x+y+z)}}$$

$$K_{s,app}^{HAp} = \frac{a(\text{Ca}^{2+})_{eq}^{5}a(\text{PO}_{4}^{3-})_{eq}^{3}a(\text{HO}^{-})_{eq}^{1-q}a(\text{HPO}_{4}^{2-})_{eq}^{3z}}{K_{a1}^{3x}\left(\frac{1}{a(\text{H}_{2}\text{PO}_{4}^{-})_{eq}}\right)^{3(x+y)}a(\text{HO}_{4}^{3-})_{eq}^{3z}a(\text{HO}_{4}^{3-})_{eq}^{4-6(x+y)+3z}}$$

The speciation of phosphate species (Figures S4A and S4B) implies that at any pH the relation x + y + z + n = 1 is verified. The last equation simplifies as:

$$K_{s,app}^{HAp} = \frac{a(\text{Ca}^{2+})_{eq}^{5} a(\text{HO}^{-})_{eq}^{1-q} a(\text{PO}_{4}^{3-})_{eq}^{3n} a(\text{HPO}_{4}^{2-})_{eq}^{3z}}{\left(\frac{a(\text{H}_{2}\text{PO}_{4}^{-})_{eq} a(\text{H}_{3}\text{O}^{+})_{eq}}{a(\text{H}_{3}\text{PO}_{4})_{eq}}\right)^{3x} \left(\frac{1}{a(\text{H}_{2}\text{PO}_{4}^{-})_{eq}}\right)^{3(x+y)} a(\text{H}_{3}\text{O}^{+})_{eq}^{q+6(x+y)+3z}}$$

$$K_{s,app}^{HAp} = \frac{a(\text{Ca}^{2+})_{eq}^{5} a(\text{HO}^{-})_{eq}^{1-q} a(\text{PO}_{4}^{3-})_{eq}^{3n} a(\text{HPO}_{4}^{2-})_{eq}^{3z} a(\text{H}_{2}\text{PO}_{4}^{-})_{eq}^{3y}}{\left(\frac{1}{a(\text{H}_{3}\text{PO}_{4})_{eq}}\right)^{3x} a(\text{H}_{3}\text{O}^{+})_{eq}^{q+9x+6y+3z}}$$

Finally, the following equation is obtained:

$$K_{s,app}^{HAp} = \frac{a(\text{Ca}^{2+})_{eq}^{5} a(\text{HO}^{-})_{eq}^{1-q} a(\text{PO}_{4}^{3-})_{eq}^{3n} a(\text{HPO}_{4}^{2-})_{eq}^{3z} a(\text{H}_{2}\text{PO}_{4}^{-})_{eq}^{3y} a(\text{H}_{3}\text{PO}_{4}^{-})_{eq}^{3x}}{a(\text{H}_{3}\text{O}^{+})_{eq}^{q+9x+6y+3z}}$$
(Eq. S15)

which accounts for the general dissolution equilibrium of HAp in a buffered medium (Eq. S3).

The parameters x, y, z, n and q are dependent on the pH at the equilibrium (Figures S4A and S4B) according to:

$$x = \frac{[H_3PO_4]_{eq}}{[P]_{tot}} \frac{1}{1 + \frac{K_{a1}}{10^{-pH_{eq}}} + \frac{K_{a1}K_{a2}}{10^{-2pH_{eq}}} + \frac{K_{a1}K_{a2}K_{a3}}{10^{-3pH_{eq}}}}$$
(Eq. S16)

$$y = \frac{[H_2PO_4^-]_{eq}}{[P]_{tot}} = \frac{\frac{K_{a1}}{10^{-pH_{eq}}}}{1 + \frac{K_{a1}}{10^{-pH_{eq}}} + \frac{K_{a1}K_{a2}}{10^{-2pH_{eq}}} + \frac{K_{a1}K_{a2}K_{a3}}{10^{-3pH_{eq}}}}$$
(Eq. S17)

$$z = \frac{[\text{HPO}_4^{2-}]_{eq}}{[P]_{tot}} = \frac{\frac{K_{a1}K_{a2}}{10^{-2pH_{eq}}}}{1 + \frac{K_{a1}}{10^{-pH_{eq}}} + \frac{K_{a1}K_{a2}}{10^{-2pH_{eq}}} + \frac{K_{a1}K_{a2}K_{a3}}{10^{-3pH_{eq}}}}$$
(Eq. S18)

$$n = \frac{[PO_4^{3-}]_{eq}}{[P]_{tot}} = \frac{\frac{K_{a1}K_{a2}K_{a3}}{10^{-3pH_{eq}}}}{1 + \frac{K_{a1}}{10^{-pH_{eq}}} + \frac{K_{a1}K_{a2}}{10^{-2pH_{eq}}} + \frac{K_{a1}K_{a2}K_{a3}}{10^{-3pH_{eq}}}}$$
(Eq. S19)

$$q = \frac{10^{-pH_{eq}}}{10^{-pH_{eq}} + 10^{pH_{eq} + pK_w}}$$
 (Eq. S20)

with  $[P]_{tot} = [H_3PO_4]_{eq} + [H_2PO_4^-]_{eq} + [HPO_4^{2-}]_{eq} + [PO_4^{3-}]_{eq}$  the total concentration in phosphate species.

The calculation of the apparent solubility product associated with the HAp mineral accounts for the influence of the pH on its solubility as  $K_{s,app}^{HAp} = f(x, y, z, n, q) = f(pH_{eq})$ .

Similarly, the apparent solubility products associated with the other calcium phosphates (ACP, OCP, DCPD and DCPA) were determined. In the case of ACP, the dissolution equilibrium in a buffered medium is defined as:

$$\operatorname{Ca}_{3}(\operatorname{PO}_{4})_{2(s)} + (6x + 4y + 2z)\operatorname{H}_{3}\operatorname{O}_{(aq)}^{+} = 3\operatorname{Ca}_{(aq)}^{2+} + 2x\operatorname{H}_{3}\operatorname{PO}_{4(aq)} + 2y\operatorname{H}_{2}\operatorname{PO}_{4(aq)}^{-}$$

$$+ 2z\operatorname{HPO}_{4(aq)}^{2-} + 2n\operatorname{PO}_{4(aq)}^{3-} + (6x + 4y + 2z)\operatorname{H}_{2}\operatorname{O}_{(1)}$$
(Eq. S21)

It is associated with an apparent equilibrium constant  $K_{s, app}^{ACP}$ :

$$K_{s, app}^{ACP} = \frac{K_s^{ACP}}{K_{a1}^{2x} K_{a2}^{2(x+y)} K_{a2}^{2(x+y+z)}}$$
 (Eq. S22)

In the case of OCP, the dissolution equilibrium in a buffered medium is defined as:

$$\operatorname{Ca_4HPO_4(PO_4)_2} \cdot \frac{5}{2} \operatorname{H_2O_{(s)}} + (9x + 6y + 3z - 1) \operatorname{H_3O_{(aq)}^+} = 4 \operatorname{Ca_{(aq)}^{2+}} + 3x \operatorname{H_3PO_{4(aq)}}$$

$$+ 3y \operatorname{H_2PO_{4(aq)}^-} + 3z \operatorname{HPO_{4(aq)}^{2-}} + 3n \operatorname{PO_{4(aq)}^{3-}} + (9x + 6y + 3z + \frac{3}{2}) \operatorname{H_2O_{(l)}} \quad \text{(Eq. S23)}$$

It is associated with an apparent equilibrium constant  $K_{s, app}^{OCP}$ :

$$K_{s,app}^{OCP} = \frac{K_s^{OCP}}{K_{a1}^{3x} K_{a2}^{3(x+y)} K_{a3}^{3(x+y+z-1)}}$$
 (Eq. S24)

In the case of DCPD, the dissolution equilibrium in a buffered medium is defined as:

$$CaHPO_4 \cdot 2H_2O_{(s)} + (2x + y - n - 1)H_3O_{(aq)}^+ = Ca_{(aq)}^{2+} + x H_3PO_{4(aq)}$$
$$+ y H_2PO_{4(aq)}^- + z HPO_{4(aq)}^{2-} + n PO_{4(aq)}^{3-} + (2x + y - n + 2) H_2O_{(l)}$$
(Eq. S25)

It is associated with an apparent equilibrium constant  $K_{s, app}^{DCPD}$ :

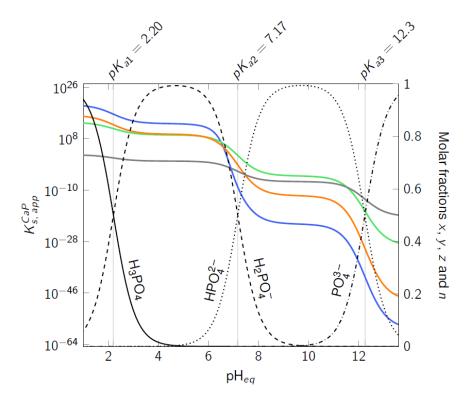
$$K_{s,app}^{DCPD} = \frac{K_s^{DCPD} K_{a3}^n}{K_{a1}^x K_{a2}^{x+y}}$$
 (Eq. S26)

Finally, in the case of DCPA, the dissolution equilibrium in a buffered medium is defined as:

$$CaHPO_{4(s)} + (2x + y - n - 1)H_3O_{(aq)}^+ = Ca_{(aq)}^{2+} + x H_3PO_{4(aq)}$$
$$+ y H_2PO_{4(aq)}^- + z HPO_{4(aq)}^{2-} + n PO_{4(aq)}^{3-} + (2x + y - n) H_2O_{(l)}$$
 (Eq. S27)

It is associated with an apparent equilibrium constant  $K_{s, app}^{DCPA}$ :

$$K_{s, app}^{DCPA} = \frac{K_s^{DCPA} K_{a3}^n}{K_{a1}^x K_{a2}^{x+y}}$$
 (Eq. S28)



**Figure S4B.** Evolution of the apparent solubility product  $K_{s,app}^{CaP}$  of HAp, ACP, OCP and DCPA as a function of the pH<sub>eq</sub> at 37 °C. As the DCPD trace is almost superimposed with that of DCPA, it has not been plotted for a readability reason. Speciation of phosphate species in aqueous solution at 37 °C: H<sub>3</sub>PO<sub>4</sub> (x, solid line), H<sub>2</sub>PO<sub>4</sub> (x, dashed line), HPO<sub>4</sub> (x, dotted line) and PO<sub>4</sub> (x, dotted line).

## S5 - Precipitation conditions of CaPs for the Ca $\rightarrow$ P and P $\rightarrow$ Ca synthesis routes

The thermodynamic approach developed in section I Thermodynamic modelling of the precipitation of calcium phosphates in open systems for the HAp material has been applied to predict the conditions for the precipitation of ACP, OCP, DCPD and DCPA. Note that the criterion  $Q \ge K_s$  (thermodynamic driving force) that was taken to estimate the precipitation conditions is the same as that commonly reported as  $S \ge 1$ , where S is the supersaturation of the solution. Indeed, for an ionic solid of stoichiometry v, these different parameters are linked according to equation S29:

$$S = \left(\frac{Q}{K_s}\right)^{\frac{1}{\nu}}$$
 (Eq. S29)

Following the  $Ca \rightarrow P$  synthesis route, the different CaPs precipitate spontaneously at the beginning of the addition step if the concentration  $[Ca^{2+}]_{min}^{CaP}$  introduced into the reactor verifies:

$$[Ca^{2+}]_{min}^{ACP} = \frac{1}{\gamma_{2\pm}} \left( \frac{K_s^{ACP}(c^{\circ})^5}{\gamma_{3\pm}^2 [PO_4^{3-}]_r^2} \right)^{\frac{1}{3}}$$
 (Eq. S30)

$$[Ca^{2+}]_{min}^{OCP} = \frac{1}{\gamma_{2+}} \left( \frac{K_s^{OCP}(c^{\circ})^8}{\gamma_{3+}^3 [PO_4^{3-}]_r^3 \gamma_+ [H_3 O^+]_r} \right)^{\frac{1}{4}}$$
 (Eq. S31)

$$[Ca^{2+}]_{min}^{DCPD} = \frac{K_s^{DCPD}(c^{\circ})^2}{\gamma_{2+}^2[HPO_4^{2-}]_r}$$
 (Eq. S32)

$$[\text{Ca}^{2+}]_{min}^{DCPA} = \frac{K_s^{DCPD}(c^{\circ})^2}{\gamma_{2\pm}^2[\text{HPO}_4^{2-}]_r}$$
 (Eq. S33)

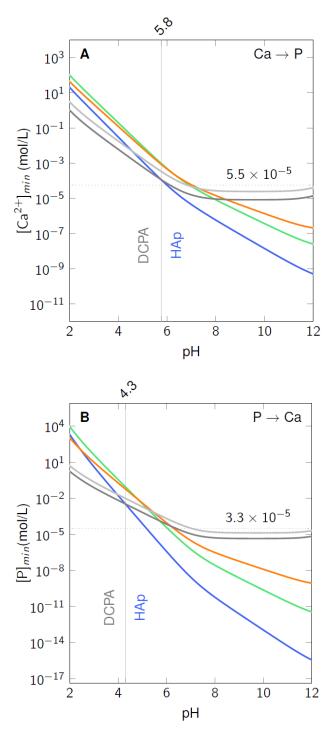
Following the  $P \to Ca$  synthesis route, the different CaPs precipitate spontaneously at the beginning of the addition step if the concentration  $[P]_{min}^{CaP}$  introduced into the reactor verifies:

$$[P]_{min}^{ACP} = \frac{1}{n\gamma_{3\pm}} \left( \frac{K_s^{ACP}(c^{\circ})^5}{\gamma_{2\pm}^3 [Ca^{2\pm}]_r^3} \right)^{\frac{1}{2}}$$
 (Eq. S34)

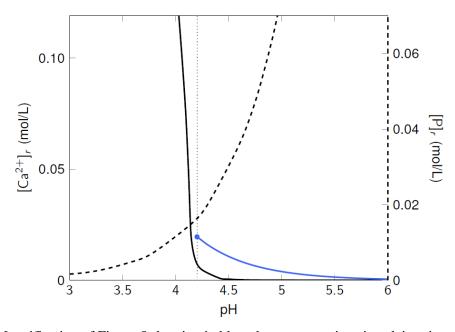
$$[P]_{min}^{OCP} = \frac{1}{n\gamma_{3\pm}} \left( \frac{K_s^{OCP}(c^\circ)^8}{\gamma_{2\pm}^3 [Ca^{2+}]_r^4 \gamma_{\pm} [H_3 O^+]_r} \right)^{\frac{1}{3}}$$
 (Eq. S35)

$$[P]_{min}^{DCPD} = \frac{K_s^{DCPD}(c^{\circ})^2}{z\gamma_{2\pm}^2[Ca^{2+}]_r}$$
 (Eq. S36)

$$[P]_{min}^{DCPA} = \frac{K_s^{DCPA}(c^{\circ})^2}{z\gamma_{2+}^2[Ca^{2+}]_r}$$
 (Eq. S37)



**Figure S5A.** Thermodynamic traces predicting the minimum concentration in precursor to be initially introduced into the reactor for both Ca  $\rightarrow$  P (**A**) and P  $\rightarrow$  Ca (**B**) routes in order to precipitate the different calcium phosphates HAp, ACP, OCP, DCPD and DCPA at 37 °C (modelling parameters: [P]<sub>r</sub> = 0.13 mol/L, and [Ca<sup>2+</sup>]<sub>r</sub> = 0.22 mol/L). Specified concentrations of [Ca<sup>2+</sup>] = 5.5 × 10<sup>-5</sup> mol/L and [P] =  $3.3 \times 10^{-5}$  mol/L are those estimated in the reactor after one drop of the corresponding precursor is added (see *Materials and Methods*).



**Figure S5B.** Magnification of Figure 5 showing in blue, the concentrations in calcium ions and phosphate species at the end of the addition step considering a targeted final ratio  $[Ca^{2+}]/[P]$  of 1.67 and assuming that the thermodynamic equilibrium is achieved. Note that the more acidic the reaction medium at the equilibrium, the greater the concentration of  $Ca^{2+}$  and  $H_xPO_4^{(3-x)-}$  ions as discussed in Section I.

## S6 - Low-angle diffractograms of HAp samples prepared at pH 6.5

The diffractograms of the samples prepared at pH 6.5 were acquired at low-angles to ascertain the absence of OCP in the corresponding samples (Figure S6).

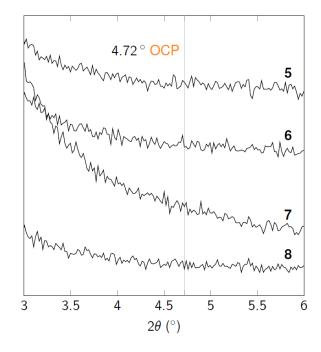
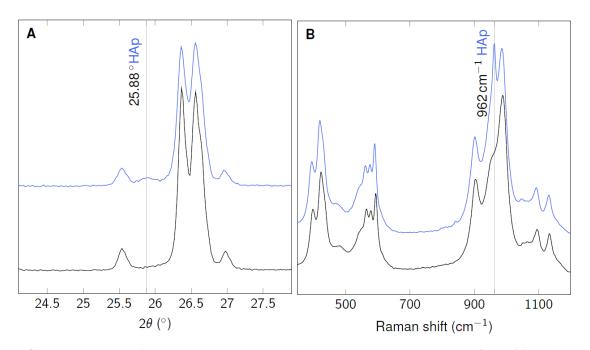


Figure S6. Low-angle XRD patterns of samples 5, 6, 7 and 8 prepared at pH 6.5 (Table 2).

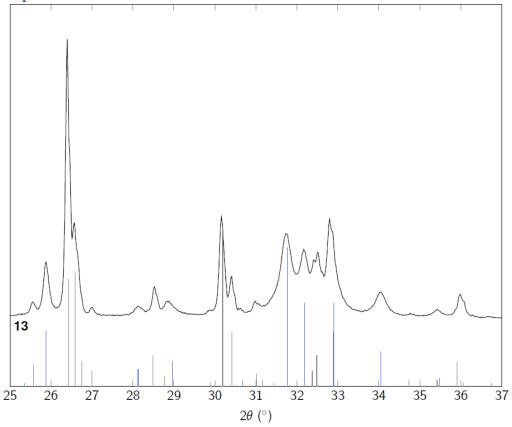
## S7 - DCPA hydrolysis into CDHAp during the washing step

Due to the slight acidity of the distilled water (pH  $\approx$  6.5) and the absence of phosphate ions, the DCPA formed in the reactor is susceptible to be hydrolyzed into HAp during the washing step (Figure S7). In that respect, the impact of the exposure time of DCPA to distilled water was investigated by comparing two procedures: centrifugation (precipitates exposed for  $\approx$  1 h to distilled water) and filtration on Büchner (precipitates exposed for  $\approx$  5 min to distilled water).



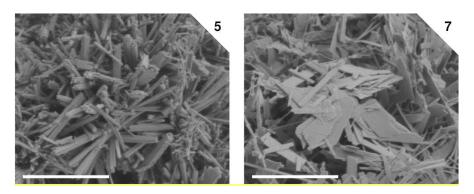
**Figure S7.** XRD patterns (**A**) and Raman spectra (**B**) of the samples prepared at pH 4.2 and 80 °C following the  $P \rightarrow Ca$  route after recovering and washing promptly with distilled water on Büchner (black line) or by centrifugation (blue line, sample 11). The longer exposure of the sample in a slightly acidic medium when recovered by centrifugation compared to filtration on Büchner allow for more pronounced hydrolysis of DCPA into (CD)HAp in the precipitates.

**S8 - Sample 13** 



**Figure S8.** XRD patterns of the sample 13 prepared at pH 4.2 and 80 °C following the Ca  $\rightarrow$  P synthesis route (Table 2) and position of the diffraction lines expected for DCPA (grey, ICDD reference card n° 00-009-0080) and for HAp (blue, ICDD reference card n° 00-009-043).

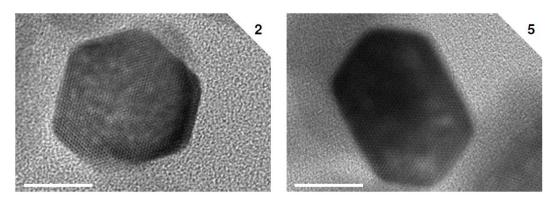
# S9 - High magnification images of rod-like and platelet/ribbon-like crystallites



**Figure S9.** Scanning electron microscopy images of samples 5 and 7 prepared at pH 6.5 following the Ca  $\rightarrow$  P and P  $\rightarrow$  Ca routes, respectively. The numbers indicated on the top right corners of the images refer to as the sample references listed in Table 2. Scale bars: 2  $\mu$ m.

### S10 - Hexagonal section of samples 2 and 5

Transmission electron microscopy (TEM) observations were done on ultrathin cuttings of crystallites in order to visualize hexagonal sections of few rods (Table 2, samples 2 and 5). The sections were prepared as follows: a few milligrams of powder were embedding in a resin. Polymerization of the resin took place at 70 °C for 48 h, then the polymerized blocks were cut with a diamond knife in slices (around 70 nm in thickness). These slices were then deposited on copper grids covered with a carbon membrane layer. TEM images were taken on a JEOL-JEM2100Plus electron microscope operating at 200 keV (LaB<sub>6</sub> gun). For samples 2 and 5, crystallites with a more or less distorted hexagonal section are observed (Figure S10).



**Figure S10.** TEM images of hexagonal sections of rods from samples 2 and 5 prepared according to the  $P \rightarrow Ca$  route at pH 9.0 and 6.5, respectively. The numbers indicated on the top right corners of the images refer to as the sample references listed in Table 2. Scale bars: 20 nm.

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